

## **HIGH MODULUS METALLIC COMPONENT FOR HIGH VIBRATORY OPERATION**

### **5 BACKGROUND OF THE INVENTION**

This invention relates to a metallic component such as an aircraft engine turbine blade that is formed with a high modulus orientation of a metal being aligned in the radial, i.e. primary, direction of the turbine blade.

10 Machines that utilize high speed components, such as an aircraft engine, produce high frequency vibrations. The high frequency vibrations are typically on the order kilohertz and impose a variety of fluctuating high cycle fatigue stresses on the high speed components of the machine. Often, the limiting factor in the life of a high speed component is high cycle fatigue stress. While the present invention is described in the context of a turbine blade, it will be recognized that the invention is  
15 not so limited.

Conventionally, the life of high speed components, such as a turbine blade employed in aircraft engine, is enhanced by designing the component to resist or minimize the imposed stresses. This involves designing the turbine blade so that the natural vibrational frequencies do not match the vibrational frequencies produced by  
20 the high speeds. A turbine blade designed in this way minimizes the stress amplitude by avoiding a resonant effect that amplifies the stresses. It is not always possible however, to design a blade that has adequately different natural vibrational frequencies from those produced by the high speed motion.

Another turbine blade design approach attempts to dampen the vibrations at  
25 critical locations on the blade. Various damping designs, such as friction damping or the application of damping coatings, are available to help reduce the stress amplitude at the critical locations. Damping is often expensive, involves highly complex analysis and experimentation, and may impair the performance of the turbine blade. Accordingly, a metallic component, such as aircraft engine turbine blade, that  
30 provides enhanced high cycle fatigue life is needed.

## SUMMARY OF THE INVENTION

In general terms, this invention is a directionally solidified metallic component, such as an aircraft engine turbine blade that is formed from a base metal that has a high modulus crystallographic orientation aligned with the radial, i.e. primary, direction of the turbine blade.

In one example, the engine turbine blade is formed from a single crystal of Ni based alloy and the  $\langle 111 \rangle$  crystallographic direction is aligned with the primary direction of the turbine. Alternatively, the engine turbine blade is formed from an alloy of Fe, Ti, Al, Co, Nb, or Mo and a high modulus direction of the alloy is aligned with the primary direction of the turbine blade.

In another example, a high modulus direction of the base metal that forms the engine turbine blade is aligned with the primary direction of a columnar grain structure and the primary direction of the columnar grain structure is aligned with the primary direction of the turbine blade.

In another example, the engine turbine blade is formed from a Ni based alloy and the  $\langle 112 \rangle$  high modulus crystallographic direction is aligned within a cone of about ten degrees of the primary direction of the turbine blade.

In another example, the engine turbine blade is formed from a Ni based alloy and the  $\langle 123 \rangle$  high modulus crystallographic direction is aligned within a cone of about ten degrees of the primary direction of the turbine blade.

In another example, the engine turbine blade is formed from a Ni based alloy and the  $\langle 110 \rangle$  crystallographic direction is aligned to within about ten degrees of the primary direction.

In another example, the Ni base metal that forms the engine turbine blade that has a high modulus direction aligned with the primary direction is a Ni superalloy.

In another example, the engine turbine blade turbine blade is heat treated to recrystallize the base metal with a high modulus direction aligned with the primary direction of the turbine blade.

In another example, the engine turbine blade that has a high modulus direction aligned with the primary direction is in an aircraft engine.

The various features and advantages of this invention will become apparent to those skilled in the art from the following detailed description of the currently preferred embodiment. The drawings that accompany the detailed description can be briefly described as follows.

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### **BRIEF DESCRIPTION OF THE DRAWINGS**

Figure 1 is a cross sectional view of an aircraft engine.

Figure 2 is a schematic cross sectional view of an aircraft engine turbine blade;

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Figure 3 is a sketch of a single crystal unit of a base metal;

Figure 4 is a microscopic sketch of an equiaxed Ni metal portion;

Figure 5 is a microscopic sketch of an anisotropic Ni metal portion;

Figure 6 is a microscopic sketch of a single crystal Ni metal portion.

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### **DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENT**

Figure 1 shows a cross sectional schematic view of an aircraft engine 2. The aircraft engine 2 includes an engine casing 3 that houses a fan 4 and compressor 5. The compressor 5 includes impellers 6 that pressurize air in the aircraft engine 2. The impellers 6 are attached to a rotatable shaft 7 that rotates around axis 8. When the shaft 7 rotates, the impellers 6 rotate. A combustor 9 is in fluid communication with pressurized gas that exits from the compressor 5. The combustor 9 combusts the pressurized gas. A turbine 10 receives the combusted pressurized gas and converts it into energy that is used to rotate the shaft 7 and power the compressor 5. The turbine 10 includes a rotor 11 that is attached to the shaft 7, and turbine blades 13 that are attached to the rotor 11.

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Inside the aircraft engine 2, the turbine blades 13 operate at high rotational speeds as the combusted pressurized gas from the combustor 9 expands. The high speeds cause vibrations in the aircraft engine 2 and impose high frequency fatigue stresses on the turbine blades 13, i.e. high cycle fatigue.

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Referring to Figure 2, the turbine blade 13 has base portion 14 and tip portion 15. The turbine blade 13 further includes an associated primary direction 16 that extends from the base portion 14 to the tip portion 15 and a substantially

perpendicular axial direction 18. The turbine blade 13 has a dimension  $L_1$  in the primary direction and a dimension  $W_1$  in the axial direction. The dimension  $L_1$  is greater than the dimension  $W_1$ . That is, the turbine blade 13 has a  $L_1$  to  $W_1$  aspect ratio greater than one. The primary direction 16 is defined as the direction of the greater dimension  $L_1$  in the turbine blade 13, or the greater dimension of any component having an aspect ratio that is greater than one.

The axial direction 18 is approximately the same direction of the axis 11 (referring to Figure 1) around which the turbine blade 13 rotates. The primary direction 16 is substantially perpendicular (i.e. radial relative to the axis 11 about which the turbine blade 13 rotates - see Figure 1). A significant amount of fatigue stress occurs in the primary direction 16 due to the high speed rotation of the turbine blade 13 around the axis 11.

The turbine blade 13 is formed from a base metal. A base metal is the primary metal of an alloy and may include substantial amounts of alloying elements. All alloys and metals are crystalline and therefore have an associated crystal structure.

The example in Figure 3 refers to a sketch of a single crystal unit of the crystal structure of a base metal used, for example, to form the aircraft engine turbine blade 13. The single crystal unit 19 has known crystallographic directions, for example the  $\langle 100 \rangle$  direction represented by the line 20,  $\langle 110 \rangle$  represented by the line 21,  $\langle 111 \rangle$  represented by the line 22,  $\langle 110 \rangle$  represented by the line 24, and  $\langle 123 \rangle$  represented by the line 26. For engineering purposes, the crystallographic direction refers to the approximate coordinate direction within about a ten degree cone angle 28 of the exact direction.

Each crystallographic direction also has an associated elastic modulus and, if at least one of the crystallographic directions has an elastic modulus that is not equal to the elastic moduli in the other crystallographic directions, the single crystal unit 19 is anisotropic with respect to elastic modulus.

In one example, a Ni based alloy is the base metal. At room temperature, in the  $\langle 100 \rangle$  crystallographic direction the Ni based alloy has an elastic modulus of about 20 Mpsi, in the  $\langle 110 \rangle$  crystallographic direction an elastic modulus of about 34 Mpsi, and in the  $\langle 111 \rangle$  crystallographic direction an elastic modulus of about 44

Mpsi. The  $\langle 100 \rangle$  is a low modulus direction because it has a lower modulus than another direction (here either the  $\langle 110 \rangle$  or  $\langle 111 \rangle$  directions) and the  $\langle 111 \rangle$  is a high modulus direction because it has a higher modulus than at least one other direction ( $\langle 110 \rangle$  or  $\langle 100 \rangle$ ). For the single crystal unit 19, the Ni based alloy is anisotropic.

Generally, despite anisotropy, an article or component formed from an anisotropic base metal will not always exhibit anisotropic properties such as for elastic modulus.

For example, Figure 4 shows a microscopic sketch of a known equiaxed Ni based alloy portion 30 of an article that was solidified in an uncontrolled manner during forming of the article, e.g. without a heat gradient. The equiaxed Ni based alloy portion 30 is comprised of grains 32. Each grain 32 has a crystallographic orientation 34 corresponding to a crystallographic direction such as referred to in Figure 1 for example. The crystallographic orientations 34 of the grains 32 are randomly oriented, i.e. equiaxed. Therefore, the elastic modulus and other properties of the equiaxed Ni based alloy portion 30 are the same in all directions.

Figure 5, however, shows an anisotropic Ni based alloy portion 36 of an article or component that was solidified in a controlled manner during forming, e.g. with a controlled heat gradient. The anisotropic Ni based alloy portion 36 is comprised of a columnar grain 38 structure having a primary direction 40 and a transverse direction 42. Each columnar grain 38 has a crystallographic orientation 44 that is aligned in the primary direction 40. The elastic modulus and other properties of the anisotropic Ni based alloy portion 36 are therefore different in the primary direction 40 than in the transverse direction 42.

As shown in Figure 6, a single crystal Ni based alloy portion 46 also exhibits anisotropic properties. The single crystal Ni based alloy portion 46 was solidified in a controlled manner using the known process of seeding, for example. The single crystal Ni based alloy portion 46 contains a single crystal 48 structure having a primary direction 50 and transverse direction 52. The single crystal 48 has a crystallographic orientation 54 that is aligned in the primary direction 50. The properties of the single crystal Ni based alloy portion 48 are therefore different in the primary direction 50 than in the transverse direction 52.

As referred to for Figure 5, the anisotropic Ni based alloy portion 36 was solidified in a controlled manner during forming. For example, the known process of investment casting may yield columnar grain 38 structure because of a cooling gradient during the solidification process. This process results naturally in the columnar grains 38 having a crystallographic orientation 44 in the  $\langle 100 \rangle$  low modulus direction (i.e. the primary direction 46). Since Ni based alloy metal has an elastic modulus of about 20 Mpsi in the  $\langle 100 \rangle$  direction, the anisotropic Ni based alloy portion 36 has an elastic modulus of about 20 Mpsi in the primary direction 40.

As referred to for Figure 6, the single crystal Ni based alloy portion 46 was solidified in a controlled manner during forming. For example, the known process of investment casting using a seed may be used to produce the single grain 48. This process results naturally in the single grain 48 having a crystallographic orientation 54 in the  $\langle 100 \rangle$  low modulus direction (i.e. the primary direction 46). Since Ni based alloy has an elastic modulus of about 20 Mpsi in the  $\langle 100 \rangle$  direction, the single crystal Ni based alloy portion 46 has an elastic modulus of about 20 Mpsi in the primary direction 50.

In one preferred example, a Ni based alloy is used as the base metal forming the turbine blade 13 and has the  $\langle 111 \rangle$  crystallographic direction aligned with the primary direction 16. Ni based alloy is preferred, but alloys of Fe, Co, Mo, Ti, Nb, and Al could alternatively be used. As is known, a high modulus direction for cubic crystal structured metals, such as Ni, is the  $\langle 111 \rangle$  direction, but one skilled in the art would recognize the high modulus directions in base metals having other crystal structures as well as other high modulus directions in cubic crystal structured metals. It should be understood that one of ordinary skill in the art who has the benefit of this disclosure would recognize the applicability of aligning a high modulus direction with a primary direction to articles other than an aircraft turbine blade such as, but not limited to, industrial gas turbines, aircraft compressor blades, and generally any high speed component having an aspect ratio greater than one.

In another example the base metal of the turbine blade 13 has a columnar grain 38 structure. A high modulus direction of the base metal is aligned with the primary direction 40 of the columnar grains 38. The primary direction 40 of the columnar grains 38 is aligned with the primary direction 16 of the turbine blade 13.

In another preferred example, the base metal has a single grain 48 structure. A high modulus direction of the base metal is aligned with the primary direction 50 of the single grain 48. The primary direction 50 of the single grain 48 is aligned with the primary direction 16 of the turbine blade 13.

5 In another example a Ni based alloy forms the aircraft engine turbine blade 13 and the  $\langle 112 \rangle$  high modulus direction is aligned to within about a ten degree cone angle 28 of the primary direction 16.

In another example a Ni based alloy forms the aircraft engine turbine blade 13 and the  $\langle 123 \rangle$  high modulus direction is aligned to within about a ten degree  
10 cone angle 28 of the primary direction 16.

In another example a Ni based alloy forms the aircraft engine turbine blade 13 and the  $\langle 110 \rangle$  high modulus direction is aligned to within about a ten degree cone angle 28 of the primary direction 16.

In another example, the Ni base metal is a known superalloy. The  
15 composition of the superalloy is 1-16% Cr, 0-3% Mo, 3-13% W, 0-8% Re, 0-14% Ta, 3-7% Al, 0-20% Co, 0-0.1% C, 0-0.02% B, 0-0.1% Zr, 0-2% Hf, 0-2% Nb, 0-1% V, 0-2% Ti, 0-10% (Ru+Rh+Pd+Os+Ir+Pt), 0-0.25% Y, and the balance Ni. In this composition 0-10% (Ru+Rh+Pd+Os+Ir+Pt) means a mixture of any or all of the six elements but not exceeding 10%. This composition is known in the aircraft  
20 industry to be adequate for forming turbine blades that have the low modulus  $\langle 100 \rangle$  direction aligned with the primary direction of the blade but not for any high modulus directions such as  $\langle 123 \rangle$ ,  $\langle 112 \rangle$  and  $\langle 111 \rangle$ . One specific superalloy for the high modulus turbine blade 13 is of the composition 5.0% Cr, 10%Co, 2.0% Mo, 6.0% W, 3.1% Re, 5.6% Al, 9.0% Ta, 0.1% Hf, and the balance Ni. Another specific  
25 superalloy for the high modulus turbine blade 13 is of the composition 2.0% Cr, 16.5% Co, 2.0% Mo, 6.0% W, 6.0% Re, 3.0% Ru, 5.65% Al, 0.15% Hf, 0.004% B, 0.05% C, and the balance Ni.

An aircraft engine turbine blade 13 that has a high modulus direction aligned in the primary direction 16 is particularly well suited to lower stresses imposed by  
30 high cycle engine vibration, i.e. high cycle fatigue. In one example, the engine turbine blade 13 is formed with a Ni based alloy and has the  $\langle 111 \rangle$  direction aligned with the primary direction 16. The engine turbine blade 13 has a higher natural

vibration frequency than a turbine blade that has the  $\langle 100 \rangle$  direction aligned with the primary direction 16. This results in a reduction in stress amplitude from high cycle vibrations and thus enhanced high cycle fatigue life.

5 An aircraft engine turbine blade 13 that has a high modulus direction aligned in the primary direction 16 is counter to the current practice in the industry. Current practice utilizes aircraft engine turbines that have the low modulus direction  $\langle 100 \rangle$  aligned with the primary direction 16 because the investment casting forming process naturally produces the  $\langle 100 \rangle$  direction aligned with the primary direction 16. Moreover, the low modulus direction  $\langle 100 \rangle$  was thought to be the best design 10 because it exhibits other favorable properties such as thermal mechanical fatigue resistance.

A known investment casting method of forming can be utilized to produce an aircraft engine turbine blade 13 that has a high modulus direction aligned in the primary direction 16. Generally, investment casting involves pouring a molten metal 15 into a mold and cooling the mold in a controlled manner so that that the molten metal solidifies in a controlled manner. This processing method can be used to align a high modulus direction with the primary direction 16. Those skilled in the art of metal forming would recognize the processing steps required to produce an engine turbine blade according to the invention. That is, a turbine blade having high 20 modulus properties is novel and inventive, however, a worker of ordinary skill in the art would know of the investment casting process to produce it.

A seed may also be used in the investment casting process to produce a component with a single crystal structure rather than a columnar grain structure. For example, a  $\langle 111 \rangle$  oriented Ni seed would be used to induce single crystal growth in 25 the  $\langle 111 \rangle$  direction for investment casting a single crystal Ni base metal turbine blade 13 that has the  $\langle 111 \rangle$  direction aligned with the primary direction 16. It should be understood that, while a turbine having high modulus properties is novel and inventive, the methods of investment casting and seeding are known to those of ordinary skill in the art of metal forming, the details of which are hereby 30 incorporated by reference.

A turbine blade 13 that has a high modulus direction aligned in the primary direction 16 can also be formed by a known machining process. In the machining

process, the aircraft engine turbine blade 13 is machined from a cast ingot. The ingot is cast, for example, with a  $\langle 100 \rangle$  direction. The primary direction 16 of the turbine blade 13 is machined at approximately a fifty-four degree angle to the ingot  $\langle 100 \rangle$  direction. This results in the  $\langle 111 \rangle$  high modulus direction being aligned with the  
5 primary direction 16.

The turbine blade 13 that has a high modulus direction aligned in the primary direction 16 can also be formed by the known process of tilting the component during casting. During solidification, the aircraft engine turbine blade 13 may be tilted at a particular angle relative to the direction of a cooling gradient so that the  
10 high modulus direction is aligned with the primary direction 16.

The turbine blade 13 that has a high modulus direction aligned in the primary direction 16 can also be formed by the known process of recrystallization. Recrystallization involves heat treating a cast or wrought turbine blade to form new grains from the grains that already existed before the heat treatment. The new grains  
15 have a high modulus direction aligned with the primary direction 16.

The invention has been described in an illustrative manner, and it is to be understood that the terminology used is intended to be in the nature of words of description rather than of limitation. Obviously, many modifications and variations of the present invention are possible in light of the above teachings. It is, therefore,  
20 to be understood that within the scope of the appended claims the invention may be practiced otherwise than as specifically described.

## CLAIMS

We claim:

1. A high modulus turbine blade comprising:  
a base portion and a tip portion;  
5 a primary direction that extends from said base portion to said tip portion; and  
said turbine blade being formed of a base metal that has a crystallographic orientation, said crystallographic orientation having a high modulus direction, wherein said high modulus direction is aligned with said primary direction.  
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2. The turbine blade as recited in claim 1, wherein said high modulus direction is aligned to within a cone of about ten degrees of said primary direction.
3. The turbine blade as recited in claim 1, wherein said base metal is a Fe  
15 based alloy.
4. The turbine blade as recited in claim 1, wherein said base metal is a Ni based alloy.
- 20 5. The turbine blade as recited in claim 4, wherein said Ni based alloy is a superalloy composition comprising 1-16% Cr, 0-3% Mo, 3-13% W, 0-8% Re, 0-14% Ta, 3-7% Al, 0-20% Co, 0-0.1% C, 0-0.02% B, 0-0.1% Zr, 0-2% Hf, 0-2% Nb, 0-1% V, 0-2% Ti, 0-10% (Ru+Rh+Pd+Os+Ir+Pt), 0-0.25% Y, and the balance Ni.
- 25 6. The turbine blade as recited in claim 1, wherein said base metal is a Co based alloy.
7. The turbine blade as recited in claim 1, wherein said base metal is a Mo based alloy.  
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8. The turbine blade as recited in claim 1, wherein said base metal is a Nb based alloy.

9. The turbine blade as recited in claim 1, wherein said base metal is an Al based alloy.

5 10. The turbine blade as recited in claim 1, wherein said base metal is a Ti based alloy.

10 11. The turbine blade as recited in claim 1, wherein said turbine blade has been heat treated to recrystallize the base metal with said high modulus direction aligned with said primary direction.

12. An aircraft engine comprising:  
a fan;  
a compressor in at least partial fluid communication with said fan;  
5 a combustor in fluid communication with said compressor portion;  
and  
a turbine in fluid communication with said combustor, said turbine comprising a rotor having at least one high modulus turbine blade that has a primary direction that extends from a base portion to a tip portion of said high modulus turbine  
10 blade, said high modulus turbine blade being formed of a base metal that has a crystallographic orientation, and said crystallographic orientation having a high modulus direction, wherein said high modulus direction is aligned with said primary direction.
13. The aircraft engine as recited in claim 12, wherein said high modulus direction is aligned within about ten degrees of said primary direction.
14. The aircraft engine as recited in claim 12, wherein said base metal is a Fe based alloy.
15. The aircraft engine as recited in claim 12, wherein said base metal is a Ni based alloy.

16. The turbine blade as recited in claim 15, wherein said Ni based alloy is a superalloy composition comprising 1-16% Cr, 0-3% Mo, 3-13% W, 0-8% Re, 0-14% Ta, 3-7% Al, 0-20% Co, 0-0.1% C, 0-0.02% B, 0-0.1% Zr, 0-2% Hf, 0-2% Nb, 0-1% V, 0-2% Ti, 0-10% (Ru+Rh+Pd+Os+Ir+Pt), 0-0.25% Y, and the balance Ni.

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17. The aircraft engine as recited in claim 12, wherein said base metal is a Co based alloy.

18. The aircraft engine as recited in claim 12, wherein said base metal is a  
10 Mo based alloy.

19. The aircraft engine as recited in claim 12, wherein said base metal is a Nb based alloy.

20. The aircraft engine as recited in claim 12, wherein said base metal is an  
15 Al based alloy.

21. The aircraft engine as recited in claim 12, wherein said base metal is a Ti  
based alloy.

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22. The turbine blade as recited in claim 12, wherein said turbine blade has been heat treated to recrystallize the base metal with said high modulus direction aligned with said primary direction.

23. A method of tuning the natural vibration frequency of a turbine blade comprising the step of:

5 increasing the elastic modulus in the primary direction of the turbine blade, wherein the primary direction is a direction that extends from a base portion to a tip portion of the turbine blade.

24. The method as recited in claim 23, further including the step of aligning a high modulus direction of a base metal within a cone of about ten degrees of the  
10 primary direction of the turbine blade.

25. The method as recited in claim 23, wherein the base metal is selected from the group comprising Fe, Ti, Co, Ni, Mo, Nb, and Al based alloys.

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