

MODIFIED POLYISOBUTYLENE POLYMER FOR RUBBER COMPOUNDING AND
RUBBER COMPOSITION COMPRISING SAME

BACKGROUND OF THE INVENTION

5

1. Technical Field

The present invention relates to a modified polyisobutylene polymer for rubber compounding and a rubber composition including the same.

10

2. Description of the Related Art

Polyisobutylene (polybutene) is a polymer obtained by polymerizing an olefin component having 4 carbon atoms (C4) obtained in a naphtha-cracking process using a Friedel-Craft-
15 type catalyst, and the number average molecular weight (Mn) thereof is about 300 to 10,000.

What remains after extracting 1,3-butadiene from C4 feedstock is called C4 raffinate-1, and includes paraffins such as isobutane, normal butane, etc., and olefins such as 1-
20 butene, 2-butene, isobutene, etc. Here, the amount of isobutene is about 30 to 50 wt%. The C4 raffinate-1 is mainly used for the preparation of methyl t-butylether (MTBE) or polyisobutylene (polybutene), serving as an octane number improver. Since the reactivity of isobutene is the highest
25 among the olefin components of C4 raffinate-1, the resulting

polyisobutylene (polybutene) is mainly composed of isobutene units. Polyisobutylene (polybutene) increases in viscosity with increased molecular weight thereof, and has a viscosity of about 4 to 40000 cSt (centi-stokes) at 100°C.

5 Korean Patent Application Publication No. 10-2011-0072253 discloses a method of improving the grip performance of a tire tread using polyisobutylene (polybutene) having high reactivity. Also, Korean Patent Application Publication No. 10-2007-0096748 discloses a method of improving wear
10 performance, fuel efficiency and braking performance using carboxylated liquid isoprene rubber. As such, efforts are being made to increase fuel efficiency and braking performance of rubber, specifically when applied to tire treads.

With regard to improving grip force, Korean Patent
15 Application Publication No. 10-2016-0002044 discloses a composition exhibiting superior grip performance and wear resistance under high-speed conditions due to the use of a masterbatch in which pellet-type plant resins such as sesame resin, sunflower resin, coconut resin and the like are added
20 to styrene butadiene rubber. As already mentioned, the grip force is a technique in which the surface of a tire is brought into close contact with the road surface, and it is advantageous for the tire to have superior elasticity if possible. However, when considering both grip force and
25 rolling resistance, rolling resistance advantageously

decreases with a decrease in closeness of contact with the road surface, so there is a tradeoff between rolling resistance and the grip force of the tire. Specifically, a tire having low rolling resistance is advantageous in terms of fuel efficiency, but may have poor contact with a road when the road is wet. Accordingly, recent tire development is progressing in a way to control both at the same time, rather than a one-dimensional method of increasing the rolling resistance or increasing the grip force.

For example, Korean Patent Application Publication No. 10-2015-0024701 and U.S. Patent No. 8,637,606 disclose the use of silica and a modified terpene phenol resin having a high softening point, in which the phenol increases compatibility with synthetic rubber to thus reduce the fluidity of the resin, thereby improving grip performance on wet surfaces without deteriorating rolling resistance. Also, Korean Patent No. 10-1591276 discloses a rubber composition including 20 to 50 parts by weight of epoxidized natural rubber having a Tg of -50 to -40°C, a Mooney viscosity of 60 to 80, and an epoxidation degree of 5 to 50%, thereby improving the braking power of tires on the wet road surface, and uniformly improving low rolling resistance or fuel efficiency and durability without deteriorating wear resistance. Despite a variety of such attempts, there is a need for a technique that has satisfactory numerical values of both rolling resistance

and grip force of tires.

[Citation List]

[Patent Literature]

Korean Patent Application Publication No. 10-2011-0072253

5 Korean Patent Application Publication No. 10-2007-0096748

Korean Patent Application Publication No. 10-2016-0002044

Korean Patent Application Publication No. 10-2015-0024701

U.S. Patent No. 8,637,606

10

SUMMARY OF THE INVENTION

An objective of the present invention is to provide a modified polyisobutylene polymer for rubber compounding, which enables the preparation of a rubber composition capable of exhibiting superior grip performance and rolling resistance as well as high dispersibility when mixed with a filler.

The present invention provides a modified polyisobutylene polymer for rubber compounding, prepared by mixing polyisobutylene having isobutylene as a main chain, 20 unsaturated dicarboxylic anhydride, and a silane compound.

More specifically, the modified polyisobutylene polymer for rubber compounding may be prepared by mixing 20 to 80 wt% of polyisobutylene having isobutylene as a main chain, 1 to 20 wt% of unsaturated dicarboxylic anhydride, and 1 to 60 wt% of

a silane compound.

In the modified polyisobutylene polymer for rubber compounding according to an embodiment of the present invention, the polyisobutylene having isobutylene as the main
5 chain may have a number average molecular weight of 350 g/mol to 6,000 g/mol.

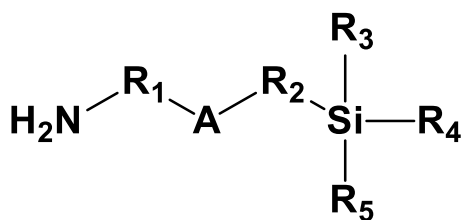
In the modified polyisobutylene polymer for rubber compounding according to an embodiment of the present invention, the polyisobutylene having isobutylene as the main
10 chain may have a polydispersity of 1 to 5.

In the modified polyisobutylene polymer for rubber compounding according to an embodiment of the present invention, the unsaturated dicarboxylic anhydride may be at least one selected from among maleic anhydride, itaconic
15 anhydride, citraconic anhydride, propenyl succinic anhydride and 2-pentendioic anhydride.

In the modified polyisobutylene polymer for rubber compounding according to an embodiment of the present invention, the silane compound may contain an amino group.

20 In the modified polyisobutylene polymer for rubber compounding according to an embodiment of the present invention, the silane compound may be represented by Chemical Formula 1 below.

[Chemical Formula 1]



In Chemical Formula 1,

R₁ and R₂ are each independently selected from among (C1-C5) alkylene, (C1-C5) aminoalkylene, carbonylene and (C1-C5) alkyl carbonylene, and are identical to or different from each other;

R₃, R₄ and R₅ are each independently selected from among hydrogen, hydroxy, (C1-C20) alkyl, (C1-C12) cycloalkyl, (C2-C14) acyloxy, (C4-C20) aryloxy, (C5-C30) araloxy, (C1-C20) amine and (C1-C12) alkoxy, and are identical to or different from each other; and

A is methylene, S_n or ((R₆)NR₇)_n, in which R₆ is hydrogen or (C1-C5) alkyl, R₇ is (C1-C5) alkylene, and n is an integer of 1 to 10.

In the modified polyisobutylene polymer for rubber compounding according to an embodiment of the present invention, the silane compound may be at least one selected from among 3-aminopropyl triethoxysilane, 3-aminopropyl trimethoxysilane, N-(2-aminoethyl)-3-aminopropyl triethoxysilane, 3-aminopropyl methyldiethoxysilane, 3-aminopropyl silanetriol, 3-aminopropyl methyldimethoxysilane, 3-(2-aminoethylamino)propyl dimethoxymethylsilane, 3-(2-

aminoethylamino)propyl trimethoxysilane, 2-ethanediamine N-(2-aminoethyl)-N'-[3-(trimethoxysilyl)propyl]-1, 1-[3-(trimethoxysilyl)propyl]urea and 1-[3-(triethoxysilyl)propyl]urea.

5 The modified polyisobutylene polymer for rubber compounding according to an embodiment of the present invention may contain nitrogen.

 The modified polyisobutylene polymer for rubber compounding according to an embodiment of the present
10 invention may contain a carbonyl group.

 The modified polyisobutylene polymer for rubber compounding according to an embodiment of the present invention may contain an amide group.

 The modified polyisobutylene polymer for rubber
15 compounding according to an embodiment of the present invention may have a Si content of 0.05 to 10 mass% based on X-ray fluorescence analysis.

 The modified polyisobutylene polymer for rubber compounding according to an embodiment of the present
20 invention may have a glass transition temperature of -50°C or less.

 The modified polyisobutylene polymer for rubber compounding according to an embodiment of the present invention may have a Brookfield viscosity at 150°C of 5 to
25 10,000 cP.

The modified polyisobutylene polymer for rubber compounding according to an embodiment of the present invention may have a number average molecular weight of 580 to 10,000 g/mol and a polydispersity of 1 to 5.

5 In addition, the present invention provides a rubber composition. The rubber composition according to the present invention includes the modified polyisobutylene polymer for rubber compounding according to an embodiment of the present invention, a rubber base, and a filler.

10 In the rubber composition according to an embodiment of the present invention, the filler may include at least one selected from among silica and carbon black.

In the rubber composition according to an embodiment of the present invention, the rubber base may include at least
15 one selected from among butadiene rubber, butyl rubber, emulsion-polymerized styrene butadiene rubber (E-SBR), solution-polymerized styrene butadiene rubber (S-SBR), epichlorohydrin rubber, nitrile rubber, hydrogenated nitrile rubber, brominated polyisobutyl isoprene-co-paramethyl styrene
20 (BIMS) rubber, urethane rubber, fluorine rubber, silicone rubber, styrene-ethylene-butadiene-styrene copolymer rubber, ethylene propylene rubber, ethylene-propylene-diene monomer rubber, Hypalon rubber, chloroprene rubber, ethylene vinyl acetate rubber and acrylic rubber.

25 The rubber composition according to an embodiment of the

present invention may include, based on 100 parts by weight of the rubber base, 50 to 150 parts by weight of silica, 5 to 20 parts by weight of carbon black, 2 to 40 parts by weight of the modified polyisobutylene polymer, and 2 to 15 parts by weight of a silane-coupling agent.

In addition, the present invention provides a tire tread, and the tire tread of the present invention may include the rubber composition according to an embodiment of the present invention.

10

According to the present invention, a modified polyisobutylene polymer for rubber compounding is prepared by mixing polyisobutylene having isobutylene as a main chain, unsaturated dicarboxylic anhydride, and a silane compound, and upon application thereof to a rubber composition, the dispersibility of a filler can be increased, and both braking performance and rolling resistance can be improved, thereby exhibiting high grip performance and low rolling resistance.

20

DESCRIPTION OF SPECIFIC EMBODIMENTS

The advantages and features of embodiments of the present invention and methods of achieving the same will be clarified with reference to examples described below in detail. However, the present invention is not limited to these

25

embodiments, but may be modified in other forms. These
embodiments are provided in order to fully convey the spirit
of the present invention to those skilled in the art so that
the contents introduced herein are thorough and complete, and
5 the present invention is only defined by the scope of the
claims. The same reference numerals refer to the same
components throughout the specification.

In describing embodiments of the present invention, when
it is determined that a detailed description of a known
10 function or configuration may unnecessarily obscure the gist
of the present invention, such a detailed description will be
omitted. Moreover, terms to be described later are terms
defined in consideration of functions in embodiments of the
present invention, and may vary depending on a user's or
15 operator's intention or practice. Therefore, the definition
should be made based on the contents throughout this
specification. For example, polyisobutylene may be referred
to as polybutene.

According to the present invention, a modified
20 polyisobutylene polymer for rubber compounding is prepared by
mixing polyisobutylene having isobutylene as a main chain,
unsaturated dicarboxylic anhydride, and a silane compound.

More specifically, the modified polyisobutylene polymer
for rubber compounding according to the present invention may
25 be prepared by mixing 20 to 80 wt% of polyisobutylene having

isobutylene as a main chain, 1 to 20 wt% of unsaturated dicarboxylic anhydride, and 1 to 60 wt% of a silane compound. Based on the results of analysis with X-ray fluorescence (XRF), the modified polyisobutylene polymer thus prepared may
5 have a Si content of 0.05 to 10 mass%, and preferably 0.5 to 5 mass%. When the modified polyisobutylene polymer is prepared by mixing individual components in amounts in the above ranges and is mixed with a rubber base and a filler to afford a rubber composition, the dispersibility of the filler in the
10 rubber composition may be increased.

In an embodiment of the present invention, the modified polyisobutylene polymer may have a glass transition temperature of -50°C or less, preferably -60 to -100°C , and more preferably -65 to -90°C . Also, the modified
15 polyisobutylene polymer may have a viscosity of 5 to 10,000 cP, and preferably 50 to 5,000 cP, when measured using a Brookfield viscometer at 150°C . When the glass transition temperature and viscosity thereof fall in the above ranges, a storage modulus difference ($\Delta G'$) due to a Payne effect may be
20 as low as 2.4 or less, and preferably 2.1 or less, upon subsequent mixing with the rubber base and the filler, indicating that the filler is uniformly dispersed. Here, $\Delta G'$ means $G'_{20\%} - G'_{0.02\%}$, which is a difference in storage modulus values measured at elongations of 0.02% and 20%.

25 In an embodiment of the present invention, the modified

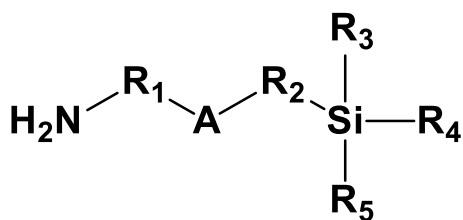
polyisobutylene polymer may have a number average molecular weight of 580 to 10,000 g/mol and a polydispersity of 1 to 5. When the modified polyisobutylene polymer satisfying the above molecular weight and polydispersity is mixed with the rubber
5 base and the filler to afford a rubber composition, both grip performance and rolling resistance performance may be improved, and the rubber thus manufactured may have a dynamic loss coefficient at 0°C of 0.25 or more and a dynamic loss coefficient at 60°C of less than 0.094. More preferably, the
10 modified polyisobutylene polymer has a number average molecular weight of 900 to 3,000 g/mol and a polydispersity of 1 to 3, and the manufactured rubber has a dynamic loss coefficient at 0°C of 0.27 or more and a dynamic loss coefficient at 60°C of 0.090 or less. Here, the dynamic loss
15 coefficient at 0°C may be an indicator of grip performance (wet grip), and the higher the dynamic loss coefficient at 0°C, the better the grip performance. The dynamic loss coefficient at 60°C is an indicator of rolling resistance, and the lower the numerical value thereof, the better the rolling
20 resistance. Specifically, according to the present invention, when the modified polyisobutylene polymer satisfying a number average molecular weight of 580 to 10,000 g/mol and a polydispersity of 1 to 5 is added to the rubber composition, both braking performance and rolling performance may be
25 improved, thus increasing tire performance.

Meanwhile, in the preparation of the modified polyisobutylene polymer according to an embodiment of the present invention, the polyisobutylene having isobutylene as the main chain may have a number average molecular weight of 5 350 g/mol to 6,000 g/mol, and preferably 700 g/mol to 3,000 g/mol, and a polydispersity of 1 to 5, and preferably 1 to 3. The viscosity at 100°C of polyisobutylene may be 2 cSt to 10,000 cSt, and preferably 100 cSt to 2,000 cSt.

In an embodiment of the present invention, the 10 polyisobutylene may have a terminal vinylidene (α -vinylidene) content of 80 mol% or more, and preferably 85 mol% or more, based on ^{13}C -NMR. If the α -vinylidene content is less than 80 mol%, the extent of improvement in both grip performance and rolling performance of rubber resulting from adding the 15 prepared modified polyisobutylene polymer to the rubber composition may become insignificant. Here, the upper limit of the α -vinylidene content is not particularly limited, and may be, for example, 99 mol% or less, and particularly 95 mol% or less.

20 In an embodiment of the present invention, the silane compound may contain an amino group, and preferably, the silane compound in the modified polyisobutylene polymer according to an embodiment of the present invention may be a compound represented by Chemical Formula 1 below.

25 [Chemical Formula 1]



In Chemical Formula 1,

R₁ and R₂ are each independently selected from among (C1-C5) alkylene, (C1-C5) aminoalkylene, carbonylene and (C1-C5) alkyl carbonylene, and are identical to or different from each other;

R₃, R₄ and R₅ are each independently selected from among hydrogen, hydroxy, (C1-C20) alkyl, (C1-C12) cycloalkyl, (C2-C14) acyloxy, (C4-C20) aryloxy, (C5-C30) araloxy, (C1-C20) amine and (C1-C12) alkoxy, and are identical to or different from each other; and

A is methylene, S_n or ((R₆)NR₇)_n, in which R₆ is hydrogen or (C1-C5) alkyl, R₇ is (C1-C5) alkylene, and n is an integer of 1 to 10.

More preferably, R₁ and R₂ are each independently (C1-C3) alkylene,

R₃, R₄ and R₅ are each independently hydrogen, hydroxy, (C1-C5) alkyl or (C1-C5) alkoxy, and

A is methylene, S_n or ((R₆)NR₇)_n, in which R₆ is hydrogen or (C1-C5) alkyl, R₇ is (C2-C4) alkylene, and n is an integer of 2 to 5.

When the modified polyisobutylene polymer is prepared

using a silane compound satisfying the above criteria, the dispersibility of the filler may be further increased.

Specifically, the silane compound may be at least one selected from among 3-aminopropyl triethoxysilane, 3-aminopropyl trimethoxysilane, N-(2-aminoethyl)-3-aminopropyl triethoxysilane, 3-aminopropyl methyldiethoxysilane, 3-aminopropyl silanetriol, 3-aminopropyl methyldimethoxysilane, 3-(2-aminoethylamino)propyl dimethoxymethylsilane, 3-(2-aminoethylamino)propyl trimethoxysilane, 2-ethanediamine N-(2-aminoethyl)-N'-[3-(trimethoxysilyl)propyl]-1, 1-[3-(trimethoxysilyl)propyl]urea and 1-[3-(triethoxysilyl)propyl]urea.

In the modified polyisobutylene polymer according to an embodiment of the present invention, the unsaturated dicarboxylic anhydride may be at least one selected from among maleic anhydride, itaconic anhydride, citraconic anhydride, propenyl succinic anhydride and 2-pentendioic anhydride.

Moreover, the modified polyisobutylene polymer prepared by mixing the polyisobutylene, the unsaturated dicarboxylic anhydride and the silane compound may contain nitrogen, and preferably a nitrogen atom joined through covalent bonding. Also, the modified polyisobutylene polymer may contain a carbonyl group, and preferably an imide group or an amide group. For example, the amide group may include a pyrrolidine-2,5-dione group.

In addition, the present invention pertains to a rubber composition.

According to the present invention, the rubber composition includes the modified polyisobutylene polymer according to an embodiment of the present invention, a rubber base, and a filler.

In an embodiment of the present invention, the rubber composition includes the modified polyisobutylene polymer, whereby the filler is uniformly dispersed therein, thus exhibiting superior grip performance and rolling resistance.

In the rubber composition according to an embodiment of the present invention, the rubber base may include at least one selected from among butadiene rubber, butyl rubber, emulsion-polymerized styrene butadiene rubber (E-SBR), solution-polymerized styrene butadiene rubber (S-SBR), epichlorohydrin rubber, nitrile rubber, hydrogenated nitrile rubber, brominated polyisobutyl isoprene-co-paramethyl styrene (BIMS) rubber, urethane rubber, fluorine rubber, silicone rubber, styrene-ethylene-butadiene-styrene copolymer rubber, ethylene propylene rubber, ethylene-propylene-diene monomer rubber, Hypalon rubber, chloroprene rubber, ethylene vinyl acetate rubber and acrylic rubber. Preferably, the rubber base includes at least one selected from among butadiene rubber, styrene butadiene rubber and butyl rubber.

More preferably, the rubber base includes styrene

butadiene rubber, and the styrene butadiene rubber that is used may be one in which a styrene content is 9 to 19% and a vinyl group content in butadiene is 10 to 54%, a styrene content is 20 to 28% and a vinyl content in butadiene is 40 to 72%, or a styrene content is 30 to 42% and a vinyl content in butadiene is 20 to 70%.

In the rubber composition according to an embodiment of the present invention, the filler may be used without limitation, so long as it is a filler typically used in a rubber composition, preferably a rubber composition for tire treads, and the present invention is not limited thereto. As a specific and non-limiting example, the filler may include at least one selected from among silica and carbon black.

Here, the silica may be used without limitation, so long as it is silica particles used in rubber, preferably rubber for tire treads. Specifically, the silica may have a specific surface area (CTAB) of 80 m²/g to 300 m²/g, preferably 110 m²/g to 220 m²/g, more preferably 150 m²/g to 180 m²/g, and most preferably 165 m²/g. If the specific surface area thereof is less than the above lower limit, reinforcing properties may decrease and thus the strength thereof may be lowered. On the other hand, if the specific surface area thereof exceeds the above upper limit, the viscosity may increase upon rubber compounding, and dispersion may also be impeded.

Also, the carbon black may be used without limitation, so

long as it is carbon black typically used in rubber for tire treads, and carbon black having a grade of 500 to 600 is preferably used. Specific and non-limiting examples thereof may include commercially available carbon black such as N110, 5 N121, N134, N220, N231, N234, N242, N293, N299, S315, N326, N330, N332, N339, N343, N347, N351, N358, N375, N539, N550, N582, N630, N642, N650, N660, N683, N754, N762, N765, N774, N787, N907, N908, N990, N991, etc., and the present invention is not limited thereto.

10 Moreover, the rubber composition according to an embodiment of the present invention may further include a silane-coupling agent. Here, examples of the silane-coupling agent may include commercially available products such as Si-69 and the like, and known materials such as bis-(3-15 triethoxysilylpropyl)tetrasulfane (TESPT), bis-(3-ethoxysilylpropyl)disulfane (ESPD), N-[2-(vinylbenzylamino)-3-aminopropyltrimethoxysilane] and the like, and the present invention is not limited thereto.

 The rubber composition according to an embodiment of the 20 present invention may include, based on 100 parts by weight of the rubber base, 50 to 150 parts by weight of silica, 5 to 20 parts by weight of carbon black, 2 to 40 parts by weight of the modified polyisobutylene polymer, and 2 to 15 parts by weight of the silane-coupling agent, and preferably, based on 25 100 parts by weight of the rubber base, 60 to 120 parts by

weight of silica, 7 to 15 parts by weight of carbon black, 5 to 20 parts by weight of the modified polyisobutylene polymer and 3 to 10 parts by weight of the silane-coupling agent.

The rubber composition according to an embodiment of the present invention may further include an additive typically used in rubber compositions. As a specific and non-limiting example, the rubber composition may further include an additive such as an antioxidant, an activator, a vulcanizing agent, a vulcanization accelerator, etc. The amount of each additive may vary depending on the type of additive, the end use of rubber to be manufactured, etc. As a specific and non-limiting example, based on 100 parts by weight of the rubber base, 0.5 to 5 parts by weight of each additive may be added, but the present invention is not limited thereto.

As a specific and non-limiting example, the vulcanizing agent that is used may be sulfur, morpholine disulfide, etc., and the vulcanization accelerator may include at least one selected from among sulfenamide-based, thiazole-based, thiuram-based, thiourea-based, guanidine-based, dithiocarbamic-acid-based, aldehyde-amine-based, aldehyde-ammonia-based, imidazoline-based and xanthate-based vulcanization accelerators.

Specifically, the sulfenamide-based vulcanization accelerator may include at least one sulfenamide-based compound selected from among CBS (N-cyclohexyl-2-

benzothiazylsulfenamide), TBBS (N-tert-butyl-2-
 benzothiazylsulfenamide), N,N-dicyclohexyl-2-
 benzothiazylsulfenamide, N-oxydiethylene-2-
 benzothiazylsulfenamide and N,N-diisopropyl-2-
 5 benzothiazolesulfenamide, the thiazole-based vulcanization
 accelerator may include at least one thiazole-based compound
 selected from among MBT (2-mercaptobenzothiazole), MBTS
 (dibenzothiazyl disulfide), 2-mercaptobenzothiazole sodium
 salt, zinc salt, copper salt, cyclohexylamine salt, 2-(2,4-
 10 dinitrophenyl)mercaptobenzothiazole and 2-(2,6-diethyl-4-
 morpholiniothio)benzothiazole, the thiuram-based vulcanization
 accelerator may include at least one thiuram-based compound
 selected from among TMTD (tetramethylthiuram disulfide),
 tetraethylthiuram disulfide, tetramethylthiuram monosulfide,
 15 dipentamethylenethiuram disulfide, dipentamethylenethiuram
 monosulfide, dipentamethylenethiuram tetrasulfide,
 dipentamethylenethiuram hexasulfide, tetrabutylthiuram
 disulfide and pentamethylenethiuram tetrasulfide, the
 thiourea-based vulcanization accelerator may include at least
 20 one thiourea-based compound selected from among thiocarbamide,
 diethylthiourea, dibutylthiourea, trimethylthiourea and
 di(ortho-tolyl)thiourea, and the guanidine-based vulcanization
 accelerator may include at least one guanidine-based compound
 selected from among diphenylguanidine, di(ortho-
 25 tolyl)guanidine, triphenylguanidine, ortho-tolyl biguanide and

diphenylguanidine phthalate, but the present invention is not limited thereto.

In addition, the present invention pertains to a tire tread including the rubber composition according to an embodiment of the present invention. The tire tread according to the present invention has superior grip performance and rolling resistance, as described above, and thus has an advantage of exhibiting excellent fuel efficiency compared to other tire treads under the same conditions. Specifically, the tire tread may be a tire tread used for a passenger car, an SUV, a bus, a truck or an electric vehicle, but the present invention is not limited thereto.

A better understanding of the present invention will be given through the following examples and comparative examples. The following examples are merely set forth to illustrate the present invention, and are not to be construed as limiting the scope of the present invention.

[Preparation Example 1] Preparation of HRPB350 modified polyisobutylene polymer

Polyisobutylene (Mn: 351 g/mol, PD=1.1, α -vinylidene: 80.6 mol% based on ^{13}C -NMR, viscosity: 3 cSt at 100°C, 500 g, 1.42 mol) and maleic anhydride (172 g, 1.75 mol) were placed in a 1 L autoclave and allowed to react at 200°C for 12 hr using a mechanical stirrer. In order to remove unreacted

maleic anhydride, nitrogen bubbling was performed for 2 hr, thus obtaining polyisobutylene succinic anhydride [PIBSA-1]. The conversion rate thereof was determined to be 79.1% through column chromatography.

5 Then, the PIBSA-1 thus prepared (500 g, 1.11 mol) and 3-aminopropyl triethoxysilane (491 g, 2.22 mol) were added to 500 mL of toluene in a 1 L autoclave and allowed to react at 120°C for 4 hr using a Dean-Stark apparatus. Unreacted compounds were removed, after which 650 g of modified
10 polyisobutylene (yield: 88.3%) was obtained. The modified polyisobutylene polymer thus obtained was determined to have a number average molecular weight of 633 g/mol, a weight average molecular weight of 837 g/mol, and a polydispersity of 1.3. The T_g thereof was measured to be -75°C using DSC, the
15 Brookfield viscosity thereof was 7 cP at 150°C, and the Si content thereof was measured to be 2.2 mass% using XRF.

[Preparation Example 2] Preparation of HRPB550 modified polyisobutylene polymer

20 Polyisobutylene (M_n: 595 g/mol, PD=1.4, α-vinylidene: 81.0 mol% based on ¹³C-NMR, viscosity: 30.1 cSt at 100°C, 300 g, 0.5 mol) and maleic anhydride (60.0 g, 0.6 mol) were placed in a 1 L autoclave and allowed to react at 230°C for 12 hr using a mechanical stirrer. In order to remove unreacted
25 maleic anhydride, nitrogen bubbling was performed for 2 hr,

thus obtaining polyisobutylene succinic anhydride (PIBSA-2). The conversion rate thereof was determined to be 84.4% through column chromatography.

Then, the PIBSA-2 thus prepared (300 g, 0.46 mol) and 3-aminopropyl triethoxysilane (214 g, 0.97 mol) were added to 200 mL of toluene in a 1 L autoclave and allowed to react at 120°C for 4 hr using a Dean-Stark apparatus. Unreacted compounds were removed, after which 340 g of modified polyisobutylene (yield: 93%) was obtained.

The modified polyisobutylene polymer thus obtained was determined to have a number average molecular weight of 819 g/mol, a weight average molecular weight of 1,448 g/mol, and a polydispersity of 1.8. The T_g thereof was measured to be -72°C using DSC, the Brookfield viscosity thereof was 85 cP at 150°C, and the Si content thereof was measured to be 1.8 mass% using XRF.

[Preparation Example 3] Preparation of HRPB750 modified polyisobutylene polymer

Polyisobutylene (M_n: 742 g/mol, PD=1.3, α-vinylidene: 87.0 mol% based on ¹³C-NMR, viscosity: 86 cSt at 100°C, 500 g, 0.67 mol) and maleic anhydride (72 g, 0.74 mol) were placed in a 1 L autoclave and allowed to react at 200°C for 12 hr using a mechanical stirrer. In order to remove unreacted maleic anhydride, nitrogen bubbling was performed for 2 hr, thus

obtaining polyisobutylene succinic anhydride (PIBSA-3). The conversion rate thereof was determined to be 84.4% through column chromatography.

Then, the PIBSA-3 thus prepared (500 g, 0.60 mol) and 3-aminopropyl triethoxysilane (268 g, 1.21 mol) were added to 500 mL of toluene in a 1 L autoclave and allowed to react at 120°C for 4 hr using a Dean-Stark apparatus. Unreacted compounds were removed, after which 560 g of modified polyisobutylene (yield: 96%) was obtained.

The modified polyisobutylene polymer thus obtained was determined to have a number average molecular weight of 1,088 g/mol, a weight average molecular weight of 2,103 g/mol, and a polydispersity of 1.9. The T_g thereof was measured to be -70°C using DSC, the Brookfield viscosity thereof was 106 cP at 150°C, and the Si content thereof was measured to be 1.5 mass% using XRF.

[Preparation Example 4] Preparation of HRPB1000 modified polyisobutylene polymer

Polyisobutylene (M_n: 992 g/mol, PD=1.4, α-vinylidene: 88.3 mol% based on ¹³C-NMR, viscosity: 193 cSt at 100°C, 500 g, 0.5 mol) and maleic anhydride (52 g, 0.5 mol) were placed in a 1 L autoclave and allowed to react at 230°C for 12 hr using a mechanical stirrer. In order to remove unreacted maleic anhydride, nitrogen bubbling was performed for 2 hr, thus

obtaining polyisobutylene succinic anhydride (PIBSA-4). The conversion rate thereof was determined to be 78.4% through column chromatography.

Then, the PIBSA-4 thus prepared (500 g, 0.45 mol) and 3-aminopropyl triethoxysilane (211 g, 0.95 mol) were added to 500 mL of toluene in a 1 L autoclave and allowed to react at 120°C for 4 hr using a Dean-Stark apparatus. Unreacted compounds were removed, after which 550 g of modified polyisobutylene (yield: 96%) was obtained.

The modified polyisobutylene polymer thus obtained was determined to have a number average molecular weight of 1,266 g/mol, a weight average molecular weight of 2,174 g/mol, and a polydispersity of 1.7. The T_g thereof was measured to be -69°C using DSC, the Brookfield viscosity thereof was 250 cP at 150°C, and the Si content thereof was measured to be 1.3 mass% using XRF.

[Preparation Example 5] Preparation of HRPB1300 modified polyisobutylene polymer

Polyisobutylene (M_n: 1,312 g/mol, PD=1.6, α-vinylidene: 89.6 mol% based on ¹³C-NMR, viscosity: 431 cSt at 100°C, 500 g, 0.38 mol) and maleic anhydride (45 g, 0.46 mol) were placed in a 1 L autoclave and allowed to react at 200°C for 12 hr using a mechanical stirrer. In order to remove unreacted maleic anhydride, nitrogen bubbling was performed for 2 hr, thus

obtaining polyisobutylene succinic anhydride (PIBSA-5). The conversion rate thereof was determined to be 86.7% through column chromatography.

Then, the PIBSA-5 thus prepared (500 g, 0.35 mol) and 3-aminopropyl triethoxysilane (156 g, 0.71 mol) were added to 500 mL of toluene in a 1 L autoclave and allowed to react at 120°C for 4 hr using a Dean-Stark apparatus. Unreacted compounds were removed, after which 550 g of modified polyisobutylene (yield: 95.4%) was obtained.

The modified polyisobutylene polymer thus obtained was determined to have a number average molecular weight of 1,929 g/mol, a weight average molecular weight of 3,452 g/mol, and a polydispersity of 1.8. The T_g thereof was measured to be -69°C using DSC, the Brookfield viscosity thereof was 375 cP at 150°C, and the Si content thereof was measured to be 1.1 mass% using XRF.

[Preparation Example 6] Preparation of HRPB1800 modified polyisobutylene polymer

Polyisobutylene (M_n: 1,792 g/mol, PD=1.5, α-vinylidene: 87.5 mol% based on ¹³C-NMR, viscosity: 1,279 cSt at 100°C, 500 g, 0.28 mol) and maleic anhydride (29 g, 0.3 mol) were placed in a 1 L autoclave and allowed to react at 230°C for 12 hr using a mechanical stirrer. In order to remove unreacted maleic anhydride, nitrogen bubbling was performed for 2 hr,

thus obtaining polyisobutylene succinic anhydride (PIBSA-6). The conversion rate thereof was determined to be 76.4% through column chromatography.

Then, the PIBSA-6 thus prepared (500 g, 0.26 mol) and 3-aminopropyl triethoxysilane (123 g, 0.56 mol) were added to 500 mL of toluene in a 1 L autoclave and allowed to react at 120°C for 4 hr using a Dean-Stark apparatus. Unreacted compounds were removed, after which 510 g of modified polyisobutylene (yield: 92.4%) was obtained.

The modified polyisobutylene polymer thus obtained was determined to have a number average molecular weight of 1,950 g/mol, a weight average molecular weight of 4,700 g/mol, and a polydispersity of 2.4. The T_g thereof was measured to be -70°C using DSC, the Brookfield viscosity thereof was 776 cP at 150°C, and the Si content thereof was measured to be 0.9 mass% using XRF.

[Preparation Example 7] Preparation of HRPB2000 modified polybutene polymer

Polyisobutylene (M_n: 1,980 g/mol, PD=1.6, α-vinylidene: 81.4 mol% based on ¹³C-NMR, viscosity: 1400 cSt at 100°C, 500 g, 0.25 mol) and maleic anhydride (28 g, 0.29 mol) were placed in a 1 L autoclave and allowed to react at 200°C for 12 hr using a mechanical stirrer. In order to remove unreacted maleic anhydride, nitrogen bubbling was performed for 2 hr,

thus obtaining polyisobutylene succinic anhydride (PIBSA-7). The conversion rate thereof was determined to be 76.8% through column chromatography.

Then, the PIBSA-7 thus prepared (500 g, 0.25 mol) and 3-aminopropyl triethoxysilane (110 g, 0.50 mol) were added to 500 mL of toluene in a 1 L autoclave and allowed to react at 120°C for 4 hr using a Dean-Stark apparatus. Unreacted compounds were removed, after which 540 g of modified polybutene (yield: 93%) was obtained.

The modified polybutene polymer thus obtained was determined to have a number average molecular weight of 2,209 g/mol, a weight average molecular weight of 5,219 g/mol, and a polydispersity of 2.4. The T_g thereof was measured to be -68°C using DSC, the Brookfield viscosity thereof was 882 cP at 150°C, and the Si content thereof was measured to be 0.8 mass% using XRF.

[Preparation Example 8] Preparation of HRPB2300 modified polyisobutylene polymer

Polyisobutylene (M_n: 2,264 g/mol, PD=1.7, α-vinylidene: 85.7 mol% based on ¹³C-NMR, viscosity: 1,477 cSt at 100°C, 500 g, 0.22 mol) and maleic anhydride (23.0 g, 0.23 mol) were placed in a 1 L autoclave and allowed to react at 230°C for 12 hr using a mechanical stirrer. In order to remove unreacted maleic anhydride, nitrogen bubbling was performed for 2 hr,

thus obtaining polyisobutylene succinic anhydride (PIBSA-8). The conversion rate thereof was determined to be 77.4% through column chromatography.

Then, the PIBSA-8 thus prepared (300 g, 0.13 mol) and 3-aminopropyl triethoxysilane (61 g, 0.27 mol) were added to 200 mL of toluene in a 1 L autoclave and allowed to react at 120°C for 4 hr using a Dean-Stark apparatus. Unreacted compounds were removed, after which 280 g of modified polyisobutylene (yield: 91.3%) was obtained.

The modified polyisobutylene polymer thus obtained was determined to have a number average molecular weight of 2,350 g/mol, a weight average molecular weight of 5,632 g/mol, and a polydispersity of 2.4. The Brookfield viscosity thereof was 900 cP at 150°C, the T_g thereof was measured to be -68°C using DSC, and the Si content thereof was measured to be 0.7 mass% using XRF.

[Preparation Example 9] Preparation of HRPB3000 modified polyisobutylene polymer

Polyisobutylene (M_n: 3,213 g/mol, PD=1.7, α-vinylidene: 85.2 mol% based on ¹³C-NMR, viscosity: 3,200 cSt at 100°C, 500 g, 0.16 mol) and maleic anhydride (20.0 g, 0.20 mol) were placed in a 1 L autoclave and allowed to react at 230°C for 12 hr using a mechanical stirrer. In order to remove unreacted maleic anhydride, nitrogen bubbling was performed for 2 hr,

thus obtaining polyisobutylene succinic anhydride (PIBSA-9). The conversion rate thereof was determined to be 77.4% through column chromatography.

Then, the PIBSA-9 thus prepared (300 g, 0.09 mol) and 3-aminopropyl triethoxysilane (40 g, 0.18 mol) were added to 200 mL of toluene in a 1 L autoclave and allowed to react at 120°C for 4 hr using a Dean-Stark apparatus. Unreacted compounds were removed, after which 290 g of modified polyisobutylene (yield: 88.1%) was obtained.

The modified polyisobutylene polymer thus obtained was determined to have a number average molecular weight of 3,862 g/mol, a weight average molecular weight of 9,948 g/mol, and a polydispersity of 2.6. The Brookfield viscosity thereof was 4,600 cP at 150°C, the T_g thereof was measured to be -65°C using DSC, and the Si content thereof was measured to be 0.1 mass% using XRF.

[Preparation Example 10] Preparation of HRPB5000 modified polyisobutylene polymer

Polyisobutylene (M_n: 5,897 g/mol, PD=1.8, α-vinylidene: 87.0 mol% based on ¹³C-NMR, viscosity: 11,322 cSt at 100°C, 500 g, 0.1 mol) and maleic anhydride (10.0 g, 0.1 mol) were placed in a 1 L autoclave and allowed to react at 230°C for 12 hr using a mechanical stirrer. In order to remove unreacted maleic anhydride, nitrogen bubbling was performed for 2 hr,

thus obtaining polyisobutylene succinic anhydride (PIBSA-10). The conversion rate thereof was determined to be 70.4% through column chromatography.

Then, the PIBSA-10 thus prepared (300 g, 0.05 mol) and 3-aminopropyl triethoxysilane (25 g, 0.1 mol) were added to 200 mL of toluene in a 1 L autoclave and allowed to react at 120°C for 4 hr using a Dean-Stark apparatus. Unreacted compounds were removed, after which 250 g of modified polyisobutylene (yield: 80.7%) was obtained.

The modified polyisobutylene polymer thus obtained was determined to have a number average molecular weight of 6,187 g/mol, a weight average molecular weight of 19,750 g/mol, and a polydispersity of 3.2. The Brookfield viscosity thereof was 9,100 cP at 150°C, the T_g thereof was measured to be -60°C using DSC, and the Si content thereof was measured to be 0.05 mass% using XRF.

[Preparation Example 11] Preparation of HRPB260 modified polyisobutylene polymer

Polyisobutylene (M_n: 260 g/mol, PD=1.2, α-vinylidene: 78.4 mol% based on ¹³C-NMR, viscosity: 5 cSt at 40°C, 500 g, 1.92 mol) and maleic anhydride (192 g, 1.96 mol) were placed in a 1 L autoclave and allowed to react at 200°C for 12 hr using a mechanical stirrer. In order to remove unreacted maleic anhydride, nitrogen bubbling was performed for 2 hr,

thus obtaining polyisobutylene succinic anhydride (PIBSA-11). The conversion rate thereof was determined to be 76.1% through column chromatography.

Then, the PIBSA-11 thus prepared (500 g, 1.39 mol) and 3-aminopropyl triethoxysilane (690 g, 3.12 mol) were added to 500 mL of toluene in a 1 L autoclave and allowed to react at 120°C for 4 hr using a Dean-Stark apparatus. Unreacted compounds were removed, after which 700 g of modified polyisobutylene (yield: 92%) was obtained.

The modified polyisobutylene polymer thus obtained was determined to have a number average molecular weight of 543 g/mol, a weight average molecular weight of 751 g/mol, and a polydispersity of 1.4. The T_g thereof was measured to be -75°C using DSC, the Brookfield viscosity thereof was 4 cP at 150°C, and the Si content thereof was measured to be 2.3 mass% using XRF.

[Preparation Example 12] Preparation of HRPB7000 modified polyisobutylene polymer

Polyisobutylene (M_n: 7,096 g/mol, PD=2.5, α-vinylidene: 80.4 mol% based on ¹³C-NMR, viscosity: 24,734 cSt at 100°C, 500 g, 0.07 mol) and maleic anhydride (10.0 g, 0.1 mol) were placed in a 1 L autoclave and allowed to react at 230°C for 12 hr using a mechanical stirrer. In order to remove unreacted maleic anhydride, nitrogen bubbling was performed for 2 hr,

thus obtaining polyisobutylene succinic anhydride (PIBSA-12). The conversion rate thereof was determined to be 75.1% through column chromatography.

Then, the PIBSA-12 thus prepared (300 g, 0.04 mol) and 3-aminopropyl triethoxysilane (18 g, 0.08 mol) were added to 200 mL of toluene in a 1 L autoclave and allowed to react at 120°C for 4 hr using a Dean-Stark apparatus. Unreacted compounds were removed, after which 250 g of modified polyisobutylene (yield: 80.7%) was obtained.

The modified polyisobutylene polymer thus obtained was determined to have a number average molecular weight of 7,641 g/mol, a weight average molecular weight of 22,812 g/mol, and a polydispersity of 3.0. The Brookfield viscosity thereof was 12,000 cP at 150°C, the T_g thereof was measured to be -60°C using DSC, and the Si content thereof was measured to be 0.05 mass% using XRF.

[Preparation Example 13] Preparation of HRPB1000 modified polyisobutylene polymer

Polyisobutylene (M_n: 997 g/mol, PD=1.4, α-vinylidene: 68.3 mol% based on ¹³C-NMR, viscosity: 197 cSt at 100°C, 500 g, 0.5 mol) and maleic anhydride (52 g, 0.5 mol) were placed in a 1 L autoclave and allowed to react at 230°C for 12 hr using a mechanical stirrer. In order to remove unreacted maleic anhydride, nitrogen bubbling was performed for 2 hr, thus

obtaining polyisobutylene succinic anhydride (PIBSA-13). The conversion rate thereof was determined to be 73.4% through column chromatography.

Then, the PIBSA-13 thus prepared (500 g, 0.45 mol) and 3-aminopropyl triethoxysilane (211 g, 0.95 mol) were added to 500 mL of toluene in a 1 L autoclave and allowed to react at 120°C for 4 hr using a Dean-Stark apparatus. Unreacted compounds were removed, after which 481 g of modified polyisobutylene (yield: 84%) was obtained.

The modified polyisobutylene polymer thus obtained was determined to have a number average molecular weight of 1,175 g/mol, a weight average molecular weight of 1,986 g/mol, and a polydispersity of 1.7. The Tg thereof was measured to be -70°C using DSC, the Brookfield viscosity thereof was 220 cP at 150°C, and the Si content thereof was measured to be 1.2 mass% using XRF.

The properties of the modified polyisobutylene polymers prepared in Preparation Examples 1 to 13 are shown in Table 1 below.

[Table 1]

	Polyisobutylene		Modified polyisobutylene polymer				
	Mn	α -vinylidene (mol%)	Mn	Polydispersity	Viscosity (cP)	Si content (mass%)	Tg (°C)
Preparation Example 1	351	80.6	633	1.3	7	2.2	-75
Preparation Example 2	595	81.0	819	1.8	85	1.8	-72

Preparation Example 3	742	87.0	1,088	1.9	106	1.5	-70
Preparation Example 4	992	88.3	1,266	1.7	250	1.3	-69
Preparation Example 5	1,312	89.6	1,929	1.8	375	1.1	-69
Preparation Example 6	1,792	87.5	1,950	2.4	776	0.9	-70
Preparation Example 7	1,980	81.4	2,209	2.4	882	0.8	-68
Preparation Example 8	2,264	85.7	2,350	2.4	900	0.7	-68
Preparation Example 9	3,213	85.2	3,862	2.6	4,600	0.1	-65
Preparation Example 10	5,897	87.0	6,187	3.2	9,100	0.05	-60
Preparation Example 11	260	78.4	543	1.4	4	2.3	-75
Preparation Example 12	7,096	80.4	7,641	3.0	12,000	0.05	-60
Preparation Example 13	997	68.3	1,175	1.7	220	1.2	-70

Preparation of rubber composition (Examples 1 to 10 and Comparative Examples 1 to 5)

[Example 1]

5 A rubber base including 48.1 parts by weight of styrene butadiene rubber 1 (styrene: 25%, vinyl: 63%, TDAE: 37.5 phr, SBR1), 45 parts by weight of styrene butadiene rubber 2 (styrene: 10%, vinyl: 39%, SBR2), and 20 parts by weight of butadiene rubber (KBR-01, Kumho Petrochemical, BR) was
10 prepared, and, based on 100 parts by weight of the rubber base, 5 parts by weight of carbon black, 70 parts by weight of silica (US7000GR, Evonik, CTAB 165 m²/g), 5.6 parts by weight of Si-69 as a silane-coupling agent, 3.0 parts by weight of zinc oxide, 2.0 parts by weight of stearic acid, 1.6 parts by
15 weight of sulfur as a vulcanizing agent (Miwon Chemical), 1.6

parts by weight of CBS (N-cyclohexyl-2-benzothiazylsulfenamide) as a vulcanization accelerator, 2 parts by weight of DPG (1,3-diphenylguanidine) as a vulcanization accelerator, and 6.875 parts by weight of the modified polyisobutylene prepared in Preparation Example 1 were mixed using an internal Banbury mixer to afford a masterbatch, after which mixed rubber was manufactured using an open twin-screw roll mill, followed by vulcanization at 165°C for 10 min, thereby manufacturing rubber. The components and amounts of the rubber composition are shown in Table 2 below.

[Example 2]

Rubber was manufactured in the same manner as in Example 1, with the exception that the modified polyisobutylene polymer prepared in Preparation Example 2 was used in lieu of the modified polyisobutylene polymer prepared in Preparation Example 1.

[Example 3]

Rubber was manufactured in the same manner as in Example 1, with the exception that the modified polyisobutylene polymer prepared in Preparation Example 3 was used in lieu of the modified polyisobutylene polymer prepared in Preparation Example 1.

[Example 4]

Rubber was manufactured in the same manner as in Example 1, with the exception that the modified polyisobutylene polymer prepared in Preparation Example 4 was used in lieu of the modified polyisobutylene polymer prepared in Preparation Example 1.

[Example 5]

10 Rubber was manufactured in the same manner as in Example 1, with the exception that the modified polyisobutylene polymer prepared in Preparation Example 5 was used in lieu of the modified polyisobutylene polymer prepared in Preparation Example 1.

15

[Example 6]

Rubber was manufactured in the same manner as in Example 1, with the exception that the modified polyisobutylene polymer prepared in Preparation Example 6 was used in lieu of the modified polyisobutylene polymer prepared in Preparation Example 1.

20

[Example 7]

Rubber was manufactured in the same manner as in Example 1, with the exception that the modified polyisobutylene

25

polymer prepared in Preparation Example 7 was used in lieu of the modified polyisobutylene polymer prepared in Preparation Example 1.

5 **[Example 8]**

Rubber was manufactured in the same manner as in Example 1, with the exception that the modified polyisobutylene polymer prepared in Preparation Example 8 was used in lieu of the modified polyisobutylene polymer prepared in Preparation
10 Example 1.

[Example 9]

Rubber was manufactured in the same manner as in Example 1, with the exception that the modified polyisobutylene
15 polymer prepared in Preparation Example 9 was used in lieu of the modified polyisobutylene polymer prepared in Preparation Example 1.

[Example 10]

20 Rubber was manufactured in the same manner as in Example 1, with the exception that the modified polyisobutylene polymer prepared in Preparation Example 10 was used in lieu of the modified polyisobutylene polymer prepared in Preparation Example 1.

25

[Comparative Example 1]

Rubber was manufactured in the same manner as in Example 1, with the exception that a rubber composition was prepared by mixing 6.875 parts by weight of rubber-compounding oil (TDAE), rather than the polymer of Preparation Example 1.

[Comparative Example 2]

Rubber was manufactured in the same manner as in Example 1, with the exception that a rubber composition was prepared by mixing 6.875 parts by weight of commercially available terpene resin (terpene phenol resin, Yasuhara Chemical T160), rather than the polymer of Preparation Example 1.

[Comparative Example 3]

Rubber was manufactured in the same manner as in Example 1, with the exception that a rubber composition was prepared by mixing the modified polyisobutylene polymer prepared in Preparation Example 11, rather than the polymer of Preparation Example 1.

20

[Comparative Example 4]

Rubber was manufactured in the same manner as in Example 1, with the exception that a rubber composition was prepared by mixing the modified polyisobutylene polymer prepared in Preparation Example 12, rather than the polymer of Preparation

Example 1.

[Comparative Example 5]

Rubber was manufactured in the same manner as in Example 1, with the exception that a rubber composition was prepared by mixing the modified polyisobutylene polymer prepared in Preparation Example 13, rather than the polymer of Preparation Example 1.

The specific components and amounts of the rubber compositions are shown in Tables 2 and 3 below.

[Table 2]

Classification (phr)		Examp e 1	Examp e 2	Examp e 3	Examp e 4	Examp e 5	Examp e 6	Examp e 7	Examp e 8	Examp e 9	Examp e 10
Polymer	SBR1	48.1	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left
	SBR2	45	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left
	BR	20	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left
Filler	Silica	70	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left
	Carbon black	5	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left
	Silane (Si-69)	5.6	Same as value	Same as value	Same as value	Same as value	Same as value	Same as value	Same as value	Same as value	Same as value

		on left	on left	on left	on left	on left	on left	on left	on left	on left	on left
Preparation Example 1	6.875	-	-	-	-	-	-	-	-	-	-
Preparation Example 2	-	6.875	-	-	-	-	-	-	-	-	-
Preparation Example 3	-	-	6.875	-	-	-	-	-	-	-	-
Preparation Example 4	-	-	-	6.875	-	-	-	-	-	-	-
Preparation Example 5	-	-	-	-	6.875	-	-	-	-	-	-
Preparation Example 6	-	-	-	-	-	6.875	-	-	-	-	-
Preparation Example 7	-	-	-	-	-	-	6.875	-	-	-	-
Preparation Example 8	-	-	-	-	-	-	-	6.875	-	-	-
Preparation Example 9	-	-	-	-	-	-	-	-	6.875	-	-
Preparation Example 10	-	-	-	-	-	-	-	-	-	-	6.875
Zinc oxide	3	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left
Stearic acid	2	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left
Final MB	Sulfur	1.6	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left
	CBS	1.6	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left	Same as value on left
	DPG	2	Same as	Same as	Same as	Same as	Same as	Same as	Same as	Same as	Same as

			value on left	value on left	value on left	value on left	value on left	value on left	value on left	value on left	value on left
--	--	--	---------------------	---------------------	---------------------	---------------------	---------------------	---------------------	---------------------	---------------------	---------------------

[Table 3]

Classification (phr)		Comparative Example 1	Comparative Example 2	Comparative Example 3	Comparative Example 4	Comparative Example 5
Polymer	SBR1	48.1	Same as value on left	Same as value on left	Same as value on left	Same as value on left
	SBR2	45	Same as value on left	Same as value on left	Same as value on left	Same as value on left
	BR	20	Same as value on left	Same as value on left	Same as value on left	Same as value on left
Filler	Silica	70	Same as value on left	Same as value on left	Same as value on left	Same as value on left
	Carbon black	5	Same as value on left	Same as value on left	Same as value on left	Same as value on left
	Silane (Si-69)	5.6	Same as value on left	Same as value on left	Same as value on left	Same as value on left
Oil & Chemical	Oil (TDAE)	6.875	-	-	-	-
	Terpene resin	-	6.875	-	-	-
	Preparation Example 11	-	-	6.875	-	-
	Preparation Example 12	-	-	-	6.875	-
	Preparation Example 13	-	-	-	-	6.875
	Zinc oxide	3	Same as value on left	Same as value on left	Same as value on left	Same as value on left
	Stearic acid	2	Same as value on left	Same as value on left	Same as value on left	Same as value on left
Final MB	Sulfur	1.6	Same as value on left	Same as value on left	Same as value on left	Same as value on left
	CBS	1.6	Same as value on left	Same as value on left	Same as value on left	Same as value on left
	DPG	2	Same as value on left	Same as value on left	Same as value on left	Same as value on left

The properties of the prepared rubber compositions were measured through the following methods. Considering that some properties of the rubber composition may vary depending on the surrounding environmental conditions, results obtained under the same conditions on the same day are shown in Tables 4 and 5 below.

Evaluation of Payne effect

The Payne effect is associated with storage modulus values measured at elongations of 0.02% and 20%, and the smaller the change, the better the dispersion of silica, whereby superior rolling resistance can be obtained and thus the general properties of the rubber can be improved. For the rubber manufactured in Examples and Comparative Examples, the Payne effect value was measured in a 0.02-20% strain sweep at a rate of 1 Hz at 60°C using a sample having a weight of 7 g or more and an RPA 2000 from ALPHA Technologies, and the difference in storage modulus values ($\Delta G' = G'_{20\%} - G'_{0.02\%}$) measured at elongations of 0.02% and 20% was determined, and the results thereof are shown in Tables 4 and 5 below.

Measurement of grip performance and rolling resistance through dynamic loss coefficient

The $\tan \delta$ value at 0°C corresponds to grip force, and the

higher this value, the better the grip force, whereas the $\tan\delta$ value at 60°C corresponds to rolling resistance, and the lower this value, the better the rolling resistance [M. J. Wang, Rubber. Chem. Technol., 71, 520 (1998)]. For the rubber manufactured in Examples and Comparative Examples, the dynamic loss coefficient at 0°C and 60°C using DMTS (dynamic mechanical thermal spectrometry; GABO, EPLEXOR 500N) and the glass transition temperature (T_g) were measured, and the results thereof are shown in Tables 4 and 5 below. Here, the measurement conditions were as follows: frequency: 10 Hz, strain (static strain: 3%, dynamic strain: 0.25%), and temperature: -60 to 70°C.

[Table 4]

Classification (phr)		Comparative Example 1	Comparative Example 2	Comparative Example 3	Comparative Example 4	Comparative Example 5
Payne Effect	$\Delta G'$ Unvulcanized	3.3	2.9	2.5	2.8	3.8
DMA	T _g	-30.1	-26.0	-30.5	-27.4	-29.2
	Tan δ @0°C	0.2465	0.2989	0.2507	0.2728	0.2656
	Tan δ @60°C	0.0940	0.1030	0.0955	0.0990	0.0971

15

[Table 5]

Classification (phr)		Example 1	Example 2	Example 3	Example 4	Example 5	Example 6	Example 7	Example 8	Example 9	Example 10
Payne Effect	$\Delta G'$ unvulcanized	2.3	2.0	2.1	2.1	2.0	2.1	1.9	1.8	2.1	2.4
DMA	T _g	-29.8	-29.6	-29.1	-28.9	-29.1	-28.7	-29.3	-29.7	-28.7	-28
	Tan δ @0°C	0.2557	0.2607	0.2707	0.2748	0.2779	0.2848	0.2879	0.2938	0.3018	0.2908
	Tan δ @60°C	0.093	0.089	0.085	0.083	0.082	0.084	0.086	0.087	0.089	0.093

		2	6	5	9	7	8	7	0	0	3
--	--	---	---	---	---	---	---	---	---	---	---

Tables 4 and 5 show the results of measurement of the properties of the rubber compositions of Comparative Examples 1 to 5 using the rubber-compounding oil, terpene resin or modified polyisobutylene polymer having a molecular weight falling out of the range proposed by the present inventors, and of Examples 1 to 10 using the modified polyisobutylene polymers having different molecular weights, respectively.

With reference to Tables 4 and 5, in the case of the $\Delta G'$ value representing the extent of dispersion of filler, such as carbon black and silica, Examples 1 to 10 using the modified polyisobutylene polymer synthesized from polyisobutylene having a number average molecular weight satisfying the range of 350 g/mol to 6,000 g/mol and including 80 mol% or more of α -vinylidene exhibited $\Delta G'$ of 2.4 or less, and the $\Delta G'$ value of Examples 2 to 9 using polyisobutylene having a molecular weight ranging from 600 g/mol to 3,200 g/mol was 2.1 or less, which is evaluated to be better.

On the other hand, in Comparative Examples 1 to 5 using the rubber-compounding oil, terpene resin or the modified polyisobutylene polymer synthesized from polyisobutylene having a number average molecular weight falling out of the range of 350 g/mol to 6,000 g/mol or having an α -vinylidene content of less than 80 mol%, high $\Delta G'$ of 2.5 or more was

manifested, indicating that the filler was not dispersed well. Therefore, it was confirmed that the use of the modified polyisobutylene polymer according to Examples of the present invention was capable of significantly improving the
5 dispersibility of silica and carbon black.

In Examples 1 to 10, $\tan\delta$ at 0°C , which indicates superior braking performance with an increase in the numerical value thereof, was 0.255 or more, and $\tan\delta$ at 60°C , which indicates superior rolling resistance performance with a
10 decrease in the numerical value thereof, was 0.094 or less, based on which it was confirmed that both braking performance and rolling resistance performance were improved compared to Comparative Examples.

In particular, in Examples 3 to 8, the modified
15 polyisobutylene polymer having a number average molecular weight of 900 to 3,000 g/mol was prepared from polyisobutylene having a number average molecular weight of 700 to 3,000 g/mol, and was added to the rubber composition, whereby $\tan\delta$ at 0°C was 0.2707 or more and $\tan\delta$ at 60°C was 0.0870 or less,
20 based on which it was confirmed that braking performance and rolling resistance performance were further improved.

Moreover, in Comparative Example 2 using the terpene resin in lieu of the modified polyisobutylene polymer, braking performance was improved but $\tan\delta$ at 60°C was increased to
25 0.1030 and thus rolling resistance performance was

deteriorated, compared to Comparative Example 1.

In Comparative Examples 3 to 5 using the modified polyisobutylene polymers (Preparation Examples 11 to 13) prepared from polyisobutylene in which the number average
5 molecular weight fell out of the range of 350 g/mol to 6,000 g/mol or the α -vinylidene content was less than 80 mol%, superior braking performance was obtained, but $\tan\delta$ at 60°C, which indicates improved fuel efficiency with a decrease in the value thereof, was measured to be 0.094 or more, based on
10 which it was confirmed that the rolling resistance performance was deteriorated compared to Examples, and also that the rolling resistance performance was hardly improved even when compared with Comparative Example 1.

Based on the above results, it can be concluded that,
15 when the modified polyisobutylene polymer according to the present invention is added to the rubber composition, the dispersibility of silica can be increased, thereby improving both braking performance and fuel efficiency performance.

Although the preferred embodiments of the present
20 invention have been disclosed for illustrative purposes, those skilled in the art will appreciate that various modifications, additions and substitutions are possible, without departing from the scope and spirit of the invention as disclosed in the accompanying claims.

WHAT IS CLAIMED IS:

1. A modified polyisobutylene polymer for rubber compounding, prepared by mixing:

5 polyisobutylene having isobutylene as a main chain;
unsaturated dicarboxylic anhydride; and
a silane compound;

2. The modified polyisobutylene polymer of claim 1,
10 wherein the modified polyisobutylene polymer is prepared by mixing 20 to 80 wt% of the polyisobutylene having isobutylene as the main chain, 1 to 20 wt% of the unsaturated dicarboxylic anhydride, and 1 to 60 wt% of the silane compound.

15 3. The modified polyisobutylene polymer of claim 1, wherein the polyisobutylene having isobutylene as the main chain has a number average molecular weight of 350 g/mol to 6,000 g/mol.

20 4. The modified polyisobutylene polymer of claim 1, wherein the polyisobutylene having isobutylene as the main chain has a polydispersity of 1 to 5.

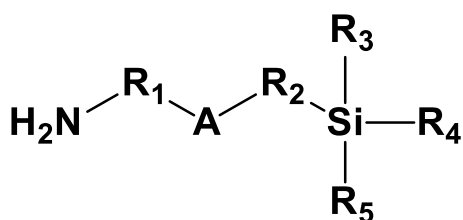
5. The modified polyisobutylene polymer of claim 1,
25 wherein the unsaturated dicarboxylic anhydride is at least one

selected from among maleic anhydride, itaconic anhydride, citraconic anhydride, propenyl succinic anhydride and 2-pentendioic anhydride.

5 6. The modified polyisobutylene polymer of claim 1, wherein the silane compound contains an amino group.

7. The modified polyisobutylene polymer of claim 6, wherein the silane compound is represented by Chemical Formula 1 below:

[Chemical Formula 1]



in Chemical Formula 1,

R₁ and R₂ are each independently selected from among (C1-15 C5) alkylene, (C1-C5) aminoalkylene, carbonylene and (C1-C5) alkyl carbonylene, and are identical to or different from each other;

R₃, R₄ and R₅ are each independently selected from among hydrogen, hydroxy, (C1-C20) alkyl, (C1-C12) cycloalkyl, (C2-20 C14) acyloxy, (C4-C20) aryloxy, (C5-C30) araloxy, (C1-C20) amine and (C1-C12) alkoxy, and are identical to or different from each other; and

A is methylene, S_n or $((R_6)NR_7)_n$, in which R_6 is hydrogen or (C1-C5) alkyl, R_7 is (C1-C5) alkylene, and n is an integer of 1 to 10.

5 8. The modified polyisobutylene polymer of claim 6,
wherein the silane compound is at least one selected from
among 3-aminopropyl triethoxysilane, 3-aminopropyl
trimethoxysilane, N-(2-aminoethyl)-3-aminopropyl
triethoxysilane, 3-aminopropyl methyldiethoxysilane, 3-
10 aminopropyl silanetriol, 3-aminopropyl methyldimethoxysilane,
3-(2-aminoethylamino)propyl dimethoxymethylsilane, 3-(2-
aminoethylamino)propyl trimethoxysilane, 2-ethanediamine N-(2-
aminoethyl)-N'-[3-(trimethoxysilyl)propyl]-1, 1-[3-
(trimethoxysilyl)propyl]urea and 1-[3-
15 (triethoxysilyl)propyl]urea.

9. The modified polyisobutylene polymer of claim 1,
wherein the modified polyisobutylene polymer contains
nitrogen.

20

10. The modified polyisobutylene polymer of claim 1,
wherein the modified polyisobutylene polymer contains a
carbonyl group.

25 11. The modified polyisobutylene polymer of claim 1,

wherein the modified polyisobutylene polymer contains an amide group.

12. The modified polyisobutylene polymer of claim 1,
5 wherein the modified polyisobutylene polymer has a Si content of 0.05 to 10 mass% based on X-ray fluorescence analysis.

13. The modified polyisobutylene polymer of claim 1,
wherein the modified polyisobutylene polymer has a glass
10 transition temperature of -50°C or less.

14. The modified polyisobutylene polymer of claim 1,
wherein the modified polyisobutylene polymer has a Brookfield
viscosity at 150°C of 5 to 10,000 cP.

15

15. The modified polyisobutylene polymer of claim 1,
wherein the modified polyisobutylene polymer has a number
average molecular weight of 580 to 10,000 g/mol.

20

16. The modified polyisobutylene polymer of claim 1,
wherein the modified polyisobutylene polymer has a
polydispersity of 1 to 5.

17. A rubber composition, comprising:
25 the modified polyisobutylene polymer of any one of claims

1 to 16;

a rubber base; and

a filler.

5 18. The rubber composition of claim 17, wherein the filler comprises at least one selected from among silica and carbon black.

10 19. The rubber composition of claim 17, wherein the rubber base comprises at least one selected from among butadiene rubber, butyl rubber, emulsion-polymerized styrene butadiene rubber (E-SBR), solution-polymerized styrene butadiene rubber (S-SBR), epichlorohydrin rubber, nitrile rubber, hydrogenated nitrile rubber, brominated polyisobutyl
15 isoprene-co-paramethyl styrene (BIMS) rubber, urethane rubber, fluorine rubber, silicone rubber, styrene-ethylene-butadiene-styrene copolymer rubber, ethylene propylene rubber, ethylene-propylene-diene monomer rubber, Hypalon rubber, chloroprene rubber, ethylene vinyl acetate rubber and acrylic rubber.

20

20. The rubber composition of claim 17, wherein the rubber composition comprises, based on 100 parts by weight of the rubber base, 50 to 150 parts by weight of silica, 5 to 20 parts by weight of carbon black, 2 to 40 parts by weight of
25 the modified polyisobutylene polymer, and 2 to 15 parts by

weight of a silane-coupling agent.