

BLOCK COPOLYMERS WITH SURFACE-ACTIVE JUNCTION GROUPS, COMPOSITIONS AND PROCESSES THEREOF

The invention relates to a novel block copolymer comprising a junction group between two blocks to which is attached at least one pendant group having a low surface energy, novel compositions comprising the novel block copolymer and processes using the novel composition for producing perpendicular domains (e.g. lamellae, cylinders etc.) formed by self-assembly (SA) or directed self-assembly (DSA) of the novel block copolymers (BCPs). The processes are useful for fabrication of electronic devices.

In conventional lithography approaches, ultraviolet (UV) radiation may be used to expose through a mask onto a photoresist layer coated on a substrate or layered substrate. Positive or negative photoresists are useful and these can also contain a refractory element such as silicon to enable dry development with conventional integrated circuit (IC) plasma processing. In a positive photoresist, UV radiation transmitted through a mask causes a photochemical reaction in the photoresist such that the exposed regions are removed with a developer solution or by conventional IC plasma processing. Conversely, in negative photoresists, UV radiation transmitted through a mask causes the regions exposed to radiation to become less removable with a developer solution or by conventional IC plasma processing. An integrated circuit feature, such as a gate, via or interconnect, is then etched into the substrate or layered substrate, and the remaining photoresist is removed. When using conventional lithographic exposure processes, the dimensions of features of the integrated circuit feature are limited. Further reduction in pattern dimensions are difficult to achieve with radiation exposure due to limitations related to aberrations, focus, proximity effects, minimum achievable exposure wavelengths and maximum achievable numerical apertures. Directed self-assembly is a promising approach which has been of interest in overcoming some of the drawback of conventional lithography as outlined above.

Specifically, directed self-assembly of block copolymers is a method useful for generating very small patterned features for the manufacture of microelectronic devices in which the critical dimensions (CD) of features usually on the order of nano scale ranging in feature size from 10 nm to 50 nm can be achieved. Achieving feature sizes below 10 nm using conventional approaches for directed self-assembly of block copolymers is challenging. Directed self-assembly methods such as those based on graphoepitaxy and chemical epitaxy of block copolymers are desirable for extending the resolution capabilities of lithographic technology.

These techniques can be employed to either enhance conventional lithographic techniques by enabling the generation of pattern with higher resolution and/ or improving CD control for EUV, e-beam, deep UV or immersion lithography. The directed self-assembly block copolymer comprises a block of etch resistant polymeric unit and a block of highly etchable polymeric unit, which when coated, aligned and etched on a substrate give regions of high resolution patterns.

Known examples of block copolymers suitable for directed self-assembly are ones capable of microphase separation and comprising a block rich in carbon (such as styrene or containing some other element like Si, Ge, and Ti) which is resistant to plasma etch, and a block which is highly plasma etchable or removable, which can provide a high resolution pattern definition. Examples of highly etchable blocks can comprise monomers which are rich in oxygen and which do not contain refractory elements, and are capable of forming blocks which are highly etchable, such as methyl methacrylate. The plasma etch gases used in the etching process of defining the self-assembly pattern typically are those used in processes to make integrated circuits (IC). In this manner very fine patterns can be created on typical IC substrates compared to conventional lithographic techniques, thus achieving pattern multiplication.

In the graphoepitaxy directed self-assembly method, the block copolymers self-organizes on a substrate that is pre-patterned with conventional lithography

(Ultraviolet, Deep UV, and e-beam, Extreme UV (EUV) exposure source) to form topographical features such as a line/space (L/S) or contact hole (CH) pattern. In an example of L/S directed self-assembly array, the block copolymer can form self-aligned lamellar regions with a sub-lithographic pitch in the trenches between sidewalls of pre-pattern, thus enhancing pattern resolution by subdividing the space in the trench between the topographical lines into finer patterns. Similarly, features such as contact holes can be made denser by using graphoepitaxy in which a suitable block copolymer arranges itself by directed self-assembly within an array of pre-patterned holes or pre-patterned posts defined by conventional lithography, thus forming a denser array of regions of etchable and etch resistant domains which when etched give rise to a denser array of contact holes. In addition, block copolymers can form a single and smaller etchable domain at the center of prepattern hole with proper dimension and provide potential shrink and rectification of the hole in prepattern. Consequently, graphoepitaxy has the potential to offer both pattern rectification and pattern multiplication.

In chemical epitaxy DSA methods, the self-assembly of the block copolymer occurs on a surface that has regions of differing chemical affinity but no or very slight topography to guide the self-assembly process. For example, the chemical prepattern could be fabricated using lithography (UV, Deep UV, e-beam, EUV) and nanofabrication process to create surfaces of different chemical affinity in a line and space (L/S) pattern. These areas may present little to no topographical difference, but do present a surface chemical pattern to direct self-assembly of block copolymer domains. This technique allows precise placement of these block copolymer domains of higher spatial frequency than the spatial frequency of the prepattern. The aligned block copolymer domains can be subsequently pattern transferred into an underlying substrate after plasma or wet etch processing. In addition, Chemical epitaxy has the advantage that the block copolymer self-assembly can rectify variations in the surface chemistry, dimensions, and roughness of the underlying chemical pattern to yield improved line-edge roughness and CD control in the final self-assembled block copolymer

domain pattern. Other types of patterns such as contact holes (CH) arrays could also be generated or rectified using chemoepitaxy.

The ability of a BCP to phase separate depends on the Flory Huggins interaction parameter (χ). PS-b-PMMA (poly(styrene-block-methyl methacrylate)) is the most promising candidate for directed self-assembly (DSA) applications. However, the minimum half-pitch of PS-b-PMMA is limited to about 10 nm because of lower interaction parameter (χ) between PS and PMMA. To enable further feature miniaturization, a block copolymer with a larger interaction parameter between two blocks (higher χ) is highly desirable.

For lithography applications, orientation of the block copolymer domains perpendicular to the substrate is desirable. For a conventional block copolymer such as PS-b-PMMA in which both blocks have similar surface energies at the BCP-air interface, this can be achieved by coating and thermally annealing the block copolymer on a layer of non-preferential or neutral material that is grafted or cross-linked at the polymer-substrate interface. Due to larger difference in the interaction parameter between the domains of higher- χ block copolymers, it is important to control both BCP-air and BCP-substrate interactions. Many orientation control strategies for generating perpendicularly oriented BCP domains have been implemented with higher- χ BCPs. For example, solvent vapor annealing has been used for orientation control of polystyrene-b-polyethylene oxide (PS-b-PEO), polystyrene-b-polydimethylsiloxane (PS-b-PDMS), polystyrene-b-poly(2-vinyl pyridine) (PS-b-P2VP), polylactide-b-poly(trimethylsilylstyrene) PLA-b-PTMSS and P α MS-b-PHOST. Introducing a solvent vapor chamber and kinetics of solvent vapor annealing may complicate DSA processing. Alternatively, the combination of neutral underlayers and topcoat materials has been applied to PS-b-P2VP, PS-b-PTMSS and PLA-b-PTMSS to achieve perpendicular orientation of the polymer domains. However, the additional topcoat materials may increase the process cost and complexity. Thus, there exists a need to have a topcoat free higher- χ BCP system using

simple thermal annealing on a range of preferential and non-preferential substrates.

The present invention relates to novel block copolymers with one or more low-surface energy groups at the junction of the two blocks. The invention further relates to a composition comprising the novel block copolymer, and also relates to novel processes comprising coating the film comprising the novel block copolymer, where the process allows for the perpendicular orientation of BCP domains in thin films (5-100 nm) by simple thermal annealing over proper underlayer.

BRIEF DESCRIPTION OF DRAWINGS

Figures 1a and 1b

Structures of block copolymers with a surface active junction group having structures (6)-(11) and wherein the junction group is comprised of an oxyalkylenecarbonylalkylene moiety.

Figure 2

Structures of block copolymers with surface active junction group having structure (12) and (14) derived from a 1,1-diphenylethylene (DPE) derivative.

Figure 3

Structures of block copolymer with surface active junction group having structures (20), (21a) (21b) wherein the junction group is derived from an aryl substituted alkyleneoxy moiety.

Figure 4

a) Schematic depiction of block copolymer of structure (1) and b) schematic depiction of self-assembly of polymer domains.

Figure 5

Schematic depiction of directed self-assembly (DSA) of the novel block copolymer of structure (1) a) shows graphoepitaxy DSA where most of the novel block copolymer is confined within the recessed region in the topographical prepatter, b) shows coating of this novel block copolymer over and within shallow, periodic topographical prepatter where the domains of this block copolymer are aligned to the topographical prepatter c) shows chemoepitaxial DSA where a film of this novel block copolymer is coated over the chemical prepatter comprised of preferential and non-preferential regions.

Figure 6

Figures 6a-6m are atomic force microscopy (AFM) and scanning electron micrograph (SEM) images at different magnifications of the self-assembled block copolymer film formed with Examples C7-C35 (comparative examples).

Figure 7

Figures 7a-7yy are AFM and SEM images at different magnifications of the self-assembled block copolymer film formed with Examples 97-118. Each image is labelled with the corresponding Example number.

Figure 8

Graphical representation for relationship of pitch v/s blends of Example 21 and 23.

Figure 9

Graphoepitaxy directed self-assembly (DSA) figures of Example 119a (Figure 9a1, 9a2); Example 120 (Figure 9b-1, 9b-2), and Example 122 (Figure 9c-1, 9c-2).

Figure 10

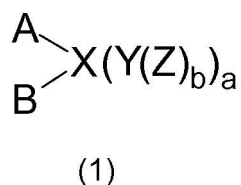
Prepattern SEM and overcoating DSA figures of AZ® Electronic Materials AZEMBLY™ NLD-089)/SiARC A940 Prepattern (Figure 10a) Example 123 (Figures 10b-1 and 10b-2), Example 124(Figure 10c-1, and 10c-2), and Example 125 (Figure 10d).

Figure 11

Chemoepitaxy directed self-assembly (DSA) figures of Example 126a (Figure 11a), Example 126b (Figure 11b), Example 127a(Figure 11c), Example 127d(Figure 11d), Example 128(Figure 11e) and Example 129(Figure 11f).

SUMMARY OF INVENTION

The invention relates to a novel block copolymer with a junction group having structure (1),



wherein,

A -is a block polymer chain,

B is a block polymer chain,

wherein, A- and B- are chemically different, covalently connected polymer chains, which are phase separable,

the moiety $\text{X}(\text{Y}(\text{Z})_b)_a$ is a junction group, which comprises a surface active pendant moiety $\text{Y}(\text{Z})_b$ wherein;

a is an integer from 1 to 4 denoting the number of surface active pendant moieties $\text{Y}(\text{Z})_b$ on X,

b is an integer from 1 to 5 denoting the number of Z moieties on the linking moiety Y,

X is a linking group between the A polymer block, the B polymer block and the moiety Y,

Y is a linking group or a direct valence bond between X and Z; and

Z is a moiety independently selected from, a fluorine containing moiety, a Si₁-Si₈ siloxane containing moiety or a hydrocarbon moiety with at least 18 carbons,

and further wherein the junction group X(Y(Z)_b)_a has a surface energy less than that of the block A and less than that of the block B.

The invention also relates to a composition comprised of the novel block copolymer and a solvent, and also relates to the process of using this composition in self-assembly and directed self-assembly processes.

DETAILED DESCRIPTION OF THE INVENTION

As used herein, unless otherwise indicated, aryl refers to an aromatic moiety with one attachment point (e.g. phenyl, anthracyl, naphthyl and the like). Apart from the attachment point, aryl group may be substituted with alkyl groups, aryl groups or halides (e.g. F, Cl, I, Br). Arylene refers to an aromatic moiety with more than 1 attachment point. Apart from the attachment points, arylene group may be substituted with alkyl groups, or halides (e.g. F, Cl, I, Br) or aryl groups. Alkyl refers to a linear, branched or cyclic alkane moiety with one attachment point (e.g. methyl, ethyl, propyl, n-butyl, tert-butyl, cyclohexyl and the like). Apart from the attachment point, alkyl groups may be substituted with alkyl groups, aryl groups or halides (e.g. F, Cl, I, Br). Alkylene refers to a linear, branched or cyclic alkane moiety with more than one attachment point. Apart from the attachment point, alkylene groups may be substituted with alkyl groups, aryl groups, or halides (e.g. F, Cl, I, Br). A fluoro moiety designates a fluoro group (i.e. F) or a partially or fully fluorinated hydrocarbon group where the hydrocarbon group is an aromatic moiety, an aliphatic moiety, or mixtures of

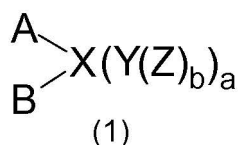
these, where these moieties may also contain heteroatoms other than fluorine as pendant group or as part of a chain or cyclic structure. Fluoroaryl, fluoroarylene, fluoroalkyl, fluoroalkylene refers to these moieties as described above which are partially or completely fluorinated. Nitrogen containing heterocyclic moieties refers to 5-8 membered cyclic hydrocarbons containing one or more nitrogen atoms within the ring, the rings may be fully saturated or contain unsaturated moieties (e.g. nitrogen containing hetero cyclic compounds, such as pyrrolidine, pyrrole, imidazolidine, imidazole, 1H-1,2,3-triazole, 2H-1,2,3-triazole, 1H-1,2,4-Triazole, 4H-1,2,4-Triazole, pyridine and the like). Oxygen containing heterocyclic moieties refers to 5-8 membered cyclic hydrocarbons containing one or more oxygen atoms within the ring, (e.g. tetrahydrofuran, furan, oxetane, pyran, dioxolane, dioxane and the like). Sulfur containing heterocyclic moieties refers to 5-8 membered cyclic hydrocarbons containing one or more sulfur atoms within the ring, (e.g. thiophane, thiophene, thiane, thiopyran, dithiane, dithiine and the like). When heterocyclic moieties containing oxygen, sulfur or nitrogen are connected through another moiety the name of the moiety which preceded the term "...containing heterocyclic moiety) under parenthesis. For instance a moiety in which an oxygen containing heterocyclic moiety is attached through a oxy (-O-) moiety will be termed and "oxy-(oxygen containing heterocyclic moiety)".

In general, fluorine containing linear hydrocarbon group, fluorine containing branched hydrocarbon group, and fluorine containing cyclic hydrocarbon group refer respectively to linear, branched or cyclic partially or fully fluorinated alkyl group which may also contain alkene unsaturations, arylene, moieties or aryl substituents. Fluorine containing linear alkyl ether group, fluorine containing branched alkyl ether group, fluorine containing cyclic alkyl ether group, refer to linear, branched or cyclic ether are linear, branched or cyclic, partially or fully fluorinated, alkyl or alkyloxy moieties containing one or more ether moieties, which may also contain alkene unsaturations, arylene, moieties or aryl substituents.

In general, a Si₁-Si₈ siloxane moiety, when referring to the Z moiety part of pendant moiety Y(Z)_b moiety refers, to a moiety containing a (--Si(alkyl)₂-O)_n-alkyl species where the alkyl moiety may be a linear, branched or cyclic alkyl moiety and n designates the number of siloxane moieties in a chain where n=1 to 8. A linear hydrocarbon moiety with a Si₁-Si₈ siloxane pendant group, a branched hydrocarbon moiety with a Si₁-Si₈ siloxane pendant group, a cyclic hydrocarbon moiety with a Si₁-Si₈ siloxane pendant group, refer to respectively a linear, branched or cyclic alkylene moiety to which is attached a Si₁-Si₈ siloxane moiety as defined above. A carbosilane, refers to an alkyl moiety in which at least one carbon atom has been replaced by a silicon atom. A linear hydrocarbon moiety with a carbosilane pendant group, a branched hydrocarbon moiety with a carbosilane pendant group, and a cyclic hydrocarbon moiety with a carbosilane pendant group, refers to respectively a linear, branched, or cyclic alkylene moiety to which is attached a carbosilane moiety as described above. A silane refers to a Si_nH_{2n+1} moiety. A linear hydrocarbon moiety with a silane pendant group, a branched hydrocarbon moiety to which is attached a silane pendant group, and a cyclic hydrocarbon moiety refers to respectively refers to respectively a linear, branched, or cyclic alkylene moiety to which is attached with a silane pendant group. The term “-b-” refers to “-block-” and designates monomer repeat units that form block copolymer. The term “P” refers to “poly” when in front of a monomer acronym it designates the polymerized monomer (e.g. PS, designates poly(styrene) because S is the defined acronym for styrene). The term “JBCP” (Block copolymers with Junction with surface active pendant moiety) is an acronym which refers to the block copolymers of this invention with a junction group having structure (1) as defined herein that contain a surface active moiety. The term “Surface Active Junction Group” refers to the X(Y(Z)_b)_a moiety in structure (1) as defined herein. The term “Phase Separable” as used to describe polymer blocks in a block copolymer which are phase separable and refers to the property of these blocks when cast as a film to be able to phase separate from each other because of a mutual insolubility of the block domains with each other. This phase separability of the polymer block domains in a film may manifest itself

spontaneously in solution, or alternatively in the cast film or by heating of a cast film. When cast as a film the phase separability may also occur at room temperature if the polymer block polymer film is plasticized by solvent vapor. The term V_f refers to the volume fraction, and when further annotated with a subscript which denotes the acronym for a polymer repeat unit block (e.g. PS for polystyrene) denotes the volume fraction of this block (e.g. V_{fPS} denotes the volume fraction of the polystyrene block in a block copolymer containing this block). The term L_0 is the domain periodicity of a self-assembled pattern of a BCP.

The present invention relates to a novel block copolymer having structure (1), wherein the block copolymer comprises a diblock copolymer with a junction group,



wherein,

A is a block polymer chain,

B is a block polymer chain,

wherein, A and B are chemically different, covalently connected polymer chains, which are phase separable;

$\text{X}(\text{Y}(\text{Z})_b)_a$ is a moiety which is a junction group comprising a surface active pendant moiety $\text{Y}(\text{Z})_b$, wherein a is an integer from 1 to 4 denoting the number of surface active pendant moieties $\text{Y}(\text{Z})_b$ attached to X, b is an integer from 1 to 5 denoting the number of Z moieties attached to the moiety Y,

X is a linking group connecting the polymer block A, the polymer block B and the moiety Y, Y is a linking group or a direct valence bond between X and Z, and

Z is a moiety independently selected from a fluorine containing moiety, a $\text{Si}_1\text{-Si}_8$ siloxane containing moiety or a hydrocarbon moiety with at least 18 carbons, and

further wherein the junction group $X(Y(Z)_b)_a$ has a surface energy less than that of block A and less than that of block B.

Specific examples of the fluorine containing moiety a), the Si₁-Si₈ siloxane moiety b) and the hydrocarbon moiety with at least 18 carbons c) are where,

a) Z is a fluorine group, a linear hydrocarbon group containing fluorine, a fluorine containing branched hydrocarbon group, a fluorine containing cyclic hydrocarbon group, a fluorine containing linear alkyl ether group, a fluorine containing branched alkyl ether group, a fluorine containing cyclic alkyl ether group or a mixture thereof,

b) Z is a Si₁-Si₈ siloxane, a linear hydrocarbon moiety with a Si₁-Si₈ siloxane pendant group, a branched hydrocarbon moiety with a Si₁-Si₈ siloxane pendant group, a cyclic hydrocarbon moiety with a Si₁-Si₈ siloxane pendant group, a carbosilane, a linear hydrocarbon moiety with a carbosilane pendant group, a branched hydrocarbon moiety with a carbosilane pendant group, a cyclic hydrocarbon moiety with a carbosilane pendant group, a silane, a linear hydrocarbon moiety with a silane pendant group, a branched hydrocarbon moiety with a silane pendant group, a cyclic hydrocarbon moiety with a silane pendant group, or a mixture thereof, and

c) Z is a linear hydrocarbon group with at least 18 carbons, a branched hydrocarbon with at least 18 carbons; and,

In one aspect of this invention the junction moiety $X(Y(Z)_b)_a$ in structure (1) has a surface energy less than about 30 mN m⁻¹. In a more specific embodiment, the junction moiety has as surface energy between 30 mN m⁻¹ and 10 mN m⁻¹, preferably between 25 mN m⁻¹ and 14 mN m⁻¹.

Typically, the block copolymers of structure (1) suitable for being used in these inventions have a weight-averaged molecular weight (M_w) in the range of about 3,000 to about 500,000 g/mol and a number averaged molecular weight (M_n) of about 1,000 to about 60,000 and a polydispersity (M_w/M_n) (PD) of about 1.01 to about 6, or 1.01 to about 2 or 1.01 to about 1.5. Further embodiments are

wherein in structure (1), a is an integer from 1 or 2. Another embodiment is where b is an integer from 1 to 2. In another embodiment of the block copolymers having structure (1) the M_n ranges from 4,000 - 150,000 and the polydispersity (PD) ranges from 1.01 to 5.0, more preferably from 1.01 to 2.0.

In another embodiment of the present invention one of block A or block B comprises repeat units which are resistant to plasma etching techniques typically employed in manufacturing IC devices relative to the other block; and the other block either etches rapidly under these same plasma etch conditions or can be removed by chemical or photochemical processes. This property enables the pattern transfer of a self-assembled domain pattern into the substrate, when these materials are formulated with a solvent, and coated on a patterned substrate. If the self-assembly occurs on a patterned substrate it is directed self-assembly. The process of directed self-assembly of the novel block copolymer may, for instance, be made under a using a graphoepitaxy approach or a chemoepitaxy approach.

Another embodiment of the present invention is where block A in structure (1) is comprised of units derived from ethylenically unsaturated polymerizable monomers, or units derived from ring-opening polymerization (ROP) of cyclic monomers, and where block B is comprised of units derived from ethylenically unsaturated polymerizable monomers, or units derived from ring-opening polymerization (ROP) of cyclic monomers, and further wherein polymeric block A and B are different from each other and are phase separable.

Another aspect of this invention is where either block A or B is derived from a vinyl polymerizable monomer.

Another embodiment of structure (1) is where the linking group X in structure (1) is a moiety with 3 to 7 attachment points selected from a multivalent heteroatom, a multivalent group comprised of hetero atoms, a multivalent organic group or a multivalent organic group containing heteroatoms. More specific examples of X moieties are a C_1 - C_{20} linear alkylene moiety, a C_3 - C_{30} branched alkylene moiety, a C_3 - C_{30} cyclic alkylene moiety, a C_1 - C_{20} linear alkylene moiety

with a C₆-C₃₀ diaryl pendant groups, a C₃-C₃₀ branched alkylene moiety with a C₆-C₃₀ diaryl pendant group, a C₃-C₃₀ cyclic alkylene moiety with a C₆-C₃₀ diaryl pendant group, a C₁-C₃₀ linear alkyleneoxy moiety, a C₃-C₃₀ branched alkyleneoxy moiety, a C₁-C₂₀ urethane- linear alkylene moiety(-N(R₁₁)-C(=O)-O-linear alkylene), a C₃-C₃₀ urethane- branched alkylene moiety (-N(R₁₁)-C(=O)-O-branched alkylene moiety), a C₃-C₃₀ urethane- cyclic alkylene moiety (-N(R₉)-C(=O)- O-cyclic alkylene), a C₁-C₂₀ urea- linear alkylene moiety(-N(R₁₁)-C(=O)-N(R₁₁)-linear alkylene), a C₃-C₃₀ urea- branched alkylene moiety (-N(R₁₁)-C(=O)-N(R₁₁)-branched alkylene moiety), a C₃-C₃₀ urea- cyclic alkylene moiety (-N(R₁₁)-C(=O)- N(R₁₁)-cyclic alkylene), a C₁-C₂₀ thiourea-linear alkylene moiety(-N(R₁₁)-C(=S)-N(R₁₁)-linear alkylene), a C₃-C₃₀ thiourea-branched alkylene moiety (-N(R₁₁)-C(=S)-N(R₁₁)-branched alkylene moiety), a C₃-C₃₀thiourea- cyclic alkylene moiety (-N(R₁₁)-C(=S)- N(R₁₁)-cyclic alkylene), where R₁₁ is hydrogen or a C₁ to C₄ alkyl, a 1,2,3-triazole moiety, a 1,2,3-triazole C₁-C₃₀ alkylene moiety, a 1,2,3-triazole C₃-C₃₀ branched alkylene moiety, a 1,2,3-triazole C₃-C₃₀cyclic alkylene moiety, a 1,2,3-triazole C₁-C₃₀ alkylene oxy moiety, a 1,2,3-triazole C₃-C₃₀ branched alkyleneoxy moiety, a 1,2,3-triazole C₃-C₃₀cyclic alkyleneoxy moiety, a C₃-C₃₀ cyclic alkoxy moiety, a C₃-C₃₀ linear alkyleneoxycarbonyl moiety, a C₅-C₃₀ branched alkyleneoxycarbonyl moiety, a C₃-C₃₀ cyclic alkoxycarbonyl moiety, a C₁-C₃₀ linear fluoroalkylene moiety, a C₃-C₃₀ branched fluoroalkylene moiety, a C₆-C₃₀ cyclic fluoroalkylene moiety, a C₆-C₃₀ arylen moiety, a C₆-C₃₀ fluoroarylen moiety, a C₅-C₃₀ oxyalkyleneoxycarbonylalkylene moiety, a C₂-C₃₀ alkyleneoxyalkylene moiety, an aryl substituted C₈-C₄₂ alkyleneoxy moiety, an aryl substituted C₈-C₄₂ alkyleneoxyalkylene moiety, a C₈-C₄₂ aryl substituted alkylene moiety, a C₇-C₃₀ alkylenearylen moiety, a C₇-C₃₀ fluoroalkylenearylen moiety where R₉ is independently chosen from a C₁-C₄ alkyl group or hydrogen and combinations thereof.

A more specific embodiment of this aspect are when the X moiety is chosen from a C₅-C₃₀ oxyalkyleneoxycarbonylalkylene moiety, a C₂-C₃₀ alkyleneoxyalkylene moiety, an aryl substituted C₈-C₄₂ alkyleneoxy moiety, an

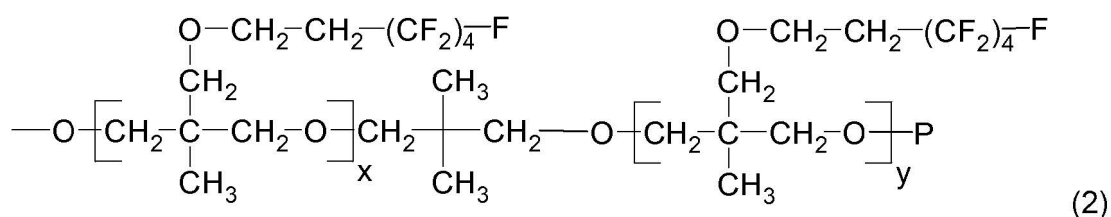
aryl substituted C₈-C₄₂ alkyleneoxyalkylene moiety, a C₈-C₄₂ aryl substituted alkylene moiety, or a C₇-C₃₀ alkylenearylene moiety.

A further embodiment of structure (1) is where the linking group Y in structure (1) is a direct valence bond, a multivalent heteroatom, a multivalent group comprised of hetero atoms, a multivalent organic group or a multivalent organic group containing heteroatoms. More specific examples of Y moieties are a direct valence bond, oxy (-O-), carboxyl (-C(=O)-) carbonyloxy (-C(=O)-O-), oxycarbonyl (-O-C(=O)-), carbonate (-O-C(=O)-O-), a C₁-C₂₀, linear alkylene, a C₃-C₂₀ branched alkylene, a C₃-C₂₀ cyclic alkylene, a C₁-C₂₀, linear fluoroalkylene, a C₃-C₂₀ branched fluoroalkylene, a C₆-C₂₀ cyclic fluoroalkylene, a C₁-C₂₀, linear oxyalkylene, a C₃-C₂₀ branched oxyalkylene, a C₆-C₂₀ cyclic oxyalkylene, carbonyloxyalkylene (-C(=O)-O-alkylene), alkenylene (-CR₁=CR₂-) where R₁ and R₂ are hydrogen or a C₁-C₅ alkyl, alkynylene (-C≡C-), amino (-N(R₃)-) where R₃ is hydrogen or an imine (-C(=N-R₄)- where R₄ is hydrogen or a C₁-C₅ alkyl, an amide (-C(=O)-N(R₅)- and -N(R₅)-C(=O)- where R₅ is hydrogen or a C₁-C₅ alkyl, imide (-C(=O)-N(R₆)-C(=O)- where R₆ is hydrogen or a C₁-C₅ alkyl, aminooxy (-O-N(R₇)- and -N(R₇)-O- where R₇ is a C₁-C₅ alkyl, sulfide (-S-), disulfide (-S-S-), thionyl (-S(=O)-), sulfone (-S(=O)₂-), sulfoneamide (-S(=O)₂-N(R₈)- where R₈ is hydrogen or a C₁-C₅ alkyl, thiocarbonyl (-C(=S)-), thiocarboxy (-C(=S)-O-), oxythiocarbonyl (-O-C(=S)-), a C₃-C₁₀-dialkylsilylene-alkylene (-Si(R₉)(R₁₀)-alkylene), where R₉ and R₁₀ are independently selected from a C₁ to C₄ alkyl moiety, a C₁-C₈ urethane-alkylene (-N(R₁₁)-C(=O)-O-alkylene), a C₁-C₈ urea-alkylene (-N(R₁₁)-C(=O)-N(R₁₁)-alkylene), a C₁-C₈ thiourea-alkylene (-N(R₁₁)-C(=S)-N(R₁₁)-alkylene), where R₁₁ is hydrogen or a C₁-C₄ alkyl dithiocarboxy (-C(=S)-S-), a C₁-C₂₀, carbonyloxyfluoroalkylene (-C(=O)-O-fluoroalkylene-), a C₁-C₂₀ fluoroalkylene, a C₆-C₂₀ arylene, a C₆-C₂₀ oxyarylene, a C₇-C₂₀ alkylenearylene, a C₇-C₂₀ fluoroalkylenearylene, a C₇-C₂₀ carbonylarylene, a C₇-C₂₀ carbonyloxyarylene, a C₇-C₂₀ oxycarbonylarylene, a C₇-C₂₀ oxycarbonyloxyarylene, a C₅-C₂₀ nitrogen containing heterocyclic moiety, a C₅-C₂₀ oxy-(nitrogen containing heterocyclic moiety), a C₆-C₂₀ alkylene-(nitrogen containing heterocyclic moiety), a C₆-C₂₀ carbonyl-(nitrogen containing

heterocyclic moiety), a C₆-C₂₀ carbonyloxy-(nitrogen containing heterocyclic moiety), a C₅-C₂₀ oxygen containing heterocyclic moiety, a C₅-C₂₀ oxy-(oxygen containing heterocyclic moiety), a C₆-C₂₀ alkylene-(oxygen containing heterocyclic moiety), a C₆-C₂₀ carbonyl-(oxygen containing heterocyclic moiety), a C₆-C₂₀ carbonyloxy-(oxygen containing heterocyclic moiety), a C₅-C₂₀ sulfur containing heterocyclic moiety, a C₅-C₂₀ oxy-(sulfur containing heterocyclic moiety), a C₆-C₂₀ alkylene-(sulfur containing heterocyclic moiety), a C₆-C₂₀ carbonyl-(sulfur containing heterocyclic moiety), a C₆-C₂₀ carbonyloxy-(sulfur containing heterocyclic moiety), a C₅-C₂₀ nitrogen containing heterocyclic moiety, a C₅-C₂₀ oxy-(nitrogen containing heterocyclic moiety), a C₆-C₂₀ alkylene-(nitrogen containing heterocyclic moiety), a C₆-C₂₀ carbonyl-(nitrogen containing heterocyclic moiety), a C₆-C₂₀ carbonyloxy-(nitrogen containing heterocyclic moiety), a $-(C_1-C_{20}alkylene)_o-(Si(C_1-C_{20} alkyl)_2-O)_p-$, a $-O-(Si(C_1-C_{20} alkyl)_2-O)_p-$, wherein o is an integer between 1 and 5, and p is a positive integer from 1 to 5, or combinations thereof. A more specific embodiment of this aspect of the invention is where Y is selected from the group consisting of oxy (-O-), carbonyl (-C(=O)-), carbonyloxy (-C(=O)-O-), oxycarbonyl (-O-C(=O)-), carbonate (-O-C(=O)-O-), carbonyloxyakylene (-C(=O)-O-akylene-), a C₁-C₂₀ carbonyloxyfluoroakylene (-C(=O)-O-fluoroakylene-), alkylene-fluoroalkylene, a C₆-C₂₀ arylene, a C₆-C₂₀ oxyarylene, a C₇-C₂₀ alkylenearylene, a C₁-C₂₀, linear alkylene, a C₃-C₂₀ branched alkylene, a C₆-C₂₀ cyclic alkylene, a C₁-C₂₀, linear fluoroalkylene, a C₃-C₂₀ branched fluoroalkylene, and a C₆-C₂₀ cyclic fluoroalkylene.

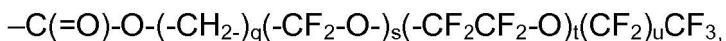
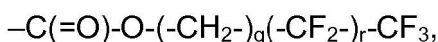
A further embodiment of structure (1) is where the moiety Z is a fluorine containing hydrocarbon group such as linear C₁-C₁₀fluoroalkyl group, C₃-C₁₀ branched fluoroalkyl groups, C₄-C₁₀ cyclic fluoroalkyl group, a alkylenefluoroalkyl group $-(CH_2)_q(CF_2)_r-CF_3$, a CFH containing alkylenefluoroalkyl group $-(CH_2)_q(CHF)_{q2}(CF_2)_r-CF_3$, a alkylenefluoroether-fluoroalkyl group $-(CH_2)_q(-CF_2-O-)_s(-CF_2CF_2-O-)_t(CF_2)_uCF_3$, a -CHF containing

alkylene fluoroether-fluoroalkyl groups $(-\text{CH}_2-)_{\text{q}}(\text{CHF})_{\text{q}2}(-\text{CF}_2-\text{O}-)_{\text{s}}(-\text{CF}_2\text{CF}_2-\text{O}-)_{\text{t}}(\text{CF}_2)_{\text{u}}\text{CF}_3$, or a moiety having structure (2)

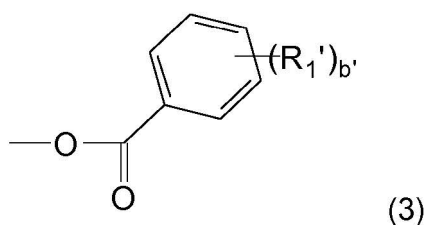


wherein q is an integer from 0 to 10, q2 is an integer from 1 to 10, r is an integer from 1 to 10, s is an integer from 0-10, t is an integer from 1 to 10, u is an integer between 0 and 10 x is 1 to 5, y is 1 to 5, and P is hydrogen, a C₁ to C₄ alkyl, or a $-\text{CH}_2-\text{CH}_2-(\text{CF}_2)_4\text{F}$ moiety.

One embodiment of the novel block copolymer of structure (1) is where the surface active pendant moiety $-\text{Y}(\text{Z})_{\text{b}}$ is



a moiety having structure (3)

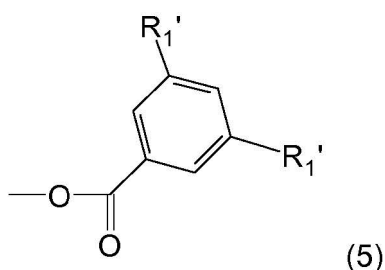


or a moiety having structure (4)



wherein $-\text{R}_1'$ and $-\text{R}_1''$ are independently selected from a fluorinated moiety.

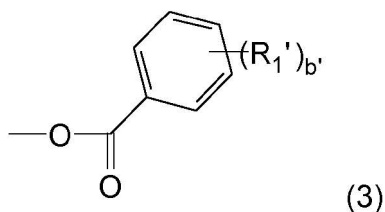
In a further embodiment, the fluorinated moiety is selected from moieties such as $-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, $-\text{O}-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, $-\text{C}(=\text{O})-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, $-\text{O}-\text{C}(=\text{O})-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, $-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$, $-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$, $-\text{O}-\text{C}(=\text{O})-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$ or $-\text{C}(=\text{O})-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$ and further wherein b' and b'' are integers from 1 to 5, q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10 and u is an integer between 0 and 10. A more specific embodiment of this aspect of the invention is where the surface active pendant moiety $-\text{Y}(\text{Z})_b$ is represented by (5)



wherein $-\text{R}_1'$ is $-\text{O}-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, and further wherein q is an integer from 0 to 10, and r is an integer from 1 to 10.

In a further embodiment of the novel structure (1), the surface active pendant moiety $\text{Y}(\text{Z})_b$ is $-\text{C}(=\text{O})-\text{O}-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, or

$-\text{C}(=\text{O})-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$ where q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0-10, and t is an integer from 1 to 10 and u is an integer between 0 and 10. Another more specific embodiment of this aspect of the invention is where the surface active pendant moiety $\text{Y}(\text{Z})_b$ has structure (3)



where R_1' is $-(CH_2)_q(-CF_2)_r-CF_3$, $-O-(CH_2)_q(-CF_2)_r-CF_3$,
 $-C(=O)-(CH_2)_q(-CF_2)_r-CF_3$, $-O-C(=O)-(CH_2)_q(-CF_2)_r-CF_3$,
 $-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$, $-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$,
 $-O-C(=O)-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$ or
 $-C(=O)-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$ and further wherein b' and b'' are integers from 1 to 5, q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10 and u is an integer between 0 and 10.

In another aspect of this invention the surface active pendant moiety $Y(Z)_b$ – in structure (1) is $-(C_1-C_{20} \text{ alkylene})_o-(Si(CH_3)_2-O)_p-$,
 $-(Si(CH_3)_2-O)_v-Si(CH_3)_3$,
or $-O-(Si(CH_3)_2-O)_p--(Si(CH_3)_2-O)_v-Si(CH_3)_3$, where o is an integer between 1 and 10, p is a positive integer from 1 to 10 and v is an integer from 2 to 10.

Another aspect of the invention is where structure (1) has an X moiety which is comprised of an oxyalkyleneoxycarbonylalkylene moiety. In this aspect of the invention the novel block copolymer may be made by atom-transfer radical-polymerization (ATRP) initiators or ring-opening polymerization (ROP) initiators. More specific examples of inventive materials where the X moiety is comprised of an oxyalkylenecarbonylalkylene moiety are shown in Figures 1 which depicts structures (6) to (11).

In Structure (6)-(9) R_2' is hydrogen or a C_1 to C_8 alkyl group, or a C_6 - C_{20} aryl group, R_3' and R_4' are independently a C_1 to C_8 alkyl group, hydrogen, bromine, chlorine, or iodine, w and w' are independently integers from 1 to 4; in structure

(7) x_a is an integer from 1 to 6, y_a is an integer from 2-16; in structure (8), x_a and x_b are independently integers from 1 to 6, y_a is an integer from 2-16, and b' is an integer from 1 to 5, in structure (9) x_a and x_b are independently integers from 1 to 6, y_a is an integer from 2-16, and b' is an integer from 1 to 5.

In Structure (10) $R_{2'}$ is a C_1 - C_4 alkyl, and $R_{3'}$ and $R_{4'}$ are independently a C_1 to C_8 alkyl group, and R_{f1} and R_{f2} are independently selected from

$-(CH_2)_q(CF_2)_rCF_3$, $-O-(CH_2)_q(CF_2)_rCF_3$, or $-(CH_2)_q(CF_2O)_s(CF_2CF_2O)_t(CF_2)_uCF_3$, and further wherein q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10, u is an integer between 0 and 10.

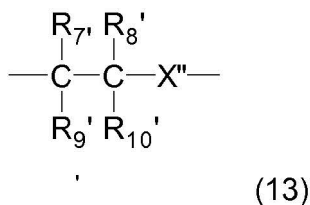
In Structure (11) $R_{2'}$ is selected from a C_1 - C_4 alkyl, and $R_{3'}$ and $R_{4'}$ are independently a C_1 to C_8 alkyl group, and R_{f1} and R_{f2} are independently selected from $-(CH_2)_q(CF_2)_rCF_3$, $-O-(CH_2)_q(CF_2)_rCF_3$, or

$-(CH_2)_q(CF_2O)_s(CF_2CF_2O)_t(CF_2)_uCF_3$, and further wherein q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10 and u is an integer between 0 and 10.

In another embodiment, when X is an oxyalkylenecarbonylalkylene moiety such as shown in Figures 1 (Structures (6) to (11)), block B is derived from a vinyl aryl monomer and block A is derived from a lactone, lactam, epoxide or cyclic carbonate monomer.

A further aspect of the novel block copolymer of structure (1) is where the junction group $X(Y(Z)_b)a$ is a moiety derived from a 1,1-diphenylethylene (DPE) derivative. Figure 2, which depicts structure (12) and (14) shows more specific examples of the novel materials containing an $X(Y(Z)_b)a$ moiety derived from DPE.

In structure (12) X' is a direct valence bond or a moiety (13)



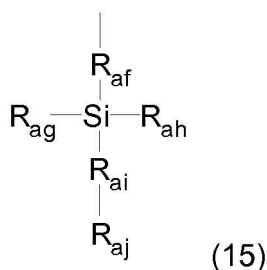
where R_5' is $-(CH_2)_q(-CF_2)_r-CF_3$, $-O-(CH_2)_q(-CF_2)_r-CF_3$, $-C(=O)-(CH_2)_q(-CF_2)_r-CF_3$, $-O-C(=O)-(CH_2)_q(-CF_2)_r-CF_3$, $-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$, $-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$, $-O-C(=O)-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$ or $-C(=O)-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$ and R_6' is $-(CH_2)_q(-CF_2)_r-CF_3$, $-O-(CH_2)_q(-CF_2)_r-CF_3$, $-C(=O)-(CH_2)_q(-CF_2)_r-CF_3$, $-O-C(=O)-(CH_2)_q(-CF_2)_r-CF_3$, $-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$, $-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$, $-O-C(=O)-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$, $-C(=O)-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$ or hydrogen, R_7' and R_9' are independently selected from hydrogen, or a C_1 - C_4 alkyl, R_8' and R_{10}' are independently selected from hydrogen, a C_1 - C_4 alkyl, or a C_2 - C_6 alkyleneoxyalkyl moiety, X'' is $-O-$ or $-S-$, and further wherein q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10, u is an integer between 0 and 10 and ba and bb are independently integers from 1 to 5.

In Structure (14) R_{ac} is $-O-(CH_2)_q(-CF_2)_r-CF_3$, and R_{ad} is

$-(CH_2)_q(-CF_2)_r-CF_3$, or hydrogen, q is an integer from 0 to 10, and r is an integer from 1 to 10.

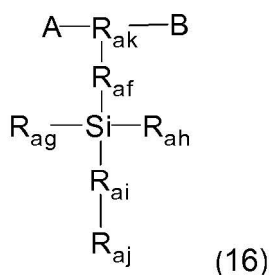
In Figure 2 for structures (12) and (14) for the phase separable blocks A and B, block A is derived from a vinyl aryl monomer, and block B is derived from a lactone, lactam, epoxide, cyclic carbonate monomer; also, block B may be derived from an ethylenically unsaturated polymerizable monomers in structure (14) and in structure (12) when X' is a direct valence bond.

In another aspect of this invention the novel block copolymer has a surface active pendant moiety $Y(Z)_b$ containing an alkyl silicon junction moiety which may be derived from a hydrosilation coupling reaction having structure (15) where R_{af} and R_{ai} are independently a C_1 - C_5 linear alkylene moiety, a C_3 - C_{10} branched alkylene moiety or a C_3 - C_{10} cyclic alkylene moiety, R_{ag} and R_{ah} are independently selected from a C_1 to C_4 alkyl moiety, and R_{aj} is $-(CH_2)_q(-CF_2)_r-CF_3$, $-O-(CH_2)_q(-CF_2)_r-CF_3$, $-C(=O)-(CH_2)_q(-CF_2)_r-CF_3$, $-O-C(=O)-(CH_2)_q(-CF_2)_r-CF_3$, $-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$, $-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$, $-O-C(=O)-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$ or $-C(=O)-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$ and further wherein b' and b'' are integers from 1 to 5, q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10 and u is an integer between 0 and 10.



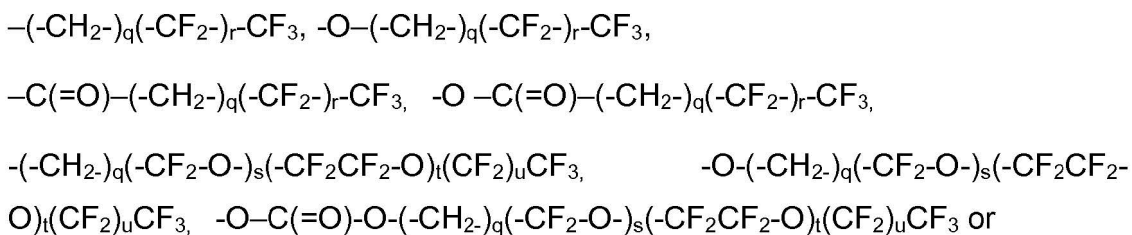
More specific examples of inventive block copolymer materials with a surface active pendant moiety containing an alkyl silicon junction moiety are shown in structure (16), where R_{af} , R_{ak} and R_{ai} are independently a C_1 - C_{10} linear alkylene moiety, a C_3 - C_{10} branched alkylene moiety, or a C_3 - C_{10} cyclic alkylene moiety, R_{ag} and R_{ah} are independently selected from a C_1 to C_4 alkyl moiety, and R_{aj} is $-(CH_2)_q(-CF_2)_r-CF_3$, $-O-(CH_2)_q(-CF_2)_r-CF_3$, $-C(=O)-(CH_2)_q(-CF_2)_r-CF_3$, $-O-C(=O)-(CH_2)_q(-CF_2)_r-CF_3$, $-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$, $-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$, $-O-C(=O)-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$ or $-C(=O)-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$

$-\text{C}(=\text{O})-\text{O}-(-\text{CH}_2-)_{\text{q}}(-\text{CF}_2-\text{O}-)_{\text{s}}(-\text{CF}_2\text{CF}_2-\text{O})_{\text{t}}(\text{CF}_2)_{\text{u}}\text{CF}_3$ and further wherein b' and b'' are integers from 1 to 5, q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10, and u is an integer between 0 and 10.



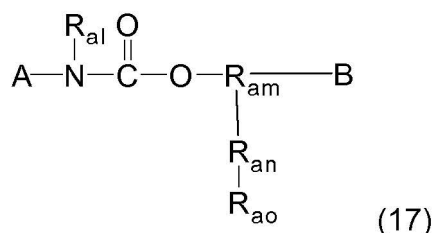
In a further aspect of the invention depicted in structure (16), the phase separable blocks A and B may be ones where A is derived from a vinyl aryl monomer and block B is derived from a lactone, lactam, epoxide, cyclic carbonate or an ethynically unsaturated monomer which is not the same as block A, and further wherein block A and B are chemically distinct and are phase separable.

In another aspect of this invention the novel block copolymer of structure (1) has a junction group $\text{X}(\text{Y}(\text{Z})_{\text{b}})_{\text{a}}$ incorporating a urethane-alkylene moiety. Structure (17) shows specific examples of novel block copolymer comprising an urethane-alkylene moiety where R_{al} is hydrogen or a $\text{C}_1\text{-C}_4$ alkyl, R_{am} and R_{an} are independently a $\text{C}_1\text{-C}_{20}$ linear alkylene moiety, a $\text{C}_3\text{-C}_{30}$ branched alkylene moiety, a $\text{C}_3\text{-C}_{30}$ cyclic alkylene moiety, and R_{ao} is



$-\text{C}(=\text{O})-\text{O}-(-\text{CH}_2-)_{\text{q}}(-\text{CF}_2-\text{O}-)_{\text{s}}(-\text{CF}_2\text{CF}_2-\text{O})_{\text{t}}(\text{CF}_2)_{\text{u}}\text{CF}_3$, b' and b'' are integers from 1 to 5, q is an integer from 0 to 10, r is an integer from 1 to 10, s is an

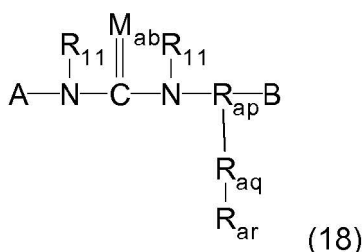
integer from 0 to 10, t is an integer from 1 to 10, and u is an integer between 0 and 10.



In a further aspect of the invention depicted in structure (17) block A is derived from a vinyl aryl monomer and block B is derived from a lactone, lactam, epoxide, cyclic carbonate or an ethynically unsaturated monomer which is not the same as block A, and further wherein block A and B are chemically distinct and are phase separable.

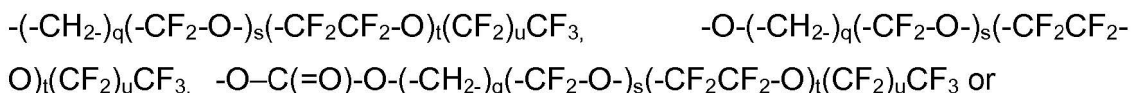
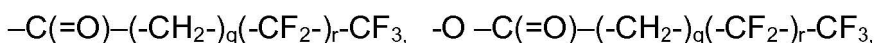
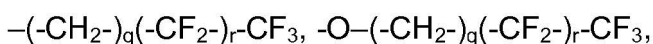
In another aspect of this invention the novel block copolymer of structure (1) has a junction group, $X(Y(Z)_b)_a$, comprising a urea-alkylene or thiourea-alkylene moiety which may arise from an isocyanates and thioisocyanate coupling reaction. Structure (18) shows specific examples of this novel block copolymer having a urea-alkylene or thiourea-alkylene moiety where R_{11} is hydrogen or a C_1 - C_4 alkyl, R_{ap} and R_{aq} are independently a C_1 - C_{20} linear alkylene moiety, a C_3 - C_{30} branched alkylene moiety, or a C_3 - C_{30} cyclic alkylene moiety, M_{ab} is O or S, and R_{ar} is $-(CH_2)_q(-CF_2)_r-CF_3$, $-O-(CH_2)_q(-CF_2)_r-CF_3$, $-C(=O)-(CH_2)_q(-CF_2)_r-CF_3$, $-O-C(=O)-(CH_2)_q(-CF_2)_r-CF_3$, $-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$, $-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$, $-O-C(=O)-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$ or $-C(=O)-O-(CH_2)_q(-CF_2-O)_s(-CF_2CF_2-O)_t(CF_2)_uCF_3$.

and further wherein b' and b'' are integers from 1 to 5, q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10 and u is an integer between 0 and 10.

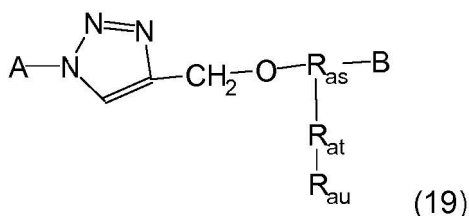


In another aspect of this embodiment, as shown in Structure (18) the phase separable block A and B are ones where block A and B are chemically distinct and block A is derived from a vinyl aryl monomer, and block B is derived from a lactone, lactam, epoxide, cyclic carbonate or an ethynically unsaturated monomer which is not the same as the vinyl aryl from which block A is derived from.

In another aspect of this invention the novel block copolymer of structure (1) has a junction group $\text{X}(\text{Y}(\text{Z})_b)_a$ comprised of a 1,2,3-triazole alkyleneoxy moiety may arise from a click chemistry coupling reaction. In another embodiment of this aspect of the invention where $\text{X}(\text{Y}(\text{Z})_b)_a$ comprises of a 1,2,3-triazole alkyleneoxy, the block copolymer may have structures as described in Structure (19) where R_{as} and R_{at} are independently a $\text{C}_1\text{-C}_{20}$ linear alkylene moiety, a $\text{C}_3\text{-C}_{30}$ branched alkylene moiety, a $\text{C}_3\text{-C}_{30}$ cyclic alkylene moiety, and R_{au} is



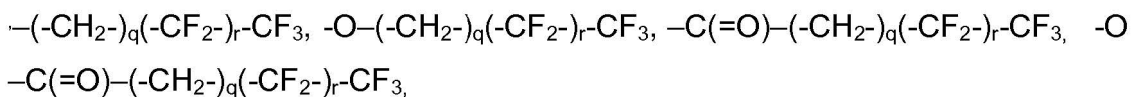
$-\text{C}(=\text{O})-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{-O})_s(\text{CF}_2\text{CF}_2\text{-O})_t(\text{CF}_2)_u\text{CF}_3$ and further wherein b' and b'' are integers from 1 to 5, q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10, u is an integer between 0 and 10.



In this embodiment where the novel block copolymer has structure (19), block A and B are chemically distinct and are phase separable where block A is derived from a vinyl aryl, and block B is derived from a lactone, lactam, epoxide, cyclic carbonate or an ethynically unsaturated monomer which is not the same as the vinyl aryl from which block A is derived.

In another aspect of this invention the novel block copolymer of structure (1) has a junction group incorporating aryl substituted alkyleneoxy moiety. $X(Y(Z)_b)_a$ which may be derived from the reaction of a poly styrenic anion derivative with an epoxide followed by reaction of the oxy anion formed with a cyclic monomer. In one example of this embodiment the novel block copolymer may have the structures as shown in Figure 3 comprising structures 20, 21a, and 21b.

In the embodiment of this aspect of the novel block copolymer as shown in Structure 20, E' and E'' are independently selected from hydrogen or methyl, R_{ax} is hydrogen, a C_1 - C_3 alkyl group, a C_3 - C_{20} trialkylsilyl group, a C_3 - C_{20} trialkylgermanium group, a C_3 - C_{20} trialkyltin group, a C_3 - C_{20} trialkyltitanium group, a C_2 - C_{20} alkylcarbonyl, a C_2 - C_{20} alkylcarbonyloxy, a C_1 - C_{20} alkoxy, a C_9 - C_{36} tris(trialkylsilyl)silyl group, a C_3 - C_{20} trialkylsilyl group, a C_3 - C_{20} trialkylgermanium group, a C_3 - C_{20} trialkyltin group, a C_3 - C_{20} trialkyltitanium group, a C_2 - C_{20} alkylcarbonyl, a C_2 - C_{20} alkylcarbonyloxy, a C_1 - C_{20} alkoxy or a C_9 - C_{36} tris(trialkylsilyl)silyl group, R_{av} is a C_1 - C_4 alkylene spacer moiety and R_{aw} is



$-(\text{CH}_2)_q(\text{CF}_2\text{-O})_s(\text{CF}_2\text{CF}_2\text{-O})_t(\text{CF}_2)_u\text{CF}_3$, $-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{-O})_s(\text{CF}_2\text{CF}_2\text{-O})_t(\text{CF}_2)_u\text{CF}_3$, $-\text{O}-\text{C}(=\text{O})-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{-O})_s(\text{CF}_2\text{CF}_2\text{-O})_t(\text{CF}_2)_u\text{CF}_3$ or $-\text{C}(=\text{O})-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{-O})_s(\text{CF}_2\text{CF}_2\text{-O})_t(\text{CF}_2)_u\text{CF}_3$ and further wherein q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10, u is an integer between 0 and 10 and ba is an integer from 1 to 5. In a further aspect of this embodiment, the phase separable block copolymers are derived in the case of block A from a vinyl aryl and for block B from a lactone, lactam, epoxide or cyclic carbonate.

In Figure 3, structures (21a) and (21b) show another embodiment of this aspect of the novel block copolymer where xb is an integer from 1 to 6 and yb is an integer from 4 to 20.

In the aspects of the novel block copolymer illustrated by Figure 3, block A and B are phase separable and where further in this aspect of the invention block A is derived from a vinyl aryl and block B is derived from a lactone, lactam, epoxide or cyclic carbonate.

The block copolymer having structure (1), and the substructures described above may be made by a variety of approaches, such as, for instance, block copolymer prepared by using atom-transfer radical-polymerization (ATRP) and ring-opening polymerization (ROP) initiators, block copolymers prepared using a 1,1-diphenylethylene (DPE) derivative as an initiator and capping agent in anionic block polymerization.

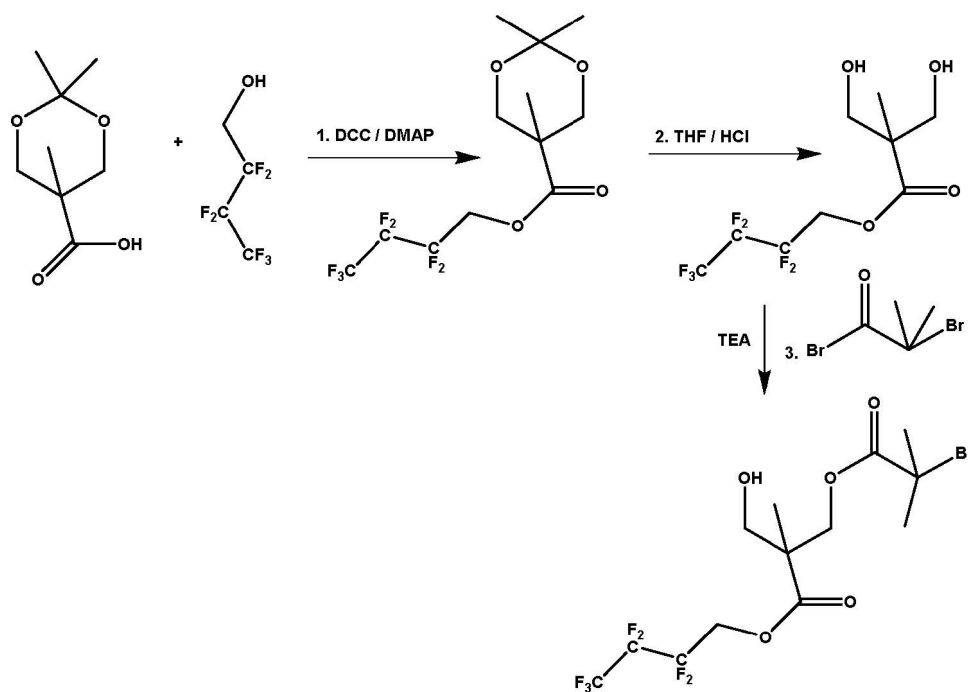
Scheme 1 shows the synthesis of an ATRP and ROP initiator comprised of an oxyalkylenecarbonylalkylene moiety. This initiator may be employed to make one example of the novel block copolymer of structure (1) depicted in Figures 1a and 1b, namely one belonging to the group having structure (7) where the pendant Z moiety is derived from 2,2,3,3,4,4,4-Heptafluoro-1-butanol (C4OL) pendant on an X moiety comprised of a oxyalkylenecarbonylalkylene moiety.

Similarly, other types of pendant Z moieties may be attached to such initiators by using suitable alcohols containing suitable Z moieties such as a

fluorine containing linear hydrocarbon group, a fluorine containing branched hydrocarbon group, a fluorine containing cyclic hydrocarbon group, a fluorine containing linear alkyl ether group, a fluorine containing branched alkyl ether group, a fluorine containing cyclic alkyl ether group or a mixture thereof, a Si₁-Si₈ siloxane moiety, a linear hydrocarbon moiety with a Si₁-Si₈ siloxane pendant group, a branched hydrocarbon moiety with a Si₁-Si₈ siloxane pendant group, a cyclic hydrocarbon moiety with a Si₁-Si₈ siloxane pendant group, a carbosilane, a linear hydrocarbon moiety with a carbosilane pendant group, a branched hydrocarbon moiety with a carbosilane pendant group, a cyclic hydrocarbon moiety with a carbosilane pendant group, a silane, a linear hydrocarbon moiety with a silane pendant group, a branched hydrocarbon moiety with a silane pendant group, a cyclic hydrocarbon moiety with a silane pendant group, or a linear hydrocarbon chain having more than 18 carbons.

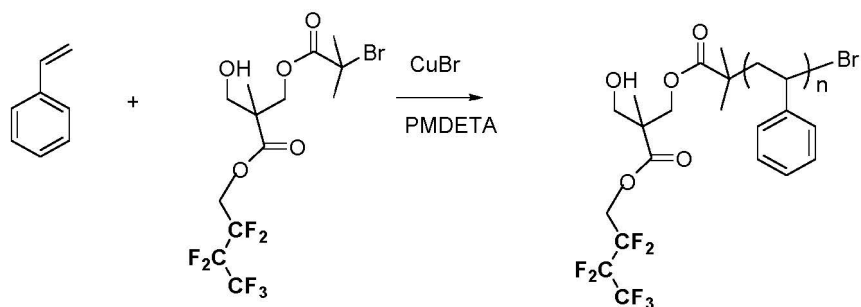
The Example section shows several specific examples of the synthesis of such initiators capable of making novel block copolymer of this invention where the X moiety contains an oxyalkylenecarbonylalkylene moiety.

A specific non-limiting Example of a full synthesis of a novel block copolymer of this invention having structure (7) based on the initiator resulting from Scheme 1 is depicted in sequence illustrated by Schemes 2 and 3. The Example section shows other examples of this approach, using different initiators containing an oxyalkylenecarbonylalkylene moiety employed to make other novel block copolymers with other monomers.



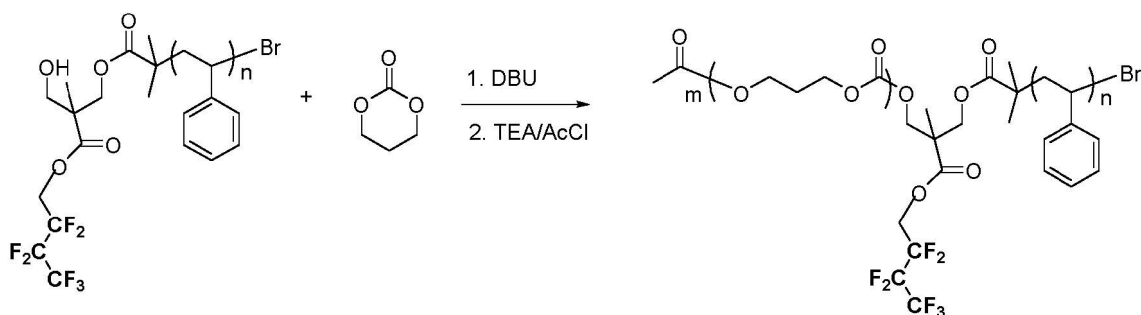
Scheme 1

Specifically, in this instance, Scheme 2 shows the reaction of initiator made in Scheme 1 with styrene.



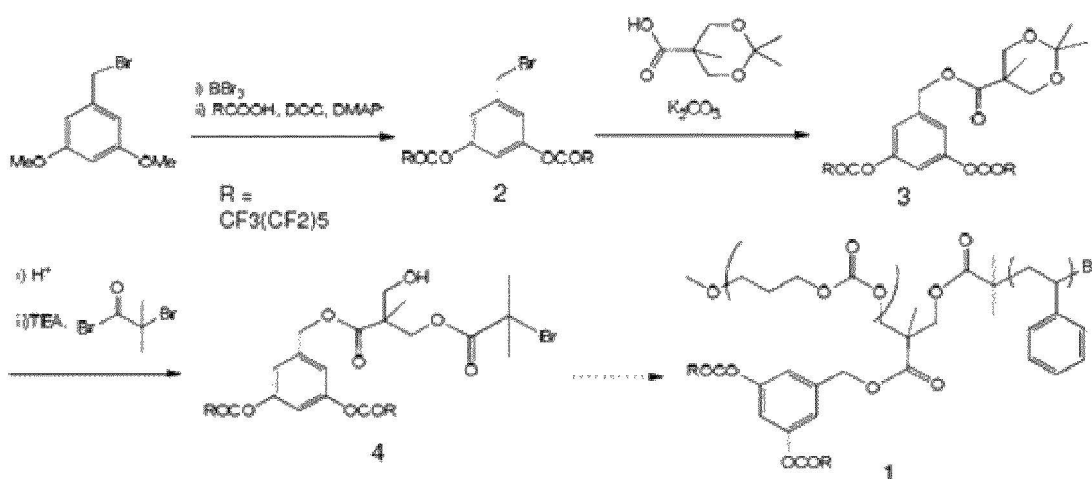
Scheme 2

In turn Scheme 3 shows the reaction of this styrene macroinitiator moiety with the cyclic monomer 1,3-propylene carbonate to form one non limiting example of an embodiment of the novel block copolymer of this invention.



Scheme 3

Scheme 4 shows another similar approach where more than one Z moiety may be introduced where the R moiety would be a Z moiety derived from an alcohol as described above.



Scheme 4

Another approach to making the novel block copolymer of structure (1) is to use 1,1-diphenylethylene (DPE) as an initiator and capping agent in anionic block polymerization of styrene and alkyl(meth)acrylate monomers. These are the materials having structures (12) and (14) depicted in Figure 2.

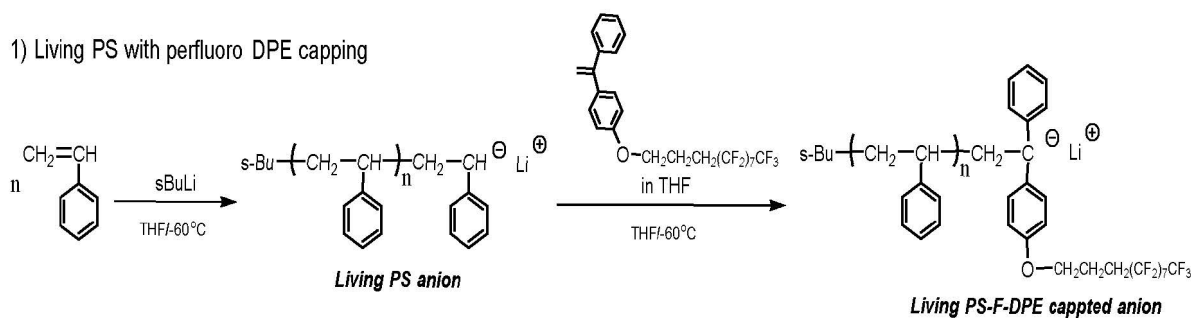
1,1-diphenylethylene (DPE) being a non-polymerizable, vinyl monomer can be used in anionic polymerization of various alkyl(meth)acrylates. It can also

be used as a reactivity modifier in styrene block copolymerization with alkyl(meth)acrylates in anionic polymerization. The advantage of DPE is that the opening of double bond via carbanion reaction results to form a carbanion, which is highly delocalized into the two phenyl substitution at alpha-carbon of the double bond. This significantly reduces the reactivity and thereby suppresses undesired attachment of carbonyl group in alkyl(meth)acrylate polymerization. An adduct of sec-BuLi and DPE is generally used as initiator for alkyl(meth)acrylate polymerization. The active center of anionic polymerization of styrene, a styryllithium, is very reactive and will not efficiently initiate alkyl(meth)acrylate to produce well-defined block copolymer.

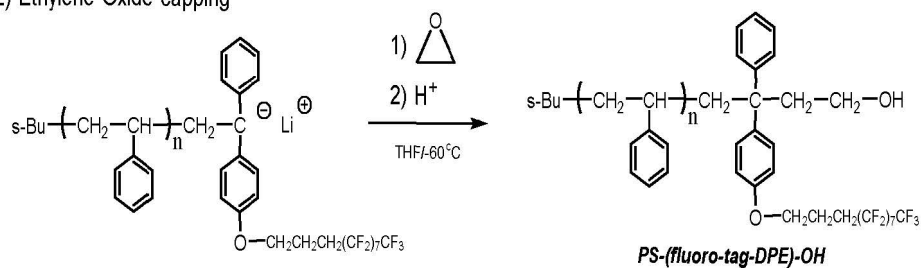
By functionalizing DPE with pendant surface active moieties it is possible to prepare the novel block copolymer of this invention having structure (1) where the surface active pendant moieties $Y(Z)_b$ is comprised of Z moieties selected from, as non-limiting examples, perfluoroalkane and perfluoroethers.

Schemes 5 and 6 show respectively non limiting examples of formation of macroinitiators capable of forming polymers having structure (1) in which the junction groups are derived from DPE having one or two Z moiety. In these schemes the X moiety is an alkyleneoxy moiety to which is attached the linking group Y an arylene moiety (in this instance phenylene) to which is attached the Z moiety which is an alkoxy group derived from a fluoroalcohol. Final polymers of structure (1) would be formed by reaction of the final product of Schemes 5 or 6 (PS-(fluoro-tag-DPE)-OH or PS-(fluoro-tag-DPE)-OH, respectively, with a base to form alkoxide macroinitiators followed by the addition of a monomer capable of undergoing ring-opening polymerization with these macroinitiators such as an epoxide, lactam, lactone, or cyclic carbonate to form the second block.

1) Living PS with perfluoro DPE capping

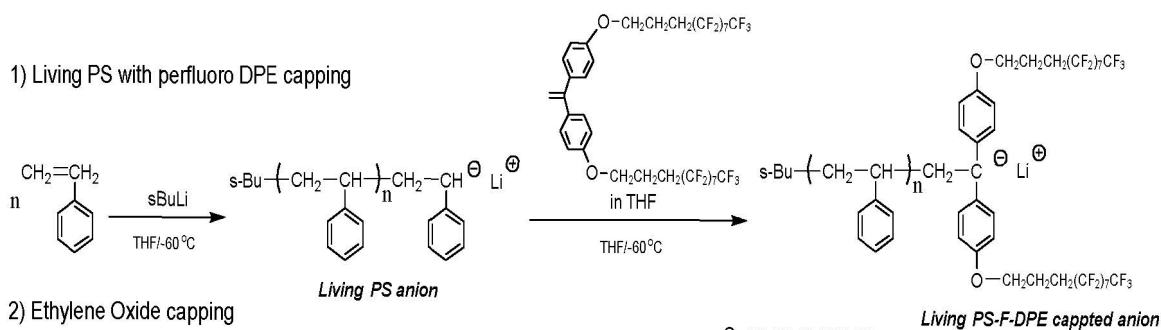


2) Ethylene Oxide capping

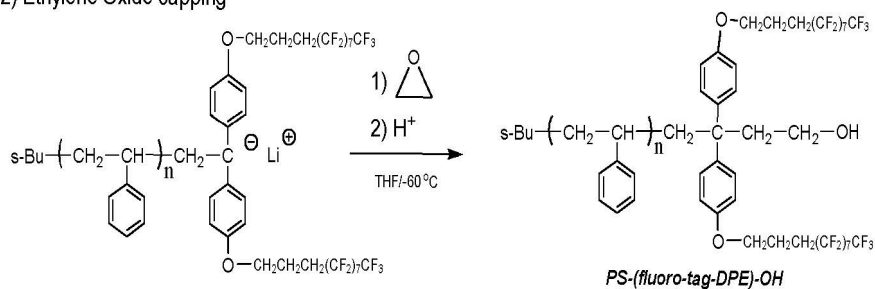


Scheme 5

1) Living PS with perfluoro DPE capping



2) Ethylene Oxide capping

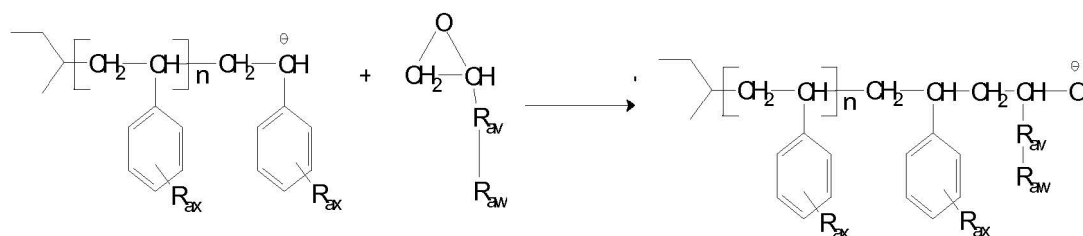


Scheme 6

The Examples sections show other non-limiting example of polymers having structure (1) which can be prepared using this approach. Using similar approaches block copolymer having structures (12) and (14) shown in Figure 2 may be prepared.

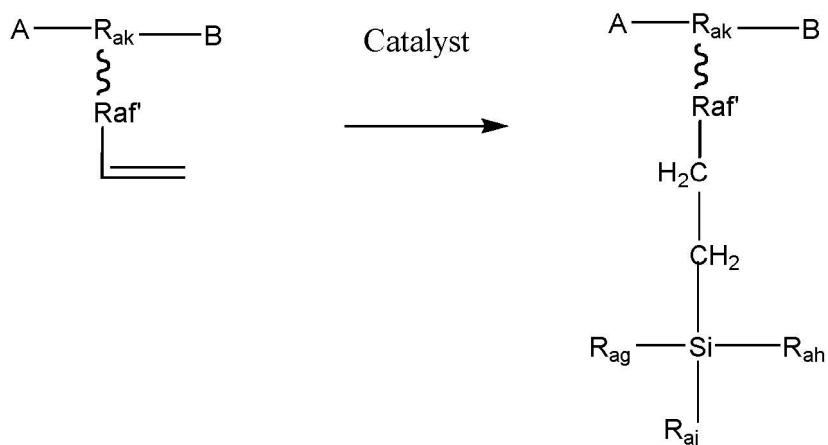
Apart from the methods described above, the novel block copolymer (1) having a junction group containing a surface active pendant group can also be prepared using other synthetic approaches such as shown in Schemes 7 to 11.

For instance, Scheme 7 shows, in a non-limiting example, the first step in a possible pathway for forming these block copolymers having structure (31) in which an intermediate polymer with an reactive alkoxide end group is formed. This reactive alkoxide end group could then be reacted with a cyclic monomer susceptible to ring-opening polymerization, such as an epoxide, lactam, lactone, or cyclic carbonate to form the second block. In this manner the novel block copolymer having structures (20), (21a) and (21b) shown in Figure 3 may be formed. The assignment of the variables denoted in Scheme 7 is the same as that in structure (20).



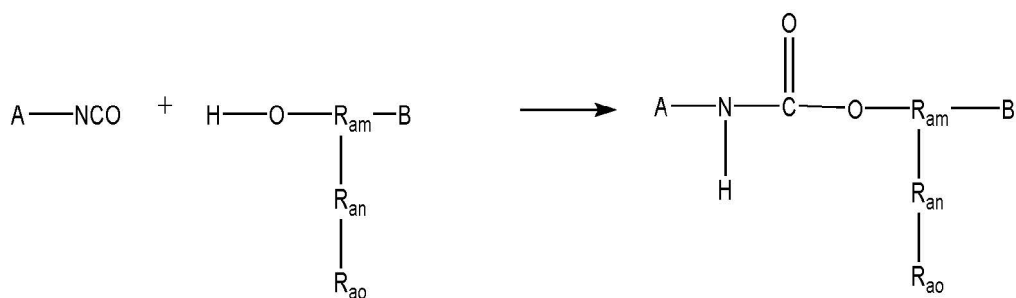
Scheme 7

Another approach as shown in **Scheme 8** to making the novel block copolymer is to employ an intermediate hydrosilation coupling reaction as shown in Scheme 8 to make structure such as (16). In scheme 8 R_{af} is a C_1 - C_8 linear alkylene moiety, a C_3 - C_8 branched alkylene moiety, a C_3 - C_8 cyclic alkylene moiety R_{ak} and R_{ai} are independently a C_1 - C_{10} linear alkylene moiety, or a C_3 - C_{10} branched alkylene moiety, or a C_3 - C_{10} cyclic alkylene moiety, and all other variables are as previously described for structure (16).

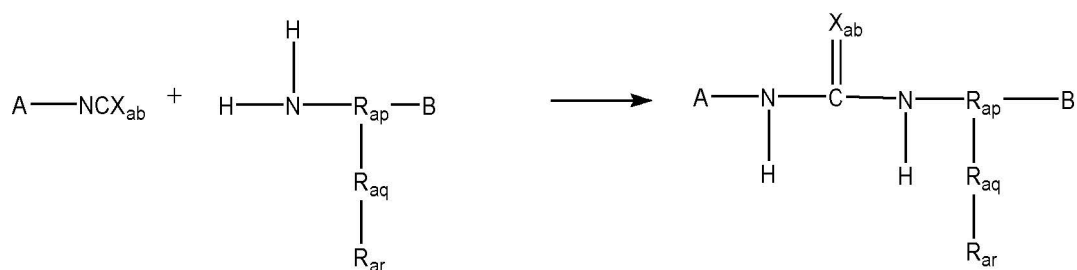


Scheme 8

In another method of preparation shown in **Schemes 9 and 10**, the novel block copolymers having junction groups containing urethane, urea or thio urea functionalities as part of the $\text{X}(\text{Y}(\text{Z})_b)_a$ junction group with structure (17), (18) with M_{ab} as oxygen or (18) with M_{ab} as sulfur respectively may be made by using coupling chemistry. Various type of coupling reactions forming urethane, urea or thiourea may be employed to form the novel block copolymers. Schemes 9 and 10 show two such non-limiting examples of reaction that may be employed. The assignment of variables in schemes 9 and 10 are the same respectively as denoted previously for structures (17) and (18).



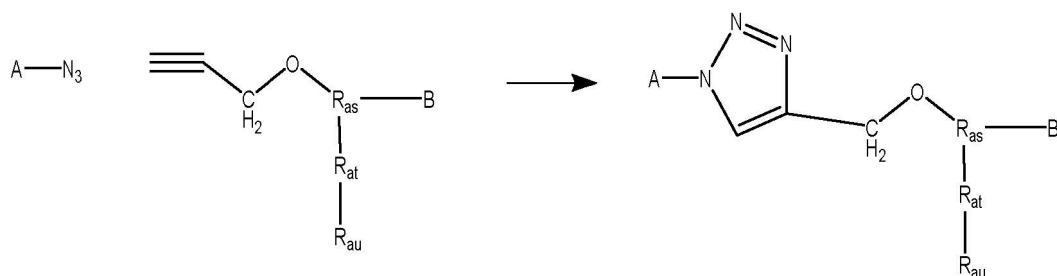
Scheme 9



Scheme 10

Other coupling reactions may be employed to join two different polymer blocks together through a linking $\text{X}(\text{Y}(\text{Z})_b)_a$ moiety of structure (1).

For instance, click coupling chemistry could be employed to make novel block copolymers having structure (29) by using a reaction pathway depicted in Scheme 11. The assignments of the variables in Scheme 11 are the same as in structure (19).



Scheme 11

In the block copolymer of structure (1), block A may be a moiety comprised of units derived from an alkylvinyl monomer, an alkyl methacrylate monomer, an alkyl acrylate monomer, a lactone monomer, an epoxide monomer, a lactam monomer, a cyclic carbonate monomer, and block B may be a moiety comprised of repeat units derived from a vinylaryl monomer and further wherein block A and B are phase separable.

Moreover, in another embodiment of this invention, block A may be comprised of units derived from an ethylenically unsaturated polymerizable

monomers, which is a vinyl aryl monomer and block B may be comprised of units derived from either an ethylenically unsaturated polymerizable monomer, selected from an acrylate or methacrylate or comprised of units derived from ring-opening polymerization (ROP) of cyclic monomers, selected from lactone, or cyclic carbonates and further wherein block A and B are phase separable.

In another embodiment the block copolymer of structure (1) has one block which is derived from a ring-opening polymerizable monomer. In another embodiment, block A and B may be derived from two different ethylenically unsaturated polymerizable monomers where block A and B are phase separable moieties.

When the polymer block in the novel block copolymer having structure (1) is derived from either a cyclic carbonyl monomer or an ethylenically unsaturated polymerizable monomer it can be a homopolymer or a random copolymer. The cyclic carbonyl monomers can be stereospecific or non-stereospecific.

The formation of polymer blocks containing units derived from cyclic monomers by ring-opening polymerization in the novel block copolymer having structure (1) may be accomplished using techniques described for other types of diblock copolymers in which one of the blocks has been derived by ring-opening polymerization such as disclosed in (US 8642086 B2) incorporated herein by reference.

Generally, in structure (1) one block in the diblock copolymer is resistant to plasma etching while the other block or etches very quickly under the same conditions allowing for pattern transfer of the self-assembled pattern formed by coating and annealing the novel polymers of structure (1) into a substrate. This self-assembly is directed on a patterned substrate using either a graphoepitaxy or chemoepitaxy approach.

Ethylenically unsaturated monomers which can be employed to form a polymer block in the novel block polymers having structure (1) may be comprised of units derived from a vinyl aryl, an alkylvinyl, monomer, an alkyl methacrylate

monomer, or an alkyl acrylate monomer. A polymer block in the novel block copolymer having structure (1) derived from unsaturated monomers can be a homopolymer or a random copolymer. Ethylenically unsaturated can be stereospecific or non-stereospecific.

Non limiting examples of alkyl vinyl compounds, alkyl acrylate or alkyl methacrylate that can be employed to make a block of the novel block copolymers of this invention are one where the alkyl group is a C₁-C₄ fluoroalkyl, a C₁-C₁₀ linear alkyl moiety, a C₁-C₄ hydroxfluoroalkylene group, a C₂-C₁₀-alkyloxyalkylene group, a C₁-C₁₀- hydroxyalkylene group, a C₃-C₂₀ branched alkyl a C₃-C₂₀ cyclic alkyl, a C₂-C₂₀ carboxylalkylene, a C₃-C₂₀ alkyloxycarboxylalkylene, or a C₃-C₂₀ alkyloxycarboxyloxyalkylene; other substituents which do not contain a refractory element are possible. The alkyl group in these examples, attached to a carboxyl moiety may also be an acid cleavable group such as a tertiary ester an acetal or ketal capable of releasing a free carboxyl moiety when reacted with a strong acid formed either by a thermal acid generator or a photoacid generator. If the block derived from an alkyl vinyl compound, alkyl acrylate or alkyl methacrylate is to be used as the etch resistant block moiety the alkyl group may be substituted with moieties containing a refractory element such as a C₃-C₂₀ trialkylsilyl group, a C₃-C₂₀ trialkylgermanium group, a C₃-C₂₀ trialkyltin group, or C₃-C₂₀ trialkyltitanium group.

Non limiting examples of vinyl aryl compounds that can be employed to make an etch resistant block of the novel block copolymers of structure (1) of this invention are unsubstituted styrene, unsubstituted vinyl naphthalenes, unsubstituted vinyl anthracite, unsubstituted vinyl pyrene and the like, or these aryl moieties containing one or more substituents such as a C₁-C₂₀ alkyl group, a C₁-C₂₀ hydroxy alkyl group, a C₂-C₁₀- alkyloxyalkylene group, a C₁-C₄ fluoroalkyl, a C₁-C₄ hydroxfluoroalkylene group, a hydroxyl group, a hydroxyl group blocked with an acid labile group, a carboxyl group, a carboxyl group blocked with an acid labile group, (these acid cleavable moieties capable of releasing respectively a free hydroxyl or free carboxyl moiety when reacted with

a strong acid present in a film formed either by a thermal acid generator or a photoacid generator), a fluoroalcohol group, a C₃-C₂₀ trialkylsilyl group, a C₃-C₂₀ trialkylgermanium group, a C₃-C₂₀ trialkyltin group, a C₃-C₂₀ trialkyltitanium group, a C₂-C₂₀ alkylcarbonyl, a C₂-C₂₀ alkylcarbonyloxy, a C₁-C₂₀ alkoxy or a C₉-C₃₆ tris(trialkylsilyl)silyl group, unsubstituted vinyl naphthalene and vinyl naphthalene substituted with, a C₁-C₂₀ alkyl group, a C₃-C₂₀ trialkylsilyl group, a C₃-C₂₀ trialkylgermanium group, a C₃-C₂₀ trialkyltin group, a C₃-C₂₀ trialkyltitanium group, a C₂-C₂₀ alkylcarbonyl, a C₂-C₂₀ alkylcarbonyloxy, a C₁-C₂₀ alkoxy, a C₉-C₃₆ tris(trialkylsilyl)silyl group, vinyl anthracene, a vinyl anthracene substituted with a C₁-C₂₀ alkyl group, a C₃-C₂₀ trialkylsilyl group, a C₃-C₂₀ trialkylgermanium group, a C₃-C₂₀ trialkyltin group, a C₃-C₂₀ trialkyltitanium group, a C₂-C₂₀ alkylcarbonyl, a C₂-C₂₀ alkylcarbonyloxy, a C₁-C₂₀ alkoxy, a C₉-C₃₆ tris(trialkylsilyl)silyl group, vinyl pyrene, a vinyl pyrene substituted with a C₁-C₂₀ alkyl group, a C₃-C₂₀ trialkylsilyl group, a C₃-C₂₀ trialkylgermanium group, a C₃-C₂₀ trialkyltin group, a C₃-C₂₀ trialkyltitanium group, a C₂-C₂₀ alkylcarbonyl, a C₂-C₂₀ alkylcarbonyloxy, a C₁-C₂₀ alkoxy or a C₉-C₃₆ tris(trialkylsilyl)silyl group or the like.

In one embodiment of the novel diblock polymer having structure (1) the two polymer blocks are derived from two different ethylenically unsaturated monomers one block of is plasma etch resistant and is derived from either a vinyl aryl compounds, a vinyl alkyl, an alkyl methacrylate or a alkyl acrylate, where the alkyl group contains a refractory element as described above, while the other block etches quickly with the same plasma chemistry such as blocks derived from alkyl vinyl, alkyl acrylate, alkyl methacrylate where the alkyl moiety contains no refractory elements .

In another embodiment of the novel diblock polymer having structure (1) one polymer block is etch resistant and derived either from a vinyl aryl compounds, a vinyl alkyl, an alkyl methacrylate or an alkyl acrylate, where the alkyl group contains a refractory element as described above, while the other block which etches quickly with the same plasma chemistry is derived from ring-

opening polymerization of a cyclic monomer, such as an epoxide, a lactone, a lactam, or a cyclic carbonate. These cyclic monomers may contain one or more substituents, which do not contain refractory element, or aromatic rings. Non limiting examples of possible substituents, are hydroxyl groups, a C₁-C₂₀ alkyl group, a C₁-C₂₀ hydroxy alkyl group, a C₂-C₁₀- alkyloxyalkylene group, halide (F, Cl, Br, I), haloalkyl, cyano, a C₁-C₄ fluoroalkyl, a C₁-C₄ hydroxfluoroalkylene group, a hydroxyl group, a hydroxyl group blocked with an acid labile group, a carboxyl group, a carboxyl group blocked with an acid labile group, a carboxyalkylene group, a carboxyalkylene group where the carboxyl group is blocked by an acid labile group, an alkyloxycarbonyl group, an alkylcarbonyl group, a alkyl group containing a double or triple carbon bound and other substituent that do not contain a refractory element.

Specific examples of lactones which may be employed as precursors for one of the blocks in the novel block copolymer of structure (1) are as follows: beta-propiolactone, gamma-butyrolactone, delta-valerolactone, epsilon-caprolactone, beta-butyrolactone, gamma-valerolactone, pivalolactone, 1,5-dioxepan-2-one, 5-(benzyloxy)oxepan-2-one a cyclic carbonate selected from trimethylene carbonate, tetramethylene carbonate, pentamethylene carbonate, glycoside, 7-oxooxepan-4-yl 2-bromo-2-methylpropanoate, 5-phenyloxepan-2-one, 5-methyloxepan-2-one, 1,4,8-trioxa(4,6)spiro-9-undecane, 5-(benzyloxymethyl)oxepan-2-one, 7-oxooxepan-4-yl 3-hydroxy-2-(hydroxymethyl)-2-methylpropanoate, (Z)-6,7-dihydrooxepin-2(3H)-one, D-lactide, L-lactide, meso-lactide or these materials containing other substituents such as hydroxyl groups, a C₁-C₂₀ alkyl group, a C₁-C₂₀ hydroxy alkyl group, a C₂-C₁₀- alkyloxyalkylene group, halide (F, Cl, Br, I), haloalkyl, cyano, a C₁-C₄ fluoroalkyl, a C₁-C₄ hydroxfluoroalkylene group, a hydroxyl group, a hydroxyl group blocked with an acid labile group, a carboxyl group, a carboxyl group blocked with an acid labile group, a carboxyalkylene group, a carboxyalkylene group where the carboxyl group is blocked by an acid labile group, an alkyloxycarbonyl group, a alkylcarbonyl group, a alkyl group containing a double or triple carbon bound a refractory element such as a C₃-C₂₀ trialkylsilyl group, a C₃-C₂₀

trialkylgermanium group, a C₃-C₂₀ trialkyltin group, or C₃-C₂₀ trialkyltitanium group.

Specific examples of carbonates which may be employed as precursors for one of the blocks in the novel block copolymer of structure (1) are as follows: trimethylene carbonate, tetramethylene carbonate, pentamethylene carbonate, 5-methyl-2-oxo-1,3-dioxane-5-carboxylic acid, methyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, tert-butyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, ethyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 5-methyl-2-oxo-1,3-dioxane-5-carbonyl chloride, benzyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 2-(tetrahydro-2H-pyran-2-yloxy)ethyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 2,2,3,3,4,4,5,5,6,6,7,7,7-tridecafluoroheptyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 2,2,2-trifluoroethyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, prop-2-ynyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, allyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 2-(2-methoxyethoxy)ethyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 3-(tert-butylthio)propyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 2-(tert-butoxycarbonylamino)ethyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 2-(pyridin-2-ylidisulfanyl)ethyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 4-(hydroxymethyl)benzyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, tert-butyl 4-(5-methyl-2-oxo-1,3-dioxane-5-carboxyloxy)piperidine-1-carboxylate, N-benzyl-5-methyl-2-oxo-1,3-dioxane-5-carboxamide, N-isopropyl-N,5-dimethyl-2-oxo-1,3-dioxane-5-carboxamide, 5,5-dimethyl trimethylene carbonate (5,5-dimethyl-1,3-dioxan-2-one), 5-methyl trimethylene carbonate (5-methyl-1,3-dioxan-2-one), 5,5-diethyl trimethylene carbonate (5,5-diethyl-1,3-dioxan-2-one), and 5-ethyl trimethylene carbonate (5-ethyl-1,3-dioxan-2-one) or any of these materials containing other substituents such as hydroxyl groups, a C₁-C₂₀ alkyl group, a C₁-C₂₀ hydroxy alkyl group, a C₂-C₁₀- alkyloxyalkylene group, halide (F, Cl, Br, I), haloalkyl, cyano, a C₁-C₄ fluoroalkyl, a C₁-C₄ hydroxfluoroalkylene group, a hydroxyl group, a hydroxyl group blocked with an acid labile group, a carboxyl group, a carboxyl group blocked with an acid labile group, a carboxyalkylene group, a carboxyalkylene group where the carboxyl group is blocked by an acid labile group, an alkyloxycarbonyl group, an alkylcarbonyl group, an alkyl group

containing a double or triple carbon bound a refractory element such as a C₃-C₂₀ trialkylsilyl group, a C₃-C₂₀ trialkylgermanium group, a C₃-C₂₀ trialkyltin group, or C₃-C₂₀ trialkyltitanium group.

Specific examples of epoxides which may be employed as precursors for one of the blocks in the novel block copolymer of structure (1) are as follows: propylene oxide, butylene oxide, ethylene oxide, styrene oxide, glycidyl ether with an R group that is C₁-C₂₀ hydrocarbon (linear, aliphatic, aromatic, branched, etc., and silicon containing epoxy monomers (e.g. with trimethylsilyl (TMS) pendant groups attached to oxygen or carbon), or any of these materials with other substituents such as hydroxyl groups, a C₁-C₂₀ alkyl group, a C₁-C₂₀ hydroxy alkyl group, a C₂-C₁₀- alkyloxyalkylene group, halide (F, Cl, Br, I), haloalkyl, cyano, a C₁-C₄ fluoroalkyl, a C₁-C₄ hydroxfluoroalkylene group, a hydroxyl group, a hydroxyl group blocked with an acid labile group, a carboxyl group, a carboxyl group blocked with an acid labile group, a carboxyalkylene group, a carboxyalkylene group where the carboxyl group is blocked by an acid labile group, an alkyloxycarbonyl group, an alkylcarbonyl group, an alkyl group containing a double or triple carbon bound a refractory element such as a C₃-C₂₀ trialkylsilyl group, a C₃-C₂₀ trialkylgermanium group, a C₃-C₂₀ trialkyltin group, or a C₃-C₂₀ trialkyltitanium group.

Specific examples of lactams which may be employed as precursors for one of the blocks in the novel block copolymer of structure (1): β -propiolactam, γ -butyrolactam, δ -valerolactam, and ϵ -caprolactam and these lactams or any of these materials with other substituents such as hydroxyl groups, a C₁-C₂₀ alkyl group, a C₁-C₂₀ hydroxy alkyl group, a C₂-C₁₀- alkyloxyalkylene group, halide (F, Cl, Br, I), haloalkyl, cyano, a C₁-C₄ fluoroalkyl, a C₁-C₄ hydroxfluoroalkylene group, a hydroxyl group, a hydroxyl group blocked with an acid labile group, a carboxyl group, a carboxyl group blocked with an acid labile group, a carboxyalkylene group, a carboxyalkylene group where the carboxyl group is blocked by an acid labile group, an alkyloxycarbonyl group, an alkylcarbonyl group, an alkyl group containing a double or triple carbon bound a refractory

element such as a C₃-C₂₀ trialkylsilyl group, a C₃-C₂₀ trialkylgermanium group, a C₃-C₂₀ trialkyltin group, or C₃-C₂₀ trialkyltitanium group.

In another embodiment the novel diblock polymer having structure (1) one polymer block is etch resistant and derived from either a vinyl aryl compounds or an etch resistant vinyl alkyl, an etch resistant alkyl methacrylate, an etch resistant alkyl acrylate, or an etch resistant substituted cyclic monomer. The etch resistance of the vinyl alkyl, alkyl methacrylate, alkyl acrylate or substituted cyclic monomer is imparted by having the alkyl group or the substituents on the cyclic monomer contain either a refractory element as described above, or contains an aryl substituent, an arylalkylene, or an alkylarylene, or a polycyclic alkyl substituent.

In another embodiment of the novel block copolymer having structure (1), the ethylenically unsaturated monomer which can be employed to form a polymer block A may be comprised of units derived from vinyl aryl monomer and polymer block B may be comprised of units derived from either an ethylenically unsaturated polymerizable monomers, a lactone or a cyclic carbonate and further wherein block A and B are phase separable. In this embodiment the ethylenically unsaturated polymerizable monomers may be selected from an acrylate or methacrylate or comprised of units derived from ring-opening polymerization (ROP) of cyclic monomers, selected from lactone, or cyclic carbonates, wherein the vinyl aryl chosen from the group consisting of styrene, 4-methylstyrene, 4-ethyl styrene, 4-n-propylstyrene, 4-isopropylstyrene, 4-n-butylstyrene, 4-trimethylsilylstyrene, 4-cyclohexylstyrene, 4-cyclopentylstyrene, 3-trimethylsilylstyrene, 4-tertbutylstyrene, 4-isoamylstyrene, 4-methoxystyrene, 4-n-propyloxystyrene, 4-isopropyloxystyrene, 4-n-butyloxystyrene, 4-cyclohexyloxystyrene, 4-cyclopentyloxystyrene, 4-trimethylsilyloxystyrene, 3-trimethyloxysilylstyrene, 4-tertbutyloxystyrene, 4-isoamyloxystyrene, 4-trifluoromethylstyrene, 3-trifluoromethylstyrene, 4-trifluoromethyloxystyrene, 3-trifluoromethyloxystyrene, 4-tert-butyloxycarbonyloxystyrene, 4-tertbutyloxycarbonylstyrene, 4-tertbutyloxycarbonyloxystyrene, 4-vinylbenzoic acid,

4-vinylbenzyl alcohol, methyl 4-vinylbenzoate, α -methylstyrene, 2,4-dimethylstyrene, 2,5-Dimethylstyrene, and 2,4,6-trimethylstyrene, and where the methacrylate, if present, may be selected from the group consisting of methyl methacrylate, ethyl methacrylate, propyl methacrylate, *n*-butyl methacrylate, isopropyl methacrylate, *t*-butyl methacrylate, cyclohexyl methacrylate, cyclopentyl methacrylate, isobornyl methacrylate, hydroxy-1-adamantyl methacrylate, 2-ethyl-2-adamantyl methacrylate, 2-adamantyl methacrylate, ethylcyclopentylmethacrylate, 2-ethyl-, adamantyloxymethyl methacrylate, tricyclo[5,2,1,0^{2,6}]deca-8-yl methacrylate, 2-oxoadamantyl methacrylate, 2-ethyl-2-diamantylmethacrylate, 2-carboxyethyl methacrylate methyl ester, 2-carboxyethyl methacrylate ethyl ester, 2-carboxyethyl methacrylate propyl ester, 2-carboxyethyl methacrylate *n*-butyl ester, 2-carboxyethyl methacrylate *tert*-butyl ester, 2-carboxyethyl methacrylate adamantyl ester, and 2-hydroxyethyl methacrylate, and acrylates if present may be selected from the group consisting of methyl acrylate, ethyl acrylate, propyl acrylate, *n*-butyl acrylate, *t*-butyl acrylate, isopropyl acrylate, cyclohexyl acrylate, cyclopentyl acrylate, isobornyl acrylate, hydroxy-1-adamantyl acrylate, 2-ethyl-2-adamantyl acrylate, 2-adamantyl acrylate, ethylcyclopentylacrylate, 2-ethyl-, adamantyloxymethyl acrylate, tricyclo[5,2,1,0^{2,6}]deca-8-yl acrylate, 2-oxoadamantyl acrylate, 2-ethyl-2-diamantylacrylate, 2-carboxyethyl acrylate methyl ester, 2-carboxyethyl acrylate ethyl ester, 2-carboxyethyl acrylate propyl ester, 2-carboxyethyl acrylate *n*-butyl ester, 2-carboxyethyl acrylate *tert*-butyl ester, 2-carboxyethyl acrylate adamantyl ester, and 2-hydroxyethyl acrylate. Also in this embodiment the lactone may be selected from the group consisting of beta-propiolactone, gamma-butyrolactone, delta-valerolactone, epsilon-caprolactone, beta-butyrolactone, gamma-valerolactone, pivalolactone, 1,5-dioxepan-2-one, 5-(benzyloxy)oxepan-2-one, 7-oxooxepan-4-yl 2-bromo-2-methylpropanoate, 5-phenyloxepan-2-one, 5-methyloxepan-2-one, 1,4,8-trioxa(4,6)spiro-9-undecane, 5-(benzyloxymethyl)oxepan-2-one, 7-oxooxepan-4-yl 3-hydroxy-2-(hydroxymethyl)-2-methylpropanoate, (Z)-6,7-dihydrooxepin-2(3H)-one, glycolide, D-lactide, L- racemic lactide, 1:1 D:L forms lactide, and

meso-lactide. Also in this embodiment the cyclic carbonate may be selected from the group consisting of trimethylene carbonate, tetramethylene carbonate, pentamethylene carbonate, 5-methyl-2-oxo-1,3-dioxane-5-carboxylic acid, methyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, tert-butyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, ethyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 5-methyl-2-oxo-1,3-dioxane-5-carbonyl chloride, benzyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 2-(tetrahydro-2H-pyran-2-yloxy)ethyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 2,2,3,3,4,4,5,5,6,6,7,7,7-tridecafluoroheptyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 2,2,2-trifluoroethyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, prop-2-ynyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, allyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 2-(2-methoxyethoxy)ethyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 3-(tert-butylthio)propyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 2-(tert-butoxycarbonylamino)ethyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 2-(pyridin-2-yldisulfanyl)ethyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, 4-(hydroxymethyl)benzyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate, tert-butyl 4-(5-methyl-2-oxo-1,3-dioxane-5-carboxyloxy)piperidine-1-carboxylate, N-benzyl-5-methyl-2-oxo-1,3-dioxane-5-carboxamide, N-isopropyl-N,5-dimethyl-2-oxo-1,3-dioxane-5-carboxamide, 5,5-dimethyl trimethylene carbonate (5,5-dimethyl-1,3-dioxan-2-one, 5-methyl trimethylene carbonate (5-methyl-1,3-dioxan-2-one), 5,5-diethyl trimethylene carbonate (5,5-diethyl-1,3-dioxan-2-one), and 5-ethyl trimethylene carbonate (5-ethyl-1,3-dioxan-2-one).

In another more specific embodiment of the novel block copolymer, having structure (1) the ethylenically unsaturated monomer which can be employed to form polymer block A may be comprised of units derived from vinyl aryl monomer and block B may be comprised of units derived from either an ethylenically unsaturated and polymer block B is comprised of units derived from either an ethylenically unsaturated polymerizable monomer, or a lactone and further wherein block A and B are phase separable. In this embodiment the ethylenically unsaturated polymerizable monomers may be. In this embodiment the ethylenically unsaturated polymerizable monomer is selected from an acrylate or methacrylate or comprised on units derived from ring-opening

polymerization (ROP) of a cyclic carbonates, where the vinyl aryl is chosen from the group consisting of styrene, 4-methylstyrene and 4-ethyl styrene, and where the methacrylate, if present, is selected from the group consisting of methyl methacrylate, ethyl methacrylate, and propyl methacrylate; and the lactone, is selected from the group consisting of beta-propiolactone, gamma-butyrolactone, delta-valerolactone, epsilon-caprolactone, beta-butyrolactone, gamma-valerolactone, pivalolactone, 1,5-dioxepan-2-one, 5-(benzyloxy)oxepan-2-one a cyclic carbonate selected from trimethylene carbonate, tetramethylene carbonate, pentamethylene carbonate, glycoside, 7-oxooxepan-4-yl 2-bromo-2-methylpropanoate, 5-phenyloxepan-2-one, 5-methyloxepan-2-one, 1,4,8-trioxa(4,6)spiro-9-undecane, 5-(benzyloxymethyl)oxepan-2-one, 7-oxooxepan-4-yl 3-hydroxy-2-(hydroxymethyl)-2-methylpropanoate, (Z)-6,7-dihydrooxepin-2(3H)-one, D-lactide, L-lactide, meso-lactide, and further wherein block A and B are phase separable.

The present invention also relates to a composition comprising the novel block copolymer of structure (1) and a solvent. Solvents suitable for dissolving either the novel block copolymer of structure (1) alone or in a composition with other components include propylene glycol monomethyl ether acetate (PGMEA), ethoxyethyl propionate, anisole, ethyl lactate, 2-heptanone, cyclohexanone, amyl acetate, n-butyl acetate, n-amyl ketone (MAK), gamma-butyrolactone (GBL), toluene, and the like. In an embodiment, specifically useful casting solvents include propylene glycol monomethyl ether acetate (PGMEA), gamma-butyrolactone (GBL), or a combination of these solvents. The wt% of the novel block copolymer having structure (1) in a solvent may range from 0.2 to 10 wt%. In another embodiment the range may be from 0.5 to 10 wt%. In yet another embodiment the range may be 0.5 to 5 wt%. In yet another more specific embodiment the range may be 0.8 to 1.2 wt%

Solutions containing the novel block copolymer of structure (1) can further comprise additional components and/or additives selected from the group consisting of: inorganic-containing polymers; additives including small molecules,

inorganic-containing molecules, surfactants, photoacid generators, thermal acid generators, quenchers, hardeners, cross-linkers, chain extenders, and the like; and combinations comprising at least one of the foregoing, wherein one or more of the additional components and/or additives co-assemble with the block copolymer to form the block copolymer assembly.

In another aspect of this invention the novel composition comprised of the first block copolymer of structure (1) may further contain a second different block copolymer block copolymer of structure (1). In this aspect of the invention the second block copolymer different from the first block copolymer may be present from 5 to 50 wt % of total solids.

In a further embodiment of the novel composition, the composition comprises the block copolymer of structure (1) and a second block copolymer which does not have a junction group but comprises blocks derived from either ethylenically unsaturated monomers or cyclic monomers as described above for the novel block copolymers. A specific example of a suitable second diblock copolymer containing a high etch resistant block and a highly etchable block would be the block copolymer poly(styrene-*b*-methylmethacrylate). Typically, the block copolymers suitable for being used in these inventions have a weight-averaged molecular weight (M_w) in the range of about 3,000 to about 500,000 g/mol and a number averaged molecular weight (M_n) of about 1,000 to about 60,000 and a polydispersity (M_w/M_n) (PD) of about 1.01 to about 6, or 1.01 to about 2 or 1.01 to about 1.5. Other specific non-limiting examples of other diblock copolymers not containing a junction group that are may be useful as additional components are poly(styrene-*b*-methyl methacrylate), poly(styrene-*b*-butadiene), poly(styrene-*b*-isoprene), poly(styrene-*b*-methyl methacrylate), poly(styrene-*b*-alkenyl aromatics), poly(styrene-*b*-(ethylene-propylene)), poly(styrene-*b*-*t*-butyl (meth)acrylate), poly(styrene-*b*-tetrahydrofuran), poly(styrene-*b*-ethyleneoxide), poly(styrene-*b*-dimethylsiloxane), poly(methyl methacrylate-*b*-dimethylsiloxane), and poly(methylmethacrylate-*b*-4-vinylpyridine)). All these polymeric materials share in common the presence of at least one block which has repeat units

resistant to plasma etching techniques typically employed in manufacturing IC devices, and at least one block which etches rapidly under these same conditions or that can be removed by chemical or photochemical processes. This allows for the directed self-assembled polymer to pattern transfer onto the substrate to affect via formation.

In this aspect of the invention the second block copolymer may be present from 1 to 20 wt%, or more preferentially from 5 to 10 wt % of total solids.

A further embodiment of this novel composition is where it further comprises a homopolymer as an additional component. This homopolymer may be a homopolymer derived from any of the ethylenically unsaturated monomers or cyclic monomer previously described above as suitable precursors for the block of the novel block copolymer having structure (1). Such a homopolymer component may be present in a composition in a level range from 0.5 to 10 wt % or more specifically 1 to 5 wt % of solids.

The molar ratio in the novel block copolymer of the repeat unit in block A to that of the repeat unit in block B is between 1.2 to 0.8 and further as described above in all its possible embodiments.

The present invention relates to forming patterns using the novel composition in processes comprising directed self assembly of the novel block copolymer. Any process that comprises the step of directed self assembly of the novel block copolymer may be used.

The composition comprising the novel block copolymer may be used to form films by applying the novel composition a substrate by processes such as spin-casting, dip-coating, doctor blading, spraying, or any other known processes. Films of the block copolymer can have a thickness of 1-1000 nm and more particularly 1-130 nm. It is particularly thicknesses film equal or greater than 25 nm in particular in the thickness range of 25 to 125 nm. Films can be optionally annealed to promote self-assembly and remove defects. Annealing processes include thermal annealing, thermal gradient annealing, solvent vapor annealing,

thermal solvent vapor annealing and the like. Thermal annealing can be carried out at a temperature above the glass transition temperature of the block copolymer and below the thermal decomposition temperature of the block copolymer. Thermal annealing may be carried out from a temperature of room temperature to about 300°C. Thermal annealing can be performed for a period of about 10 sec to about 100 hours, more particularly 30 seconds to 1 hour.

The block copolymers of the present invention can form self-assembled domains with various morphologies including lamella, cylinders, and spheres. The size (e.g. width) of these domains may be from 1nm to 100nm, from 2 nm to 30 nm, or more particularly from 3 nm to 20 nm. The present invention further relates to novel processes which employ the novel block copolymer of structure (1). One of the unexpected results imparted by the novel block copolymers of structure (1) is that, when they are coated on a substrate and made to undergo self-assembly, the underlayer does not need to be non-preferential (e.g. neutral). This is due to the larger process latitude (with respect to the underlayer surface affinity) for perpendicular alignment of the novel block copolymer domains imparted by the presence of the surface active junction group in structure (1). Another unexpected results imparted by the novel block copolymers of structure (1) is that they afford block copolymer domains with perpendicular alignment relative to the substrate for much thicker block copolymer films than analogous block copolymers without the novel surface-active junction group. Specifically, ranges from 25 nm to 125 nm are possible.

As a non-limiting example of how this may be used in pattern transfer, when the novel block copolymer is coated on an underlayer (i.e. substrate) and further processed the block copolymer forms microphase-separated domains that comprise cylindrical microdomains oriented perpendicular to the underlayer surface. This is because neither domain has any preference to associate with the underlayer, and which further provide parallel line/space patterns in the block copolymer assembly. The domains, so oriented, are thermally stable under further processing conditions. Thus, after forming a coating layer of the novel

block copolymer on an underlayer, and causing it to self-assemble by baking and/or annealing, the domains of the block copolymer will form on and remain perpendicular to the underlayer surface.

One of the domains may be selectively removed in the presence of the other domain to generate an etched domain pattern. This selective removal may be accomplished by wet or dry processes. In one example wet or plasma etching could be used with optional UV exposure. Wet etching could be with acetic acid. Standard plasma etch process, such as a plasma comprising oxygen may be used; additionally argon, carbon monoxide, carbon dioxide, CF_4 , CHF_3 , may be present in the plasma. In the case of a thermally decomposable polymer block, the selective removal may be accomplished by a thermal bake. In another example, one of the block copolymer domains may be selectively modified after self-assembly to increase its etch resistance. For example, etch resistant metal or inorganic species may be introduced by chemical infiltration from solution or vapor. Domains or functional groups may be selectively reacted with reactive compounds such as silanes or silyl chlorides. Alternatively, cycles of atomic layer deposition (e.g. sequential infiltration synthesis) can be used to infiltrate etch-resistant materials into one domain. After modification, wet or dry etch processes may be used to remove the less resistant domain.

After selective removal process, the resulting etched domain pattern can be further pattern transferred in the substrate layers. These self-assembled block copolymer patterns are transferred into the underlying substrate using known techniques.

In some cases, it may be advantageous to invert the tone of the pattern prior to transfer into the substrate or after transfer into a layer of the substrate. This may be accomplished by standard tone-inversion processes such as backfilling an organic pattern with an etch-resistant spin-on dielectric tone-inversion material.

Figure 4a and b show respectively a schematic depiction of the novel block copolymer of structure (1) and a schematic depiction of its self assembly. Specifically, Figure 4a shows a schematic depiction of the novel block copolymer.

Figure 4b shows a schematic depiction of the block copolymer depicted schematically in Figure 4a when coated on an underlayer (i.e. substrate) undergoing two type of self-assembly. In the top view of Figure 4a the film of the block copolymer forms domains which orient to form vertically oriented lamellae. In the bottom view of Figure 4b these domains in the film orient to form vertically oriented cylinder domains.

Figure 5a-c shows schematics of the directed self-assembly (DSA) of the novel block copolymer of structure (1). Specifically Figure 5a shows graphoepitaxy DSA where most of the novel block copolymer is confined within the recessed region in the topographical prepattern, the domains of this novel block copolymer subdivide the recessed region. Figure 5b shows coating of this novel block copolymer over and within shallow, periodic topographical prepattern where the domains of this block copolymer are aligned to the topographical prepattern. Finally, Figure 5c shows chemoepitaxial DSA where a film of this novel block copolymer is coated over the chemical prepattern comprised of preferential and non-preferential regions (such as the non-limiting example of a neutral region) with domains of this novel block copolymer aligned to the wetting features. Unlike, conventional block polymers, the inventive materials of structure (1) with a surface active junction group, do not require the presence of substantially non-preferential regions and therefore provide the wider process latitude with respect to underlayer surface affinity. That is, these novel materials can form domains which orient perpendicular to the substrate during self-assembly for a wider range of underlayer surface property. This can enable more robust SA performance across surfaces with imperfect or non-uniform surface properties, for example. The imperfect or non-uniform surface properties may be the result of damage, contamination, imperfect

deposition/grafting conditions, impurities, poor compositional uniformity of the underlayer material, or other reasons.

In the above processes, and the following inventive processes employing the novel composition comprised of the novel block copolymer of structure (1), other type of substrate may be used. As an example, a substrate which has a coating of high carbon underlayer and a silicon antireflective coating may be used as a substrate. The high carbon underlayer can have coating thickness of about 20 nm to about 2 microns. Over this is coated a silicon antireflective coating of about 10 nm to about 100 nm. In instances where an orientation of the self-assembled cylinder of the novel block copolymer perpendicular to the substrate is desired a neutral layer may be employed.

This invention may be employed also in a variety of other substrates such as those prepared by chemically vapor deposition (CVD), physical vapor deposition (PVD) and atomic layer deposition (ALD). The preparation of substrates using these CVD and PVD is discussed in the following reference: "HANDBOOK OF THIN-FILM DEPOSITION PROCESSES AND TECHNIQUES, Principles, Methods, Equipment and Applications, Second Edition, Edited by Krishna Seshan, Intel Corporation, Santa Clara, California, Copyright © 2002 by Noyes Publications, Library of Congress Catalog Card Number: 2001135178, ISBN: 0-8155-1442-5, Noyes Publications / William Andrew Publishing, 13 Eaton Avenue Norwich, NY 13815 ,Chapter 1 title: Deposition Technologies and Applications: Introduction and Overview, page 11-page 43, Chapter 1 Authors: Werner Kern and Klaus K. Schuegraf." The preparation of substrates by ALD is described in the following reference: "Chemical Review 2010, 110, page111-131."

In another aspect of this invention the novel composition comprised of the block copolymer having structure (1) is employed in a method for vertically orienting first and second block copolymer domains over an unpatterned substrate using a layer of a block copolymer having a periodicity of L_0 comprising steps a) and b); wherein,

- a) forming a coating layer of a block copolymer from the novel composition on said unpatterned substrate; and
- b) annealing the layer of the block copolymer to generate a non-zero positive integer number of first and second block copolymer domains, vertically oriented on said unpatterned substrate.

In this inventive method the unpatterned substrate may be selected from the group consisting of a polymer brush layer, a cross-linked polymer layer, a self-assembled monolayer, a layer of anti-reflection coating, a layer deposited by chemically vapor deposition (CVD), a layer of carbon, a layer deposited by physical vapor deposition (PVD), a layer deposited by atomic layer deposition (ALD).

In another aspect of this invention the novel composition comprised of the block copolymer having structure (1) is employed in a method of vertically orienting first and second block copolymer domains over a first patterned substrate where the height of topography of the pattern on the substrate is at least 0.7 times L_0 and aligning the domains with the pattern, using a coating comprised of a block copolymer having a periodicity of L_0 comprising steps a) and b); wherein,

- a) forming a coating layer of the block copolymer with a junction group with the composition of the aforementioned novel composition comprised of the block copolymer having structure (1) on said first topographical substrate wherein the average thickness of the coating layer of the block copolymer is less than the height of the topography of the first topographical substrate, wherein the block copolymer layer is laterally confined by the topography; and,
- b) annealing the block copolymer layer to generate first and second block copolymer domains, which are vertically oriented on said first patterned substrate, and confined within the recessed region.

In this aspect of the invention the first patterned substrate is a topography-forming material over an underlayer where the topography-forming material may

be selected from the group consisting of resist materials, a cross-linked polymer layer, a layer of anti-reflection coating, a layer deposited by chemical vapor deposition (CVD), a layer deposited by physical vapor deposition (PVD), a layer deposited by atomic layer deposition (ALD), and further where the underlayer is selected from the group consisting of a polymer brush layer, a cross-linked polymer layer, a self-assembled monolayer, a layer of anti-reflection coating, a layer deposited by chemical vapor deposition (CVD), a layer deposited by physical vapor deposition (PVD), a layer deposited by atomic layer deposition (ALD).

Furthermore, in this aspect of the invention the resist material may be selected from the group consisting of i-line photoresists, g-line photoresists 284 nm photoresists, 193 nm photoresist, Extreme Ultraviolet photoresists, and electron beam photoresists. Another embodiment is where the antireflective coating material is selected from the group consisting of bottom antireflective coatings, silicon antireflective coatings, and titanium antireflective coatings.

In another aspect of this invention the novel composition comprised of the block copolymer having structure (1) is employed in a method of vertically orienting, first and second block copolymer domains with a periodicity of L_0 over a second patterned substrate having a topographical pattern of which the height of topography is larger than 0.7 times L_0 and a pitch P_1 where the pitch P_1 is a non-zero positive integer multiplied by L_0 , and aligning the domains with the pattern, using the novel coating comprised of a block copolymer comprising steps a) and b); wherein,

a) forming a coating layer from block copolymer with a surface-active junction group with aforementioned the composition of the novel composition comprised of the block copolymer having structure (1) on said second patterned substrate, where the thickness of the coating layer of the block copolymer is more than the height of the topography of the second patterned substrate; and,

b) annealing the block copolymer layer to generate a non-zero positive integer number of first and second block copolymer domains vertically oriented on said

second patterned substrate and aligning them to the second patterned substrate where the sum of vertically oriented domains is equal or larger than the pitch P_1 the of topographical pattern.

In this aspect of the invention the first patterned substrate is prepared by treating topographical organic features on top of inorganic layer with plasma which selectively trims the organic features and modifies the exposed surface region of the inorganic layer. In a further aspect of this embodiment the material of topographical organic features is selected from the group consisting of patterned i-line photoresists, g-line photoresists 248 nm photoresists, 193 nm photoresist, Extreme Ultraviolet photoresists, and electron beam photoresists over a thin underlayer. In a final aspect of this embodiment the underlayer may be an antireflective coating, or an inorganic layer material is selected from silicon antireflective coatings, or titanium antireflective coatings.

In another aspect of this invention the novel composition comprised of the block copolymer having structure (1) is employed in a method of vertically orienting first and second block copolymer domains over a substrate having a surface chemical prepattern having a pitch P_2 , where the pitch P_2 is a non-zero positive integer multiplied by L_0 and aligning the domains with the prepattern, using a coating comprised of a block copolymer where the block copolymer has a periodicity of L_0 comprising steps a) and b)

- a) forming a coating layer of the block copolymer with a surface-active junction group with the novel composition containing the block copolymer of structure (1) on the substrate having a surface chemical prepattern,
- b) annealing the block copolymer layer to generate vertically oriented first and second block copolymer domains aligned with the substrate having a surface chemical prepattern having a pitch P_2 .

In this aspect of the invention the substrate having a surface chemical prepattern is comprised of alternating preferential wetting regions and orientation control regions where the sum of the width of preferential wetting regions and the

width of the orientation control region is equal to the prepattern pitch P_2 , where the materials used in the orientation control region supports vertical orientation of the block copolymer domains, and the preferential wetting region has a lower interfacial energy to one of the block copolymer domain compared to the other block copolymer domain.

The following specific examples will provide detailed illustrations of the methods of producing and utilizing compositions of the present invention. These examples are not intended, however, to limit or restrict the scope of the invention in any way and should not be construed as providing conditions, parameters or values which must be utilized exclusively in order to practice the present invention.

EXAMPLES

I. Synthesis of block copolymers and relevant materials

I-0. Instruments and chemicals

Chemicals not specified in Table 1 were obtained from the Aldrich Chemical Company (Sigma-Aldrich Corp St. Louis, MO, USA) and used as received. Unless otherwise indicated spinning and baking of films was done with Laurel WS-650-23B spin coater and a Tokyo Electron Ltd. Clean Track ACT-8. Nordson (300 Nordson Dr. M/S 47 Amherst, OH 44001 U.S.A. MARCH etcher was used for plasma etching of non-Si containing domain. Top down and cross section images were respectively taken on an AMAT (Applied Materials, Inc. 3050 Bowers Avenue P.O. Box 58039 Santa Clara, CA 95054-3299 U.S.A.) NanoSEM 3D and a Hitachi (Hitachi High Technologies America Inc. 10 North Martingale Road, Suite 500 Schaumburg, Illinois 60173-2295) S-5500. Table 1 gives a list of abbreviations for compounds and solvents used. In this Table "Sigma-Aldrich" refers to the "Sigma-Aldrich Corporation" located at 3050 Spruce St., St. Louis, Missouri 63103; "Synquest" refers to "SynQuest Laboratories, Inc." located at 13201 Rachael Blvd, Rt 2054, "Exfluor" refers to

“Exfluor Research Corporation” located at 2350 Double Creek Dr, Round Rock, TX 78664, “Perstorp Chemicals” refers to “Perstorp Specialty Chemicals” located at Neptunigatan 1, 211 20 Malmö, Sweden, “Richman Chemicals” refers to “Richman Chemicals Custom Solutions” located at 768 North Bethlehem Pike, PA 19002. The coating SiArc solution (Shin Etsu SHB A-940 L35) was obtained from the Shin-Etsu Chemical Co., Ltd (Tokyo 100-0004, Japan).

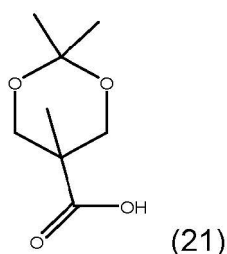
The coating solutions, AZEMBLY™ NLD-089, AZEMBLY™ NLD-208D, AZEMBLY™ NLD-128, and AZEMBLY™ NLD-127 were obtained from AZ Electronic Materials (Somerville, NJ).

Table 1

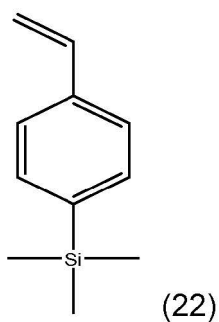
ABBREVIATION	DESCRIPTION	SUPPLIER
DBU	1,8-Diazabicyclo[5,4,0]undec-7-ene	Sigma Aldrich
DCM	Dichloromethane	Sigma Aldrich
AcCl	Acetyl Chloride	Sigma Aldrich
TEA	Triethylamine	Sigma Aldrich
BriBr	α -Bromoisobutyryl bromide	Sigma Aldrich
THF	Tetrahydrofuran	Sigma Aldrich
MeOH	Methanol	Sigma Aldrich
ACN	Acetonitrile	Sigma Aldrich
S	Styrene	Sigma Aldrich
EtAc	Ethyl acetate	Sigma Aldrich
Tol	Toluene	Sigma Aldrich
C8OL	1H,1H-Perfluoro-1-heptanol	Synquest
C10GOL	Fluorinated triethylene glycol monobutyl ether	Exfluor
C9OL	1H,1H-Perfluoro-1-nonanol	Synquest
C4OL	2,2,3,3,4,4,4-Heptafluoro-1-butanol	Sigma Aldrich
BISMPA	Dimethylol propionic acid	Perstorp Chemicals
ODBr	Octadecyl bromide	Sigma Aldrich
Chol	Cholesterol	Sigma Aldrich
TMC	Trimethylene carbonate	Richman Chemicals
CuBr	Copper (I) bromide	Sigma Aldrich
PMDTA	N,N,N',N',N''-pentamethyldiethylenetriamine	Sigma Aldrich

Anisole	Anisole	Sigma Aldrich
Si Gel	Silica Gel	Sigma Aldrich
DCC	<i>N,N'</i> -Dicyclohexylcarbodiimide	Sigma Aldrich
DMAP	4-Dimethylaminopyridine	Sigma Aldrich

Acetal-protected BISMPA monomer (AcBISMPA) (21) was prepared as previously reported (ACS Macro Letters, 2(1), 19-22; 2013).

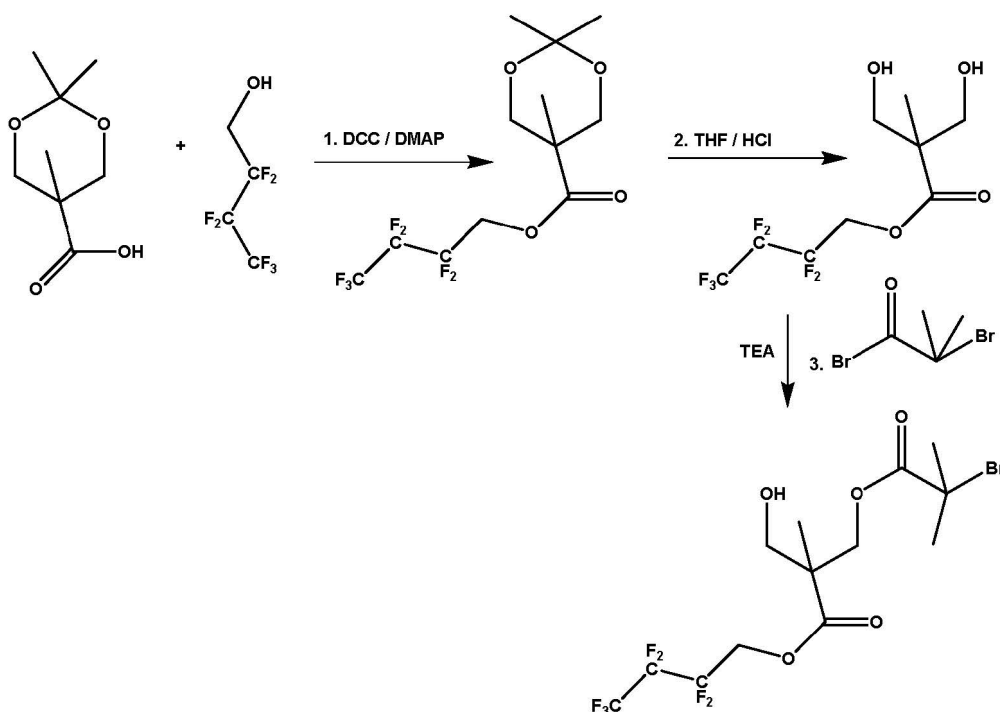


Trimethyl silylstyrene (TMSS) (22) was synthesized as previously reported (Macromolecules 2009, 42, 4614–4621).



I-1. Synthesis of block copolymers with surface-active junction groups (JBCPs) using macroinitiators prepared by atom transfer radical polymerization (ATRP)

Example 1. Synthesis of ATRP and ROP initiator with pendant group C4OL



Scheme 12

To a 100 ml round bottom flask (RBF) equipped with a magnetic stir bar, 1.0 (5.74 mm) of AcBISMPA, 1.15 g (5.74 mm) of C4OL, 0.7g (5.74 mm) of DMAP were added to 30 ml of dry DCM. The reaction was stirred under N₂ for 10 min at room temperature after which point 1.18g (5.74 mm) of DCC in 10 ml of DCM was added drop wise over 10 minutes. The reaction was stirred for 24 hours and was stopped the solution was filtered to remove the dicyclohexylurea. DCM was evaporated under vacuum and the resulting crude product was purified by passing through a silica gel column with Hexane: EtOAc 90:10 as the eluent to obtain the acetal-protected C4OL ester. The product was confirmed by ¹H NMR and ¹⁹F NMR. ¹H NMR (400 MHz, CDCl₃ 23°C, ppm): 1.21 (3H), 1.46 (6H), 3.71 (2H), 4.21 (2H), and 4.69 (2H). ¹⁹F NMR (400 MHz, CDCl₃ 23°C, ppm): -81.00, -120.48, -127.79.

In the next step, 1.5 g of the intermediate was added to 15 ml of THF in a 250 ml RBF equipped with a magnetic stir bar. To this solution, 20 ml of 1 M HCl was added and the reaction was allowed to stir for 24 hours at room temperature.

After this time, THF was removed under vacuum and 50 ml of EtOAc was added to the acidic water solution. The organic layer was washed with saturated K_2CO_3 and brine and collected over magnesium sulfate. Pure diol was collected by filtering the magnesium sulfate and evaporating the EtOAc under vacuum. The product was confirmed by 1H NMR. 1H NMR (400 MHz, $CDCl_3$ 23°C, δ): 1.13 (3H), 2.68 (2H), 3.80 (2H), 3.96 (2H), and 4.70 (2H).

To obtain the ATRP and ROP initiator with pendant C4OL chain, the 1.0 g (3.16 mm) diol from above step was dissolved in 20 ml of DCM in a 100 ml RBF equipped with a magnetic stir bar. To this solution, 0.32 g (3.16 mm) of TEA was added and the reaction mixture was stirred under N_2 flow for 10 min at room temperature. 0.54 g (2.37 mm) of BriBr in 10 ml of DCM was then added dropwise over 30 min. The resulting mixture was further stirred at r.t. for 24 hours at which point the reaction was stopped by filtering the solution to remove the TEA.HCl salt. The filtrate was collected and washed with 1X saturated K_2CO_3 , 1X 1 M HCl, 1X deionized water and 1X brine solution. The organic layer was collected over magnesium sulfate and the crude product was obtained by filtering the DCM solution and evaporating the DCM layer under vacuum. Pure ATRP and ROP initiator with pendant C4OL chain was obtained by purifying the crude product with column chromatography using Si gel column with Hex: EtOAc 90:10 as the eluant. The product was confirmed by 1H NMR and ^{19}F NMR. 1H NMR (400 MHz, $CDCl_3$ 23°C, ppm): 1.32 (3H), 1.94 (6H), 2.31 (1H), 3.80 (2H), 3.36-3.46 (2H), and 4.67 (2H). ^{19}F NMR (400 MHz, $CDCl_3$ 23°C, ppm): -80.80, -120.45, -127.57.

Example 2. Synthesis of ATRP and ROP initiator with pendant group C8OL

The same reaction conditions used for Example 1 were employed, except C8OL was used instead of C4OL. In the first step, the product acetal protected C8OL ester was confirmed by 1H NMR. 1H NMR (400 MHz, $CDCl_3$ 23°C, ppm): 1.21 (3H), 1.46 (6H), 3.71 (2H), 4.21 (2H), and 4.69 (2H). In the next step the product was confirmed by 1H NMR. 1H NMR (400 MHz, $CDCl_3$ 23°C, δ): 1.08 (3H), 2.54

(2H), 3.77 (2H), 3.94 (2H), and 4.51 (2H). In the final step the ATRP and ROP initiator with pendant C8OL group was confirmed by ^1H NMR and ^{19}F NMR. ^1H NMR (400 MHz, CDCl_3 , 23°C , ppm): 1.28 (3H), 1.94 (6H), 2.36 (1H), 2.53 (2H), 3.77 (2H), 4.33 (1H), and 4.45-4.50 (3H).

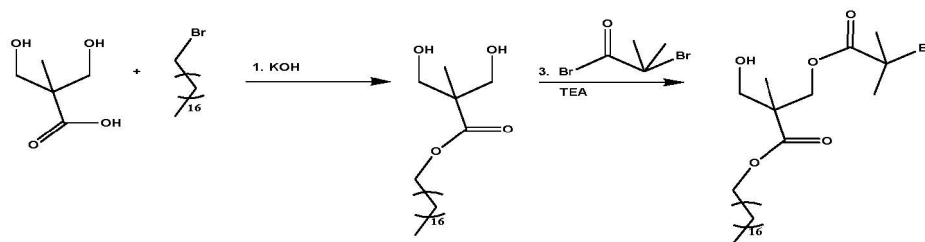
Example 3. Synthesis of ATRP and ROP initiator with pendant group C9OL

The same reaction conditions used for Example 1 were employed, except C9OL was used instead of C4OL in the same molar amount. In the first step, the acetal-protected C9OL ester was confirmed by ^1H NMR. ^1H NMR (400 MHz, CDCl_3 , 23°C , ppm): 1.21 (3H), 1.46 (6H), 3.69 (2H), 4.20 (2H), and 4.71 (2H). In the next step the diol was confirmed by ^1H NMR. ^1H NMR (400 MHz, CDCl_3 , 23°C , δ): 1.14 (3H), 3.81 (2H), 3.97 (2H), and 4.71 (2H). In the final step to obtain the ATRP and ROP initiator with pendant C9OL chain, the product was confirmed by ^1H NMR and ^{19}F NMR. ^1H NMR (400 MHz, CDCl_3 , 23°C , ppm): 1.32 (3H), 1.94 (6H), 2.24 (1H), 3.80 (2H), 4.37-4.46 (2H), 4.69 (2H). ^{19}F NMR (400 MHz, CDCl_3 , 23°C , ppm): -199.41, -121.93, -122.72, -123.21, and -126.11.

Example 4. Synthesis of ATRP and ROP initiator with pendant group C10GOL

The same reaction conditions as for Example 1 were employed, except C10GOL was used instead of C4OL. In the first step the C10GOL acetal protected ester was confirmed by ^1H NMR. ^1H NMR (400 MHz, CDCl_3 , 23°C , ppm): 1.21 (3H), 1.46 (6H), 3.71 (2H), 4.23 (2H), and 4.57 (2H). In the next step the pure diol was confirmed by ^1H NMR. ^1H NMR (400 MHz, CDCl_3 , 23°C , δ): 1.12 (3H), 2.63 (2H), 3.81 (2H), 3.95 (2H), and 4.58 (2H). In the final step to obtain the ATRP and ROP initiator with pendant C10GOL chain, the product was confirmed by ^1H NMR and ^{19}F NMR. ^1H NMR (400 MHz, CDCl_3 , 23°C , ppm): 1.32 (3H), 1.94 (6H), 2.24 (1H), 3.80 (2H), 4.37-4.46 (2H), 4.69 (2H). ^{19}F NMR (400 MHz, CDCl_3 , 23°C , ppm): -199.41, -121.93, -122.72, -123.21, and -126.11.

Example 5. Synthesis of ATRP and ROP initiator with pendant group ODBr



Scheme 13

To a 100 ml RBF equipped with a magnetic stir bar and a condenser, 2.0 g (14.91 mm) of BISMPA, 5.22 g (15.65 mm) of ODBr, 1.0 g (17.90 mm) of KOH and 50 ml of DMF were taken. The reaction was stirred at 100°C for 16 hours. The reaction was then cooled to room temperature and DMF was removed under vacuum. 100 ml of EtOAc was added to the remaining compound and was rinsed 2X with 100 ml deionized water and 1X with 50 ml saturated brine solution. The organic layer was dried over magnesium sulfate and the crude solids were obtained by evaporating the solvent under vacuum. Pure diol with octadecyl pendant group was obtained by passing the compound through a Si gel column with Hexane:EtOAc 60:40 as the eluent. The product was confirmed by ¹H NMR. ¹H NMR (400 MHz, CDCl₃ 23°C, ppm) 0.9 (3H), 1.08 (3H), 1.19-1.46 (30H), 1.69 (2H), 2.83 (2H), 3.76 (2H), 3.92 (2H), 4.19 (2H).

To obtain the ATRP and ROP initiator with pendant octadecyl chain, the 1.5 g (3.88 mm) diol from the above step was dissolved in 30 ml of DCM in a 250 ml RBF equipped with a magnetic stir bar. To this solution, 0.78 g (7.76 mm) of TEA was added and the reaction mixture was stirred under N₂ flow for 10 min at room temperature. 0.89 g (3.88 mm) of BriBr in 10 ml of DCM was then added dropwise over 30 min. The resulting mixture was further stirred at r.t. for 48 hours at which point the reaction was stopped by filtering the solution to remove the TEA.HCl salt. The filtrate was collected and washed with 1X saturated K₂CO₃, 1X 1 M HCl, 1X deionized water and 1X brine solution. The organic layer was

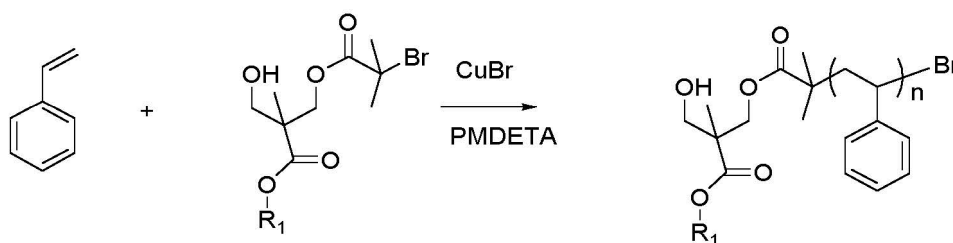
collected over magnesium sulfate and the crude product was obtained by filtering the DCM solution and evaporating the DCM layer under vacuum. Pure ATRP and ROP initiator was pendant C9OL chain was obtained by purifying the crude product with column chromatography using Si gel column with Hexane: EtOAc 80:20 as the eluent. The product was confirmed by ^1H NMR. ^1H NMR (400 MHz, CDCl_3 23°C, ppm): 0.9 (3H), 1.21-1.42 (33H), 1.68 (2H), 1.94 (6H), 2.45 (1H), 3.74 (2H), 4.16 (2H), 4.27-4.46 (2H).

Table 2. summarizes various ATRP and ROP dual initiators containing different pendant groups

Table 2

Example	Pendant Group
1	$\text{CH}_2\text{C}_3\text{F}_7$
2	$\text{C}_2\text{H}_4\text{C}_6\text{F}_{13}$
3	$\text{CH}_2\text{C}_8\text{F}_{17}$
4	$\text{CH}_2\text{CF}_2(\text{OCF}_2\text{CF}_2)_3\text{C}_2\text{F}_5$
5	$\text{C}_{18}\text{H}_{37}$

Example 6. Synthesis of hydroxyl-functional polystyrene with Initiator from Example 1



Scheme 14

To a 50 ml Schlenk flask equipped with a stir bar, 7.53 g, (7.25 mmol) of styrene(ST), 0.30 g (0.657 mmol) ATRP initiator from **Example 1**, 94 mg (0.657

mm) CuBr and 7.5g of Anisole were added. The flask was sealed with a rubber septum and was purged by bubbling nitrogen for one hour. At this point, 0.12 g (0.657 mmol) of PMDETA was added and the reaction flask was placed in an oil-bath set at 100°C for 150 minutes. The reaction was stopped by placing the Schlenk flask in an ice-water bath. The flask was opened to air at which time the reaction mixture turned dark green. The mixture was diluted by adding THF (100 ml) and was passed through a short silica-gel column to remove copper catalyst. The resulting solution was concentrated under vacuum and was precipitated twice in methanol from THF. The polymer was collected in a frit funnel and was dried under vacuum for 24 hours at 50°C. $M_n = 4.8k$, $M_w = 5.1k$, PDI = 1.06.

Example 7. Synthesis of hydroxyl-functional polystyrene with initiator from Example 2

The same conditions were employed as for Example 6 except that the ATRP initiator from **Example 2** was employed in the same molar amounts. The polymer obtained had $M_n = 5.2k$, $M_w = 5.6k$, PDI = 1.07.

Example 8. Synthesis of hydroxyl-functional polystyrene with initiator from Example 3

The same conditions were employed as for Example 6 except that the ATRP initiator from **Example 3** was employed in the same molar amounts. The polymer obtained had $M_n = 4.7k$, $M_w = 5.0k$, PDI = 1.07.

Example 9. Synthesis of hydroxyl-functional polystyrene with initiator from Example 3

The same conditions were employed as for Example 6 except that the ATRP initiator from **Example 3** was employed in the same molar amounts. The polymer obtained had $M_n = 9.4k$, $M_w = 10.1k$, PDI = 1.07. (AV7-47-1)

Example 10. Synthesis of hydroxyl-functional polystyrene with initiator from Example 4

The same conditions were employed as for Example 6 except that the ATRP initiator from **Example 4** was employed in the same molar amounts. The polymer obtained had $M_n = 6.9k$, $M_w = 7.4k$, $PDI = 1.07$.

Example 11. Synthesis of hydroxyl-functional polystyrene with initiator from Example 5

The same conditions were employed as for Example 6 except that the ATRP initiator from **Example 5** was employed in the same molar amounts. The polymer obtained had $M_n = 4.2k$, $M_w = 4.5k$, $PDI = 1.06$.

Example 12. Synthesis of hydroxyl-functional poly(trimethyl silylstyrene) with initiator from Example 2

The same procedures was employed as for Example 6 except that the following reagents were employed : 1.83 g, (10.41 mmol) of TMSS, 0.10 g (0.170mmol) ATRP initiator from **Example 2**, 30 mg (0.208 mmol) CuBr, 1.83 g of Anisole, and 36 mg (0.208 mmol) of PMDETA. The polymer obtained had $M_n = 7.5k$, $M_w = 8.1k$, $PDI = 1.09$.

Example 13. Synthesis of hydroxyl-functional poly(trimethyl silylstyrene) with initiator from Example 3

The same procedure was employed as for Example 6 except that the following reagents were employed: 3.0 g, (17.0 mmol) of TMSS, 0.13 g (0.170mmol) ATRP initiator from **Example 3**, 24 mg (0.170 mmol) CuBr, 3.0 g of Anisole and 30 mg (0.170 mmol) of PMDETA. The polymer had $M_n = 6.8k$, $M_w = 7.4k$, $PDI = 1.09$.

Example 14. Synthesis of hydroxyl-functional poly(trimethylsilylstyrene-r-styrene) with initiator from Example 4

The same procedure was employed as for Example 6 except that the following reagents were employed: 0.89 g, (5.08 mmol) of TMSS, 2.11 g (20.3 mmol) of styrene, 0.15 g (0.188mmol) ATRP initiator from **Example 4**, 27 mg (0.170 mmol) CuBr, 3.0 g of Anisole and 33 mg (0.188 mmol) of PMDETA. The polymer had $M_n = 4.7k$, $M_w = 5.1k$, $PDI = 1.09$. The ratio of Styrene:TMSS by 1H NMR = 74:26

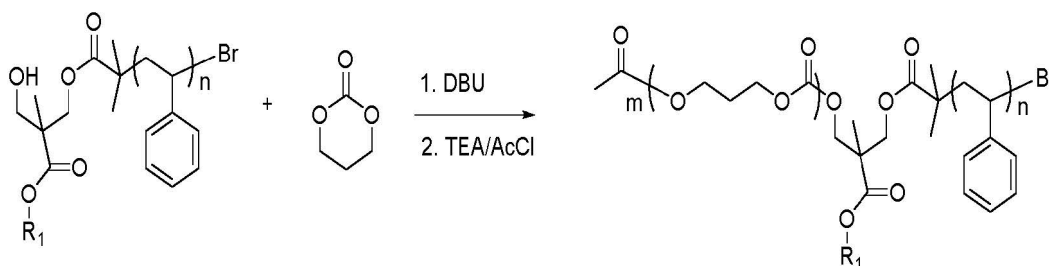
Example 15. Synthesis of hydroxyl-functional polystyrene with initiator from Example 3

The same procedure was employed as for Example 6 except that the following reagents were employed: 9.20 g, (88.20 mmol) of styrene, 0.15 g (0.196 mmol) ATRP initiator from **Example 3**, 28 mg (0.196 mm) CuBr, 9.20 g of 34 mg (0.196 mmol) of PMDETA The polymer had $M_n = 15.6k$, $M_w = 16.9k$, $PDI = 1.08$.

Table 3

Example	Monomer	Catalyst	Initiator	Pendant Group	Temp °C	Time (hours)	M_n	M_w	PDI
6	Styrene	CuBr/PMDETA	Example 1	$R_1 = CH_2C_3F_7$	100	2.5	4.8K	5.1K	1.06
7	Styrene	CuBr/PMDETA	Example 2	$R_2 = C_2H_4C_6F_{13}$	100	1	5.2K	5.6K	1.07
8	Styrene	CuBr/PMDETA	Example 3	$R_3 = CH_2C_8F_{17}$	100	3	4.7K	5.0K	1.07
9	Styrene	CuBr/PMDETA	Example 3	$R_3 = CH_2C_8F_{17}$	100	3	9.5K	10.1K	1.07
10	Styrene	CuBr/PMDETA	Example 4	$R_4 = CH_2CF_2(OCF_2CF_2)_3C_2F_5$	100	2.5	6.9K	7.4K	1.07
11	Styrene	CuBr/PMDETA	Example 5	$R_5 = C_{18}H_{37}$	100	2	4.2K	4.5K	1.07
12	Trimethyl silylstyrene	CuBr/PMDETA	Example 2	$R_2 = C_2H_4C_6F_{13}$	100	0.67	7.5K	8.1K	1.07
13	Trimethyl silylstyrene	CuBr/PMDETA	Example 3	$R_3 = CH_2C_8F_{17}$	100	0.5	6.8K	7.4K	1.09
14	Styrene and Trimethyl silylstyrene	CuBr/PMDETA	Example 4	$R_4 = CH_2CF_2(OCF_2CF_2)_3C_2F_5$	100	1	4.7	5.1K	1.09
15	Styrene	CuBr/PMDETA	Example 3	$CH_2C_8F_{17}$	100	2.33	15.6K	16.9K	1.08

Example 16. Synthesis of PS-b-PTMC with macroinitiator of Example 6 (JB4-F)



Scheme 15

Synthesis of PS-b-PTMC with pendant group CH₂C₃F₇. To an oven dried 4 ml glass vial equipped with a magnetic stir bar, 0.20 g (0.0454 mmol) hydroxyl-functional polystyrene with CH₂C₃F₇ pendant group from **Example 6**, 0.52 g (5.17 mmol) of TMC and 1.70 g of DCM were added. The reaction mixture was stirred until the macroinitiator and TMC were completely dissolved in DCM, upon which DBU (about 12 mg, 0.078 mmol) was added. The reaction mixture was stirred at room temperature for 5.5 hours in a glove box. After 5.5 hours, the reaction was stopped by bringing the reaction vial out of the glove box and adding DCM (2ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The reaction was further stirred for two hours at room temperature. The resulting polymer was isolated by precipitating the reaction mixture in methanol. The product was collected in a frit funnel by removing methanol under vacuum and the resulting solids were redissolved in THF to form a 20 wt% solution and reprecipitated in methanol. The solids were collected in a frit funnel and dried under vacuum at 40°C for two hours to obtain the resulting compound. This compound was dissolved in THF to form a 20 wt% solution and the polymer was precipitated in methanol: acetonitrile (200 ml, 60:40 v:v). The precipitated solids and the solvents were collected in a centrifuge tube and the solids was collected by centrifuging at 4000 RPM at 0°C followed by decanting the solvent and drying

the solids in a vacuum oven at 40°C for two hours. M_n (GPC) = 14.2k, M_w = 15.7k, PDI = 1.10. M_n (NMR) = PS(4.8k)-b-PTMC(5.8k), V_{fPTMC} = 0.49.

Example 17. Synthesis of PS-b-PTMC with macroinitiator of Example 6 (JB8-F)

The same procedure was employed as for Example 16 except that the following reagents were employed: 0.15 g (0.0312 mmol) hydroxyl-functional polystyrene with $\text{CH}_2\text{C}_3\text{F}_7$ pendant group from **Example 6**, 0.208 g (2.08 mmol) of TMC, 0.67 g of DCM, DBU (12 mg, 0.078 mmol), DCM (1ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The polymer had M_n (GPC) = 9.5k, M_w = 10.5k, PDI = 1.10. M_n (NMR) = PS(4.8k)-b-PTMC(2.6k), V_{fPTMC} = 0.30.

Example 18. Synthesis of PS-b-PTMC with macroinitiator of Example 7 (JB11F)

The same procedure was employed as for Example 16 except that the following reagents were employed 0.3 g (0.0577 mmol) hydroxyl-functional polystyrene with $\text{C}_2\text{H}_4\text{C}_6\text{F}_{13}$ pendant group from **Example 7**, 0.78 g (7.64 mmol) of TMC, 2.55 g of DCM, DBU (12 mg, 0.078 mmol), DCM (2ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The polymer had M_n (GPC) = 14.6k, M_w = 15.5k, PDI = 1.16. M_n (NMR) = PS(5.2k)-b-PTMC(6.6k), V_{fPTMC} = 0.50.

Example 19. Synthesis of PS-b-PTMC with macroinitiator of Example 7 (JB12F)

The same procedure was employed as for Example 16 except that the following reagents were employed: 0.2 g (0.0384 mmol) hydroxyl-functional polystyrene with $\text{C}_2\text{H}_4\text{C}_6\text{F}_{13}$ pendant group from **Example 7**, 0.217 g (2.12 mmol) of TMC, 0.7 g of DCM, DBU (12 mg, 0.078 mmol), DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The polymer had M_n (GPC) = 9.2k, M_w = 9.8k, PDI = 1.06. M_n (NMR) = PS(5.2k)-b-PTMC(2.5k), V_{fPTMC} = 0.28.

Example 20. Synthesis of PS-b-PTMC with macroinitiator of Example 8 (JB1F)

The same procedure was employed as for Example 16 except that the following reagents were employed: 0.2 g (0.0425 mmol) hydroxyl-functional polystyrene with $\text{CH}_2\text{C}_8\text{F}_{17}$ pendant group from **Example 8**, 0.260 g (2.55 mmol) of TMC, 0.85 g of DCM, DBU (12 mg, 0.078 mmol) DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The reaction was further stirred for two hours at room temperature. The resulting polymer had M_n (GPC) = 10.2k, M_w = 11.1k, PDI = 1.08. M_n (NMR) = PS(4.7k)-b-PTMC(2.3k), $V_{f_{\text{PTMC}}}$ = 0.28.

Example 21. Synthesis of PS-b-PTMC with macroinitiator of Example 8 (JB2F)

The same procedure was employed as for Example 16 except that the following reagents were employed: 0.2 g (0.0425 mmol) hydroxyl-functional polystyrene with $\text{CH}_2\text{C}_8\text{F}_{17}$ pendant group from **Example 8**, 0.43 g (4.25 mmol) of TMC, 1.4 g of DCM, DBU (12 mg, 0.078 mmol), DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The polymer had M_n (GPC) = 14.7k, M_w = 15.9k, PDI = 1.07, M_n (NMR) = PS(4.7k)-b-PTMC(5.3k), $V_{f_{\text{PTMC}}}$ = 0.47. .

Example 22. Synthesis of PS-b-PTMC with macroinitiator of Example 8 (JB2IIF)

Similar reaction conditions as described in Example 21 were used to make PS-b-PTMC with $\text{CH}_2\text{C}_8\text{F}_{17}$ pendant group. M_n (GPC) = 15.3k, M_w = 16.7k, PDI = 1.08. M_n (NMR) = PS(4.7k)-b-PTMC(5.7k), $V_{f_{\text{PTMC}}}$ = 0.49.

Example 23. Synthesis of PS-b-PTMC with macroinitiator of Example 9 (JB9F)

The same procedure was employed as for Example 16 except that the following reagents were employed: 0.2 g (0.021 mmol) hydroxyl-functional polystyrene with $\text{CH}_2\text{C}_8\text{F}_{17}$ pendant group from **Example 9**, 0.515 g (5.05 mmol) of TMC, 1.65 g of DCM, upon which DBU (about 12 mg, 0.078 mmol) was added, DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The polymer had M_n (GPC) = 24.3k, M_w = 27.6k, PDI = 1.13, M_n (NMR) = PS(9.5k)-b-PTMC(11.8k), $V_{f\text{PTMC}}$ = 0.49.

Example 24. Synthesis of PS-b-PTMC with macroinitiator of Example 9 (JB10F)

The same procedure was employed as for Example 16 except that the following reagents were employed: 0.15 g (0.0158 mmol) hydroxyl-functional polystyrene with $\text{CH}_2\text{C}_8\text{F}_{17}$ pendant group from **Example 9**, 0.167 g (1.64 mmol) of TMC, 0.73 g of DCM, DBU (about 12 mg, 0.078 mmol), DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The polymer had M_n (GPC) = 18.0k, M_w = 20.0k, PDI = 1.11. M_n (NMR) = PS(9.5k)-b-PTMC(4.8k), $V_{f\text{PTMC}}$ = 0.28.

Example 25. Synthesis of PS-b-PTMC with macroinitiator of Example 10 (JB13F)

The same procedure was employed as for Example 16 except that the following reagents were employed: 0.20 g (0.0290 mmol) hydroxyl-functional polystyrene with $\text{CH}_2\text{CF}_2(\text{OCF}_2\text{CF}_2)_3\text{C}_2\text{F}_5$ pendant group from **Example 10**, 0.212 g (2.08 mmol) of TMC, 1.1 g of DCM, DBU (about 12 mg, 0.078 mmol), DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The polymer had M_n (GPC) = 12.4k, M_w = 13.7k, PDI = 1.11. M_n (NMR) = PS(6.9k)-b-PTMC(3.2k), $V_{f\text{PTMC}}$ = 0.27.

Example 26. Synthesis of PS-b-PTMC with macroinitiator of Example 10 (JB14F)

The same procedure was employed as for Example 16 except that the following reagents were employed: 0.20 g (0.0290 mmol) hydroxyl-functional polystyrene with $\text{CH}_2\text{CF}_2(\text{OCF}_2\text{CF}_2)_3\text{C}_2\text{F}_5$ pendant group from **Example 10**, 0.502 g (4.93 mmol) of TMC, 1.6 g of DCM, DBU (about 12 mg, 0.078 mmol), DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The polymer had M_n (GPC) = 19.3k, M_w = 21.0k, PDI = 1.09. M_n (NMR) = PS(6.9k)-b-PTMC(8.3k), $V_{f\text{PTMC}}$ = 0.49.

Example 27. Synthesis of PS-b-PTMC with macroinitiator of Example 11 (JB6F)

The same procedure was employed as for Example 16 except that the following reagents were employed: 0.15 g (0.0357 mmol) hydroxyl-functional polystyrene with $\text{C}_{18}\text{H}_{37}$ pendant group from **Example 11**, 0.382 g (3.75 mmol) of TMC, 1.24 g of DCM, DBU (12 mg, 0.078 mmol), DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The resultant polymer had M_n (GPC) = 10.8k, M_w = 11.6k, PDI = 1.06. M_n (NMR) = PS(4.2k)-b-PTMC(5.0k), $V_{f\text{PTMC}}$ = 0.48.

Example 28. Synthesis of PS-b-PTMC with macroinitiator of Example 11 (JB7F)

The same procedure was employed as for Example 16 except that the following reagents were employed: 0.15 g (0.0357 mmol) hydroxyl-functional polystyrene with $\text{C}_{18}\text{H}_{37}$ pendant group from **Example 11**, 0.172 g (1.68 mmol) of TMC, 0.56 g of DCM, DBU (12 mg, 0.078 mmol), DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The resultant polymer had M_n (GPC) = 8.1k, M_w = 8.6k, PDI = 1.06. M_n (NMR) = PS(4.2k)-b-PTMC(2.9k), $V_{f\text{PTMC}}$ = 0.35.

Table 4 summarizes the polystyrene-b-polycarbonate block copolymers with different pendant group at the junction

Table 4

Example	Pendant Group	Catalyst	Macroinitiator	Temp	Time (hours)	GPC			NMR (M_n , kDa)		Vf (PC block)
						M_n (kDa)	M_w (kDa)	PDI	PS block	PC block	
16	$R_1 = \text{CH}_2\text{C}_3\text{F}_7$	DBU	Example 6	rt	5.5	14.2K	15.7K	1.1	4.8K	5.8K	0.49
17	$R_1 = \text{CH}_2\text{C}_3\text{F}_7$	DBU	Example 6	rt	2	9.5K	10.6K	1.1	4.8K	2.6K	0.3
18	$R_2 = \text{C}_2\text{H}_4\text{C}_6\text{F}_{13}$	DBU	Example 7	rt	8	14.6K	15.5K	1.06	5.2K	6.6K	0.5
19	$R_2 = \text{C}_2\text{H}_4\text{C}_6\text{F}_{13}$	DBU	Example 7	rt	1.5	9.2K	9.8K	1.06	5.2K	2.5K	0.28
20	$R_3 = \text{CH}_2\text{C}_6\text{F}_{17}$	DBU	Example 8	rt	1.5	10.2K	11.1K	1.09	4.7K	2.3K	0.28
21	$R_3 = \text{CH}_2\text{C}_6\text{F}_{17}$	DBU	Example 8	rt	2	14.7K	15.9K	1.07	4.7K	5.3K	0.47
22	$R_3 = \text{CH}_2\text{C}_6\text{F}_{17}$	DBU	Example 8	rt	2.3	15.2K	16.6K	1.09	4.7K	5.7K	0.49
23	$R_3 = \text{CH}_2\text{C}_6\text{F}_{17}$	DBU	Example 9	rt	8	24.3K	27.6K	1.13	9.5K	11.8K	0.49
24	$R_3 = \text{CH}_2\text{C}_6\text{F}_{17}$	DBU	Example 9	rt	3.5	18.0K	20.0K	1.11	9.5K	4.8K	0.28
25	$R_4 = \text{CH}_2\text{CF}_2(\text{OCF}_2\text{CF}_2)_3\text{C}_2\text{F}_5$	DBU	Example 10	rt	3	12.4K	13.7K	1.11	6.9K	3.2K	0.27
26	$R_4 = \text{CH}_2\text{CF}_2(\text{OCF}_2\text{CF}_2)_3\text{C}_2\text{F}_5$	DBU	Example 10	rt	7	19.3K	21.0K	1.09	6.9K	8.3K	0.49
27	$R_6 = \text{C}_{18}\text{F}_{37}$	DBU	Example 11	rt	3.5	10.8K	11.6K	1.06	4.2K	5.0K	0.48
28	$R_6 = \text{C}_{18}\text{F}_{37}$	DBU	Example 11	rt	1.5	8.1K	8.7K	1.06	4.2K	2.9K	0.35

Example 29. Synthesis of PTMSS-b-PTMC with macroinitiator of Example 12 (PERL)

The same procedure was employed as for Example 16 except that the following reagents were employed: 0.05 g (0.0067 mmol) hydroxyl-functional PTMSS with $C_2H_4C_6F_{13}$ pendant group from **Example 12**, 0.134 g (1.31 mmol) of TMC, 0.87 g of DCM, 14 mg (0.039 mmol) of 1-[3,5-Bis(trifluoromethyl)phenyl]-3-cyclohexylthiourea (TU) and DBU (6 mg, 0.039 mmol). The resultant polymer had M_n (GPC) = 17.0k, M_w = 18.9k, PDI = 1.11. M_n (NMR) = PS(6.9k)-b-PTMC(8.3k), V_{fPTMC} = 0.49. .

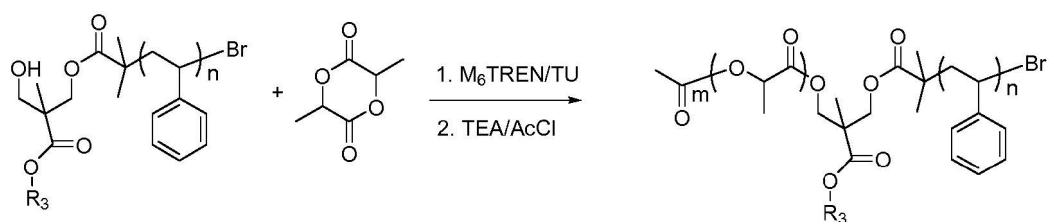
Example 30. Synthesis of PTMSS-b-PTMC with macroinitiator of Example 13 (PERL)

The same procedure was employed as for Example 16 except that the following reagents were employed: 0.05 g (0.0067 mmol) hydroxyl-functional PTMSS with $C_2H_4C_6F_{13}$ pendant group from **Example 13**, 0.39 g (3.85 mmol) of TMC, 1.75 g of DCM, DBU (18 mg, 0.12 mmol), DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The resultant polymer had M_n (GPC) = 18.6k, M_w = 20.2k, PDI = 1.08. M_n (NMR) = PS(6.8k)-b-PTMC(8.4K), V_{fPTMC} = 0.48.

Table 5

Example	Pendant Group	Catalyst	Macroinitiator	Temp	Time (hours)	GPC			NMR (M _n , kDa)		Monomer %Conv.	Vf (PC block)
						M _n (kDa)	M _w (kDa)	PDI	PS block	PC Block		
29	C ₂ H ₄ C ₆ F ₁₃	DBU/TU	Example 12	rt	3	17.0K	18.9K	1.11	6.9k	8.3k	~50	0.50
30	CH ₂ C ₈ F ₁₇	DBU	Example 13	rt	9	18.6k	20.2	1.08	6.8k	8.4k	~50	0.486

Example 31. Synthesis of PS-b-PLA with macroinitiator of Example 9 (SFE-1-II)



Scheme 16

Synthesis of PS-b-PLA with $\text{CH}_2\text{C}_8\text{F}_{17}$ pendant group. To an oven dried 4 ml glass vial equipped with a magnetic stir bar, 0.1 g (0.0105 mmol) hydroxyl-functional polystyrene with $\text{CH}_2\text{C}_8\text{F}_{17}$ pendant group from **Example 9**, 0.23 g (1.59 mmol) of D,L-lactide (PLA) and 2.1 g of DCM were added. The reaction mixture was stirred until the macroinitiator and D,L-lactide were completely dissolved in DCM, upon which 20 mg (0.0525 mmol) of TU and tris[2-(dimethylamino)ethyl]amine (M6TREN) (12 mg, 0.0526 mmol) was added. The reaction mixture was stirred at room temperature for 3 hours in a glove box. After 3 hours, the reaction was stopped by bringing the reaction vial out of the glove box and adding DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The reaction was further stirred for two hours at room temperature. The resulting polymer was isolated by precipitating the reaction mixture in methanol. The product was collected in a frit funnel by removing methanol under vacuum and the resulting solids were redissolved in THF to form a 20 wt% solution and reprecipitated in methanol. The solids were collected in a frit funnel and dried under vacuum at 40°C for two hours to obtain the resulting compound. M_n (GPC) = 24.0K, M_w = 26.0k, PDI = 1.09. M_n (NMR) = PS(9.5k)-b-PLA(9.6k), $V_{f\text{PLA}}$ = 0.47.

Example 31a. The compound from Example 31 was dissolved in THF to form a 20 wt% solution and the polymer was precipitated in methanol: acetonitrile (20ml, 75:25 v:v). The precipitated solids and the solvents were collected in a centrifuge tube and the solids was collected by centrifuging at 4000 RPM at 0°C followed by decanting the solvent and drying the solids in a vacuum oven at 40°C for two hours. M_n (GPC) = 24.6k, M_w = 26.8k, PDI = 1.09. M_n (NMR) = PS(9.5k)-b-PLA(8.0K), $V_{f_{PLA}}$ = 0.42

Example 32. Synthesis of PS-b-PLA with macroinitiator of Example 9 (SFE-2)

The same procedure was employed as for Example 31 except that the following reagents were employed: 0.1 g (0.0105 mmol) hydroxyl-functional polystyrene with $\text{CH}_2\text{C}_8\text{F}_{17}$ pendant group from **Example 9**, 0.1 g (0.69 mmol) of D,L-lactide, 0.92 g of DCM, 20 mg (0.00525 mmol) of TU, M6TREN (12 mg, 0.0526 mmol, DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The resultant polymer had M_n (GPC) = 18.5K, M_w = 20.0k, PDI = 1.08. M_n (NMR) = PS(9.5k)-b-PLA(5.4k), $V_{f_{PLA}}$ = 0.32.

Example 33. Synthesis of PS-b-PLA with macroinitiator of Example 15 (SFE-3)

The same procedure was employed as for Example 31 except that the following reagents were employed: 0.1 g (0.0064 mmol) hydroxyl-functional polystyrene with $\text{CH}_2\text{C}_8\text{F}_{17}$ pendant group from **Example 15**, 0.115 g (0.80 mmol) of D,L-lactide, 1.0 g of DCM 12 mg (0.032 mmol) of TU, M6TREN (7 mg, 0.026 mmol), DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). M_n (GPC) = 35.6K, M_w = 38.7k, PDI = 1.08. M_n (NMR) = PS(15.6k)-b-PLA(16.6k), $V_{f_{PLA}}$ = 0.485.

Example 33a. The compound from Example 33 was dissolved in THF to form a 20 wt% solution and the polymer was precipitated in methanol: acetonitrile (20ml, 70:30 v:v). The precipitated solids and the solvents were collected in a centrifuge tube and the solids was collected by centrifuging at 4000 RPM at 0°C followed by decanting the solvent and drying the solids in a vacuum oven at 40°C for two hours. M_n (GPC) = 37.3k, M_w = 40.2k, PDI = 1.07. M_n (NMR) = PS(15.6k)-b-PLA(13.5K), $V_{f_{PLA}}$ = 0.435

Example 34. Synthesis of PS-b-PLA with macroinitiator of Example 15 (SFE-4)

The same procedure was employed as for Example 31 except that the following reagents were employed: 0.1 g (0.0064 mmol) hydroxyl-functional polystyrene with $\text{CH}_2\text{C}_8\text{F}_{17}$ pendant group from **Example 15**, 0.46 g (0.80 mmol) of D,L-lactide, 0.6 g of DCM, 12 mg (0.032 mmol) of TU and M6TREN (7 mg, 0.026 mmol) was added, DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The resultant polymer had M_n (GPC) = 25.5K, M_w = 27.7k, PDI = 1.08. M_n (NMR) = PS(15.6k)-b-PLA(6.3k), $V_{f_{PLA}}$ = 0.27.

Table 6 summarizes different polystyrene-b-poly lactide block copolymers with various pendant groups at the junction

Table 6

Example	Polymer Name	Pendant Group	Catalyst	Macroinitiator	Temp	Time (hours)	GPC			NMR (M_n , kDa)		V_f (PLA block)
							M_n (kDa)	M_w (kDa)	PDI	PS block	PLA block	
31	SFE-1-II	$R_3 = \text{CH}_2\text{C}_8\text{F}_{17}$	M6TREN / TU	Example 9	rt	1.5	24.0K	26.0K	1.09	9.5K	9.9K	0.47
31a	SFE-1-IIF	$R_3 = \text{CH}_2\text{C}_8\text{F}_{17}$	M6TREN / TU	Example 9	rt	1.5	24.6k	26.8k	1.09	9.5K	8.0K	0.42
32	SFE-2	$R_3 = \text{CH}_2\text{C}_8\text{F}_{17}$	M6TREN / TU	Example 9	rt	0.25	18.5K	20.0K	1.08	9.5K	5.4K	0.32
33	SFE-3	$R_3 = \text{CH}_2\text{C}_8\text{F}_{17}$	M6TREN / TU	Example 15	rt	2.25	35.6K	38.7K	1.08	15.6K	16.6K	0.48
33a	SFE-3F	$R_3 = \text{CH}_2\text{C}_8\text{F}_{17}$	M6TREN / TU	Example 15	rt	0.75	37.3k	40.2k	1.07	15.6K	13.5	0.435
34	SFE-4	$R_3 = \text{CH}_2\text{C}_8\text{F}_{17}$	M6TREN / TU	Example 15	rt	0.75	25.5K	27.7K	1.09	15.6K	6.3K	0.27

Example 35. Synthesis of PTMSS-b-PLA with macroinitiator of Example 12 (PERL)

The same procedure was employed as for Example 31 except that the following reagents were employed: 0.05 g (0.0067 mmol) hydroxyl-functional PTMSS with $\text{C}_2\text{H}_4\text{C}_6\text{F}_{13}$ pendant group from **Example 12**, 0.060 g (0.412 mmol) of Lactide, 1.10 g of DCM, 29 mg (0.078 mmol) of TU, M6TREN (about 18 mg, 0.078 mmol), DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The resultant polymer had M_n (GPC) = 15.5k, M_w = 16.9k, PDI = 1.09. M_n (NMR) = PS(6.3k)-b-PLA(4.0k), $V_{f_{\text{PLA}}} = 0.35$.

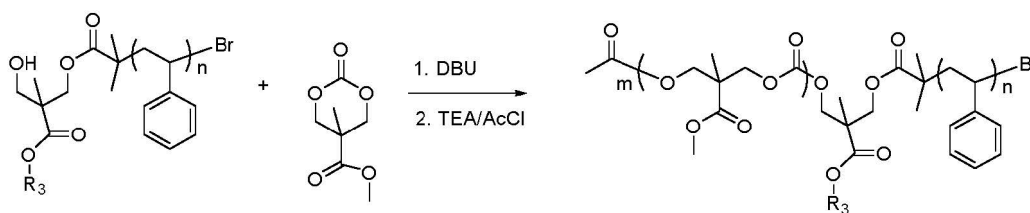
Example 36. Synthesis of PTMSS-b-PLA with macroinitiator of Example 12 (PERL)

The same procedure was employed as for Example 31 except that the following reagents were employed: 0.05 g (0.0067 mmol) hydroxyl-functional PTMSS with $\text{C}_2\text{H}_4\text{C}_6\text{F}_{13}$ pendant group from **Example 12**, 0.060 g (0.412 mmol) of lactide, 1.10 g of DCM, 29 mg (0.078 mmol) of TU, M6TREN (18 mg, 0.078 mmol), DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The resultant polymer had M_n (GPC) = 17.3k, M_w = 19.1k, PDI = 1.10. M_n (NMR) = PS(6.3k)-b-PLA(6.2k), $V_{f_{\text{PLA}}} = 0.45$.

Table 7

Example	Pendant Group	Catalyst	Macroinitiator	Temp	Time (minutes)	GPC			NMR (M_n , kDa)		Vf (PLA block)
						M_n (kDa)	M_w (kDa)	PDI	PS block	PLA block	
35	$C_2H_4C_6F_{13}$	M6TREN / TU	Example 12	rt	20	15.5k	16.9k	1.08	6.3k	4.0k	0.35
36	$C_2H_4C_6F_{13}$	M6TREN / TU	Example 12	rt	45	17.3k	19.1k	1.10	6.3k	6.2k	0.45

Example 37. Synthesis of PS-b-PMeCAR



Scheme 17

Synthesis of PS-b-PMeCAR with CH₂C₈F₁₇ pendant group. To an oven dried 4 ml glass vial equipped with a magnetic stir bar, 0.1 g (0.0212 mmol) hydroxyl-functional polystyrene with CH₂C₈F₁₇ pendant group from **Example 8**, 0.24 g (1.37 mmol) of methyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate (MeCAR) and 1.80 g of DCM were added. The reaction mixture was stirred until the macroinitiator and MeCAR were completely dissolved in DCM, upon which DBU (3.27 mg, 0.0212 mmol, of a 10 wt% stock solution in toluene) was added. The reaction mixture was stirred at room temperature for forty minutes in a glove box. After forty minutes, the reaction was stopped by bringing the reaction vial out of the glove box and adding DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The reaction was further stirred for two hours at room temperature. The resulting polymer was isolated by precipitating the reaction mixture in methanol. The product was collected in a frit funnel by removing methanol under vacuum and the resulting solids were redissolved in THF to form a 20 wt% solution and reprecipitated in methanol. The solids were collected in a frit funnel and dried under vacuum at 40°C for two hours to obtain the resulting compound. This compound was dissolved in THF to form a 20 wt% solution and the polymer was precipitated in methanol: acetonitrile (20ml, 85:15 v:v). The precipitated solids and the solvents were collected in a centrifuge tube and the solids were collected by centrifuging at 4000 RPM at 0°C followed by decanting the solvent and drying the solids in a vacuum oven at 40°C for two hours. M_n

(GPC) = 13.0k, M_w = 14.0k, PDI = 1.07. M_n (NMR) = PS(4.70k)-b-PMeCAR(6.2K), $V_{f_{PMeCAR}}$ = 0.51.

Examples 38-40. Synthesis and purification of PS-b-PMeCAR with $CH_2C_8F_{17}$ pendant group with different molecular weights. These polymers were prepared using the general procedure of Example 37, summarized in Table 8 along with example 37.

Table 8

Ex	Macroinitiator Example	Macroinitiator (g, mmol)	MeCAR (g, mmol)	DCM (g)	DBU (mg)	Time (hours)	Mn (NMR)		$V_{f_{PMeCAR}}$	GPC		
							PS	PMeCAR		Mn	Mw	PDI
37	Example 8	0.1, 0.0212	0.24, 1.378	1.80	3.27	0.67	4.7k	6.2k	0.51	13.0k	14.0k	1.07
38	Example 9	0.10, 0.0105	0.25, 1.430	1.86	1.6	2	9.5k	8.8k	0.43	20.0k	21.9k	1.09
39	Example 9	0.10, 0.0105	0.44, 2.52	3.36	1.6	2.5	9.5k	11.7k	0.495	21.5k	23.5k	1.09
40	Example 15	0.1, 0.0064	0.23, 1.320	1.75	1.0	4.2	15.6k	19k	0.49	30.1k	32.7k	1.08

Example 41. Synthesis of PS-b-PEtCAR

The same procedure was employed as for Example 37 except that the following reagents were employed: 0.1 g (0.0105 mmol) hydroxyl-functional polystyrene with $CH_2C_8F_{17}$ pendant group from **Example 8**, 0.22 g (1.17 mmol) of ethyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate (EtCAR), 1.50 g of DCM, DBU (1.6 mg, 0.0105 mmol, 10 wt% stock solution in toluene), DCM (0.5 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The resultant polymer had M_n (GPC) = 23.3k, M_w = 26.7k, PDI = 1.14. M_n (NMR) = PS(9.5k)-b-PEtCAR(11.1), $V_{f_{PEtCAR}}$ ~ 0.48.

Example 42. Synthesis of PS-b-PPCAR

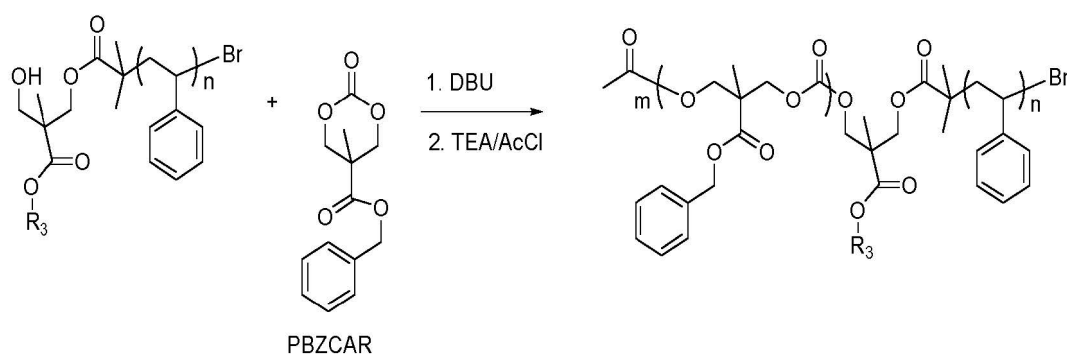
The same procedure was employed as for Example 37 except that the following reagents were employed: 0.1 g (0.0105 mmol) hydroxyl-functional polystyrene with $\text{CH}_2\text{C}_8\text{F}_{17}$ pendant group from **Example 9**, 0.34 g (1.68 mmol) of Propyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate (PCAR), 2.23 g of DCM, DBU (1.6 mg, 0.0105 mmol, 10 wt% stock solution in toluene), DCM (0.5 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The resultant polymer had M_n (GPC) = 23.3k, M_v = 25.7k, PDI = 1.10. M_n (NMR) = PS(9.5k)-b-PPCAR(11.5), $V_{f\text{PCAR}} \sim 0.49$.

Example 43 Synthesis of PS-b-PPCAR with different molecular weight.

This polymer was prepared using the general procedure of Example 42, summarized in Table 9.

Table 9

Ex.	Macroinitiator Example	Macroinitiator (g, mmol)	PPCAR (g, mmol)	DCM (g)	DBU (mg)	Time (hours)	M _n (NMR)		V _f PMeCAR	GPC		
							PS	PMeCAR		M _n	M _w	PDI
42	Example 9	0.10, 0.0105	0.34, 1.68	2.23	1.6	3.5	9.5k	11.5k	0.49	23.3	25.7	1.10
43	Example 15	0.1, 0.0064	0.33, 1.64	2.17	1.0	8.5	15.6k	17.5	0.47	30.2	33.7	1.11

Example 44. Synthesis of PS-b-PBzCAR**Scheme 18**

The same procedure was employed as for Example 37 except that the following reagents were employed: 0.15 g (0.0096 mmol) hydroxyl-functional polystyrene with CH₂C₈F₁₇ pendant group from **Example 15**, 0.20 g (0.80 mmol) of benzyl 5-methyl-2-oxo-1,3-dioxane-5-carboxylate (PBZCAR), 1.06 g of DCM, DBU (1.6 mg, 0.0105 mmol, 10 wt% stock solution in toluene), DCM (0.5 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The resultant polymer had M_n (GPC) = 26.0k, M_w = 28.3k, PDI = 1.09. M_n (NMR) = PS(15.6k)-b-PPCAR(14.9), V_f_{PBZCAR} ~ 0.44.

Example 45. Synthesis of hydroxyl-functional polystyrene with initiator from Example 3

The same procedure was employed as for Example 6 except that the following reagents were employed: 27.50 g, (0.266 mol) of styrene, 1.20 g (1.56 mmol) ATRP initiator from **Example 3**, 0.225 g (1.56 mmol) CuBr, and 27.5 g of Anisole, 0.271 g (1.56 mmol) of PMDETA. The final reaction mixture was diluted in adding THF (100 ml) and was passed through a short silica-gel column to remove copper catalyst. The resulting solution was concentrated under vacuum and was precipitated once in methanol from THF. The polymer was collected and redissolved in THF and precipitated once in methoxyperfluorobutane(HFE-7100). The resulting polymer was dried under vacuum and redissolved in THF and reprecipitated in methanol. The polymer was collected in a frit funnel and was dried under vacuum. $M_n = 8.5k$, $M_w = 9.2k$, PDI = 1.08.

Example 46. Synthesis of PS-b-PLA with macroinitiator of Example 45

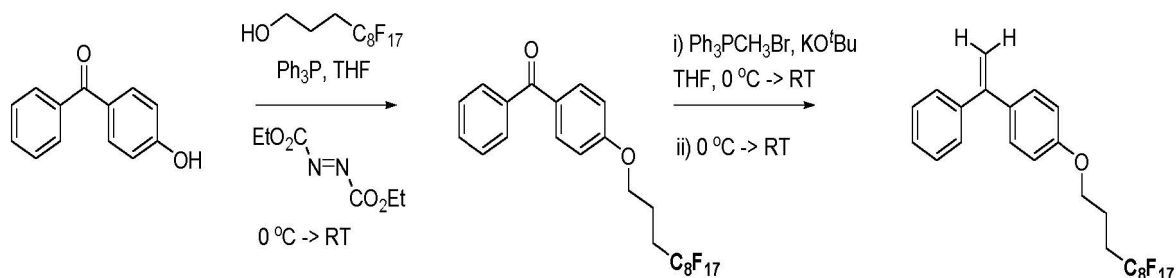
The same procedure was employed as for Example 31 and Example 31a except that the following reagents were employed: 0.15 g (0.0176 mmol) hydroxyl-functional polystyrene with $CH_2C_8F_{17}$ pendant group from **Example 45**, 0.23 g (1.58 mmol) of D,L-lactide (LA), 2.1 g of DCM, 32 mg (0.088 mmol) of TU, M6TREN (20 mg, 0.088 mmol), DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The resultant polymer had M_n (GPC) = 19.9K, $M_w = 21.4k$, PDI = 1.07. M_n (NMR) = PS(8.5k)-b-PLA(9.2k), $V_{f_{PLA}} = 0.48$.

Example 47. Synthesis of PS-b-PMeCAR

The same procedure was employed as for **Example 31** except that the following reagents were employed: 0.25 g (0.029 mmol) hydroxyl-functional polystyrene with $CH_2C_8F_{17}$ pendant group from **Example 45**, 0.80 g (4.60 mmol) of MeCAR, 6.50 g of DCM, DBU (4.5 mg, 0.0294 mmol, 10 wt% stock solution in toluene), DCM (1 ml), acetyl chloride (0.1 g, 0.988 mmol), and TEA (0.1 g, 1.28 mmol). The polymer had M_n (GPC) = 17.2k, $M_w = 18.5k$, PDI = 1.07. M_n (NMR) = PS(8.5k)-b-PMeCAR(10.2K), $V_{f_{PMeCAR}} = 0.49$.

I-2. Synthesis of JBCPs by anionic polymerization

Example 48. Synthesis of monofunctional perfluoroalkane-substituted DPE ($C_3H_6C_8F_{17}$ -DPE)

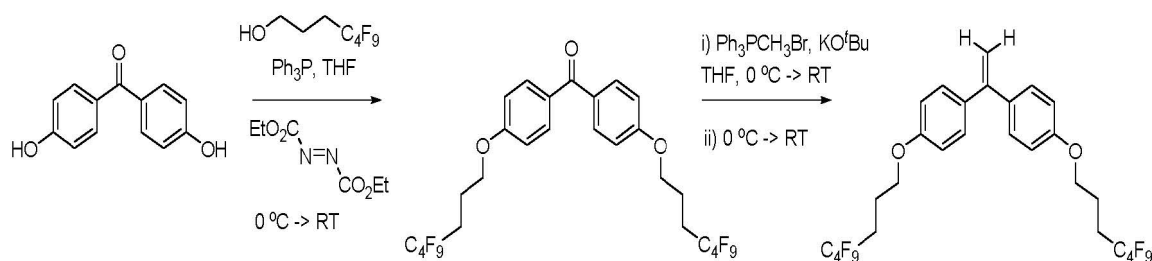


Scheme 19: Synthesis of perfluoro alkane substituted 1,1'-diphenylethylene ($C_3H_6C_8F_{17}$ -DPE).

To a mixture of 4-hydroxybenzophenone (6 g, 30.2 mmol), (perfluorooctyl)propanol (17.4 g, 36.4 mmol) and triphenylphosphine (10.2 g, 36.4 mmol) in 100 mL of THF, 17.7 mL of diethyl azodicarboxylate (36.4 mmol, 40 wt% in toluene) was drop wise added at ice bath temperature. After stirring for 2 hours at RT, most of solvent was evaporated and then about 100 mL of methanol was poured into reaction mixture and its slurry mixture was stirred for one hour to dissolve byproducts and reagents. White solid product was filtered and washed with cold methyl alcohol several times. Resulting 18.1g of product was dried under vacuum oven at 50°C (91%). 1H NMR (300 MHz, $CDCl_3$):(ppm) 7.83 (2H, d, $J=8.7$ Hz), 7.76 (2H, d, $J=6.9$ Hz), 7.60-7.45 (m, 3H), 6.96 (2H, d, $J=8.7$ Hz), 4.13 (2H, t, $J=5.7$ Hz), 2.41-2.13 (4H, m). Potassium *t*-butoxide (4.01g, 35.75 mmol) was added into methyltriphenylphosphonium bromide (11.8 g, 33 mmol) in 30 mL of THF at 0°C under nitrogen and the solution was stirred at RT for 1 hr. This reacting mixture was slowly transferred into $C_3H_6C_8F_{17}$ -BP (18.1 g, 27.5 mmol) in 30 mL of THF under nitrogen by cannula at 0°C and was left at RT for 1hr. 1~2 mL of water was added to quench the reaction and then methanol (~100 mL) was poured to precipitate product and to dissolve other byproducts. The mixture was stored in refrigerator for overnight and then product was filtered

and washed with cold methyl alcohol several times. The product was dissolved into chloroform and filtered through silica column with eluent of $\text{CHCl}_3/\text{Hexane}$ = 1/10. After solvent evaporation and vacuum dry, 16.4 g of product was obtained (91%). ^1H NMR (300 MHz, CDCl_3): (ppm) 7.35-7.25 (7H, m), 6.85 (2H, d, J =8.7 Hz), 6.38 (2H, d, J =10.2 Hz), 4.06 (2H, t, J =5.7 Hz), 2.41-2.06 (4H, m); ^{19}F NMR (282 MHz, CDCl_3):(ppm) -80.82 (3F), -114.34 (2F), -121.70—121.91 (6F), -122.70 (2F), -123.41 (2F), -126.10 (2F).

Example 49. Synthesis of difunctional perfluoroalkane-substituted DPE ($\text{diC}_3\text{H}_6\text{C}_4\text{F}_9\text{-DPE}$)

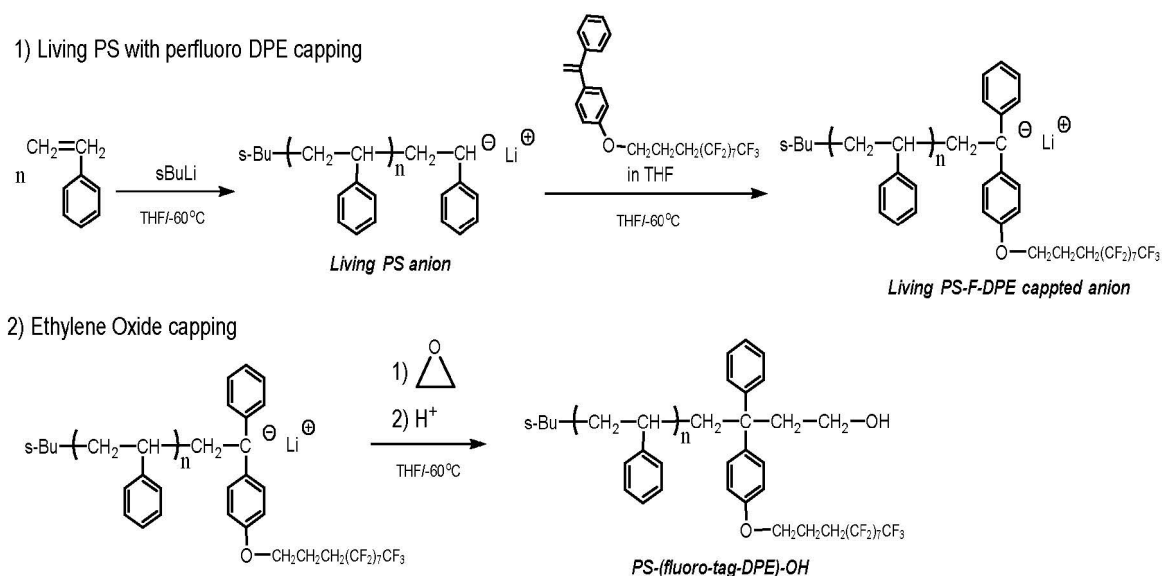


Scheme 20: Synthesis of perfluoro alkane disubstituted 1,1'-diphenylethylene ($\text{diC}_3\text{H}_6\text{C}_4\text{F}_9\text{-DPE}$)

4,4'-Dihydroxybenzophenone and (perfluorobutyl)propanol were used for di-substituted BP and its synthesis was accomplished with the similar method of synthesis as $\text{C}_3\text{H}_6\text{C}_8\text{F}_{17}\text{-BP}$ using same reagents and reaction condition. White solid powder was obtained with 94% yield. ^1H NMR (300 MHz, CDCl_3): (ppm) 7.82 (4H, d, J = 8.7 Hz), 6.96 (4H, d, J = 8.7 Hz), 4.13 (4H, t, J = 5.7 Hz), 2.41-2.11 (8H, m). The reaction and purification were followed by the same method as **Example 48** by replacing appropriate perfluoroalkane. A white solid product was obtained with 90% yield, which was further purified using column chromatography. ^1H NMR (300 MHz, CDCl_3):(ppm) 7.26 (4H, d, J = 9.0 Hz), 6.85 (4H, d, J = 9.0 Hz), 4.06 (4H, t, J = 6 5.7 Hz Hz), 2.41-2.07 (8H, m); ^{19}F NMR (282 MHz, CDCl_3):(ppm) -81.05 (3F), -114.57 (2F), -123.40 (2F), -126.02 (2F).

Example 50. Synthesis of hydroxyl polystyrene macroinitiator containing perfluoroalkane-substituted DPE via anionic polymerization (DB-PS(F)-OH-20)

Required amount of monomers were distilled in the presence of dehydrating agents into calibrated ampules and stored under N₂. Liquids were transferred into the reactor either via ampule or using stainless steel cannula under N₂. Into a dry 500 mL round bottom reactor equipped with side arms for connecting ampules, magnetic stir bar, nitrogen/vacuum three-way septum adapter, was added 200 mL dry tetrahydrofuran. The temperature of the reactor was reduced to -60°C using dry ice-acetone bath. Then, 1 mL (0.0014 moles) of *sec*-butyllithium (*sec*-BuLi) (1.4 M) was added into the reactor. Subsequently, a known amount of styrene (14.75 mL) was added from ampule into the reactor under fast stirring. The reaction solution turned into orange and the reaction was stirred over 10 min. Subsequently, 1.2 g of perfluoroalkane-substituted 1,1'-diphenylethylene (C₃H₆C₈F₁₇-DPE) of **Example 48** in 8 ml of dry toluene was added via ampule into the reactor. The orange color of the reaction mixture turned into dark brick-red indicating conversion of styryllithium active centers to delocalized carbanion resulting from C₃H₆C₈F₁₇-DPE adduct formation. After 2 min of stirring, 2 mL of ethylene oxide (EO), which distilled over *sec*-BuLi, was added into the reactor via ampule and the dark brick-red color disappeared immediately upon addition of EO and became a faint-pink in color. The reaction was terminated after 15 min with 1 mL of degassed methanol. The polymer was recovered by precipitation in excess methanol (4 times of the polymer solution) containing 10 % water, filtered, and dried at 55 °C for 12 h under vacuum giving 14.1 g of PS(F)-OH (98 % yield).



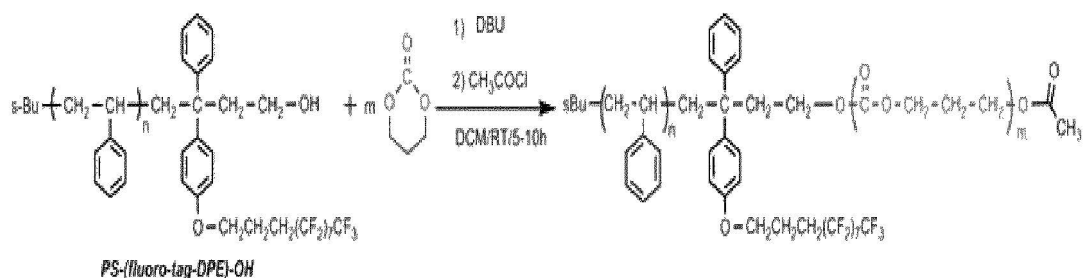
Scheme 21: Synthesis of hydroxyl with perfluoroalkane-functionalized PS macroinitiator from anionic polymerization using $\text{C}_3\text{H}_6\text{C}_8\text{F}_{17}\text{-DPE}$.

Gel permeation chromatography equipped with 100Å, 500 Å, 10^3 Å, 10^5 Å and 10^6 Å μ -ultrastayragel columns showed that the polymer had $M_{n,\text{PS}} = 10,100$ g/mol and $M_w/M_n = 1.06$ with respect to PS calibration standards. After subtracting molecular weight of $\text{C}_3\text{H}_6\text{C}_8\text{F}_{17}\text{-DPE}$ (656 g/mol), the molecular weight of PS is 9,400 g/mol.

Example 50a. Synthesis of hydroxyl polystyrene macroinitiator containing perfluoroalkane-substituted DPE via anionic polymerization (DB-PS(F)-OH-3)

Similar experiment as given in Example 50 was performed using 13.75 ml of styrene and keeping all other reagents concentration same. The obtained polymer had, $M_{n,\text{PS}} = 9,700$ g/mol and $M_w/M_n = 1.05$ with respect to PS calibration standards. After subtracting molecular weight of $\text{C}_3\text{H}_6\text{C}_8\text{F}_{17}\text{-DPE}$ (656 g/mol), the molecular weight of PS is 9,000 g/mol.

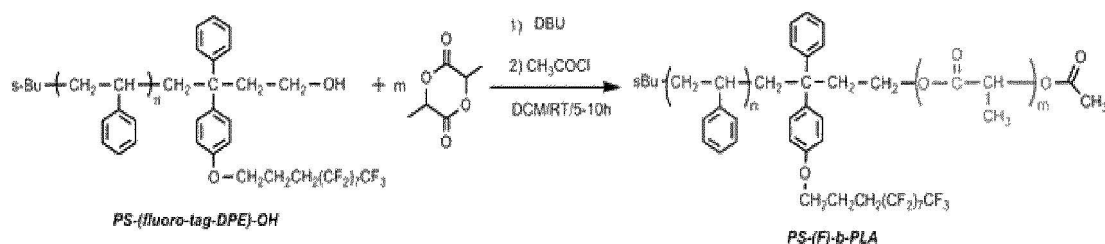
Example 51. Synthesis of PS-*b*-PTMC using macroinitiator of Example 50a (AZHC22)



Scheme 22: Ring-opening polymerization of TMC using perfluoroalkane functionalized hydroxy PS macroinitiator using base catalyst.

To an oven dried 20 ml glass vial equipped with a magnetic stir bar, 1.0 g (0.106 mmol) hydroxyl-functional polystyrene with perfluoro-pendant group (C_8F_{17}) from **Example 50a**, 2.5951 g (25.42 mmol) of 1,3-dioxan-2-one or trimethylcarbonate (TMC) and 6.35 ml of DCM were added. The reaction mixture was stirred until the macroinitiator and TMC were completely dissolved in DCM, upon which, 1,8-diazabicyclo[5.4.0]undec-7-ene (DBU) (90 μL , 0.515 mmol) was added. The reaction mixture was stirred at room temperature for 8 h in a glove box. After 8 h, the reaction vial was taken out of the glove box and benzoic acid (0.075g, 0.602 mmol) in DCM (15 ml) was added all at once to terminate the reaction. The reaction was further stirred for 10 min at room temperature. The resulting polymer was isolated by precipitating the reaction mixture in methanol. The product was collected in a frit funnel by removing methanol under vacuum and the resulting solids were redissolved and reprecipitated in methanol. The solids were collected in a frit funnel and dried under vacuum at 50°C overnight to obtain 2.34 g of BCP. This material as 20 wt. % acetone solution was precipitated in methanol: acetonitrile (375 ml, 60:40 v:v). The precipitated solid was collected in a frit funnel than dried-under vacuum at 50°C for 10 h. Yield: 2.1 g (58 %); M_n (GPC) = 19,400 g/mol, M_w = 21,700 g/mol, PDI = 1.12. Molecular weight of PTMC block based on the integration of PS block in ^1H NMR is $M_{n\text{-PTMC(NMR)}}$ = 11,100 grams/mol; PS(9.0k)-*b*-PTMC(11.1k), $V_{f\text{-PTMC}}$ = 0.49.

Example 52. Synthesis of PS-*b*-PLA with macroinitiator of Example 50 (AZHC-39)



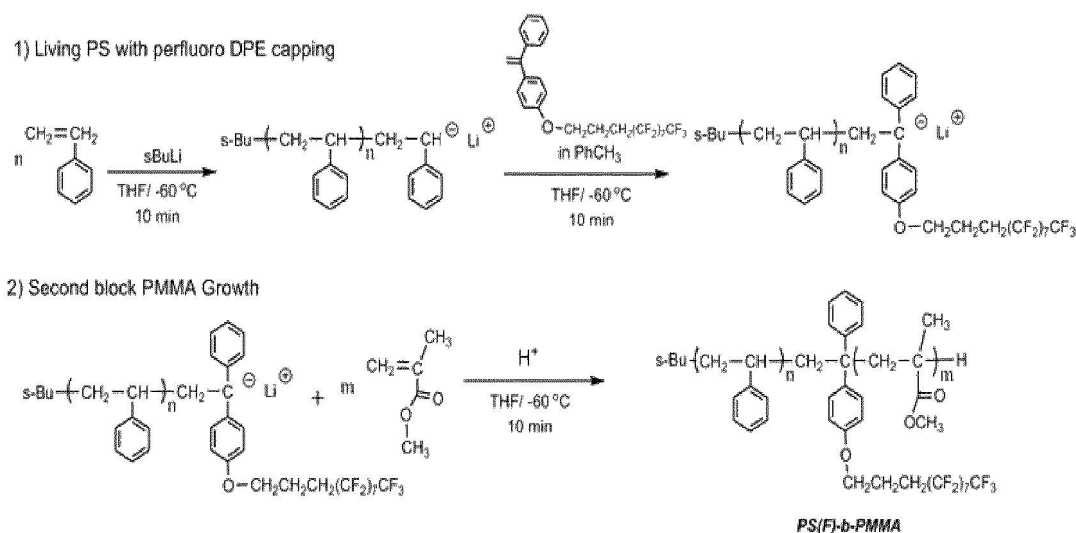
Scheme 23: Ring-opening polymerization of LA using perfluoroalkane functionalized hydroxy PS macroinitiator (**Example 50**) using base catalyst.

To an oven dried 50 ml round bottom flask equipped with a magnetic stir bar, 2.0017 g (0.197 mmol) hydroxyl-functional polystyrene with perfluoro-pendant group (C_8F_{17}) from **Example 50**, 2.9616 g (20.55 mmol) of D,L-Lactide and 17 ml of DCM were added. The reaction mixture was stirred until the macroinitiator and D,L-Lactide were completely dissolved in DCM, upon which 1-[3,5-Bis(trifluoromethyl)phenyl]-3-cyclohexylthiourea (TU) (0.2586 g, 0.699 mmol) followed by tris[2-(dimethylamino)ethyl]amine (M_6TREN) (186 μL , 0.696 mmol) were added. The reaction mixture was stirred at room temperature for 3 h and 7 min in a glove box. Then, the reaction vial was taken out of the glove box and to it, benzoic acid (0.488 g, 4 mmol) in DCM (10 ml) was added all at once. The reaction was further stirred for 10 min at room temperature. The resulting polymer was isolated by precipitating the reaction mixture in methanol. The solids were collected in a frit funnel and dried under vacuum at 50°C overnight to obtain 3.67g of BCP. This compound was dissolved in DCM (15 ml) then triethylamine, (TEA, 105 μL , 0.753 mmol) and acetyl chloride (53.6 μL , 0.753 mmol) were added and stirred for 7 h at room temperature. The resulting acetylated polymer was filtered than dried in vacuum oven at 50°C for 3 h. This material as 20 wt. % acetone solution was precipitated in methanol: acetonitrile (375 ml, 75:25 v:v). The precipitated solid was collected in a frit funnel than dried-under vacuum at

50°C for 10 h. Yield: 3.36g (67.7%); M_n (GPC) = 27.1k, M_w = 27.8k, PDI = 1.03. Molecular weight of PLA block based on the integration of PS block in ^1H NMR is $M_{n\text{-PLA(NMR)}} = 10,700$ grams/mol; PS(9.4k)-*b*-PLA(10.7k), $V_{f\text{-PTMC}} = 0.49$.

Example 53. Synthesis of PS_{18.6K}(F)-*b*-PMMA_{22K} with monofunctional perfluoroalkane-substituted DPE (C₃H₆C₈F₁₇-DPE) at the junction (DB-SFM-4)

Required amount of monomers were distilled in the presence of dehydrating agents into calibrated ampules and stored under N₂. Liquids were transferred into the reactor either via ampule or using stainless steel cannula under N₂. Into a dry 500 mL round bottom reactor equipped with side arms for connecting ampules, magnetic stir bar, nitrogen/vacuum three-way septum adapter, was added 200 mL dry tetrahydrofuran. The temperature of the reactor was reduced to -60°C using dry ice-acetone bath. Then, 3.5 mL (0.00042 moles) of *sec*-butyllithium (*sec*-BuLi, 0.12 M) was added into the reactor. Subsequently, a known amount of styrene (10 mL) was added from ampule into the reactor under fast stirring. The reaction solution turned into orange and the reaction was stirred over 10 min. Subsequently, 0.390 g of perfluoroalkane substituted 1,1'-diphenylethylene (C₃H₆C₈F₁₇-DPE) of **Example 48** in 10 ml of dry toluene was added via ampule into the reactor. The orange color of the reaction mixture turned into dark brick-red indicating conversion of styryllithium active centers to delocalized carbanion resulting from C₃H₆C₈F₁₇-DPE adduct formation. After 2 min of stirring, a small amount (2 mL) of the reaction mixture was withdrawn for PS block molecular weight analysis. Then, 10.75 mL of methyl methacrylate (MMA), distilled over a small amount of triethylaluminum (1M in toluene), was added via ampule. The reaction was terminated after 15 min with 1 mL of degassed methanol. The block copolymer was recovered by precipitation in excess methanol (4 times of the polymer solution) containing 10 % water, filtered, and dried at 55°C for 12 h under vacuum giving 19.2 g of PS(F)-*b*-PMMA (100 % yield).



Scheme 24: PS-*b*-PMMA diblock copolymer synthesis using perfluoro alkane substituted 1,1'-diphenylethylene ($\text{C}_3\text{H}_6\text{C}_8\text{F}_{17}$ -DPE) of **Example 48** as junction unit.

Gel permeation chromatography equipped with 100Å, 500 Å, 10^3 Å, 10^5 Å and 10^6 Å μ -ultrastyrigel columns showed that the 1st PS block had $M_{n,\text{PS}}$ (GPC) = 19,300 g/mol and $M_w/M_n = 1.06$ with respect to PS calibration standards. After subtracting $\text{C}_3\text{H}_6\text{C}_8\text{F}_{17}$ -DPE molecular weight (656 g/mol), the molecular weight of PS is 18,600 g/mol. The diblock copolymer molecular weight obtained from GPC is $M_{n,\text{PS-}b\text{-PMMA}} = 34,300$ g/mol and $M_w/M_n = 1.09$. The block copolymer had a small amount of low molecular weight hump, which resulted from the termination side reaction of carbanion with C-F bond of DPE derivative. The diblock copolymer was purified using Soxhlet extraction procedure using cyclohexane to remove homopolystyrene (*h*PS) and the apparent molecular weight and distribution with respect to PS standards are; $M_{n,\text{PS-}b\text{-PMMA}} = 35,900$ g/mol and $M_w/M_n = 1.07$. The molecular weight of PMMA block calculated by NMR intensities of $-\text{OCH}_3$ protons and aromatic protons based on $M_{n,\text{PS}}$ (GPC) block is: $M_{n,\text{PMMA}}(\text{NMR}) = 22,000$ g/mol. This corresponds to $V_{f,\text{PS}} = 0.49$ in the diblock copolymer.

Example 54. Synthesis of PS_{36.8K}(F)₂-b-PMMA_{38.3K} with difunctional perfluoroalkane-substituted DPE (diC₃H₆C₄F₉-DPE) of Example 49 at the junction (AZ-160)

Experiment was performed similar to **Example 53** as given above using diC₃H₆C₄F₉-DPE of **Example 49** with appropriate amount of monomers to obtain PS_{36.8K}(F)₂-b-PMMA_{38.3K}. Gel permeation chromatography equipped with 100Å, 500 Å, 10³ Å, 10⁵ Å and 10⁶ Å μ -ultrastayragel columns showed that the 1st PS block had M_{n,PS} (GPC) = 37,500 g/mol and M_w/M_n = 1.06 with respect to PS calibration standards. After subtracting C₃H₆C₈F₁₇-DPE molecular weight (656 g/mol), the molecular weight of PS is 36,800 g/mol. The diblock copolymer molecular weight obtained from GPC is M_{n,PS-b-PMMA} = 57,700 g/mol and M_w/M_n = 1.08. The block copolymer had a small amount of low molecular weight hump (5.7%). The diblock copolymer was purified using Soxhlet extraction procedure using cyclohexane to remove *h*PS. The molecular weight of PMMA block calculated by NMR intensities of –OCH₃ protons and aromatic protons based on M_{n,PS}(GPC) block is: M_{n,PMMA}(NMR) = 38,300 g/mol.

I-3. Synthesis of relevant block copolymers without surface-active junction groups [Comparative examples]

Example C1. Synthesis of hydroxyl polystyrene macroinitiator (M_n = 6,200) prepared by anionic polymerization

Required amount of monomers were distilled in the presence of dehydrating agents into calibrated ampules and stored under N₂. Liquids were transferred into the reactor either via ampule or using stainless steel cannula under N₂. Into a dry 500 mL round bottom reactor equipped with side arms for connecting ampules, magnetic stir bar, nitrogen/vacuum three-way septum adapter, was added 200 mL dry tetrahydrofuran. The temperature of the reactor was reduced to -60°C using dry ice-acetone bath. Then, 2.6 mL (0.0036 moles) of sec-butyllithium (sec-BuLi) (1.4 M) was added into the reactor. Subsequently, a known amount of styrene (14.75 mL) was added from ampule into the reactor under fast stirring.

The reaction solution turned into orange and the reaction was stirred over 10 min. After this, 2 mL of ethylene oxide (EO), which was distilled over sec-BuLi, was added into the reactor via ampule. The reaction was terminated after 15 min with 1 mL of degassed methanol. The polymer was recovered by precipitation in excess methanol (4 times of the polymer solution) containing 10 % water, filtered, and dried at 55°C for 12 h under vacuum giving 14.1 g of PS-OH with M_n (GPC) = 6,200 g/mol and M_w/M_n = 1.05 (98 % yield).

Example C2. Synthesis of hydroxyl polystyrene macroinitiator (M_n = 10,000) prepared by anionic polymerization

This polymer was prepared along the method of **Example C1**. M_n (GPC) = 10000, PDI = 1.05.

Example C3. Synthesis of PS_{6.2K}-b-PTMC_{8.4K} without surface-active junction group

To an oven dried 20 mL round bottom flask equipped with a magnetic stir bar, PS-OH (0.70 g, 0.113 mmol, M_n = 6200, PDI = 1.02, n = 57), from **Example C1**, TMC (1.76 g, 17.25 mmol), and DCM (2.94 mL) were added. The reaction mixture was stirred until the PS-OH macroinitiator and TMC were completely dissolved in DCM, upon which catalyst diphenylphosphate (DPP, 400 mg, 1.6 mmol) was added. The reaction mixture was stirred at room temperature (r.t.) for 16 hours in a glove box. The reaction flask brought out of the glove box and cooled at 0°C by immersing it in an ice-water bath. The reaction was stopped by adding DCM (6 mL), TEA (0.7 mL, 0.272 mmol) and acetyl chloride (0.25 mL, 3.52 mmol). The reaction was further stirred for two hours at room temperature. The resulting polymer was isolated by precipitating the reaction mixture in methanol. The product was collected in a frit funnel by removing methanol under vacuum and the resulting solids were redissolved in THF to form a 20 wt% solution and reprecipitated in methanol. The solid was collected in a frit funnel and dried under vacuum at 40°C for two hours to obtain the resulting compound. The

resulting polymer was dissolved in THF to form a 20 wt% solution and the polymer was precipitated in methanol: acetonitrile (200 mL, 60:40 v/v). The precipitated solids and the solvents were collected in a centrifuge tube and the solids were collected by centrifuging at 4000 RPM at 0°C followed by decanting the solvent and drying the solids in a vacuum oven at 40°C for two hours to give the polymer; M_n (GPC) = 19,300, M_w = 19,900, PDI = 1.02; M_n (NMR) = PS(6.2k)-b-PTMC(8.4k), n = 57, m = 82.

Example C4. Synthesis of PS_{6.2K}-b-PTMC_{2.9K} with non-surface active junction group

Along the procedure outlined in **Example C3**, this diblock copolymer was prepared with PS-OH macroinitiator of **Example C1** (0.30 g, 0.048 mmol, M_n = 6200, n = 57), TMC (0.296 g, 2.90 mmol), and DPP (0.12 g, 0.483 mmol). M_n (GPC) = 10800, M_w = 11200, PDI = 1.03; M_n (NMR) = PS(6.2k)-b-PTMC(2.9k), n = 57, m = 28.

Example C5. Synthesis of PS_{10K}-b-PLA_{11.4K} with non-surface active junction group

To an oven dried 4 mL round bottom flask equipped with a magnetic stir bar, PS-OH macroinitiator (0.20 g, 0.02 mmol, M_n = 10000, PDI = 1.05, n = 96) of **Example C2**, Lactide (0.254 g, 1.76 mmol), and DCM (2.00 mL) were added. The reaction mixture was stirred until the PS-OH macroinitiator and Lactide were completely dissolved in DCM, upon which M6TREN (23 mg, 0.1 mmol) and TU (37 mg, 0.1 mmol) were added. The reaction mixture was stirred at room temperature (r.t.) for 1 hour 40 minutes in a glove box. The reaction flask brought out of the glove box and was stopped by adding DCM (1 mL), TEA (0.1 mL) and acetyl chloride (0.1 mL). The reaction was further stirred for two hours at room temperature. The resulting polymer was isolated by precipitating the reaction mixture in methanol. The product was collected in a frit funnel by removing methanol under vacuum and the resulting solids were redissolved in THF to form

a 20 wt% solution and reprecipitated in methanol. The solid was collected in a frit funnel and dried under vacuum at 40°C for two hours to obtain the resulting compound. M_n (GPC) = 29,500, M_w = 31,100, PDI = 1.05; M_n (NMR) = PS(10k)-PLA(11.4k), n = 57, m = 79;

Example C6. Synthesis of PS_{36.8K}-b-PMMA_{38.3K}

Along the procedure of **Example 54** was performed by 1-1-diphenylethylene without functionalization in the same molar quantities as a reference PBCP for comparison. The polymer obtained has as similar $M_{n,PS-b-PMMA}$ = 57,700 g/mol and M_w/M_n = 1.08.

Example C7. Synthesis of hydroxyl polystyrene macroinitiator (M_n =4,200) prepared by anionic polymerization

This polymer was prepared along the method of **Example C1**. M_n (GPC) = 4200, PDI = 1.05

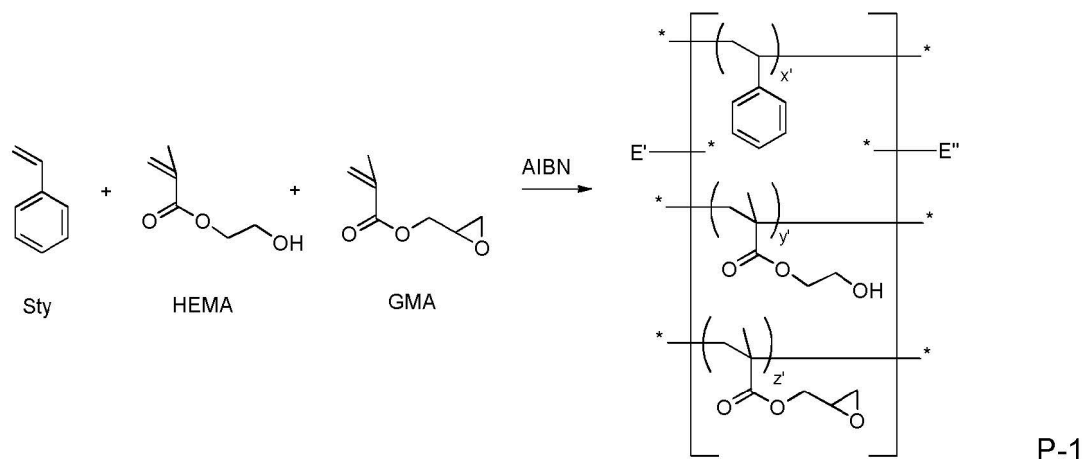
Example C8. Synthesis of PS_{4.2K}-b-PTMC_{5.3K} with non-surface active junction group

Along the procedure outlined in **Example C3**, this diblock polymer was prepared with PS-OH macroinitiator of **Example C7** (0.1 g, 0.0238 mmol, M_n = 4200, PDI = 1.05, TMC (0.133 g, 1.31 mmol), DCM (0.4 ml), and DPP (60 mg, 0.238 mmol). . M_n (GPC) = 13.7k, M_w = 14.1k, PDI = 1.02; M_n (NMR) = PS(4.2k)-b-PTMC(5.3k). .

I-4. Synthesis of underlayer materials (ULs)

Underlayer random copolymers based on styrene, TMC, and GMA (STGH). These random copolymers have the prefix "G1" in the name.

Example 55. Synthesis of macroinitiator random copolymer P-1, P(S-r-HEMA-r-GMA)



Scheme 25

In the above notation, the vertical stacking of the repeat units within the square brackets indicates a random distribution of the repeat units in the polymer chain. End group E' is linked to one of the starred bonds overlapping the left square bracket. End group E'' is linked to one of the starred bonds overlapping the right square bracket. It should be understood that for a given repeat unit, a starred bond that overlaps the left square bracket can be linked to a different repeat unit at the position indicated by the right starred bond overlapping the right square bracket, or to end group E'. Likewise, for a given repeat unit, a starred bond that overlaps the right square bracket can be linked to a different repeat unit at the position indicated by the left starred bond overlapping the left square bracket, or to end group E''. Unless otherwise indicated, subscripts x', y', and z' average number of the corresponding parenthesized repeat unit in the polymer. For P-1, the end groups E' and E'' are not shown.

Styrene (S, 14.4 g, 138.0 mmol), hydroxy ethyl methacrylate (HEMA, 1.0 g, 7.68 mmol), glycidyl methacrylate (GMA, 1.09 g, 7.66 mmol), THF (50 g), and azobisisobutyronitrile (AIBN, 0.757 g, 4.61 mmol, 3 mol% based on total moles of vinyl monomers) were combined in a 250 mL round bottom flask (RBF) equipped with a magnetic stir-bar and an overhead condenser. The reaction mixture was stirred at 70°C for 18 hours and was stopped by cooling the reaction to room

until the macroinitiator and TMC were completely dissolved in DCM, upon which 1,8-diazabicyclo[5.4.0]undec-7-ene (DBU, ~10 mg) was added. The reaction mixture was stirred at room temperature (r.t.) for 1 hour in a glove box. The reaction was stopped by bringing the reaction vial out of the glove box and by adding DCM (0.5ml), triethylamine (TEA, 0.27 g, 2.72 mmol) and acetyl chloride (~60 mg, 0.764 mmol). The reaction was further stirred for two hours at room temperature. The resulting polymer was isolated by precipitating the reaction mixture in methanol. The product was collected in a frit funnel by removing methanol under vacuum and the resulting solids were redissolved in THF to form a 20 wt% solution and reprecipitated in methanol for two more times. The solids were collected in a frit funnel and dried under vacuum at 40°C for two hours to obtain random graft copolymer G1-1, where the Styrene:TMC mole ratio $x':y'a'$ = 73:27 was determined by ^1H NMR. The average value of a' was calculated as follows:

$$\begin{aligned} x'/y' &= 50.5/3.5 \text{ (DP ratio of the P-1 macroinitiator),} \\ x'/(y'a') &= 73/27 \text{ (Styrene:TMC mole ratio of G1-1 by } ^1\text{H NMR),} \\ \text{rearranging, } x'/y' &= 73a'/27 \\ \text{substituting, } 50.5/3.5 &= 73a'/27, \text{ and} \\ \text{solving, } a' &= 5.33. \end{aligned}$$

Therefore, G1-1 has a side chain polycarbonate having an average number of carbonate repeat units $a' = 5.33$, based on the M_n of G1-1.

Examples 57-66. Synthesis of series of TMC-functional random graft copolymers, G1-2 to G1-11

Along the general procedure of **Example 56**, these polymers were prepared using macroinitiator P-1 (**Example 55**) at various Styrene:TMC molar ratios. G1-2 to G1-11 differ from G1-1 by a' , the average number of repeat units of TMC in the side chain. The results are summarized in Table 10.

Table 10

Ex.	Random Graft Polymer Name	TMC (g)	DCM (g)	DBU (mg)	P-I Initiator (g)	Temp	Time (hours)	GPC			S:TMC Mole Ratio	a ¹	Monomer % Conversion.
								M _n	M _w	PDI			
56	G1-1	0.06	0.2	~10	0.2	r.t.	1	-	-	-	73:27	5.3	>99
57	G1-2	0.1	0.32	~10	0.2	r.t.	1.5	-	-	-	60:40	9.6	>99
58	G1-3	0.46	1.50	~10	0.2	r.t.	2	23600	44000	1.87	24:76	45.7	>99
59	G1-4	0.035	0.52	~10	0.2	r.t.	0.45	-	-	-	88:12	2.0	>99
60	G1-5	0.06	0.37	~13	0.2	r.t.	1	9300	13500	1.44	73:27	5.3	>99
61	G1-6	0.1	0.45	~10	0.2	r.t.	1.5	11400	18000	1.57	59:41	10.0	>99
62	G1-7	0.06	0.72	~10	0.2	r.t.	1	-	-	-	73:27	5.3	>99
63	G1-8	0.1	0.9	~10	0.2	r.t.	1.4	-	-	-	66:34	7.4	~90
64	G1-9	0.15	0.9	~16	0.2	r.t.	1	12100	15900	1.31	62:38	8.8	~50
65	G1-10	0.15	0.9	~14	0.2	r.t.	0.75	-	-	-	62:38	8.8	~50
66	G1-11	0.15	0.9	~14	0.2	r.t.	1.4	-	-	-	43:57	19.1	~65

¹ a' is based on M_n of the random graft copolymer, without correcting for end groups

Example 67. Synthesis of random copolymer P(S-r-MMA-r-GMA)

Poly(styrene-co-methyl methacrylate-co-glycidyl methacrylate) random copolymer, P(S-r-MMA-r-GMA), referred to as N3, was prepared using the procedure of US8623458 by free-radical polymerization of styrene, methyl methacrylate, and glycidyl methacrylate in a molar feed ratio S:MMA:GMA 74:24:2. The isolated polymer composition as determined by ¹³C NMR was S:MMA:GMA 58:40:2 (mole ratio), M_n = 12100,

Examples 68-70. Synthesis of other random copolymers of P(S-r-MMA-r-GMA)

Along the general procedure of **Example 67**, these polymers were prepared... The results are summarized in Table 11.

Table 11

Example	UL Name	Polymer M _n	S:MMA:GMA Mole Ratio
---------	---------	------------------------	----------------------

67	N3	12100	58:40:2
68	N6	9500	70:28:2
69	N20		48:50:2
70	N21		38:60:2

II. Self-assembly of block copolymers with surface-active junction groups and relevant materials

II-0. Preparation of UL-modified unpatterned substrates

Examples 71-77. Substrate fabrication using random graft copolymers G1-2, G1-3, G1-7, G1-8 and G1-10

The following general procedure was used to prepare a thin film underlayer on a clean unpatterned silicon substrate. A solution was prepared by dissolving the random graft copolymer (95 parts by weight) and p-nitrobenzyl 4-methylbenzenesulphonate (p-NBT, 5 parts by weight) in propylene glycol monomethyl ether acetate (PGMEA, 10,000 parts by weight) to form a 1.0 wt% solution based on total dry solids. p-NBT is a thermal acid generator and was added to promote the grafting and partial crosslinking of a thin film of the random graft copolymer on the silicon wafer substrate when baked (annealed). The solutions were passed through a 0.2 mm polytetrafluoroethylene (PTFE) filter prior to spin coating the solution on a silicon wafer at 2000 rpm spin rate. After forming the thin film, the coated wafer was baked at 200°C for 3 minutes and cooled to room temperature. The initial baked thin film (underlayer) had a thickness of 20 nm, measured with a Nanospec Reflectometer. The underlayer was then given a solvent rinse by casting PGMEA on top of the coated wafer, letting the solvent puddle for 30 seconds, and spin drying the treated wafer at 2000 rpm for 30 seconds. The rinse was intended to remove any excess random graft copolymer that was not cross-linked or grafted to the wafer surface. The final film thickness of the underlayer was 10 nm after the solvent rinse. Table 12 summarizes the underlayer films UL-1 to UL-7 prepared with the random copolymers having a "G1" prefix in the name.

Table 12. Substrate fabrication by series of the random graft copolymers

Example	Synthetic Example	S:TMC mole ratio	DP ¹		
			S (x')	HEMA (y')	TMC (a')
71	G1-3	24:76	50.5	3.5	45.7
72	G1-2	60:40	50.5	3.5	9.6
73	G1-10	62:38	50.5	3.5	8.8
74	G1-8	66:34	50.5	3.5	7.4
75	G1-7	73:27	50.5	3.5	5.3
76	G1-11	43:57	50.5	3.5	8.8
77	G1-4	88:12	50.5	3.5	19.1

¹ x', y', and a' are based on M_n of the random graft copolymer

Examples 78-81. Substrate fabrication using N series

Underlayer substrates using N series polymers (**Examples 67-70**) were formed using various P(S-MMA-GMA) random copolymers as shown below in Table 13. A coating solution was prepared by dissolving the solid P(S-MMA-GMA) material and p-NBT (para-nitrobenzyl 4-methylbenzylsulfonate) in 95/5 w/w ratio in PGMEA at 1.0 wt% solids. p-NBT is a thermal acid generator and was added to promote the grafting and partial crosslinking of the random copolymer on a silicon wafer substrate when baked in the form of thin film. The solutions were passed through a 0.2 micrometer PTFE filter. The resulting compositions were then spin coated on the silicon wafer at 3000 rpm spin rate. After the thin film was formed, the wafer was baked at 215°C for 2 minutes and cooled to room temperature. A solvent rinse was then conducted by casting PGMEA on top of the baked wafer and letting it puddle for 30 seconds prior to spin drying at 3000 rpm for 30 seconds to form underlayers of **Examples 78 – 81**.

Table 13. UL fabrication by Nseries

Example	Synthetic Example	S:MMA:GMA mole	UL Name
78	67	58:40:2	N3
79	68	70:28:2	N6
80	69	48:50:2	N20
81	70	38:60:2	N21

Example 82. Substrate fabrication using P(S-r-MMA) random copolymer underlayer material (AZEMBL Y NLD-328J)

A 20%PS-containing P(S-r-MMA) underlayer material was received from AZ Electronic Materials (Somerville NJ) in the form of a PGMEA solution (solution code AZEMBL Y NLD-328J). The solution was used as received. The polymer solution was spin coated at 1500 rpm on a silicon wafer then baked at 250°C for 2 minutes prior to a solvent rinse to form the underlayer.

Examples 83-90. Substrate fabrication using other brush materials

Except for **Example 86a**, the general procedure of **Example 82** was applied to prepare substrates of other brush types of P(S-r-MMA) underlayers. For **Example 86a**, a cross-linking type of P(S-r-MMA) underlayer material (AZEMBL Y NLD-175) was spin-coated and baked by the same conditions as **Example 82** and used as baked. The data is summarized below in Table 14.

Table 14. Fabrication of underlayers using AZEMBLY NLD series

Example	Underlayer Name	PS:PMMA
82	AZEMBLY NLD-328J	20:80
83	AZEMBLY NLD-320	30:70
84	AZEMBLY NLD-321	40:60
85	AZEMBLY NLD-322	47:53
86	AZEMBLY NLD-208D	60:40
86a	AZEMBLY NLD-175	60:40
87	AZEMBLY NLD-307	72:28
88	AZEMBLY NLD-308	80:20
89	AZEMBLY NLD-169	100:0
90	AZEMBLY NLD-303	0:100

Example 91. Substrate fabrication using Si-containing antireflective coating material (SiARC)

The silicon-containing antireflection layer was prepared by spin coating SiArc solution (Shin Etsu SHB A-940 L35) on a silicon wafer at 1500 rpm. The resulted film was then baked at 220°C for one minute to let the material cross-linked. Final film of SiArc was 35 nm as measured by Nanospec Reflectometer.

II-1. Self-assembly of block copolymers without surface-active junction groups [Comparative examples]**Example C7. Self-assembly of block copolymer of Example C3 on UL of Example 76**

BCP of **Example C3** (PS block M_n 6200, PTMC block M_n 8400) has a TMC volume fraction of 0.52. This block copolymer represents a lamellae forming

block copolymer. A solution was prepared by dissolving the BCP in PGMEA at a concentration of 1.2 wt% based on total weight of the solution. The solution was filtered using a 0.2 micrometer PTFE filter and was spin coated at 2000 rpm rate on each of the underlayer film of **Example 76**. The resulting composite film layer containing the BCP film layer on the was subsequently annealed at 140°C for 5 minutes and immediately cooled to room temperature. The BCP film was characterized by atomic force microscopy (AFM) using a Digital Instruments 3100 AFM with a 1 N/m spring constant silicon nitride cantilever operated in a tapping mode. Scan size and speed were set at 2 μm x 2 μm (μm = micrometer or micron) area and 1 Hz respectively.

Examples C8-35. Self-assembly of block copolymers of Examples C4-C8 on ULs from Examples 71-90

Following general procedure of **Example C7** was used to form composite film layers of block copolymers of Examples C4-C8 on underlayer substrates of Examples 71-90. The results are summarized below in Table 15.

Table 15.

Example	UL Example	UL	BCP Example	BCP (M_n)	Annealing Conditions			Thin Film Morphology
					Spin Speed (rpm)	Spin time (sec)	Temperature (°C). and time (min)	
C7	Example 76	-	C3	PS-b-PTMC (6.2k-8.4k)	2000	30	140, 5	I/H
C8	Example 72	-						I/H
C9	Example 74	-						I/H
C10	Example 75	-						I/H
C11	Example 90	NLD-303						I/H
C12	Example 83	NLD-320						I/H
C13	Example 84	NLD-321						I/H
C14	Example 85	NLD-322						I/H
C15	Example 86	NLD-208D						I/H
C16	Example 87	NLD-307						I/H
C17	Example 89	NLD-169						I/H
C18	Example 75	-	C4	PS-b-PTMC (6.2k-2.9k)				II cylinders
C19	Example 77	-						II cylinders
C20	Example 76	-	C5	PS-b-PLA (10k-11.4k)	3000		170, 5	I/H
C21	Example 72	-						I/H
C22	Example 74	-						I/H
C23	Example 75	-						I/H
C24	Example 90	NLD-303						Flat
C25	Example 83	NLD-320						Flat
C26	Example 84	NLD-321						Flat
C27	Example 85	NLD-322						I/H

C28	Example 86	NLD-208D						I/H
C29	Example 87	NLD-307						I/H
C30	Example 89	NLD-169						I/H
C31	Example 86a	NLD-175	C6	PS-b-PMMA (22k-b-22k)	-	30	250, 2	Lam
C32	Example 86a	NLD-175	C6		-	30	250, 2	Lam
C33	Example 86a	NLD-175	C6		-	30	250, 2	Lam
C34	Example 86a	NLD-175	C6		-	30	250, 2	Lam
C35	Example 86a	NLD-175	C6		-	30	250, 2	Dots.

Self-assembled block copolymers of **Examples C3-C5** have a single, highly preferential air-polymer interface, where the PS block always wets the air interface owing to its low interfacial energy to air. Hence, the air-polymer interface is non-neutral with respect to self-assembled block copolymers (i.e., the air interface is preferential to the PS block). This results in parallel orientation relative to the wafer surface of the lamellae or cylinder domains under each of the test conditions. Thin-films (**Examples C31-C35**) of PS-b-PMMA without surface active junction group of Example C6 are self-assembled on neutral underlayer **Example 86a**. In this set of examples, **Example C31** has the thinnest self-assembled film ($\sim 24 \text{ nm} = 1L_o$) of perpendicular oriented domains whereas **Example C35** has the thickest film ($\sim 60 \text{ nm} = 2.5 L_o$). The film thickness of these self-assembled domains was controlled by spin-coating BCP solutions at different RPM. The results indicate that the PS-b-PMMA shows perpendicularly oriented domains only on thinner films. Once the thickness is above $\sim 2L_o$, dot-like features (undesirable) are observed.

II-2. Self-assembly of lamellae-forming block copolymers with surface-active junction groups

Example 92a. Self-assembly of PS-b-PTMC (Example 21) with surface-active junction group on underlayer of Example 75

A solution of BCP of **Example 21**, PS-b-PTMC with $\text{CH}_2\text{C}_8\text{F}_{17}$ junction group, was prepared by dissolving the BCP in PGMEA at a concentration of 1.0 wt% based on total weight of the solution. The solution was filtered using a 0.2 micrometer PTFE filter and was spin coated at 1500 rpm rate on the underlayer film of **Example 75**. The resulting composite film layer containing the BCP film layer on the was subsequently annealed at 140° C for 5 minutes and immediately cooled to room temperature. The BCP film was characterized by atomic force microscopy (AFM) using a Digital Instruments 3100 AFM with a 1 N/m spring constant silicon nitride cantilever operated in a tapping mode. Scan size and speed were set at 2 μm x 2 μm (μm = micrometer or micron) area and 1 Hz respectively.

Examples 92b-104e. Self-assembly of various block copolymers with surface-active junction groups on underlayers of Examples 71-91.

Solutions of block copolymers were prepared using the general procedure outlined in **Example 92a**. These solutions were spin coated on underlayers of **Examples 71-91** and were annealed at desired temperatures and characterized by AFM to study the BCP morphology. The results are summarized below in Table 16. "I/H" means islands/holes (not desirable). "Partially $\perp$ lamellae" means about 20% to less than 70% of the regions of the film contained perpendicular lamellae (not desirable). "Mostly $\perp$ lamellae" means 70% to less than 95% of the regions of the film contained perpendicular lamellae. " $\perp$ lamellae" means 95% to 100% of the regions of the film contained perpendicular lamellae (most desirable). "Flat" means there was no discernible structure in the film.

Table 16.

Example	UL Example	UL	BCP Example	BCP	BCP M_n	Pendant Group	BCP Conc. In PGMEA (Wt%)	Annealing Conditions			Thin Film Morphology	BCP Pitch (nm)
								Spin Speed (rpm)	Spin time (sec)	Temp (°C), time (min)		
92a	Example 75	-	21	PS-b-PTMC	4.7k-b-5.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ lamellae	~17.0
92b	Example 72	-	21	PS-b-PTMC	4.7k-b-5.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ lamellae	~17.0
92c	Example 76	-	21	PS-b-PTMC	4.7k-b-5.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ lamellae	~17.0
92d	Example 81	N21	21	PS-b-PTMC	4.7k-b-5.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ lamellae	~17.0
92e	Example 80	N20	21	PS-b-PTMC	4.7k-b-5.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ lamellae	~17.0
92f	Example 78	N3	21	PS-b-PTMC	4.7k-b-5.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ lamellae	~17.0
92g	Example 79	N6	21	PS-b-PTMC	4.7k-b-5.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ lamellae	~17.0
92h	Example 89	NLD-169	21	PS-b-PTMC	4.7k-b-5.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ lamellae	~17.0
92i	Example 90	NLD-303	21	PS-b-PTMC	4.7k-b-5.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ lamellae	
92j	Example 91	SiARC	21	PS-b-PTMC	4.7k-b-5.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ lamellae	~17.0
93a	Example 75	-	23	PS-b-PTMC	9.5k-b-11.8k	$R_3 = CH_2C_6F_{17}$	1	1500	30	170, 5	Mostly ⊥ lamellae	~25.6
93b	Example 72	-	23	PS-b-PTMC	9.5k-b-11.8k	$R_3 = CH_2C_6F_{17}$	1	1500	30	170, 5	Mostly ⊥ lamellae	
93c	Example 76	-	23	PS-b-PTMC	9.5k-b-11.8k	$R_3 = CH_2C_6F_{17}$	1	1500	30	170, 5	Mostly ⊥ lamellae	
94a	Example 75	-	18	PS-b-PTMC	5.2k-b-6.6k	$R_2 = C_2H_4C_6F_{13}$	1	1500	30	170, 5	⊥ Lamellae	
94b	Example 72	-	18	PS-b-PTMC	5.2k-b-6.6k	$R_2 = C_2H_4C_6F_{13}$	1	1500	30	170, 5	⊥ Lamellae	

94c	Example 76	-	18	PS-b-PTMC	5.2k-b-6.6k	$R_2 = C_2H_4C_6F_{13}$	1	1500	30	170, 5	$\perp$ Lamellae	
94d	Example 89	NLD-169	18	PS-b-PTMC	5.2k-b-6.6k	$R_2 = C_2H_4C_6F_{13}$	1	1500	30	170, 5	I/H	
94e	Example 91	SIARC	18	PS-b-PTMC	5.2k-b-6.6k	$R_2 = C_2H_4C_6F_{13}$	1	1500	30	170, 5	Mostly $\perp$ lamellae	
95a	Example 75	-	16	PS-b-PTMC	4.8k-b-5.8k	$R_1 = CH_2C_3F_7$	1	1500	30	140, 5	Partially $\perp$ Lamellae	
95b	Example 72	-	16	PS-b-PTMC	4.8k-b-5.8k	$R_1 = CH_2C_3F_7$	1	1500	30	140, 5	Partially $\perp$ Lamellae	
95c	Example 76	-	16	PS-b-PTMC	4.8k-b-5.8k	$R_1 = CH_2C_3F_7$	1	1500	30	140, 5	Partially $\perp$ Lamellae	
95d	Example 89	NLD-169	16	PS-b-PTMC	4.8k-b-5.8k	$R_1 = CH_2C_3F_7$	1	1500	30	140, 5	Flat	
96a	Example 75	-	26	PS-b-PTMC	6.9k-b-8.3k	$R_1 = CH_2CF_2(OCF_2C_F_2)_3C_2F_5$	1	1500	30	140, 5	$\perp$ Lamellae	~19.0
96b	Example 13	-	26	PS-b-PTMC	6.9k-b-8.3k	$R_1 = CH_2CF_2(OCF_2C_F_2)_3C_2F_5$	1	1500	30	140, 5	$\perp$ Lamellae	~19.0
96c	Example 72	-	26	PS-b-PTMC	6.9k-b-8.3k	$R_1 = CH_2CF_2(OCF_2C_F_2)_3C_2F_5$	1	1500	30	140, 5	$\perp$ Lamellae	~19.0
96d	Example 76	-	26	PS-b-PTMC	6.9k-b-8.3k	$R_1 = CH_2CF_2(OCF_2C_F_2)_3C_2F_5$	1	1500	30	140, 5	$\perp$ Lamellae	~19.0
96e	Example 89	NLD-169	26	PS-b-PTMC	6.9k-b-8.3k	$R_1 = CH_2CF_2(OCF_2C_F_2)_3C_2F_5$	1	1500	30	140, 5	$\perp$ Lamellae	~19.0
96f	Example 90	NLD-303	26	PS-b-PTMC	6.9k-b-8.3k	$R_1 = CH_2CF_2(OCF_2C_F_2)_3C_2F_5$	1	1500	30	140, 5	$\perp$ Lamellae	~19.0
96g	Example 91	SIARC	26	PS-b-PTMC	6.9k-b-8.3k	$R_1 = CH_2CF_2(OCF_2C_F_2)_3C_2F_5$	1	1500	30	140, 5	Mostly $\perp$ lamellae	
97a	Example 90	NLD-303	31	PS-b-PLA	9.5k-b-9.6k	$R_3 = CH_2C_6F_{17}$	1.2	3000	30	190, 5	Mostly $\perp$ lamellae	
97b	Example 82	NLD-328J	31	PS-b-PLA	9.5k-b-9.6k	$R_3 = CH_2C_6F_{17}$	1.2	3000	30	190, 5	$\perp$ Lamellae	~19.0
97c	Example 83	NLD-320	31	PS-b-PLA	9.5k-b-9.6k	$R_3 = CH_2C_6F_{17}$	1.2	3000	30	190, 5	$\perp$ Lamellae	~19.0
98a	Example 73	-	37	PS-b-PMeCAR	4.7k-b-6.2k	$R_3 = CH_2C_6F_{17}$	1.2	2000	30	140, 5	$\perp$ Lamellae	~12.5
98b	Example 90	NLD-303	37	PS-b-PMeCAR	4.7k-b-6.2k	$R_3 = CH_2C_6F_{17}$	1.2	2000	30	140, 5	$\perp$ Lamellae	~12.5

98c	Example 90	NLD-303	47	PS-b-PMeCAR	8.5k-b-10.2k	$R_3 = CH_2C_6F_{17}$	1.2	2000	30	200, 5	⊥ Lamellae	~15.7
99	Example 85	NLD-322	46	PS-b-PLA	8.5k-b-9.2k	$R_3 = CH_2C_6F_{17}$	1.2	2000	30	170, 5	⊥ Lamellae	~15.6
100a	Example 76	-	27	PS-b-PTMC	4.2k-b-5.0k	$R_6 = C_{18}H_{37}$	1	1500	30	140, 5	Partially ⊥ Lamellae	-
100b	Example 72	-	27	PS-b-PTMC	4.2k-b-5.0k	$R_6 = C_{18}H_{37}$	1	1500	30	140, 5	Partially ⊥ Lamellae	-
100c	Example 74	-	27	PS-b-PTMC	4.2k-b-5.0k	$R_6 = C_{18}H_{37}$	1	1500	30	140, 5	Mostly ⊥ Lamellae	-
100d	Example 75	-	27	PS-b-PTMC	4.2k-b-5.0k	$R_6 = C_{18}H_{37}$	1	1500	30	140, 5	Partially ⊥ Lamellae	-
100e	Example 89	NLD-169	27	PS-b-PTMC	4.2k-b-5.0k	$R_6 = C_{18}H_{37}$	1	1500	30	140, 5	Dots	-
100f	Example 91	SIARC	27	PS-b-PTMC	4.2k-b-5.0k	$R_6 = C_{18}H_{37}$	1	1500	30	140, 5	Dots	-
101a	Example 83	NLD-320	33	PS-b-PLA	15.6k-b-16.6k	$R_3 = CH_2C_6F_{17}$	1	3000	30	170, 5	⊥ Lamellae	~25
101b	Example 84	NLD-321	33	PS-b-PLA	15.6k-b-16.6k	$R_3 = CH_2C_6F_{17}$	1	3000	30	170, 5	⊥ Lamellae	~25
102	Example 90	NLD-303	51	PS-b-PTMC	9.5k-b-11.6k	$C_3H_6C_6F_{17}$	1	2000	30	170, 5	⊥ Lamellae	~24.8
103	Example 90	NLD-303	52	PS-b-PLA	9.4k-b-10.7k	$C_3H_6C_6F_{17}$	1	2000	30	170, 5	⊥ Lamellae	~20
104a	Example 86a	NLD-175	53	PS-b-PMMA	18.6k-b-22k	$C_3H_6C_6F_{17}$			30	250, 2	⊥ Lamellae	~24.0
104b	Example 86a	NLD-175	53	PS-b-PMMA	18.6k-b-22k	$C_3H_6C_6F_{17}$			30	250, 2	⊥ Lamellae	~24.0
104c	Example 86a	NLD-175	53	PS-b-PMMA	18.6k-b-22k	$C_3H_6C_6F_{17}$			30	250, 2	⊥ Lamellae	~24.0

104d	Example 86a	NLD- 175	53	PS-b-PMMA	18.8k-b- 22k	$C_3H_6C_6F_{17}$			30	250, 2	⊥ Lamellae	$\sim$ 24.0
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Self-assembled thin-films of **Examples 92a-101b** demonstrate that having a surface-active junction group enables perpendicular orientation of lamellae for high- χ block copolymers on non-preferential and preferential underlayers. Compared to the block copolymers without a surface-active junction group where there PS block is always at the BCP-air interface (parallel lamellae, undesirable), the inventive examples show perpendicular morphologies (desirable) throughout the thickness of the film. Another unexpected result of the inventive examples is that the perpendicular orientation can be achieved on both preferential and non-preferential substrates. For PS-b-PTMC lamellae BCP (**Example C3**), the underlayer substrate of **Example 72** is a non-preferential (neutral), whereas the underlayer substrate of **Example 90** is preferential (non-neutral). **Examples 92a and 92i** demonstrate perpendicularly orientated BCP domains of PS-b-PTMC with surface-active junction group on both these underlayer substrates, indicating an unexpected result. Similarly, thin film of **Example 104e** (~60 nm, 2.5Lo) for PS-b-PMMA with surface-active junction group shows perpendicularly oriented domains throughout the entire thickness of the film. In contrast, the thin film of PS-b-PMMA without surface-active junction group of the same film thickness and underlayer composition shows dot-like features (**Example C35**). This indicates that JBCPs enable perpendicular orientation of much thicker films.

II-3. Self-assembly of cylinder-forming block copolymers with surface-active junction groups

Examples 105a – 109h. Self-assembly of cylinder-forming block copolymers with surface-active junction groups.

Solutions of block copolymers were prepared using the general procedure outlined in **Example 92a**. These solutions were spin coated on underlayers of **Examples 71-91** and were annealed at desired temperatures and characterized by AFM to study the BCP morphology. The results are summarized below in Table 17. "Mostly $\parallel$ Cylinders" means about 20% to less than 70% of the regions of the film contained perpendicular cylinders (not desirable). "Mostly $\perp$ Cylinders"

means 70% to less than 95% of the regions of the film contained perpendicular cylinders. "⊥ Cylinders" means 95% to 100% of the regions of the film contained perpendicular cylinders (most desirable).

Table 17

Example	UL Example	UL	BCP Example	BCP	BCP M_n	Pendant Group	BCP Conc. In PGMEA (Wt%)	Annealing Conditions			Thin Film Morphology	BCP Pitch (nm)
								Spin Speed (rpm)	Spin time (sec)	Temp (°C. and time (min)		
105a	Example 76	-	20	PS-b-PTMC	4.7k-b-2.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ Cylinders	~11.0
105b	Example 72	-	20	PS-b-PTMC	4.7k-b-2.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ Cylinders	~11.0
105c	Example 74	-	20	PS-b-PTMC	4.7k-b-2.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ Cylinders	~11.0
105d	Example 75	-	20	PS-b-PTMC	4.7k-b-2.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ Cylinders	~11.0
105e	Example 89	NLD-169	20	PS-b-PTMC	4.7k-b-2.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ Cylinders	~11.0
	Example 91	SIARC	20	PS-b-PTMC	4.7k-b-2.3k	$R_3 = CH_2C_6F_{17}$	1	1500	30	140, 5	⊥ Cylinders	-
105g	Example 72	-	20	PS-b-PTMC	4.7k-b-2.3k	$R_3 = CH_2C_6F_{17}$	1.5	2000	30	140, 5	⊥ Cylinders	~11.0
105h	Example 72	-	20	PS-b-PTMC	4.7k-b-2.3k	$R_3 = CH_2C_6F_{17}$	1.5	1500	30	140, 5	⊥ Cylinders	~11.0
105i	Example 72	-	20	PS-b-PTMC	4.7k-b-2.3k	$R_3 = CH_2C_6F_{17}$	1.5	1200	30	140, 5	⊥ Cylinders	~11.0
106a	Example 76	-	19	PS-b-PTMC	5.2k-b-2.5k	$R_2 = C_2H_4C_6F_{13}$	1	1500	30	170, 5	⊥ Cylinders	~11.0
106b	Example 72	-	19	PS-b-PTMC	5.2k-b-2.5k	$R_2 = C_2H_4C_6F_{13}$	1	1500	30	170, 5	⊥ Cylinders	~11.0
106c	Example 74	-	19	PS-b-PTMC	5.2k-b-2.5k	$R_2 = C_2H_4C_6F_{13}$	1	1500	30	170, 5	⊥ Cylinders	~11.0
106d	Example 75	-	19	PS-b-PTMC	5.2k-b-2.5k	$R_2 = C_2H_4C_6F_{13}$	1	1500	30	170, 5	⊥ Cylinders	~11.0
106e	Example 89	NLD-169	19	PS-b-PTMC	5.2k-b-2.5k	$R_2 = C_2H_4C_6F_{13}$	1	1500	30	170, 5	Mostly ⊥ Cylinders	-

106f	Example 91	SiARC	19	PS-b-PTMC	5.2k-b- 2.5k	$R_2 = C_2H_4C_8F_{13}$	1	1500	30	140, 5	II Cylinders	-
107a	Example 76	-	17	PS-b-PTMC	4.8k-b- 5.8k	$R_1 = CH_2C_3F_7$	1	1500	30	140, 5	Mostly $\perp$ Cylinders	$\sim$ 11.0
107b	Example 72	-	17	PS-b-PTMC	4.8k-b- 2.4k	$R_1 = CH_2C_3F_7$	1	1500	30	140, 5	Mostly $\perp$ Cylinders	$\sim$ 11.0
107c	Example 74	-	17	PS-b-PTMC	4.8k-b- 2.4k	$R_1 = CH_2C_3F_7$	1	1500	30	140, 5	Mostly $\perp$ Cylinders	$\sim$ 11.0
107d	Example 75	-	17	PS-b-PTMC	4.8k-b- 2.4k	$R_1 = CH_2C_3F_7$	1	1500	30	140, 5	Mostly $\perp$ Cylinders	$\sim$ 11.0
107e	Example 89	NLD- 169	17	PS-b-PTMC	4.8k-b- 2.4k	$R_1 = CH_2C_3F_7$	1	1500	30	140, 5	II Cylinders	-
107f	Example 91	SiARC	17	PS-b-PTMC	4.8k-b- 2.4k	$R_1 = CH_2C_3F_7$	1	1500	30	140, 5	II Cylinders	-
108a	Example 76	-	25	PS-b-PTMC	6.9k-b- 3.2k	$R_4 =$ $CH_2CF_2(OCF_2$ $CF_2)_3C_2F_5$	1	1500	30	140, 5	$\perp$ Cylinders	$\sim$ 13.0
108b	Example 73	-	25	PS-b-PTMC	6.9k-b- 3.2k	$R_4 =$ $CH_2CF_2(OCF_2$ $CF_2)_3C_2F_5$	1	1500	30	140, 5	$\perp$ Cylinders	$\sim$ 13.0
108c	Example 89	NLD- 169	25	PS-b-PTMC	6.9k-b- 3.2k	$R_4 =$ $CH_2CF_2(OCF_2$ $CF_2)_3C_2F_5$	1	1500	30	140, 5	$\perp$ Cylinders	$\sim$ 13.0
108d	Example 90	NLD- 303	25	PS-b-PTMC	6.9k-b- 3.2k	$R_4 =$ $CH_2CF_2(OCF_2$ $CF_2)_3C_2F_5$	1	1500	30	140, 5	$\perp$ Cylinders	$\sim$ 13.0
108e	Example 91	SiARC	25	PS-b-PTMC	6.9k-b- 3.2k	$R_4 =$ $CH_2CF_2(OCF_2$ $CF_2)_3C_2F_5$	1	1500	30	140, 5	Mostly $\perp$ Cylinders	-
109a	Example 76	-	24	PS-b-PTMC	9.5k-b- 4.8k	$R_3 = CH_2C_8F_{17}$	1	1500	30	140, 5	$\perp$ Cylinders	$\sim$ 19.0
109b	Example 72	-	24	PS-b-PTMC	9.5k-b- 4.8k	$R_3 = CH_2C_8F_{17}$	1	1500	30	140, 5	$\perp$ Cylinders	$\sim$ 19.0
109c	Example 74	-	24	PS-b-PTMC	9.5k-b- 4.8k	$R_3 = CH_2C_8F_{17}$	1	1500	30	140, 5	$\perp$ Cylinders	$\sim$ 19.0
109d	Example 75	-	24	PS-b-PTMC	9.5k-b- 4.8k	$R_3 = CH_2C_8F_{17}$	1	1500	30	140, 5	$\perp$ Cylinders	$\sim$ 19.0
109e	Example 89	NLD- 169	24	PS-b-PTMC	9.5k-b- 4.8k	$R_3 = CH_2C_8F_{17}$	1	1500	30	140, 5	Mostly $\perp$ Cylinders	$\sim$ 19.0
109f	Example 91	SiARC	24	PS-b-PTMC	9.5k-b- 4.8k	$R_3 = CH_2C_8F_{17}$	1	1500	30	140, 5	$\perp$ Cylinders	$\sim$ 19.0

109g	Example 74	-	24	PS-b-PTMC	9.5k-b- 4.8k	$R_3 = CH_2C_8F_{17}$	1	1500	30	170, 1	⊥ Cylinders	~ 19.0
109h	Example 74	-	24	PS-b-PTMC	9.5k-b- 4.8k	$R_3 = CH_2C_8F_{17}$	1	1500	30	190, 1	⊥ Cylinders	~ 19.0
110a	Example 76	-	28	PS-b-PTMC	4.2k-b- 2.9k	$R_5 = C_{18}H_{37}$	1	1500	30	140, 5	Cylinders	-
110b	Example 72	-	28	PS-b-PTMC	4.2k-b- 2.9k	$R_5 = C_{18}H_{37}$	1	1500	30	140, 5	Cylinders	-
110c	Example 74	-	28	PS-b-PTMC	4.2k-b- 2.9k	$R_5 = C_{18}H_{37}$	1	1500	30	140, 5	Cylinders	-
110d	Example 75	-	28	PS-b-PTMC	4.2k-b- 2.9k	$R_5 = C_{18}H_{37}$	1	1500	30	140, 5	Cylinders	-
110e	Example 89	NLD- 169	28	PS-b-PTMC	4.2k-b- 2.9k	$R_5 = C_{18}H_{37}$	1	1500	30	140, 5	Cylinders	-
110f	Example 91	SiARC	28	PS-b-PTMC	4.2k-b- 2.9k	$R_5 = C_{18}H_{37}$	1	1500	30	140, 5	Cylinders	-

Self-assembled thin-films of **Examples 105a-109h** demonstrate that having a surface-active junction group enables perpendicular orientation of cylinders for high- χ block copolymers on non-preferential and preferential underlayers. Compared to the block copolymers without a surface-active junction group where there PS block is always at the BCP-air interface (parallel cylinders, undesirable), the inventive examples show perpendicular morphologies (desirable) throughout the thickness of the film. An unexpected result of the inventive examples is that the perpendicular orientation can be achieved on both preferential and non-preferential substrates. For PS-b-PTMC cylinder BCP (**Example C4**), the underlayer substrate of **Example 72** is a preferential (non-neutral), whereas **Example 105b** demonstrates perpendicularly orientated BCP domains of PS-b-PTMC with surface-active junction group on this underlayer, indicating an unexpected result.

II-4. Self-assembly of block copolymer blends - Binary blends of two lamellae-forming block copolymers with a surface active junction group

Examples 111 –113. Blends of JBCPs of **Examples 21** and **23**.

Blending of two or more JBCPs is another method to achieve perpendicular morphologies. This method also allows fine-tuning the effective pitch of the BCP formulation for self-assembly and directed self-assembly applications. 1.5 wt% solutions of **Examples 21** and **23** were made by dissolving the appropriate solid JBCP in PGMEA. The resulting solutions were passed through a 0.2 μm PTFE filter. Formulations of Examples 110-113 are summarized in below in Table 18.

Table 18

Example	Example 21:Example 23 Blend Ratio (w/w)	Formulation			
		PGMEA (g)	1.5 wt% Example 21 solution (g)	1.5 wt% Example 23 solution (g)	Overall wt % of dry solids in Blends
111	30:70	0.33	0.020	0.046	1
112	50:50	0.33	0.033	0.033	1
113	70:30	0.33	0.046	0.020	1

Examples 114a–116d. Self-assembly of thin-films of JBCP blends on various underlayers.

Solutions of JBCP blends were spin-coated on various underlayers and annealed at desired temperatures to form self-assembled thin-film layers. Top-down images of the thin-films were collected by SEM and these images were used to calculate the pitch of the thin-films using FFT. Results of the thin-films are summarized below in Table 19.

Table 19.

Example	UL Example	UL	Blend Example	Example 21:Example 23 Blend Ratio (w/w)	Pendant Group	Blend Conc. In PGMEA (Wt%)	Annealing Conditions			Thin Film Morphology	Blend Pitch (nm)
							Spin Speed (rpm)	Spin time (sec)	Temp (°C. and time (min)		
114a	Example 72	-	110	30:70	$R_3 = CH_2C_8F_{17}$	1	1500	30	170, 5	⊥ Lamellae	~25
114b	Example 74	-	110	30:70	$R_3 = CH_2C_8F_{17}$	1	1500	30	170, 5	⊥ Lamellae	
114c	Example 75	-	110	30:70	$R_3 = CH_2C_8F_{17}$	1	1500	30	170, 5	⊥ Lamellae	
115a	Example 72	SiARC	111	50:50	$R_3 = CH_2C_8F_{17}$	1	1500	30	170, 5	⊥ Lamellae	~20
115b	Example 74	-	111	50:50	$R_3 = CH_2C_8F_{17}$	1.5	1500	30	170, 5	⊥ Lamellae	
115c	Example 75	-	111	50:50	$R_3 = CH_2C_8F_{17}$	1.5	1500	30	170, 5	⊥ Lamellae	
116a	Example 72	-	112	70:30	$R_2 = C_2H_4C_8F_{13}$	1	1500	30	170, 5	⊥ Lamellae	~19.3
116b	Example 74	-	112	70:30	$R_2 = C_2H_4C_8F_{13}$	1	1500	30	170, 5	⊥ Lamellae	
116c	Example 75	-	112	70:30	$R_2 = C_2H_4C_8F_{13}$	1	1500	30	170, 5	⊥ Lamellae	

II-5. Self-assembly of block copolymer blends - Binary blends of a lamellae-forming JBCP (BCP with a surface active junction group) and a lamellae-forming BCP (without surface-active junction group)

Example 117. Blend of JBCP of Example 22 and Comparative Example C8

In 0.132 g of PGMEA was dissolved 0.010 g of PS-b-PTMC of **Example C8** (4.2k-5.3k, no surface-active junction group) This solution was added to a 0.067 g of 1.5% PGMEA solution of PS-b-PTMC of **Example22** (4.2k-b-5.7k, with surface-active junction group, $\text{CH}_2\text{C}_8\text{F}_{17}$). This solution was filtered through a 0.45 μm pore PTFE membrane filter to obtain a 1.0% PGMEA solution of a 50/50 (w/w) blend.

Example 118a-118f. Thin-film self-assembly of Example 118 on various underlayers.

Formulation of **Example 117** was spin-coated on various underlayers and annealed to form phase-separated domains. Annealed thin-films were characterized by AFM. Results are summarized below in Table 20.

Table 20

Example	UL Example	UL	Formulation Example	Example 22: Example C8 Blend Ratio (w/w)	Pendant Group of Example 22	Blend Conc. In PGMEA (Wt%)	Annealing Conditions			Thin Film Morphology	Blend Pitch (nm)
							Spin Speed (rpm)	Spin time (sec)	Temp (°C. and time (min)		
118a	Example 76	-	117	50:50	$R_3 = CH_2C_8F_{17}$	1	1500	30	170, 20	Mostly II Lamellae	-
118b	Example 72	-	117	50:50	$R_3 = CH_2C_8F_{17}$	1	1500	30	170, 20	⊥ Lamellae	~15.3
118c	Example 74	-	117	50:50	$R_3 = CH_2C_8F_{17}$	1	1500	30	170, 20	⊥ Lamellae	~15.3
118d	Example 75	-	117	50:50	$R_3 = CH_2C_8F_{17}$	1	1500	30	170, 20	Mostly II Lamellae	-
118e	Example 89	NLD-169	117	50:50	$R_3 = CH_2C_8F_{17}$	1	1500	30	170, 20	Flat	-
118f	Example 91	SiARC	117	50:50	$R_3 = CH_2C_8F_{17}$	1	1500	30	170, 20	Flat	-

III. Directed self-assembly of block copolymers with surface-active junction groups

III-1. Graphoepitaxy of block copolymers with surface-active junction groups

The guiding prepatterns for graphoepitaxy comprise both topography and underlayer. The underlayer in the prepattern that is exposed to subsequent JBCP coating is used to support vertical orientation of JBCP domains. The topography in the guiding prepattern was used to confine at least the majority of JBCP in the recessed region. Typical height of the topography is at least larger than $0.5 L_o$ (L_o is the pitch of the domains of JBCP). The topography was generated by a variety lithography methods and pattern transfer methods. In general, the topographical prepattern were stable under the solvent and annealing condition employed for application of the JBCPs.

Graphoepitaxy of lamellae and cylinder-forming JBCPs with a pitch of L_o were carried out by coating and annealing a layer of JBCP on such topographical prepatterns comprising underlayer and topography. Typically in graphoepitaxy, the thickness of the JBCP film on the topographical prepattern was thinner than, or equal to than the height of the topography. The majority of JBCPs were laterally confined within the recessed region by the sidewall of the topography and therefore, after annealing, the spatial arrangement of the self-assembled, vertically oriented JBCP domains was mainly directed by the topography and the sidewall surface chemistry of the topography.

Preparation of prepatterns

Graphoepitaxy was carried out on two types of topographical prepatterns, PR7210/G1-2 prepattern and/or PR3538/SiARC prepatterns, where various JBCPs were spin-coated and baked to generate aligned domains. The samples were cleaved and cross-sectionally exposed to CF_4+H_2 plasma to selectively remove PTMC or PLA domains from JBCPs used. Each prepattern was prepared as follows.

PR7210/UL prepattern: A layer of **Example 72** was coated on SiNx substrate then baked at 205°C for 3 min followed by PGMEA rinse to generate the underlayer. On top of the G1-2/SiNx stack, a 60 nm-thick film of a ArF photoresist (JSR ArF AIM7210JN-8) was prepared and exposed with 193 nm interference lithography, then developed in the negative tone with an organic developer (2-heptanone) to produce trench pre-patterns comprising resist sidewalls and the underlayer-modified bottoms. The guiding prepattern was hard-baked at 200°C for 3 min. **PR 3538/SiARC prepattern:** On an organic hardmask substrate was processed SiARC (Shin-Etsu SHB-A940 L35) at a standard condition. On top of the SiARC/hardmask stack, a 50 nm-thick film of a ArF photoresist (JSR ArF AX3538) was prepared and exposed with 193 nm immersion lithography, then developed in the positive tone with a standard aqueous alkaline developer (0.26N TMAH) to produce trench prepattern comprised of resist sidewalls and the SiARC bottoms. The developed resist prepattern was hardened by UV flood exposure with a total exposure dose of 60 mJ/cm² followed by post-exposure bake at 115°C for 60 sec and 185°C for 120 sec.

Processing of directed self-assembly

Example 119a: Example 21 in PR7210/UL prepattern

A 0.8% PGMEA solution of **Example 21** was spin-coated at 4000 rpm on a coupon of the PR7210/G1-2 substrate. The coated substrate was baked at 170°C for 5 min to generate vertically oriented lamellae aligned in parallel to the trench sidewalls. **Figure 9a-1** and **9a-2** show the plane-view and cross-section SEM images of the sample after a perpendicular etch.

Example 119b: Example 21 in PR3538/SiARC prepattern

A 0.8% PGMEA solution of **Example 21** was spin-coated at 2000 rpm on a coupon of the PR3538/SiARC substrate. The coated substrate was baked at

170°C for 5 min to generate vertically oriented lamellae aligned in parallel to the trench sidewalls.

Example 120: Example 31 in PR7210/G1-2 prepattern

A 0.8% PGMEA solution of **Example 31** was spin-coated at 2000 rpm on a coupon of the PR7210/G1-2 substrate. The coated substrate was baked at 170°C for 5 min to generate vertically oriented lamellae aligned in parallel to the trench sidewalls. **Figure 9b-1** and **9b-2** shows the plane-view and cross-section SEM images of the sample after a perpendicular etch.

Example 121: Example 20 in PR7210/G1-2 prepattern

A 0.8% PGMEA solution of **Example 20** was spin-coated at 4000 rpm on a coupon of the PR7210/G1-2 substrate. The coated substrate was baked at 170°C for 5 min to generate vertically oriented cylinders aligned in parallel to the trench sidewalls.

Example 122: Example 24 in PR3538/SiARC prepattern

A 0.8% PGMEA solution of **Example 24** was spin-coated at 4000 rpm on a coupon of the PR3538/SiARC substrate. The coated substrate was baked at 170°C for 5 min to generate vertically oriented cylinders aligned in parallel to the trench sidewalls. **Figure 9c-1** and **9c-2** shows the plane-view and cross-section SEM images of the sample after a perpendicular etch.

III-2. Overcoating of block copolymers with surface-active junction groups on topographical substrate

The guiding prepatterns for overcoating JBCPs comprise both topography and underlayer. The underlayer in the prepattern that was exposed to subsequent JBCP coating was used to support vertical orientation of JBCP domains. The topography in the guiding prepattern was used to confine at least some of JBCP in the recessed region. Typical height of the topography was at least larger than

0.7 L_o (L_o is the pitch of JBCP). The pitch of the guiding prepattern was $n L_o$ (commensurate with the pitch of JBCP). Typical BCP coating thickness was significantly larger than the height of the topographical features on the guiding prepattern. In general, the topographical prepattern were stable under the solvent and annealing condition of JBCPs.

Preparation of prepatterns

Topographical guiding prepattern for overcoating of JBCP was made by O_2 plasma treatment of the photoresist/underlayer/SiARC pattern. A 60 nm-thick film of ArF photoresist (JSR ArF AIM7946JN-8) was coated on top of the underlayer (AZ Electronic materials AZEMBLTMY NLD-089)/SiARC A940 stack, patternwise exposed and then developed in the negative tone with an organic developer (2-heptanone) to produce resist trench pattern. Subsequently, the prepattern was treated by O_2 plasma for 15 sec to remove the NLD-089 layer that was not covered by the patterned resist and to oxidize the SiARC surface, which was followed by 200°C bake for 3 min to harden the remaining trench walls comprising the resist and NLD-089 underneath. Figure 10a shows the prepattern described above.

Processing of directed self-assembly

Directed self-assembly of lamellae-forming and cylinder-forming block copolymers with a pitch of L_o were carried out by coating and annealing a layer of BCP over the topographical prepatterns comprising the oxidized resist lines and oxidized SiARC substrate surface as shown in **Figure 10a**. The height of the oxidized resist line is ~15 nm and the pitch of the organic line is 84 nm (~5 L_o). Typically, for overcoating, the thickness of the BCP film on the topographical prepattern was thicker than the height of the topography. Majority of JBCPs were coated over SiARC and topographical guiding lines and, after annealing, the spatial arrangement of the self-assembled, vertically-oriented BCP domains were directed by the topography and surface chemistry of the etched resist and

underlayer. After coating and annealing thick BCP over the plasma treated organic line/ SiARC substrate, the overcoated samples were cleaved and cross-sectionally exposed to a CF_4+H_2 plasma to partially remove the PTMC domains or PLA domains from each BCP.

Example 123: Example 22 over O_2 -plasma treated organic lines / SiARC prepattern. A 1% PGMEA solution of **Example 22** was spin-coated at 800 rpm on a coupon of O_2 -plasma treated organic lines / SiARC prepattern. The coated substrate was baked at 170°C for 10 min in the ambient atmosphere to form vertically-oriented lamellae aligned to the underlying organic lines. **Figure 10b-1 and 10b-2** show the plane-view and cross-section SEM images of the sample after a perpendicular etch.

Example 124: Example 31 over O_2 -plasma treated organic lines / SiARC prepattern. A 1% PGMEA solution of **Example 31** was spin-coated at 800 rpm on a coupon of O_2 -plasma treated organic lines / SiARC prepattern. The coated substrate was baked at 170°C for 10 min in the ambient atmosphere to form vertically-oriented lamellae aligned to the underlying organic lines. **Figure 10c-1 and 10c-2** show the plane-view and cross-section SEM images of the sample after a perpendicular etch.

Example 125: Example 24 over O_2 -plasma treated organic lines / SiARC prepattern. A 1% PGMEA solution of **Example 24** was spin-coated at 1500 rpm on a coupon of O_2 -plasma treated organic lines / SiARC prepattern. The coated substrate was baked at 170°C for 5 min in the ambient atmosphere to form vertically-oriented cylinder arrays where the close-packed rows align to the underlying organic lines. **Figure 10d** shows the plane-view SEM image of the sample.

III-3. Chemoepitaxy of block copolymers with surface-active junction groups on chemical prepatterns

Chemoepitaxy requires a chemical prepattern to guide the self-assembly of JBCP. The chemical prepattern consists of two regions of different surface chemistry: “wetting and neutral”. The wetting region is selectively wetted by one domain and the neutral region supports vertical orientation of both domains. The pitch of the chemical prepattern is defined by the distance of the one wetting region to the nearest wetting region and is an integer number times the pitch of the repeating distance of block copolymers. Both wetting region and neutral region are made from materials that compatible to solvent and process conditions of the DSA processes. For a JBCP with a pitch of L_0 , the pitch of the chemical prepattern is nL_0 . The topography of the chemical epitaxy prepattern is minimized: the height difference between the neutral region and wetting region is less than $0.5 L_0$. Typical thickness of JBCP coating is significantly larger than the height difference (topography) of the chemical prepattern. Films of JBCP are coated over the chemical prepattern and one of the JBCP domain preferentially wets the wetting region and creates a lateral ordering of JBCP domains.

Preparation of prepatterns

Chemoepitaxy was carried out on two types of topographical prepatterns, HSQ/SMMA and/or xPS/SMMA prepatterns, where various JBCPs were spin-coated and baked to generate aligned domains. Each chemical prepattern was prepared as follows.

HSQ/SMMA chemical prepattern: A solution of SMMA underlayer, NLD-208D (AZ Electronic materials AZEMBLY™ NLD-208D), was coated on top of silicon nitride and baked, followed by a PGMEA rinse to form a brush layer over silicon nitride. HSQ (diluted Dow Corning XR-1541 E-beam resist in methylisobutylketone) was coated and baked at 130°C / 2 min to form a thin HSQ layer over SMMA brush. The HSQ pattern was written by electron beam lithography using Leica-Vistec VB6 and developed by 0.26N TMAH. These HSQ features serves as the wetting region and the exposed SMMA brush layer serves as the neutral region. The typical height of HSQ features after development is

between 3 nm - 7 nm which is less than 0.5 Lo of JBCP in the experiments and the pitches of HSQ features are around 2 Lo , 3 Lo and 4Lo, commensurable to the pitch of JBCP. **xPS/SMMA chemical prepattern (A):** A fully cross-linkable polystyrene (xPS) solution (AZ Electronic Materials, AZEMBLY[™] NLD-128) was spin coated on top of a SiN/amorphous carbon (α -C) stack and baked at 315°C /5min under a N₂ environment. An ArF immersion resist was then coated on the xPS/SiN/ α -C stack, exposed with a 193 scanner and developed with a standard developer, generating grating patterns with 84 nm pitch. The patterned resist on the xPS/SiN/ α -C stack was then etched by RIE to remove the xPS layer that was not covered by the resist. The remaining resist was subsequently removed by 0.26N TMAH. As a result, 8 nm-thick xPS gratings with 84 nm pitch were generated on top of the SiN/ α -C stack. On the pattern of the xPS gratings, a PMMA brush material (AZ Electronic Materials, AZEMBLY[™] NLD-303) was coated and baked at 250°C / 2 min followed by a PGMEA rinse, resulting in formation of alternate stripes of xPS and the brush material. The brush in this case selectively grafted on the open SiN surface so it did not interfere with the top and sidewall surfaces of the xPS gratings. In this prepattern, the xPS features serve as the wetting region and the exposed SMMA brush layer serves as the neutral region. The typical height of xPS / SMMA features is between 3 nm - 5 nm which is less than 0.5 Lo of JBCP in the experiments and the pitch of xPS features is around 5 Lo , commensurable to the pitches of JBCPs. **xPS/SMMA chemical prepattern (B):** A fully cross-linkable polystyrene (xPS) solution (AZ Electronic Materials, AZEMBLY[™] NLD-128) was spin coated on top of a SiN/amorphous carbon (α -C) stack and baked at 315°C /5min under a N₂ environment. An ArF immersion resist was then coated on the xPS/SiN/ α -C stack, exposed with a 193 scanner and developed with a standard developer, generating grating patterns with 84 nm pitch. The patterned resist on the xPS/SiN/ α -C stack was then etched by RIE to remove the xPS layer that was not covered by the resist. The remaining resist was subsequently removed by 0.26N TMAH. As a result, 8 nm-thick xPS gratings with 118 - 120 nm pitch were generated on top of the SiN/ α -C stack. On the pattern of the xPS gratings, a

SMMA brush material (AZ Electronic Materials, AZEMBLY™ NLD-127) was coated and baked at 250°C / 2 min followed by a PGMEA rinse, resulting in formation of alternate stripes of xPS and the brush material. The brush in this case selectively grafted on the open SiN surface so it did not interfere with the top and sidewall surfaces of the xPS gratings. In this prepattern, the xPS features serve as the wetting region and the exposed SMMA brush layer serves as the neutral region. The typical height of xPS / SMMA features is less than 0.5 Lo of JBCP in the experiments and the pitch of xPS features is around 5Lo, commensurable to the pitches of BCPs. **xPS/SMMA chemical prepattern (C):** On a clean Si wafer crosslinkable polystyrene (xPS) solution (AZEMBLY™ NLD-128) spin-coated at 1500 rpm and baked at 315°C / 5 min under a N₂ environment to yield a 8 nm thick xPS film. On top of the xPS surface, JSR EUV photoresist (VEUVJ2107JE) was then coated at 1500 rpm to form a 40 nm thick resist film. The resist film was exposed with an EUV lithography tool (SEMATECH Berkeley MET) and the exposed regions were developed with 0.26N TMAH to generate a grating prepattern with a pitch of 44 nm. The prepattern was then etched with O₂ RIE to remove the xPS layer that was not covered by the resist gratings. The remaining resist was removed by immersing the etched prepattern to a standard TMAH solution, generating gratings of xPS. A solution of SMMA brush material (AZ Electronic Materials, AZEMBLY™ NLD-127) was subsequently coated at 1500 rpm on the prepattern and baked at 250°C for 2 min, followed by a PGMEA rinse, resulting in formation of alternate stripes of xPS and the brush material. The brush in this case selectively grafted on the open SiN surface so it did not interfere with the top and sidewall surfaces of the xPS gratings. In this prepattern, the xPS features serve as the wetting region and the exposed SMMA brush layer serves as the neutral region. The typical height of xPS / SMMA features is between 2 nm - 3 nm which is less than 0.5 Lo of JBCP in the experiments and the pitch of xPS features is around 3Lo, commensurable to the pitch of JBCP.

Processing of directed self-assembly

Example 126a: Example 21 over HSQ/SMMA chemical prepattern (for 5X frequency multiplication)

A 1 wt% PGMEA solution of **Example 21** is coated at 1500 rpm over the HSQ/SMMA chemical prepattern and annealed at 170°C for 10 min to generate aligned PS and PTMC domains. DSA of lamellae-forming JBCPs with a pitch of L_o were carried out by coating and annealing a layer of JBCP over the chemical prepatterns comprising HSQ lines over SMMA brush layer. Typical thickness of the JBCP layer is around $1 L_o$ to $1.5 L_o$. The PTMC domains of JBCP wets the HSQ lines and forms laterally aligned lamellae pattern over the chemical prepattern. **Figure 11a** is the SEM plane view of aligned PS and PTMC domains from **Example 21** to the HSQ lines from the HSQ/SMMA prepattern with a pitch of ~ 85 nm ($\sim 5L_o$) for 5X frequency multiplication.

Example 126b: Example 21 over HSQ/SMMA chemical prepattern (for 6X frequency multiplication)

Along the same process conditions as **Example 126a**, a 1 wt% PGMEA solution of **Example 21** was coated over the HSQ/SMMA chemical prepattern and annealed. **Figure 11b** is the SEM plane view of aligned PS and PTMC domains from **Example 21** to the HSQ lines from the HSQ/SMMA prepattern with a pitch of ~ 100 nm ($\sim 6L_o$) for 6X frequency multiplication.

Example 127a: Example 31 over xPS/PMMA chemical prepattern (for 3X frequency multiplication)

A 1 wt% PGMEA solution of **Example 31** was coated at 2000 rpm over the xPS/PMMA chemical prepattern (**C**) and annealed at 190°C for 5 min followed by PGMEA rinse and recoating of the solution at 3000 rpm. The film was then annealed at 190°C for 5 min to generate aligned PS and PLA domains. **Figure 11c** is the SEM plane view of aligned PS and PLA domains from **Example 31** to the xPS lines from the xPS/PMMA prepattern with a pitch of ~ 52 nm ($\sim 3L_o$) for 3X frequency multiplication.

Example 127b: Example 31 over xPS/PMMA chemical prepattern (for 5X frequency multiplication)

Along the same process conditions as **Example 127a**, a 1 wt% PGMEA solution of **Example 31** was coated at 2000 rpm over the xPS/PMMA chemical prepattern (**A**), annealed at 190°C for 5 min followed by PGMEA rinse followed by recoating of the solution. The film was then annealed again at 190°C for 5 min to generate aligned PS and PLA domains. **Figure 11d** is the SEM plane view of aligned PS and PLA domains from **Example 31** to the xPS lines from the xPS/PMMA prepattern with a pitch of ~ 85 nm (~5Lo) for 5X frequency multiplication.

Example 128: Example 46 over xPS/SMMA chemical prepattern

A 1.2 % w/w filtered solution of **Example 46** was coated over the xPS/SMMA chemical prepattern (**C**) at 3000 rpm and the sample was baked at 200°C for 5 min and cooled down to room temperature. **Figure 11e** showed that PS-b-PLA formed vertically oriented lamellae that aligned parallel to the prepattern direction, creating a pitch multiplication of 3X.

Example 129: Example 47 over xPS/SMMA chemical prepattern

A 1.2% w/w solution of **Example 47** was coated over the xPS/SMMA chemical prepattern (**C**) at 3000 rpm and the sample was baked at 200°C for 5 min. **Figure 11f** showed that the PS-b-PMeCAR formed vertically oriented lamellae that aligned parallel to the prepattern direction, creating a pitch multiplication of 3X.

Example 130a: Example 53 over xPS/SMMA chemical prepattern

A 1 % w/w PGMEA solution of **Example 53** was coated over the xPS/SMMA chemical prepattern (**B**) at 3000 rpm and baked at 250°C for 5 min under N₂. After PMMA removal and pattern transfer into substrate by dry etching, LER/LWR/SWR was measured by SEM.

Example 130b: Example C6 over xPS/SMMA chemical prepattern

By the same process conditions, A thin film of **Example C6** was prepared over the xPS/SMMA chemical prepattern (**B**) and treated by the same dry etch technique for the measurement of LER/LWR/SWR. The performance of directed self-assembly of **Example 53** and **Example C6** were summarized in Table 21.

Table 21. Comparison of LER/LWR/SWR between Example C6 and Example 53

	Example C6		Example 53	
Measurement mode	Normal	Compressive	Normal	Compressive
LER L & R	L: 3.22nm R: 3.22nm	L: 3.87nm R: 3.84nm	L: 2.67nm R: 2.64nm	L: 2.9nm R: 2.98nm
LWR	2.55nm	2.18nm	2.0nm	1.82nm
SWR	2.79nm	2.40nm	2.38nm	2.24nm

Materials used to prepare the prepatterns tested are summarized below in Table 22.

Table 22. Materials used for prepatterns

Materials	Use	Details.
G1-2	Underlayer for DSA	Example 57
PR 3538	Guiding topography	JSR ArF resist
PR 7210	Guiding topography	JSR ArF resist
PR 7946	Guiding topography	JSR ArF resist
HSQ	Chemical prepattern	Dow Corning XR-1541 resist
SiARC	Substrate for graphoepitaxy	Shin-Etsu SHB-A940 L35
NLD Series, N3	Underlayer for DSA	Underlayers of Examples 82-90 and Examples 67-70
NLD-128	x-linkable polystyrene for DSA	Random copolymer of styrene and VBCB

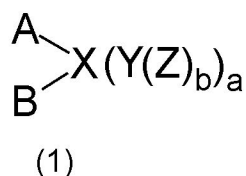
Block copolymers tested for directed self-assembly are summarized below in Table 23.

Table 23. Block Copolymers used for Directed Self-Assembly

BCP		Morphology	Pitch
Example 21	PS-CH ₂ C ₈ F ₁₇ -PTMC	Lamellae	17 nm
Example 22	PS-CH ₂ C ₈ F ₁₇ -PTMC	Lamellae	15.5 nm
Example 20	PS-CH ₂ C ₈ F ₁₇ -PTMC	Cylinders	9.5 nm
Example 24	PS-CH ₂ C ₈ F ₁₇ -PTMC	Cylinders	16 nm
Example 31	PS-CH ₂ C ₈ F ₁₇ -PLA	Lamellae	17.5 nm
Example 46	PS-CH ₂ C ₈ F ₁₇ -PLA	Lamellae	15.7 nm
Example 31	PS-CH ₂ C ₈ F ₁₇ -PLA	Lamellae	17 nm
Example 47	PS-CH ₂ C ₈ F ₁₇ -PMeCAR	Lamellae	15.6 nm
Example 53	PS-DPE-CH ₂ C ₈ F ₁₇ -PMMA	Lamellae	24 nm
Example C6	PS-DPE-PMMA	Lamellae	24 nm

Claims

1. A block copolymer with a junction group having structure (1),



wherein,

A is a block polymer chain comprising units derived from ethylenically unsaturated polymerizable monomers, or units derived from ring-opening polymerization (ROP) of cyclic monomers,

B is a block polymer chain comprising units derived from ethylenically unsaturated polymerizable monomers, or units derived from ring-opening polymerization (ROP) of cyclic monomers,

wherein A and B are chemically different covalently connected polymer chains which are phase separable;

moiety $\text{X}(\text{Y}(\text{Z})_b)_a$ is a junction group which comprises a surface active pendant moiety $\text{Y}(\text{Z})_b$, wherein:

a is an integer from 1 to 4, b is an integer from 1 to 5;

X is a linking group between the A polymer block, the B polymer block and the moiety Y, wherein X is selected from the group consisting of a C₅-C₃₀ oxyalkyleneoxycarbonylalkylene moiety, a C₂-C₃₀ alkyleneoxyalkylene moiety, an aryl substituted C₈-C₄₂ alkyleneoxy moiety, an aryl substituted C₈-C₄₂ alkyleneoxyalkylene moiety, a C₈-C₄₂ aryl substituted alkylene moiety, or a C₇-C₃₀ alkylenearylene moiety;

Y is a linking group or a direct valence bond between X and Z, and Y is selected from the group consisting of a direct valence bond, oxy (-O-), carbonyl (-C(=O)-), carbonyloxy (-C(=O)-O-), oxycarbonyl (-O-C(=O)-), carbonate (-O-C(=O)-O-), carbonyloxyalkylene (-C(=O)-O-alkylene-), a C₁-C₂₀ carbonyloxyfluoroalkylene (-C(=O)-O-fluoroalkylene-), alkylene-fluoroalkylene, a C₆-C₂₀ arylene, a C₆-C₂₀

oxyarylene, a C₇-C₂₀ alkylenearylene, a C₁-C₂₀, linear alkylene, a C₃-C₂₀ branched alkylene, a C₆-C₂₀ cyclic alkylene, a C₁-C₂₀, linear fluoroalkylene, a C₃-C₂₀ branched fluoroalkylene, and a C₆-C₂₀ cyclic fluoroalkylene; and

Z is selected from the group consisting of a fluorine group, a linear hydrocarbon group containing fluorine, a fluorine containing branched hydrocarbon group, a fluorine containing cyclic hydrocarbon group, a fluorine containing linear alkyl ether group, a fluorine containing branched alkyl ether group, a fluorine containing cyclic alkyl ether group or a mixture thereof,

and further wherein the junction group X(Y(Z)_b)_a has a surface energy less than that of block A and less than that of the block B.

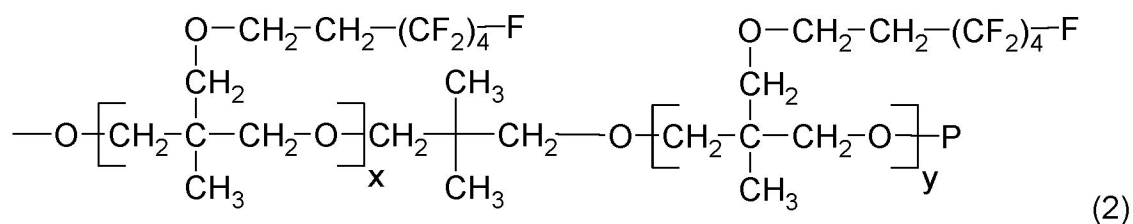
2. The block copolymer of claim 1 wherein,

(i) the polymer chain of block A comprises units derived from ethylenically unsaturated polymerizable monomers, or units derived from ring-opening polymerization (ROP) of cyclic monomers, and

(ii) the polymer chain in block B comprises units derived from ethylenically unsaturated polymerizable monomers, or units derived from ring-opening polymerization (ROP) of cyclic monomers, and further wherein polymeric block A and B are chemically different and are phase separable.

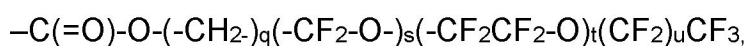
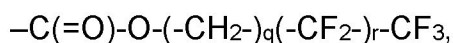
3. The block copolymer of claim 1 or 2 wherein Z, the fluorocarbon containing moiety, is a fluorine containing hydrocarbon group which is chosen from a linear C₁-C₁₀ fluoroalkyl groups, a C₃-C₁₀ branched fluoroalkyl groups, a C₄-C₁₀ cyclic fluoroalkyl groups, a

alkylenefluoroalkyl groups $-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, CFH containing alkylenefluoroalkyl groups $-(\text{CH}_2)_q(\text{CHF})_{q2}(\text{CF}_2)_r\text{CF}_3$, alkylenefluoroether-fluoroalkyl groups $-(\text{CH}_2)_q(-\text{CF}_2\text{O})_s(-\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$, - CHF containing alkylenefluoroether-fluoroalkyl groups $(-\text{CH}_2)_q(\text{CHF})_{q2}(-\text{CF}_2\text{O})_s(-\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$, or a moiety having structure (2)

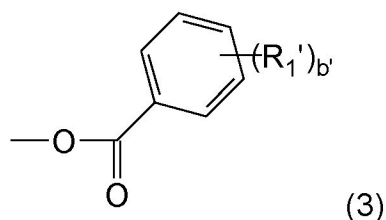


wherein q is an integer from 0 to 10, q2 is an integer from 1 to 10, r is an integer from 1 to 10, s is an integer from 0-10, t is an integer from 1 to 10, u is an integer between 0 and 10, x is 1 to 5, y is 1 to 5, P is hydrogen, a C₁ to C₄ alkyl, or a -CH₂-CH₂-(CF₂)₄F moiety.

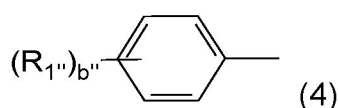
4. The block copolymer of any of claims 1 to 3 wherein the surface active pendant moiety -Y(Z)_b is chosen from the group consisting of:



a moiety having structure (3)



and a moiety having structure (4)

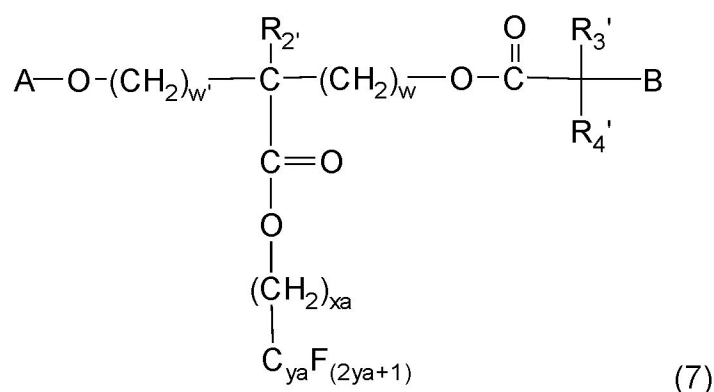


wherein -R₁' and -R₁'' are independently selected from $-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-)_{\text{r}}-\text{CF}_3$, $-\text{O}-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-)_{\text{r}}-\text{CF}_3$, $-\text{C}(=\text{O})-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-)_{\text{r}}-\text{CF}_3$, $-\text{O}-\text{C}(=\text{O})-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-)_{\text{r}}-\text{CF}_3$,

$-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-\text{O}-)_{\text{s}}(-\text{CF}_2\text{CF}_2-\text{O}-)_{\text{t}}(\text{CF}_2)_{\text{u}}\text{CF}_3$, $-\text{O}-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-\text{O}-)_{\text{s}}(-\text{CF}_2\text{CF}_2-\text{O}-)_{\text{t}}(\text{CF}_2)_{\text{u}}\text{CF}_3$, $-\text{O}-\text{C}(=\text{O})-\text{O}-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-\text{O}-)_{\text{s}}(-\text{CF}_2\text{CF}_2-\text{O}-)_{\text{t}}(\text{CF}_2)_{\text{u}}\text{CF}_3$ and

$-\text{C}(=\text{O})-\text{O}-(-\text{CH}_2-)_{\text{q}}(-\text{CF}_2-\text{O}-)_{\text{s}}(-\text{CF}_2\text{CF}_2-\text{O}-)_{\text{t}}(\text{CF}_2)_{\text{u}}\text{CF}_3$ and further wherein b' and b'' are integers from 1 to 5, q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10, u is an integer between 0 and 10.

5. The block copolymer of any of claims 1 to 4 where the block copolymer with a junction group has structure (7)



wherein,

$\text{R}_{2'}$ is hydrogen, a C_1 to C_8 alkyl group, or a C_6 - C_{20} aryl group,

$\text{R}_{3'}$ and $\text{R}_{4'}$ are independently a C_1 to C_8 alkyl group, hydrogen, bromine, chlorine, or iodine,

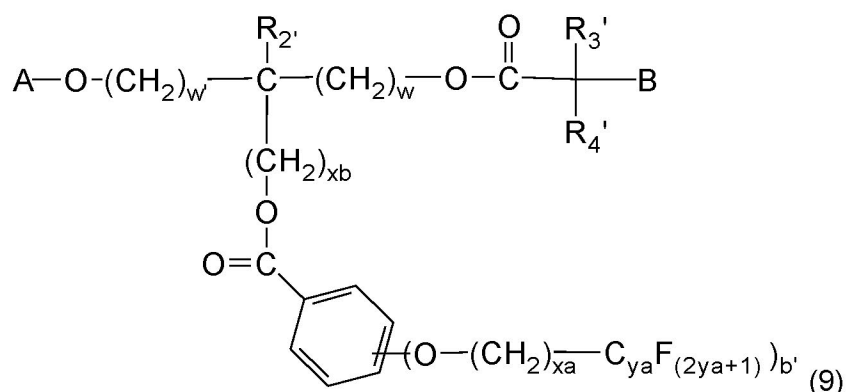
xa is an integer from 1 to 6,

ya is an integer from 2-16, w and w' are independently integers from 1 to 4,

block A is derived from a lactone, epoxide or cyclic carbonate and

block B is derived from a vinylaryl monomer where the aryl group is substituted with a silicon containing moiety selected from the group consisting of a tri(C_1 - C_8 alkyl)silyl group, a tri(C_1 - C_8 alkyl)silyl(C_1 - C_8 alkylene) group, and a tri(C_1 - C_8 alkyl)silyl group(C_2 - C_8 oxyalkylene) group and further wherein block A and B are phase separable, or

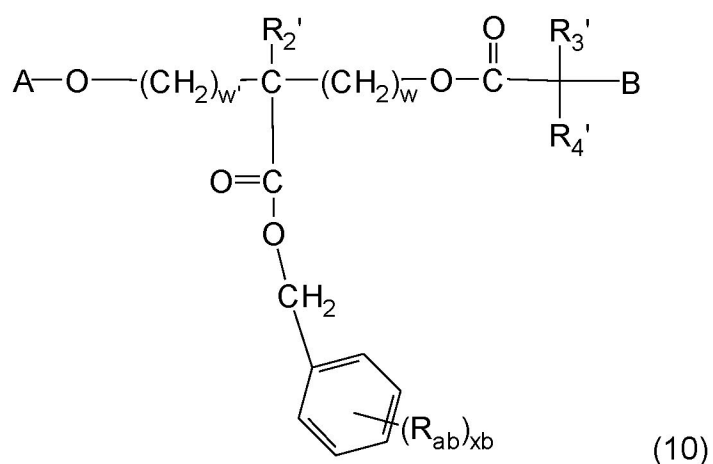
the block copolymer with a junction group has structure (9),



wherein, R_2' is hydrogen, a C_1 to C_8 alkyl group, or a C_6 - C_{20} aryl group, R_3' and R_4' are independently a C_1 to C_8 alkyl group, hydrogen, bromine, chlorine, or iodine, x_a and x_b are independently integers from 1 to 6, y_a is an integer from 2-16, w and w' are independently integers from 1 to 4, and b' is an integer from 1 to 5,

block A is derived from a lactone, epoxide or cyclic carbonate and block B is derived from a vinylaryl monomer where the aryl group is substituted with a silicon containing moiety selected from the group consisting of a tri(C_1 - C_8 alkyl)silyl group, a tri(C_1 - C_8 alkyl)silyl(C_1 - C_8 alkylene) group, and a tri(C_1 - C_8 alkyl)silyl group(C_2 - C_8 oxyalkylene) group and further wherein block A and B are phase separable, or

the block copolymer with a junction group has structure (10),

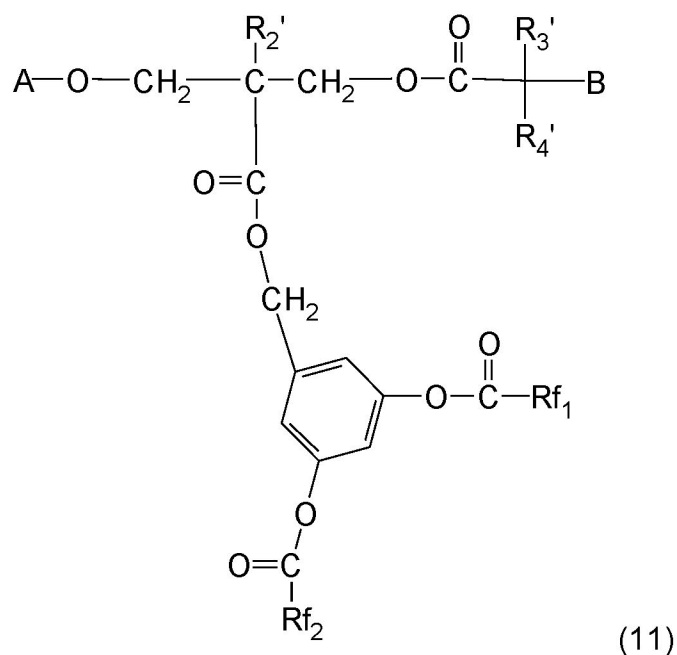


wherein R_2' is selected from a C_1 - C_4 alkyl, and R_3' and R_4' are independently a C_1 to C_8 alkyl group, and R_{ab} is selected from the group consisting of

$-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, $-\text{O}-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, $-\text{C}(=\text{O})-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, $-\text{O}-\text{C}(=\text{O})-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$,

$-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$, $-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$, $-\text{O}-\text{C}(=\text{O})-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$ and

$-\text{C}(=\text{O})-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$ and further wherein q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10, u is an integer between 0 and 10 and xb, w and w' are independently integers from 1 to 5 and further where block B is derived from a vinyl aryl and block A is derived from a lactone, lactam, epoxide or cyclic carbonate and further wherein block A and B are phase separable, or the block copolymer with a junction group has structure (11),



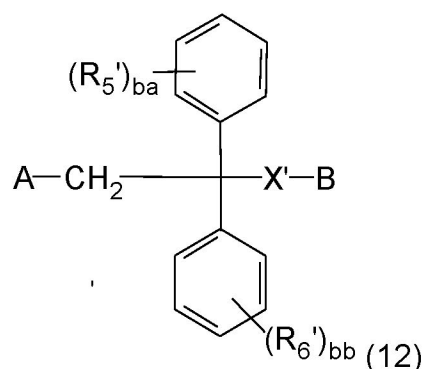
wherein R_2' is selected from a C_1 - C_4 alkyl, and R_3' and R_4' are independently a C_1 to C_8 alkyl group, and Rf_1 and Rf_2 are independently selected from

$-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, $-\text{O}-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, or

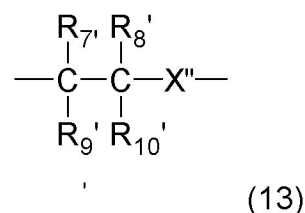
$-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$, and further wherein q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an

integer from 1 to 10, u is an integer between 0 and 10, and further wherein the block B is derived from a vinyl aryl and block A is derived from a lactone, lactam, epoxide or cyclic carbonate and further wherein block A and B are chemically distinct and are phase separable,

the block copolymer with a junction group has structure (12),



wherein X' is a direct valence bond or a moiety (13)



where R_{5'} is $-(CH_2)_q(CF_2)_rCF_3$, $-O-(CH_2)_q(CF_2)_rCF_3$, $-C(=O)-(CH_2)_q(CF_2)_rCF_3$, $-O-C(=O)-(CH_2)_q(CF_2)_rCF_3$,

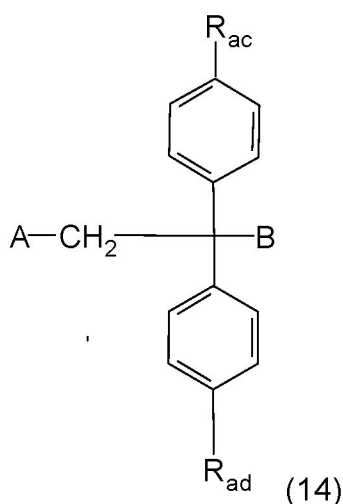
$-(CH_2)_q(CF_2O)_s(CF_2CF_2O)_t(CF_2)_uCF_3$, $-O-(CH_2)_q(CF_2O)_s(CF_2CF_2O)_t(CF_2)_uCF_3$, $-O-C(=O)-O-(CH_2)_q(CF_2O)_s(CF_2CF_2O)_t(CF_2)_uCF_3$ or

$-C(=O)-O-(CH_2)_q(CF_2O)_s(CF_2CF_2O)_t(CF_2)_uCF_3$, and R_{6'} is $-(CH_2)_q(CF_2)_rCF_3$, $-O-(CH_2)_q(CF_2)_rCF_3$, $-C(=O)-(CH_2)_q(CF_2)_rCF_3$, $-O-C(=O)-(CH_2)_q(CF_2)_rCF_3$,

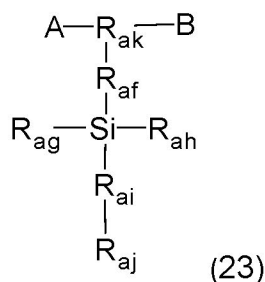
$-(CH_2)_q(CF_2O)_s(CF_2CF_2O)_t(CF_2)_uCF_3$, $-O-(CH_2)_q(CF_2O)_s(CF_2CF_2O)_t(CF_2)_uCF_3$, $-O-C(=O)-O-(CH_2)_q(CF_2O)_s(CF_2CF_2O)_t(CF_2)_uCF_3$,

$-\text{C}(=\text{O})-\text{O}-(-\text{CH}_2)_q(-\text{CF}_2-\text{O})_s(-\text{CF}_2\text{CF}_2-\text{O})_t(\text{CF}_2)_u\text{CF}_3$ or hydrogen, R_7' and R_9' are independently hydrogen, and a $\text{C}_1\text{-C}_4$ alkyl, R_8' and R_{10}' are independently selected from hydrogen, a $\text{C}_1\text{-C}_4$ alkyl, or a $\text{C}_2\text{-C}_6$ alkyleneoxyalkyl moiety, X'' is $-\text{O}-$ or $-\text{S}-$, and wherein b_a and b_b are integers from 1 to 5, q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10, u is an integer between 0 and 10.

and b' and b'' are independently integers from 1 to 5 and further wherein block A is derived vinyl aryl, and block B is derived from a lactone, lactam, epoxide or cyclic carbonate or when X' is a direct valence bond an ethylenically unsaturated monomer, and further wherein block A and B are phase separable, or the block copolymer with a junction group has structure (14)



where R_{ac}' is $-\text{O}-(-\text{CH}_2)_q(-\text{CF}_2)_r-\text{CF}_3$, and R_{ad} is $-(\text{CH}_2)_q(-\text{CF}_2)_r-\text{CF}_3$, or hydrogen, and further wherein q is an integer from 0 to 10, and r is an integer from 1 to 10 and further wherein block A is derived from a vinyl aryl, and block B is derived from a lactone, lactam, epoxide, cyclic carbonate or ethylenically unsaturated monomer and further wherein block A and B are phase separable, or the block copolymer with a junction group has structure (23)



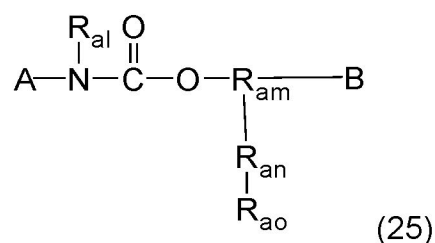
where R_{af} , R_{ak} and R_{ai} are independently a C_1 - C_{10} linear alkylene moiety, a C_3 - C_{10} branched alkylene moiety, a C_3 - C_{10} cyclic alkylene moiety, R_{ag} and R_{ah} are independently selected from a C_1 to C_4 alkyl moiety, and R_{aj} is $-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, $-\text{O}-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$,

$-\text{C}(=\text{O})-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$, $-\text{O}-\text{C}(=\text{O})-(\text{CH}_2)_q(\text{CF}_2)_r\text{CF}_3$,

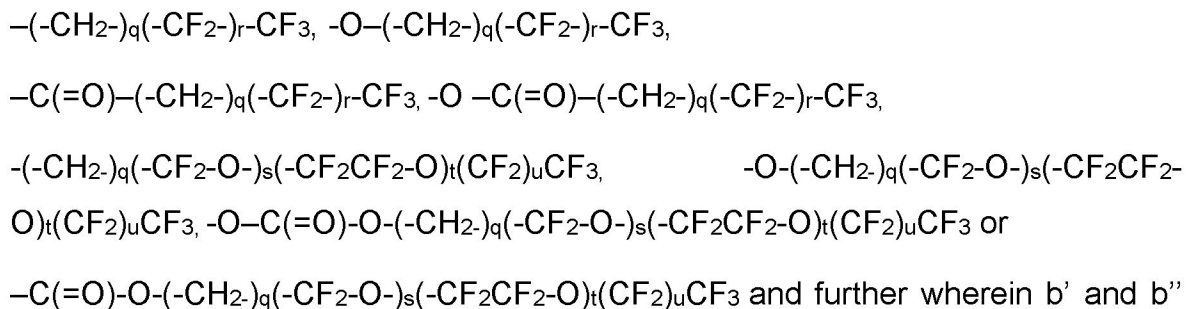
$-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$, $-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$, $-\text{O}-\text{C}(=\text{O})-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$ or

$-\text{C}(=\text{O})-\text{O}-(\text{CH}_2)_q(\text{CF}_2\text{O})_s(\text{CF}_2\text{CF}_2\text{O})_t(\text{CF}_2)_u\text{CF}_3$ and further wherein b' and b'' are integers from 1 to 5, q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10, u is an integer between 0 and 10, and further wherein the block A is derived from a vinyl aryl and block B is derived from a lactone, lactam, epoxide, cyclic carbonate or an ethynically unsaturated monomer which is not the same as block A is derived and further wherein block A and B are chemically distinct and are phase separable.

6. The block copolymer of any of claims 1 to 4, wherein the block copolymer with a junction group has a structure (25)

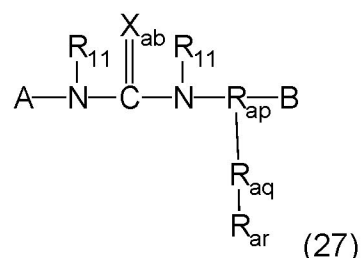


where R_{al} is hydrogen or a C_1 - C_4 alkyl, R_{am} and R_{an} are independently a C_1 - C_{20} linear alkylene moiety, a C_3 - C_{30} branched alkylene moiety, or a C_3 - C_{30} cyclic alkylene moiety, and R_{ao} is

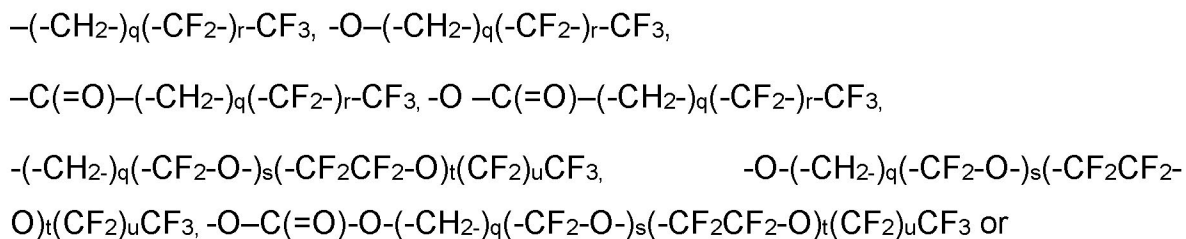


are integers from 1 to 5, q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10, u is an integer between 0 and 10, and further wherein block A is derived from a vinyl aryl and chain B is derived from a lactone, lactam, epoxide, cyclic carbonate or an ethynically unsaturated monomer which is not the same as block A is derived and further wherein block A and B are chemically distinct and are phase separable, or

The block copolymer with a junction group has structure (27)

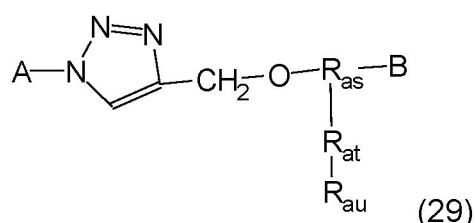


where R_{11} is hydrogen or a C_1 - C_4 alkyl, R_{ap} and R_{aq} are independently a C_1 - C_{20} linear alkylene moiety, a C_3 - C_{30} branched alkylene moiety, or a C_3 - C_{30} cyclic alkylene moiety, X_{ab} is O or S, and R_{ar} is



$-\text{C}(=\text{O})-\text{O}-(-\text{CH}_2-)_{\text{q}}(-\text{CF}_2-\text{O}-)_{\text{s}}(-\text{CF}_2\text{CF}_2-\text{O}-)_{\text{t}}(\text{CF}_2)_{\text{u}}\text{CF}_3$ and further wherein b' and b'' are integers from 1 to 5, q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10, u is an integer between 0 and 10, and where block A is derived from a vinyl aryl, and block B is derived from a lactone, lactam, epoxide, cyclic carbonate or an ethynically unsaturated monomer which is not the same as the vinyl aryl from which block A is derived and further wherein block A and B are chemically distinct and are phase separable, or

the block copolymer with a junction group has structure (29)



where R_{as} and R_{at} are independently a C_1 - C_{20} linear alkylene moiety, a C_3 - C_{30} branched alkylene moiety, a C_3 - C_{30} cyclic alkylene moiety, and R_{au} is

$-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-)_{\text{r}}\text{CF}_3$, $-\text{O}-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-)_{\text{r}}\text{CF}_3$,

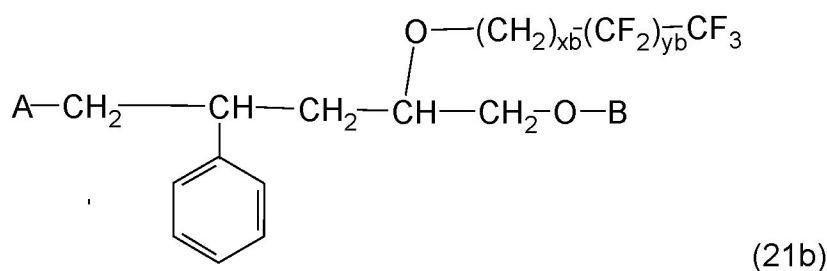
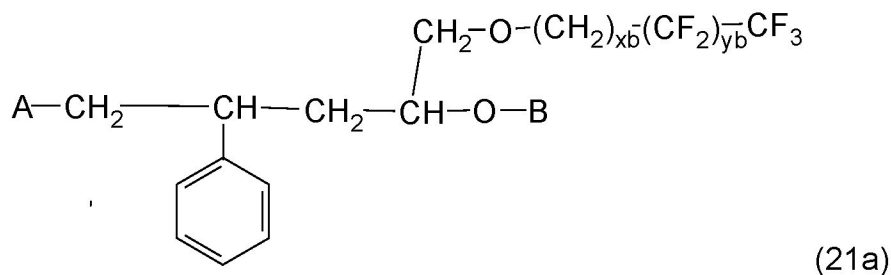
$-\text{C}(=\text{O})-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-)_{\text{r}}\text{CF}_3$, $-\text{O}-\text{C}(=\text{O})-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-)_{\text{r}}\text{CF}_3$,

$-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-\text{O}-)_{\text{s}}(-\text{CF}_2\text{CF}_2-\text{O}-)_{\text{t}}(\text{CF}_2)_{\text{u}}\text{CF}_3$, $-\text{O}-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-\text{O}-)_{\text{s}}(-\text{CF}_2\text{CF}_2-\text{O}-)_{\text{t}}(\text{CF}_2)_{\text{u}}\text{CF}_3$, $-\text{O}-\text{C}(=\text{O})-\text{O}-(\text{CH}_2-)_{\text{q}}(-\text{CF}_2-\text{O}-)_{\text{s}}(-\text{CF}_2\text{CF}_2-\text{O}-)_{\text{t}}(\text{CF}_2)_{\text{u}}\text{CF}_3$ or

$-\text{C}(=\text{O})-\text{O}-(-\text{CH}_2-)_{\text{q}}(-\text{CF}_2-\text{O}-)_{\text{s}}(-\text{CF}_2\text{CF}_2-\text{O}-)_{\text{t}}(\text{CF}_2)_{\text{u}}\text{CF}_3$ and further wherein b' and b'' are integers from 1 to 5, q is an integer from 0 to 10, r is an integer from 1 to 10, s is an integer from 0 to 10, t is an integer from 1 to 10, u is an integer between 0 and 10 and, where block A is derived from a vinyl aryl, and block B is derived from a lactone, lactam, epoxide, cyclic carbonate or an ethynically unsaturated monomer which is not the same as the vinyl aryl from which block A is derived and further wherein block A and B are chemically distinct and are phase separable, or



(1) $\mathcal{C}_1$ is a $\mathcal{C}_2$ -subalgebra of $\mathcal{C}_1$ if and only if $\mathcal{C}_1$ is a $\mathcal{C}_2$ -subalgebra of $\mathcal{C}_1$.



wherein x_b is an integer from 1 to 6 and y_b is an integer from 4 to 20 and further where block A is derived from a vinyl aryl, and block B is derived from a lactone, lactam, epoxide or cyclic carbonate and further wherein block A and B are chemically distinct and are phase separable.

8. The block copolymer of any of claims 1 wherein the junction moiety $X(Y(Z)_b)_a$ has a surface energy less than about 30 mN m^{-1} .
9. A composition comprising the block copolymer of any of claims 1 to 8 and a solvent.
10. The composition of claim 9 further comprising another block copolymer.
11. The composition of claim 9 or 10 further comprising a homopolymer.
12. A method of vertically orienting first and second block copolymer domains over an unpatterned substrate using a layer of a block copolymer having a periodicity of L_o comprising the steps of:

- a) forming a coating layer of a block copolymer with a junction group with the composition of any of claims 9 to 11 on said unpatterned substrate; and,
- b) annealing the layer of the block copolymer to generate a non-zero positive integer number of first and second block copolymer domains, vertically oriented on said unpatterned substrate.

13. A method of vertically orienting first and second block copolymer domains over a first patterned substrate where the height of topography of the pattern on the substrate is at least 0.7 times L_0 and aligning the domains with the pattern, using a coating comprised of a block copolymer having a periodicity of L_0 comprising the steps of:

- a) forming a coating layer of the block copolymer with a junction group with the composition of any of claims 9 to 11 on said first topographical substrate, wherein the thickness of the average thickness of the coating layer of the block copolymer is less than the height of the topography of the first topographical substrate, wherein the block copolymer layer is laterally confined by the topography; and,
- b) annealing the block copolymer layer to generate first and second block copolymer domains, vertically oriented on said first patterned substrate, and confined within the recessed region.

14. A method of vertically orienting, first and second block copolymer domains with a periodicity of L_0 over a second patterned substrate having a topographical pattern with the height of topography larger than 0.7 times L_0 and a pitch P_1 where the pitch P_1 is a non-zero positive integer multiplied by L_0 , and aligning the domains with the pattern comprising the steps of:

- a) forming a coating layer of the block copolymer with a surface-active junction group with the composition of any of claims 9 to 11 on said second patterned

substrate, where the thickness of the coating layer of the block copolymer is more than the height of the topography of the second patterned substrate; and,

b) annealing the block copolymer layer to generate a non-zero positive integer number of first and second block copolymer domains vertically oriented on said second patterned substrate and aligning them to the second patterned substrate where the sum of vertically oriented domains is equal or larger than the pitch P_1 the of topographical pattern.

15. A method of vertically orienting first and second block copolymer domains over a substrate having a surface chemical prepattern having a pitch P_2 , where the pitch P_2 is a non-zero positive integer multiplied by L_0 and aligning the domains comprising the steps of:

a) forming a coating layer of the block copolymer with a surface-active junction group with the composition of any of claims 9 to 11 on the substrate having a surface chemical prepattern; and,

b) annealing the block copolymer layer to generate vertically oriented first and second block copolymer domains aligned with the substrate having a surface chemical prepattern having a pitch P_2 .