

TIP CAP HOLE BRAZING AND OXIDATION
RESISTANT ALLOY THEREFOR

BACKGROUND OF THE INVENTION

This invention relates to a brazing method for closing holes and, more particularly, to a brazing method and alloy for closing high pressure turbine blade tip cap holes to maximize filling of the holes, minimize porosity, improve oxidation resistance, and improve reliability.

During the manufacture of high pressure turbine blades of the type as shown in U.S. Pat. 5,232,343, tip cap holes are formed to locate casting cores. It is important to securely close such holes because otherwise cooling air leaks out of the tip rather than being directed over airfoil surfaces. The tip cap holes have a diameter on the order of 0.04 to 0.05 inch (about 0.1 cm to 0.13 cm). Heretofore such holes have been filled by injection with a 50% dense slurry of a fill alloy consisting of a relatively high melting point ("high melt") component (H1) (about 45 wt%) intermixed with a relatively low melting point ("low melt") component (L1) (about 55 wt%). The compositions of high melt and low melt components employed in prior practice are as follows:

<u>High Melt Comp. (H1)</u>		<u>Low Melt Comp. (L1)</u>	
C	0.15-0.19 wt%	C	0.05 max wt%
Cr	13.7-14.3	Cr	14.8-15.8
Co	9.0-10.0	Co	9.5-11.0
Ti	4.8-5.2	Ta	3.0-3.8
Al	2.8-3.2	Al	3.2-3.7
W	3.7-4.3	B	2.1-2.5
Mo	3.7-4.3	Ni	Balance
W+Mo	7.7 min	Plus incidental impurities	
Ni	Balance		
Plus incidental impurities			

During brazing the low melt component particles melt to a liquid and, upon solidification, densify and bond the high melt component particles in the tip cap hole. A disadvantage of this approach has been that the cooled filling has a high internal porosity level which results in susceptibility to oxidation in service, and that a lack of fill condition exists due to a meniscus-type void that has formed upon cooling.

BRIEF SUMMARY OF THE INVENTION

Briefly, in accordance with this invention, a tip cap hole is loaded with a first composition comprising particles of a first alloy having a solidus temperature above the brazing temperature. The first composition is covered with a second composition comprising particles of a second alloy having a liquidus temperature below the brazing temperature. The second composition is heated to the brazing temperature to cause particles of the second alloy to melt to form a liquid of the second alloy which is carried into spaces between the particles of the first alloy by capillarity. The liquid of the second alloy is

cooled to form a solid securely bonding the particles of the first alloy.

According to an aspect of the present invention, there is provided a method for closing a high pressure turbine blade tip cap hole by brazing at a brazing temperature, the method comprising:

loading said tip cap hole with a first composition comprising particles of a first alloy having a solidus temperature above said brazing temperature;

covering said first composition with a second composition comprising particles of a second alloy having 10 a liquidus temperature below said brazing temperature;

heating said second composition to said brazing temperature to cause said particles of said second alloy to melt to form a liquid of said second alloy which is carried into spaces between said particles of said first alloy by capillarity; and

cooling said liquid of said second alloy to form a solid securely bonding said particles of said first alloy.

According to a further aspect of the present invention, there is provided a method for closing a high pressure turbine blade tip cap hole by brazing at a brazing temperature, the method comprising:

filling said tip cap hole with a fill composition comprising particles of at least two distinct alloy compositions comprising particles of a first alloy and particles of a second alloy wherein said first alloy has a 20 solidus temperature above said brazing temperature and said second alloy has a liquidus temperature below said brazing temperature, said particles of said first alloy and said

particles of said second alloy defining two distinct layers in said tip cap hole;

Other features of the present invention will be in part apparent and in part pointed out hereinafter.

BRIEF DESCRIPTION OF THE DRAWINGS

Fig. 1 is a schematic cross section of a tip cap hole illustrating a prior art method for filling tip cap holes.

Fig. 2 is a schematic cross section of a tip cap hole after filling by the prior art method.

Fig. 3 is a schematic cross section of a tip cap hole illustrating the filling method of the invention.

Fig. 4 is a schematic cross section of a tip cap hole after filling by the method of the invention.

Figs. 5 through 10 are 50x magnification 15 photomicrographs of cross sections of samples after an oxidation test.

Figs 11 and 12 are photomicrographs of the fill material of the invention.

DETAILED DESCRIPTION OF THE PREFERRED EMBODIMENT

In accordance with this invention, tip cap holes are brazed closed with a two-component brazing material at a particular brazing temperature at which just one of the two components of the material melts. Tip cap holes are injected at least about 80% full, preferably completely

full, with a first composition which comprises a high melt material having a solidus temperature greater than the brazing temperature used for subsequent brazing. Injection is accomplished using a hollow needle with air pressure behind it or by other suitable means. A relatively equal amount of a second composition comprising low melt material is then applied to the top of the high melt component. The low melt material has a liquidus temperature lower than the brazing temperature. At the brazing temperature, therefore, the low melt material melts while the high melt material does not.

Figure 3 is a cross section of a tip cap hole after filling with powder and before brazing which illustrates the hole filled with high melt component and with the low melt component applied to the top of the high melt component, the two components forming two distinct layers. This is contrasted with the prior art method illustrated in Fig. 1 which shows the high melt component initially intermixed with the low melt component. The high melt and low melt components are shown with different cross-hatching in these figures. Returning to Fig. 3, it can be appreciated that during brazing according to the invention, the low melt component particles melt and the low melt component moves downward through all of the high melt component particles as a result of capillarity and gravity. The result upon solidification is a hole which is substantially 100% filled (Fig. 4) as compared to a hole filled (Fig. 2) under the prior approach. Figure 2 illustrates the detrimental meniscus effect which occurs at the bottom of holes filled by the prior method, which

effect is avoided with the invention as illustrated in Fig. 4. The resulting filled holes are fuller as a result of the new brazing method, and are more resistant to oxidation as a result of the new brazing method in combination with the new alloy.

In one preferred embodiment of the invention which is a variation from the technique described above, the low melt component applied to the top of the filled hole is intermixed with a portion of high melt component. Because the high melt component has greater oxidation resistance than the low melt component, this aspect enhances the oxidation resistance at the top of the filled tip cap hole.

Furthermore, the high melt component particles intermixed with the low melt particles do not melt and do not wick down into the high melt material below. Thus the intermixed high melt particles have a build up effect which minimizes meniscus formation at the top of the filled hole during brazing. Accordingly, the hole is injected completely with high melt component and then a mixture of high melt and low melt component is applied on top of the high melt component. This mixture preferably comprises on the order of up to about 50% by weight of the high melt component, more preferably about 10% to about 50% by weight of the high melt component, still more preferably about 10% to about 25% by weight of the high melt component. In one particularly preferred embodiment, this mixture comprises about 15% of the high melt component and about 85% of the low melt component. The weight ratio of low melt to high melt component in this mixture is between about 1:1 and about 9:1, preferably between about 3:1 and about 9:1. In

one particularly preferred embodiment, this weight ratio is between about 5:1 and about 6:1.

The liquidus temperature of the low melt material is lower than the brazing temperature, and preferably no higher than about 2200°F (about 1200°C), more preferably no higher than about 2175°F (about 1190°C). In one preferred embodiment, the liquidus temperature of the low melt material is between about 2000°F (about 1090°C) and about 2200°F (about 1200°C). Specific preferred embodiments have liquidus temperatures of about 2065°F (about 1130°C) and about 2165°F (about 1185°C), respectively. The solidus temperature of the high melt material is greater than the brazing temperature and is preferably no less than about 2300°F (about 1260°C), more preferably no less than about 2375°F (about 1300°C). In one preferred embodiment, the solidus temperature of the high melt material is between about 2350°F (about 1285°C) and about 2450°F (about 1345°C).

A water-based binder such as one available under the trade name Vitta Gel from Vitta Corp. of Bethel, Connecticut, is preferably mixed with each of the high and low melt materials to help hold them in place prior to brazing. The ratio of binder to metal is on the order of about 1:7 by weight, such that the materials are contained within an approximately 50% dense slurry after some evaporation occurs. The binder volatilizes upon brazing.

It has been discovered that improved oxidation resistance achieved in accordance with this invention is enhanced by selection of a low melt material which has a

significant content of Si as a melting point depressant in a Ni- or Co-based matrix. Silicon has been found to be especially advantageous in such applications because it enhances oxidation resistance. Accordingly, the low melt alloy of the invention contains at least about 3% Si, and preferably between about 4% and about 11% Si. One preferred embodiment contains between about 3% and about 5% Si, more preferably between about 4.2% and about 4.8% Si. Another preferred embodiment contains between about 9% and 11% Si, more preferably between about 9.8% and about 10.3% Si.

The low melt alloy of the invention is further characterized by a lack of intentionally added B, and by maintaining the incidental B content below about 1%, preferably below about 0.5%, more preferably below about 0.1%. It has been discovered that, inasmuch as B has deleterious effects on oxidation resistance, minimizing B enhances oxidation resistance. In contrast, prior low melt brazing alloys for use in tip cap hole filling applications have had a significant amount (e.g., at least about 2%) of B as a melting point depressant.

The low melt component of the invention further comprises Cr in the range of about 10% to about 25% to enhance oxidation resistance.

The high melt alloy of the invention contains Cr and Al for oxidation resistance. It preferably contains at least about 5%, preferably from about 5% to about 7%, more preferably from about 5.5% to about 6.5% Al, which provides an Al oxide scale which imparts enhanced oxidation

resistance.

It has been discovered that the performance of the fill material is further enhanced by using a high melt further containing Hf, preferably at least about 0.5%, more preferably about 1% to about 3%, most preferably from about 1% to about 2% Hf, to enhance the adherence of oxidation resistant Al and Cr oxide scales.

It has also been discovered that such performance is enhanced if the high melt also contains Re, at least about 1%, preferably from about 2% to about 4% Re due to its effect as a solid solution strengthener, particularly because Re does not have as deleterious an effect on oxidation resistance as other solid solution strengtheners such as Mo, Ta and W.

And it has further been discovered that by eliminating intentional additions of Ti, and by maintaining Ti content preferably below about 0.5%, the protective Cr and Al oxide scales adhere better, because Ti has a tendency to interfere with such adherence.

Exemplary high melt and low melt compositions in accordance with this invention are as follows:

<u>High Melt Comp. HA</u>		<u>Low Melt Comp. LA</u>		<u>Low Melt Comp. LB</u>	
C	0.10-0.14 wt%	C	0.13-0.19 wt%	C	0.01 max
Hf	1.3-1.7	Si	4.2-4.8	Si	9.8-10.3
Re	2.6-3.0	Cr	13.7-14.3	Cr	18.5-19.5
Cr	6.6-7.0	Co	9.0-10.0	B	0.03 max
Ta	6.2-6.5	Ti	4.6-5.2	Ni	Balance
Al	5.94-6.3	Al	2.8-3.2		plus incid.
W	4.7-5.0	Mo	3.7-4.3		impurities
Mo	1.3-1.7	W	3.7-4.3		
Co	11.45-12.05	B	0.5-0.8		
Ti	0.2 max	Ni	Balance plus		
Ni	Balance plus		incidental impurities		
	incidental impurities				

The following examples further illustrate the invention.

Example 1

Cyclic oxidation tests were conducted comparing samples filled with the high melt/low melt mixtures of the invention (HA/LA and HA/LB) to samples filled with the prior high melt/low melt mixtures (H1/L1). Cyclic oxidation tests were conducted at up to 2000°F with a five-minute hold time for up to 2400 cycles. Comparison of the photomicrographs of Figs. 5 through 10 reveals that the fill material HA/LA (Figs. 5 and 6) and HA/LB (Figs. 7 and 8) suffered substantially less oxidative attack than the fill material H1/L1 (Figs. 9 and 10). After 1800 cycles the HA/LA (Fig. 5) and HA/LB (Fig. 7) mixtures showed virtually no oxidative attack while the H1/L1 (Fig. 9) was nearly completely oxidized away. After 2400 cycles the HA/LA (Fig. 6) and HA/LB (Fig. 8) mixtures showed only minimal oxidative surface attack while the H1/L1 (Fig. 10) mixture was nearly completely oxidized away.

The actual maximum temperatures at which each of the

tested mixtures showed acceptable levels of attack were recorded and are presented in Table 1.

Table 1

<u>Mixture</u>	<u>Cycles</u>	Maximum Temperature (F) of Acceptable <u>Oxidative Attack</u>
H1/L1 (50/50)	1800	1500 - 1600
	2400	1600
HA/LA (50/50)	1800	2000
	2400	1900
HA/LB	1800	2000
	2400	1700 - 1800

Example 2

Tensile strength tests were performed according to the procedures of ASTM E8 and E21 to compare samples prepared according to the invention with samples prepared according to the prior method. The results are presented in Table 2.

Table 2

1900°F Tensile

<u>Test #</u>	<u>Material</u>	<u>UTS</u>	<u>%EL</u>
T36	H1/L1	19900	1.70
T37	H1/L1	19700	1.40
T3	HA/LB**	18300	1.20
T4	HA/LB**	16600	1.50
T15	HA/LA**	36500	0.70
T16	HA/LA**	36200	0.90
T25	HA/LA*	13600	0.70
T26	HA/LA*	16700	0.60

2000°F Tensile

<u>Test #</u>	<u>Material</u>	<u>UTS</u>	<u>%EL</u>
T38	H1/L1	11200	1.00
T39	H1/L1	7600	1.00
T5	HA/LB**	5500	1.90
T6	HA/LB**	4500	2.50
T17	HA/LA**	19400	1.00
T18	HA/LA**	19300	0.70
T27	HA/LA*	7400	1.00
T28	HA/LA*	7300	0.80

*Blended mixtures **HA Filled joints w/braze on top

It can be seen from these data that the HA/LA combinations performed best, and that the tests performed on samples prepared according to the brazing method of the invention (braze on top) performed better than tests performed on samples prepared using blended mixtures.

Example 3

Comparative 2000°F stress rupture tests were conducted according to the procedure of ASTM E139 and the results presented in Table 3.

Table 3

<u>Test #</u>	<u>Material</u>	<u>KSI</u>	<u>Life</u>	<u>%EL</u>
T40	H1/L1	1	1141.3	7.2
T41	H1/L1	1	1106.4	
T7	HA/LB**	1	685.3	
T8	HA/LB**	1	211	
T19	HA/LA**	1	414.5	
T20	HA/LA**	1	371.8	
T29	HA/LA*	1	54.2	2.10
T30	HA/LA*	1	35	1.80
T42	H1/L1	2	15.3	2.50
T43	H1/L1	2	14.9	1.90
T9	HA/LB**	2	5.3	8.70
T10	HA/LB**	2	1.1	5.40
T21	HA/LA**	2	27.9	2.20
T22	HA/LA**	2	19.7	1.80
T31	HA/LA*	2	29.3	2.70
T32	HA/LA*	2	6.7	1.50

*Blended mixtures **HA Filled joints w/braze on top

These data reveal that at 1 ksi load all samples performed adequately and that at 2 ksi load the HA/LA samples prepared using the brazing method of the invention performed best.

Example 4

Comparative nil strength tests were conducted as a measure of the highest use temperature and the results are presented in Table 4. These tests involved hanging a 300 psi dead weight load from a flat sheet sample. The sample is hung within a clam-shell furnace and the temperature is monitored. The temperature is increased and the temperature at which the bar breaks is recorded.

Table 4

<u>Test #</u>	<u>Material</u>	<u>Load (psi)</u>	<u>Temp (Deg. F)</u>
T-1	HA/LB**	300	2103.8
T-2	HA/LB**	300	2096.3
T-12	HA/LA**	300	2122
T-23	HA/LA*	300	2104.7
T-24	HA/LA*	300	2086.9
T-34	H1/L1	300	2033.4
T-35	H1/L1	300	2045.6

*Blended mixtures **HA Filled joints w/braze on top

These data reveal that the average nil strength temperature of specimens prepared using the alloys and method of the invention was 2107.4°F as compared to 2039.5°F for specimens prepared using the prior alloy and method.

In view of the foregoing examples it is evident that the brazing method of the invention with either of the new material combinations tested (HA/LA and HA/LB) provides a more oxidation resistant product with acceptable mechanical properties as compared to the prior method. While in some instances the mechanical strength appears to be less with the new method and materials, in all instances the mechanical strength is acceptable for tip cap hole fill applications. The material must have sufficient strength to hold itself against centrifugal force and thermal loading during service, but it bears no further significant load. The more important characteristic for this application is oxidation resistance. The mechanical strength and oxidation resistance are attributed in large part to the relatively low amount of void space in the fill material, as shown in Figs. 11 and 12.

Figure 11 shows a hole filled using the method of the invention with material HA followed by an 85/15 mixture of LA/HA; Fig. 12 shows a hole filled using the same method with material HA followed by LB.

As various changes could be made in the above constructions without departing from the scope of the invention, it is intended that all matter contained in the above description or shown in the accompanying drawings shall be interpreted as illustrative and not in a limiting sense.

WHAT IS CLAIMED IS:

1. A method for closing a high pressure turbine blade tip cap hole by brazing at a brazing temperature, the method comprising:

loading said tip cap hole with a first composition comprising particles of a first alloy having a solidus temperature above said brazing temperature;

covering said first composition with a second composition comprising particles of a second alloy having a liquidus temperature below said brazing temperature;

heating said second composition to said brazing temperature to cause said particles of said second alloy to melt to form a liquid of said second alloy which is carried into spaces between said particles of said first alloy by capillarity; and

cooling said liquid of said second alloy to form a solid securely bonding said particles of said first alloy.

2. The method of claim 1 wherein said first composition is a first slurry consisting essentially of said particles of said first alloy and a water-based binder, and wherein said second composition is a second slurry consisting essentially of said particles of said second alloy and a water-based binder.

3. The method of claim 1 or 2 wherein said second alloy has a Si content in the range of about 3% to about 11% by weight.

4. The method of claim 1, 2 or 3 wherein said second alloy has a B content which is less than about 1% by weight.

5. The method of any one of preceding claims wherein said second composition additionally comprises particles of a third alloy having a solidus temperature above said brazing temperature.

6. The method of claim 5 wherein said third alloy has a composition which is substantially the same as said first alloy.

7. The method of claim 5 or 6 wherein said second alloy has a weight ratio to said third alloy of between about 1:1 and about 9:1.

8. The method of claim 7 wherein said second alloy has a weight ratio to said third alloy of between about 5:1 and about 6:1.

9. A method for closing a high pressure turbine blade tip cap hole by brazing at a brazing temperature, the method comprising:

filling said tip cap hole with a fill composition comprising particles of at least two distinct alloy compositions comprising particles of a first alloy and particles of a second alloy wherein said first alloy has a solidus temperature above said brazing temperature and said second alloy has a liquidus temperature below said brazing temperature, said particles of said first alloy and said particles of said second alloy defining two distinct layers in said tip cap hole;

heating said fill composition to said brazing temperature to cause said particles of said second alloy to melt to form a liquid of said second alloy which is carried into spaces between said particles of said first alloy by capillarity; and

cooling said fill composition to solidify said second alloy to form a solid securely bonding said particles of said first alloy.

10. The method of claim 9 wherein said fill composition additionally comprises a water-based binder.

11. The method of claim 9 or 10 wherein said second alloy has a Si content in the range of about 3% to about 11% by weight and no more than about 1% by weight B.

12. The method of claim 9, 10 or 11 wherein said particles of said second alloy are intermixed with particles of a third alloy having a solidus temperature above said brazing temperature and wherein said second alloy has a weight ratio to said third alloy of between about 1:1 and about 9:1.

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ABSTRACT OF THE DISCLOSURE

A tip cap hole is loaded with a first composition comprising particles of a first alloy having a solidus temperature above the brazing temperature. The first composition is covered with a second composition comprising particles of a second alloy having a liquidus temperature below the brazing temperature. The second composition is heated to the brazing temperature to cause particles of the second alloy to melt to form a liquid of the second alloy which is carried into spaces between the particles of the first alloy by capillarity. The liquid of the second alloy is cooled to form a solid securely bonding the particles of the first alloy. By weight, the second alloy has no more than 1% B and from 3% to 11% Si. The first alloy has Cr and at least about 5% Al, at least about 0.5% Hf, no more than 0.5% Ti.

Fig. 3