

A TRANSPARENT THERMOPLASTIC BLEND OF A CYCLOOLEFIN COPOLYMER AND A THERMOPLASTIC POLYURETHANE

FIELD OF INVENTION

The present invention relates to a transparent blend of a thermoplastic elastomer such as a polyurethane based inherently dissipative polymer (TPU-IDP), a cycloolefin copolymer, and a compatibilizing agent. The present invention also relates to blends of thermoplastic polyurethanes (TPU) and cycloolefin copolymer which are not transparent and to other inherently dissipative polymers and cycloolefin copolymers.

BACKGROUND OF THE INVENTION

Inherently dissipative polymers (IDPs) are a class of polyether elastomers which have inherent volume resistivity in the 1×10^5 to 1×10^{12} ohm-cm. Examples include polyethylene oxide-based polyether urethanes, polyether amides and polyether esters, and copolymers of ethylene oxide such as ethylene oxide/propylene oxide or ethylene oxide/epichlorohydrin. IDPs are used in alloys with other thermoplastics as a means to impart a level of conductivity sufficient to render the plastics static dissipative (surface and volume resistivities in the 1×10^5 to 1×10^{12} range). For example, U.S.-A- 5,574,104 relates to polymer compositions comprising a polyether polyurethane IDP which can be blended with one or more base polymers. The subject blends have useful static dissipative properties and exceptional cleanliness, making them suitable for handling of sensitive electronic components and devices. However, the blends are generally not transparent.

U.S.-A- 6,140,405 relates to a low molecular weight polyether oligomer which is modified with a salt, preferably containing lithium, during a chain extension reaction of the polyether oligomer to form IDP products such as polyurethanes, polyether amide block copolymers and polyether-ester block copolymers. The reaction product polymers exhibit lower surface and volume resistivities and static decay times, relative to other IDPs, yet are free of excessive amounts of extractable anions, particularly chlorine, nitrate, phosphate and sulfate.

U.S. -A- 4,332,919, 4,302,558 and 4,384,078 describe inherently dissipative acrylic polymers made by a core/shell process. This class of IDP polymers is

transparent, but suffers from high resistivity, slow static decay times and high off-gassing of volatile impurities, making it less desirable for sensitive electronic handling applications.

U.S. -A- 6,225,407 relates to a polymer blend comprising one or more cycloolefin copolymers and one or more types of core-shell particles or one or more copolymers which are composed to some extent of rubbers with low glass transition temperatures, or a combination of one or more types of core-shell particles and of one or more copolymers which are composed to some extent of rubbers with low glass transition temperatures.

U.S. -A- 6,054,533 relates to a compatibilized blend of a thermoplastic elastomer and a polyolefin. The compatibilizer is a thermoplastic polyurethane formed by the reaction of a substantially hydrocarbon intermediate such as a polybutadiene polyol, a diisocyanate such as MDI, and an amine or diol chain extender such as neopentyl glycol. The compatibilizer has high amounts of soft segments therein and imparts improved properties to blends of a thermoplastic elastomer and polyolefin such as good impact resistance, good tensile strength, good tear resistance, and good delamination resistance. These compositions are not transparent.

US-A-5,614,589 discloses that transparent methacrylate-acrylonitrile-butadiene-styrene (MABS) thermoplastic copolymers may be provided with electrostatic dissipative (ESD) properties, while maintaining or enhancing physical properties of the MABS, such as impact strength, by blending the MABS with an inherently antistatic thermoplastic urethane copolymer which is the reaction product of a hydroxy-terminated polyalkylene ether oligomer, a C2-C6 alkanediol and an aromatic diisocyanate, and compatibilizing the blend of MABS and urethane with an acrylonitrile-butadiene-styrene rubbery copolymer produced by emulsion polymerization containing at least 40 weight percent butadiene monomer and a ratio of styrene to acrylonitrile which will produce a rubbery copolymer having a refractive index in the range of about 1.53 to 1.55.

JP-A-10-050816 discloses to obtain stable volume resistivity in an antistatic region by constituting this material of olefin resin and antistatic thermoplastic urethane wherein a specified amount of metallic salt is dissolved in ion state in a fixed amount of thermoplastic urethane by using a specified amount of both thereof, respectively. As

metallic salt, 3.75wt.% of lithium perchlorate is mixed to 100wt.% of ether thermoplastic urethane for preparing the antistatic thermoplastic urethane is prepared.

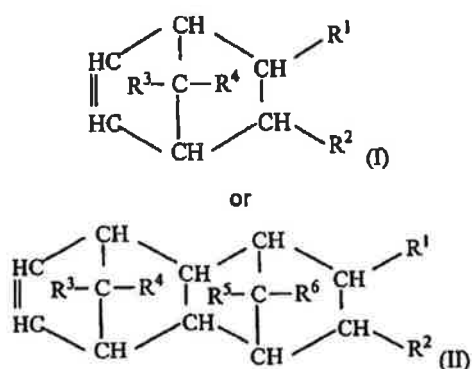
JP-A-2003-026752 discloses to provide a polyurethane-based shock-absorbing material satisfying both high hardness and high attenuation that are usually contradictory characteristics to each other. The shock-absorbing material comprises as its ingredient a polyurethane obtained by a polyaddition reaction of a polyol compound having, as a recurring unit, a specific cyclic-structure exemplified by the formula to a polyisocyanate compound.

SUMMARY OF THE INVENTION

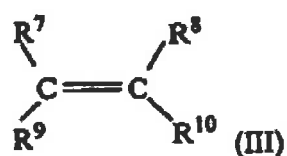
Blends of cycloolefin copolymers and thermoplastic polyurethanes (TPU) are also provided which have excellent physical properties and can be made to have electrical dissipative properties by the selection of the TPU. The cycloolefin copolymers are also blended with other inherently dissipative polymers (non-TPU) to give electrical dissipative properties. Examples of other IDP polymers which are blended with the cycloolefin copolymers include polyether amides, polyether esters, copolymers of ethylene oxide and propylene oxide, and copolymers of ethylene oxide and epichlorohydrin.

In particular, the present invention relates to an electrostatic dissipating thermoplastic composition, comprising:

(a) a cycloolefin copolymer derived from at least one polycyclic olefin monomer and at least one acyclic 1-olefin monomer, said polycyclic olefin monomer having the formula

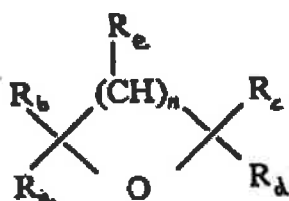


wherein each R^1 , R^2 , R^3 , R^4 , R^5 , and R^6 , independently, are the same or different, and is a hydrogen atom or a hydrocarbon radical, and wherein said acyclic 1-olefin has the formula



wherein each R^7 , R^8 , R^9 , and R^{10} , independently, are the same or different, and is a hydrogen atom or a C_6 - C_{10} aryl group or a C_1 - C_8 alkyl group.

(b) a thermoplastic polyurethane derived from one or more cyclic ether monomers, at least one diisocyanate, and at least one diol, wherein said cyclic ether monomer has the formula



wherein each R_a , R_b , R_c , R_d , and R_e , independently, is a hydrogen atom, an unsubstituted or substituted alkyl, cycloalkyl, cycloalkenyl, aryl, aralkyl or alkaryl, and wherein said substituents which can be substituted within the foregoing are OR_f , SR_f , CN or a halogen, wherein R_f is hydrogen, alkyl, cycloalkyl, cycloalkenyl, aryl, aralkyl, alkaryl, or carboxyl, and n is 0, 1, 2, or 4, and

(c) a compatibilizing agent comprising a thermoplastic polyurethane derived from a diisocyanate, a diol, and a substantially hydrocarbon intermediate containing at least 20 carbon atoms between non-carbon atoms in the intermediate backbone or a hydrocarbon intermediate derived from one or more dienes having a total of from 4 to 8 carbon atoms, and wherein said thermoplastic polyurethane has a surface resistivity of from 1×10^6 to 1×10^{12} ohm/square as measured by ASTM D-257.

Preferred embodiments are apparent from the dependent claims.

DETAILED DESCRIPTION

The thermoplastic polyurethane (TPU) of the present invention can be conventional TPUs which generally have poor electrostatic dissipating properties or other TPUs which are an inherently dissipative polymer (TPU-IDP).

Conventional TPU

Suitable typical or conventional TPUs which are utilized as a blend polymer are made by reacting a hydroxyl terminated polyester intermediate, or preferably a hydroxyl terminated polyether; at least one polyisocyanate; and one or more chain extenders. The hydroxyl terminated polyester intermediate polymer is generally a linear polyester having a number average molecular weight of from 300 to 10,000 and preferably from 500 to 5,000. The molecular weight is determined by assay of the terminal functional groups. The polymers are produced by (1) an esterification reaction of one or more glycols with one or more dicarboxylic acids or anhydrides or (2) by transesterification reaction, i.e., the reaction of one or more glycols with esters of dicarboxylic acids. Mole ratios generally in excess of more than one mole of glycol to acid are preferred so as to obtain linear chains having a preponderance of terminal hydroxyl groups.

A suitable reaction for the formation of a polyester intermediate also includes a ring opening polymerization which can utilize various lactones such as epsilon-caprolactone and can be initiated with a bifunctional initiator such as diethylene glycol.

The dicarboxylic acids of the polyester intermediate can be aliphatic, cycloaliphatic, aromatic, or combinations thereof. Suitable dicarboxylic acids which may be used alone or in mixtures generally have a total of from 4 to 15 carbon atoms and include: succinic, glutaric, adipic, pimelic, suberic, azelaic, sebacic, dodecanedioic, isophthalic, terephthalic, Cyclohexanoic. Anhydrides of the above dicarboxylic acids such as phthalic anhydride, tetrahydrophthalic anhydride, can also be used. Adipic acid is the preferred acid. The glycols which are reacted to form a desirable polyester intermediate can be aliphatic, aromatic, or combinations thereof, and have a total of from 2 to about 12 carbon atoms, and include ethylene glycol, propylene-1,2-glycol, 1,3-propanediol, butylene-1,3-glycol, 1,4-butanediol, 1,5-pentanediol, 1,6-hexanediol, 2,2-dimethylpropane-1,3-diol, 2,2-diethylene-1,3-diol, 1,4-cyclohexanedimethanol, decamethylene glycol, dodecamethylene glycol. 1,4-butanediol is the preferred glycol.

The preferred polyether polyol intermediates are derived from a diol or polyol having a total of from 2 to 15 carbon atoms, for example, an alkyl diol or glycol which is reacted with an ether comprising an alkylene oxide having from 2 to 6 carbon atoms, typically ethylene oxide or propylene oxide or mixtures thereof. For example, hydroxyl functional polyether can be produced by first reacting propylene glycol with propylene oxide followed by subsequent reaction with ethylene oxide. Primary hydroxyl groups resulting from ethylene oxide are more reactive than secondary hydroxyl groups and thus are preferred. Useful commercial polyether polyols include poly(ethylene glycol) comprising ethylene oxide reacted with ethylene glycol, poly(propylene glycol) comprising propylene oxide reacted with propylene glycol, poly(propylene-ethylene glycol) comprising propylene oxide and ethylene oxide reacted with propylene glycol, poly(tetramethylene ether glycol) comprising water reacted with tetrahydrofuran (PTMEG), glycerol adduct comprising glycerol reacted with propylene oxide, trimethylpropane adduct comprising trimethylolpropane reacted with propylene oxide, pentaerythritol adduct comprising pentaerythritol reacted with propylene oxide, and

similar hydroxyl functional polyethers. The various polyether intermediates generally have a number average molecular weight, as determined by assay of the terminal functional groups of from 200 to 10,000 and preferably from 500 to 5,000.

The desired thermoplastic polyurethane (blend polymer) is made from the above-noted intermediate such as a hydroxyl terminated polyester or polyether which is further reacted with a polyisocyanate, preferably a diisocyanate, along with extender glycol. Examples of suitable diisocyanates generally have the formula $R(NCO)_n$ wherein n equals 2, 3, or 4 with 2 being highly preferred. Mixtures of various polyisocyanates can also be utilized and thus need not be an integer. R is an aliphatic, an aromatic, or combinations thereof having a total of from 2 to 30 carbon atoms with from 6 or 8 to 15 being preferred. Examples of suitable diisocyanates include non-hindered aromatic diisocyanates such as: 4,4'-methylenebis-(phenyl isocyanate) (MDI); isophorone diisocyanate (IPDI), m-xylylene diisocyanate (XDI), toluene diisocyanate, phenylene-1,4-diisocyanate, naphthalene-1,5-diisocyanate, as well as non-hindered cyclic aliphatic diisocyanates such as 1,4-cyclohexyl diisocyanate (CHDI), decane-1,10-diisocyanate, diphenylmethane-3,3'-dimethoxy-4,4'-diisocyanate, dicyclohexylmethane-4,4'-diisocyanate, and cyclohexyl-1,4-diisocyanate. MDI is highly preferred.

Examples of suitable extender glycols (i.e., chain extenders) are lower aliphatic or short chain glycols having from 2 to 10 carbon atoms and include for instance ethylene glycol, diethylene glycol, propylene glycol, dipropylene glycol, 1,4-butane diol (highly preferred), 1,6-hexane diol, 1,3-butane diol, 1,5-pentane diol, 1,4-cyclohexane-dimethanol, neopentyl glycol, hydroquinone di(hydroxyethyl)ether and 2-methyl-1,3-propanediol. Amine chain extenders are avoided inasmuch as they generally do not result in good properties.

While a two-step reaction can be utilized such as reacting the intermediate with generally an equivalent weight amount of a diisocyanate and subsequently chain extending the same, the one-shot process is preferred. That is, the one or more intermediates, the one or more diisocyanates, and the one or more chain extenders are added to a reaction vessel and heated in the presence of a suitable catalyst to a

temperature above 100°C and usually above 125°C. Inasmuch as the reaction is exothermic, the reaction temperature increases to 200°C to 260°C or 290°C wherein the various components react with one another. The catalysts are conventional and include tin catalysts such as stannous octoate, dibutyl tin dilaurate, dibutyl tin dioctate as well as other metal carboxylate compounds. On a mole basis, the amount of extender glycol for each mole of the polyol intermediate is from 0.1 to 3.0, desirably from 0.2 to 2.0 and preferably from 0.5 to 1.5 moles. Inasmuch as amine chain extenders are not desired, the amount thereof is low, for example 0.5 moles or less, desirably 0.2 moles or less, and preferably 0.1 moles or less and most preferably nil, that is none at all. On a mole basis, the high molecular weight polyurethane polymer produced by the one-shot process comprises from 0.96 to 1.04 and preferably from 0.98 to 1.02 moles of the diisocyanate for every 1.0 total moles of both the chain extender and the intermediate, e.g., polyester or polyether.

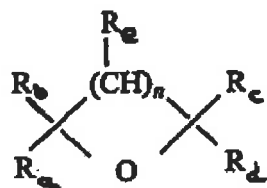
As noted, the preferred intermediate is a polyether, while MDI is the preferred isocyanate and 1,4-butane diol is the preferred chain extender.

The above conventional thermoplastic polyurethanes as well as the below TPU-IDPs are generally clear and have an index of refraction of from 1.48 to 1.58, desirably from 1.50 to 1.56 and preferably from 1.52 to 1.54.

TPU-IDP

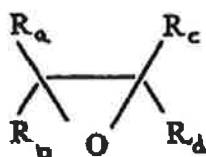
Often, as noted above, it is desirable to use a thermoplastic polyurethane based inherently dissipative polymer, TPU-IDP, composition which can contain an electrostatic dissipating agent. The TPU-IDP composition comprises a low molecular weight polyether oligomer having two reactive moieties which is reacted with a non-hindered diisocyanate and also with a chain extender, generally simultaneously, (i.e. a one-shot polymerization process).

The polyether oligomer of the TPU-IDP composition generally is derived from one or more copolymerizable cyclic ether monomers having the formula:



wherein R_a , R_b , R_c , R_d , and R_e are independently hydrogen, unsubstituted or substituted alkyl, cycloalkyl, cycloalkenyl, aryl, aralkyl or alkaryl, and wherein the substituents which can be substituted within the foregoing are OR_f , SR_f , CN or halogens, where R_f is hydrogen, alkyl, cycloalkyl, cycloalkenyl, aryl, aralkyl, alkaryl, or carboxyl, and further wherein the reactive moieties are OH, NH_2 , or NHR_f , and n is 0, 1, 2, or 4. The reaction of the low molecular weight oligomer with a diisocyanate will furnish a polymer having melt indices from 0.05 to 110 grams per ten minutes. The preferred melt index range of the polymer will be from 1.0 to 65 grams/10 minutes. In general, the melt index is determined according to ASTM D-1238 Procedure A at a barrel temperature of 190°C and an 8,700 gram piston load.

In a preferred embodiment the low molecular weight oligomer employed is a polymer of cyclic ether monomers having the formula:



In a more preferred embodiment the low molecular weight oligomer employed is a polymer of cyclic ether monomers wherein R_a , R_b and R_c are hydrogen and R_d is H, CH_3 , or $CH_2 X$ wherein X is a halogen, OR_f , or $COOR_f$, and R_f is defined herein-above.

The most preferred cyclic ether is ethylene oxide.

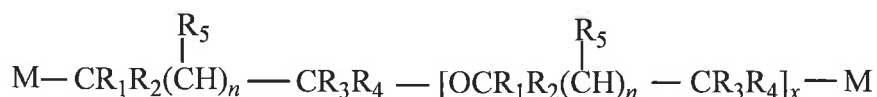
In an alternative embodiment the low molecular weight polyether oligomer is end capped with ethylene oxide thereby providing an oligomer which will have two primary moieties.

The alkyl groups of the above cyclic ether formulas can have from 1 to 6 carbon atoms, be straight or branched chain and may be unsubstituted or substituted. The alkenyl groups can have from 1 to 6 carbon atoms, be straight or branched chain, have 1 or 2 double bonds and be unsubstituted or substituted.

The cycloalkyl and cycloalkenyl groups can have from 3 to 8 ring carbon atoms and from 1 to 3 rings. The cycloalkenyl groups can have 1 or 2 double bonds.

The aryl groups can have from 6 to 10 ring carbon atoms and one or two rings.

Useful polyether oligomers are linear polymers having the general formula:



wherein X+1 is the number of repeating ether units, each M is a reactive moiety, n is 0,1,2, or 4 and R₁, R₂, R₃, R₄ and R₅ are as defined hereinabove. The most common M is the OH group. For the subject invention X+1 is at least 4 and between 4 and 250. On a molecular weight basis, the useful range of polyether oligomers have a number average molecular weight from 200 to 10,000 and preferably from 500 to 5,000. A highly preferred oligomer is polyethylene glycol. Commercially available polyethylene glycols useful in this invention are typically designated as polyethylene glycol 600, polyethylene glycol 1,450, and polyethylene glycol 4,000.

The polyether oligomer can be a homopolymer or a copolymer of two or more copolymerizable monomers. Some examples of comonomers are ethylene oxide, propylene oxide, 1,2-butylene oxide, epichlorohydrin, allyl glycidyl ether, n-butyl glycidyl ether, glycidyl acrylate, glycidyl methacrylate, 2-ethylhexyl glycidyl ether, tetrahydrofuran, or styrene oxide.

In accordance with the present invention, the low molecular weight polyether oligomer intermediate and the non-hindered diisocyanate are co-reacted simultaneously in a one-shot polymerization process at a temperature above 100°C and usually 120°C, whereupon the reaction is exothermic and the reaction temperature is increased to 200°C. to 285°C.

The glycol chain extender can be any diol (i.e., glycol) or combination of diols, containing 2 to 10 carbon atoms, such as ethylene glycol, 1,3-propanediol, propylene glycol, 1,4-butanediol, 1,3-butanediol, 1,5-pentanediol, 1,6-hexanediol, diethylene glycol, 1,4-cyclohexane dimethanol, neopentyl glycol, hydroquinone bis(2-hydroxyethyl) ether, or any aliphatic or aromatic molecule containing two reactive hydroxyl groups. The preferred chain extender is 1,4-butanediol.

The hydroxyl terminated polyols described above can be blended with a glycol chain extender before the blend is reacted with a polyisocyanate or the polyol and the chain extender can be brought to the reaction zone simultaneously. Less desired, the polyol can be reacted with the diisocyanate, and then the prepolymer is reacted with the chain extender. Stabilizers such as antioxidants can be added prior to the reaction or during the reaction.

The amount of glycol chain extender is generally from 0 or 0.1 to 35 moles and desirably from 0 or 0.1 to 20 moles for every mole of low molecular weight polyether oligomer. Generally, the number of moles of diisocyanate per total of the number of moles of the low molecular weight polyether oligomer plus the number of moles of chain extender is from 0.95 to 1.06 and preferably from 0.97 to 1.03.

In an alternative procedure two or more of the polyether oligomers can be reacted with a diisocyanate to furnish an oligomer dimer or trimer. These dimers or trimers can then be chain extended under similar conditions to form the high molecular weight polymer. This procedure can be used to produce a high molecular weight chain extended polymer with varying polyisocyanate groups.

Conventional diisocyanate or polyisocyanate type components are molecules having two functional groups (reactive sites) which will react with the reactive moieties of the polyethers.

The reactive moieties typically occur at the ends of the polyether oligomers as a result of routine synthesis, however the reactive moieties can be located at locations other than the ends. The reactive moieties most useful for the present invention are OH, NH₂ and NHR₆. In a preferred form the reactive moieties are OH, NH₂ or NHR₆ and are on primary carbon atoms. The most preferred reactive moiety is OH.

Any conventional diisocyanate can be used, either aliphatic or aromatic. The polyisocyanates generally have the formula $R(NCO)_n$ wherein n is 2, 3, or 4, or mixtures of polyisocyanates wherein n need not be an integer and preferably is 2. R is an aliphatic, aromatic, or combinations thereof having a total of from 2 to 30 carbon atoms with from 6 or 8 to 15 carbon atoms being preferred. In a preferred embodiment diisocyanates are utilized. Suitable diisocyanates include, for example, 1,4-diisocyanatobenzene (PPDI), 4,4'-methylenebis(phenyl isocyanate) (MDI), 4,4'-methylenebis(3-methoxy phenyl isocyanate), isophorone diisocyanate (IPDI), 1,5-naphthalene diisocyanate (NDI), phenylene-1,4-diisocyanate, toluene diisocyanate (TDI), m-xylene diisocyanate (XDI), 1,4-cyclohexyl diisocyanate (CHDI), 1,10-diisocyanatonaphthylene, and 4,4'-methylenebis-(cyclohexyl isocyanate) (H_{12} MDI). The most preferred diisocyanate is MDI.

Preferred TPU-IDP compositions thus contain a polyether intermediate which is polyethyleneoxide glycol, a diisocyanate which preferably is MDI and a chain extender which preferably is butane diol. Preferred TPU-IDP compositions are also set forth in U.S. -A- 5,574,104.

The TPU-IDP compositions can contain a small amount of a salt such as an electrostatic dissipating agent and such compositions are generally preferred. The salt can generally be added to any existing TPU-IDP composition such as those set forth herein above.

Accordingly, it is an important aspect of the present invention to utilize an effective amount of a salt, a salt complex or a salt compound formed by the union of a metal ion with a non-metallic ion or molecule. These salts are preferably added during the one-shot polymerization process. While the exact mechanism of attachment and/or attraction of the salt to the TPU-IDP reaction product is not completely understood, the salt unexpectedly improves the surface and volume resistivities of the resulting polymer without the presence of unacceptably high levels of extractable anions. Moreover, the static decay times remain in an acceptable range. Examples of salts useful in the subject invention include but are not limited to: $LiClO_4$, $LiN(CF_3SO_2)_2$, $LiPF_6$, $LiAsF_6$, LiI , $LiBr$, $LiSCN$, $LiSO_3CF_3$, $LiNO_3$, $LiC(SO_2CF_3)_3$, Li_2S , and $LiMR_4$ where M is Al or B, and R is a halogen, alkyl or aryl group. The preferred salt is $LiN(CF_3SO_2)_2$, which is

commonly referred to as lithium trifluoromethane sulfonimide. The effective amount of the selected salt added to the one-shot polymerization is at least 0.10 parts to 10 parts by weight based on 100 parts by weight of the TPU-IDP composition, desirably at least 0.25 parts to 7 parts by weight and preferably at least 0.75 parts to 5 parts by weight.

In accordance with another important feature of the present invention, it has been discovered that the amount of salt utilized can be unexpectedly lowered when the salt is added in conjunction with an effective amount of a co-solvent and that lower surface and volume resistivities are obtained. That is, the polyether polyol intermediate is considered to be a solvent. Examples of co-solvents suitable for this purpose include but are not limited to ethylene carbonate, propylene carbonate, dimethyl sulfoxide, tetramethylene sulfone, tri- and tetra ethylene glycol dimethyl ether, gamma butyrolactone, and N-methyl-2-pyrrolidone. Ethylene carbonate is preferred. Although the addition of one of the co-solvents is optional, in some applications lower amounts of the salt may be desirable. The effective amount of co-solvent required to achieve the desired result of lower salt usage while still attaining the desired properties in the TPU-IDP, is at least 0.10 parts to 20 parts by weight based on 100 parts of the TPU-IDP, preferably at least 0.50 parts to 15 parts, and most preferably at least 1.0 parts to 7 parts by weight.

The TPU-IDP composition also has small amounts of impurities such as anions therein. For example the total amount of extractable anions of chlorine anions, nitrate anions, phosphate anions, and sulfate anions, is generally less than 8,000 parts, desirably less than 7,000 parts, and preferably less than 6,000 parts by weight per one billion total parts by weight of total extractable solution; generally the parts by weight of extracted chlorine anions per billion parts by weight of total extractable solution is less than 1,000 parts, desirably less than 700 parts, and preferably less than 350 by weight; generally the parts by weight of extractable nitrate ions per billion parts by weight of extractable solution is less than 100 parts, desirably less than 90 parts, and preferably less than 75 parts by weight; generally the parts by weight of extractable phosphate ions per billion parts by weight of extractable solution is less than 6,000 parts, desirably less than 5,500 parts, and preferably less than 5,000 parts by weight; and generally the parts by weight of extractable sulfate ions per billion parts by weight of extractable solution is

less than 1,000 parts, desirably less than 750 parts, and preferably less than 500 parts by weight per billion parts by weight. The extractable ions was determined by placing an 8X2 cm sample in 10 ml of water for 60 minutes at 80°C. The ion content was analyzed via ion chromatography. Hence, the above results are reported by parts per billion per total extracted water.

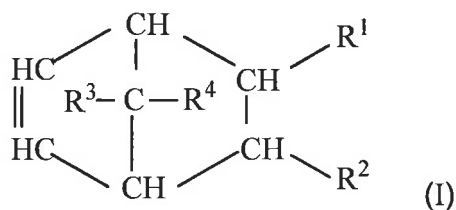
The use of lithium salts either alone or in association with a solvent is described in detail in U.S. -A- 6,140,405.

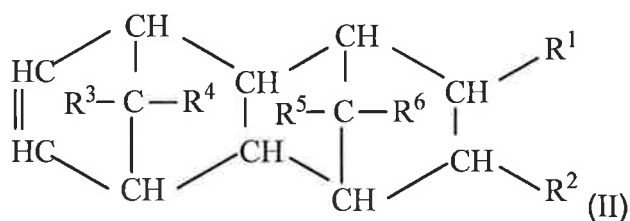
The electrostatic dissipative compositions of the present invention which preferably are thermoplastic urethanes have good surface resistivity and volume resistivity as measured by ASTM D-257 For example, the above TPU-IDP compositions whether or not they contain a lithium salt generally have a surface resistivity of from 1×10^6 to 1×10^{12} ohm/square, desirably from 1×10^7 to 1×10^{11} ohm/square and preferably from 1×10^8 to 1×10^{10} ohm/square and a volume resistivity of from 1×10^6 to 1×10^{12} ohm-centimeter, desirably from 1×10^7 to 1×10^{11} ohm-centimeter and preferably from 1×10^9 to 5×10^{10} ohm-centimeter.

IDP polymers other than TPU-IDP, as described above, may be used in the blends of this invention. IDP polymers such as polyether amides (commercially available as Pebax[®] from Atofina), polyether esters, copolymers of ethylene oxide and propylene oxide, and copolymers of ethylene oxide and epichlorohydrin may be used. The most preferred IDP is a TPU-IDP as described above, especially if a transparent blend is desired.

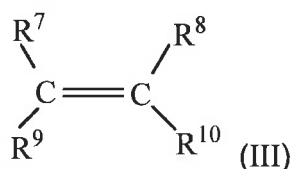
COC

The cycloolefin copolymers of the present invention are prepared by polymerizing from 0.1% to 99.9% by weight, based on the total amount of the monomers, of at least one polycyclic olefin of the formula I and/or II





where each R^1 , R^2 , R^3 , R^4 , R^5 , and R^6 are identical or different and are a hydrogen atom or a hydrocarbon radical, where the same radicals in the different formulae may have a different meaning; and from 0.1 to 99% by weight, based on the total amount of the monomers, of at least one acyclic 1-olefin of the formula III



wherein each R^7 , R^8 , R^9 and R^{10} are identical or different and are a hydrogen atom or a hydrocarbon radical, preferably a C_6 - C_{10} -aryl radical or a C_1 - C_8 alkyl radical, with ethylene or propylene being preferred.

Preference is given to cycloolefins of the formulae I or II where each R^1 , R^2 , R^3 , R^4 , R^5 , R^6 are identical or different and are a hydrogen atom or a hydrocarbon radical, in particular a (C_6-C_{10}) -aryl radical or a (C_1-C_8) -alkyl radical, where the same radicals in the different formulae may have a different meaning.

Particularly preferred polycyclic olefins are norbornene and tetracyclododecene, where these can optionally have C_1 - C_6 -alkyl substitution. They are preferably copolymerized with ethylene.

The amount of the one or more acyclic one-olefin monomers is generally from 0.1% to 99%, desirably from 45% to 85%, more desirably from 55% to 80%, and preferably from 60% to 70 mole% based upon the total number of moles of said one or more acyclic one-olefin monomers and said one or more polycyclic olefin monomers of Formulas I and/or II.

The novel polymer blend is characterized in that the cycloolefin copolymer(s) present are prepared by the process described below. The process for preparing the

cycloolefin copolymers present in the novel polymer blend is described in detail in DE-A-196 52 340.

The process according to the invention for preparing a cycloolefin copolymer encompasses the polymerization of at least one polycyclic olefin monomer with at least one acyclic 1-olefin, in the presence of a catalyst system. The catalyst system to be used for preparing the cycloolefin copolymer present in the novel polymer blend comprises at least one transition metal compound. Preference is given to the use of one or more metallocenes as transition metal compound.

The polymerization is carried out in the liquid cycloolefin itself or in a cycloolefin solution. The pressure is usually 1 bar.

The catalyst system to be used in preparing the cycloolefin copolymer present in the novel polymer blend may moreover comprise one or more cocatalysts.

The catalyst system to be used for preparing the cycloolefin copolymer present in the novel polymer blend is a high-activity catalyst for olefin polymerization. Preference is given to using a metallocene and a cocatalyst. It is also possible to use mixtures of two or more metallocenes, particularly for preparing reactor blends or polyolefins with a broad or multimodal molar mass distribution.

The process for preparing the cycloolefin copolymer present in the novel polymer blend, and also the catalyst system to be used for this process, are described in detail in DE-A-1 96 52 340.

The cocatalyst present in the catalyst system to be used for preparing the cycloolefin copolymer present in the novel polymer blend preferably comprises an aluminoxane.

Examples of the metallocenes to be used according to the invention are:
isopropylene(1-indenyl)(3-isopropylcyclopentadienyl)zirconium dichloride,
diphenylmethylene(1-indenyl)(3-isopropylcyclopentadienyl)zirconium dichloride,
methylphenylmethylene(1-indenyl)(3-isopropylcyclopentadienyl)zirconium dichloride,
isopropylene(1-indenyl)(3-tert-butylcyclopentadienyl)zirconium dichloride,
diphenylmethylene(1-indenyl)(3-tert-butylcyclopentadienyl)zirconium dichloride,
methylphenylmethylene(1-indenyl)(3-tert-butylcyclopentadienyl)zirconium dichloride,
isopropylene(4,5,6,7-tetrahydro-1-indenyl)(3-isopropylcyclopentadienyl)-zirconium

dichloride, diphenylmethylen(4,5,6,7-tetrahydro-1-indenyl)(3-isopropylcyclopentadienyl)zirconium dichloride, methylphenylmethylen(4,5,6,7-tetrahydro-1-indenyl)(3-isopropylcyclopentadienyl)zirconium dichloride, isopropylene(1-indenyl)(3-trimethylsilylcyclopentadienyl)zirconium dichloride, diphenylmethylen(1-indenyl)(3-trimethylsilylcyclopentadienyl)zirconium dichloride, methylphenylmethylen(1-indenyl)(3-trimethylsilylcyclopentadienyl)-zirconium dichloride, isopropylene(4,5,6,7-tetrahydro-1-indenyl)(3-tert-butylcyclopentadienyl)-zirconium dichloride, diphenylmethylen(4,5,6,7-tetrahydro-1-indenyl)(3-tert-butylcyclopentadienyl)zirconium dichloride, methylphenylmethylen(4,5,6,7-tetrahydro-1-indenyl)(3-tert-butylcyclopentadienyl)zirconium dichloride, isopropylene(4,5,6,7-tetrahydro-1-indenyl)(3-trimethylsilylcyclopentadienyl)-zirconium dichloride, diphenylmethylen(4,5,6,7-tetrahydro-1-indenyl)(3-trimethylsilylcyclopentadienyl)zirconium dichloride, methylphenylmethylen(4,5,6,7-tetrahydro-1-indenyl)(3-trimethylsilylcyclopentadienyl)zirconium dichloride.

Another possible embodiment of the process according to the invention uses a salt-type compound of the formula $R_x NH_{4-x} BR'_4$ or of the formula $R_3 PHBR'_4$ as cocatalyst instead of or in addition to an aluminoxane.

Here, $x=1, 2$ or 3 , R =alkyl or aryl, identical or different, and R' =aryl, which may also have been fluorinated or partially fluorinated. In this case the catalyst is composed of the reaction product of a metallocene with one of the compounds mentioned (EP-A-0 277 004).

Any solvent added to the reaction mixture is a common inert solvent, such as an aliphatic or cycloaliphatic hydrocarbon, a gasoline fraction or hydrogenated diesel oil fraction, or toluene.

The metallocenes are preferably used in the form of their racemates. The metallocene is preferably used at a concentration, based on the transition metal, of from 10^{-1} to 10^{-8} mol, preferably from 10^{-2} to 10^{-7} mol, particularly preferably from 10^{-3} to 10^{-7} mol, of transition metal per dm^3 of reactor volume. The aluminoxane is used at a concentration of from 10^{-4} to 10^{-1} mol, preferably from 10^{-4} to $2 \cdot 10^{-2}$ mol, per dm^3 of reactor volume, based on the aluminum content. In principle, however, higher concentrations are also possible.

While the cycloolefin copolymers can have glass transition temperatures of up to 250°C, preferably they have a processing temperature range such that it is compatible with the TPU polymer or the TPU-IDP polymer. Suitable cycloolefin copolymers for blending with a TPU or TPU-IDP polymer generally have a T_g of from minus 25°C to 150°C, desirably from 50°C to 125°C, and preferably from 60°C to 115°C.

The COCs suitable for the purposes of the invention have viscosity numbers (determined in decalin at 135°C) of from 25 to 200 ml/g, preferably from 40 to 120 ml/g, particularly preferably from 40 to 100 ml/g.

The cycloolefin copolymers have a particular structure, which has been described in detail in a dissertation by J. Ruchatz, Dusseldorf 1997.

Accordingly, the cycloolefin copolymers present in the novel polymer blend may have sequences of two norbornene units incorporated one after the other. Two norbornene units also correspond to the maximum possible sequence length of the cycloolefin copolymers present in the novel polymer blend.

The amount of the cycloolefin copolymer is generally from 45% to 90% by weight, desirably from 60% to 85% by weight, and preferably from 70% to 80% by weight based upon the total weight of the one or more cycloolefin copolymers and the one or more TPU, TPU-IDP, and/or other IDP polymers.

- As noted above, it is an important aspect of the present invention to produce transparent blends of the cycloolefin polymer and the thermoplastic urethane. Accordingly, a cycloolefin copolymer is utilized which generally has an index of refraction of from 1.48 to 1.58, desirably from 1.50 to 1.56 and preferably from 1.52 to 1.54. The TPU and/or TPU-IDP have a similar index of refraction so that the blend is transparent. Generally, the index of the refraction of these two components are similar and the difference between them is thus less than 0.05, desirably less than 0.03, and preferably less than 0.01. As also noted above, the TPU component or the TPU-IDP component has an index of refraction of generally from 1.48 to 1.58, desirably from 1.50 to 1.56, and preferably from 1.52 to 1.54. The light transmission of the transparent blend

for a 3.175 mm (125 mil) plaque is desirably greater than 50%, and preferably greater than 75% as measured according to ASTM D-1003-61.

Suitable cycloolefin copolymers (COC) are commercially available under the name Topas[®] from Ticona GmbH. The proper selection of the particular grade of COC will depend on the IDP polymer being blended with the COC. To obtain a clear blend, the index of refraction of the COC should be similar to the IDP polymer used. Also, the melt processing temperatures should be close enough to achieve proper intimate blending of the polymers. The proper selection of the COC is determinable by one skilled in the art of polymers compounding without undue experimentation. Additionally, the properties of the COC phase and thus the properties of the present invention can be tailored as needed by combining COC resins that have different thermal and rheological properties. Since the COC resins are miscible, properties such as glass transition temperature and viscosity can be moved by adjusting their proportions.

Compatibilizing Agent

The blends of the present invention comprise the thermoplastic polyurethane (TPU) such as an inherently dissipative thermoplastic polyurethane (TPU-IDP) composition or polymer, and the cycloolefin copolymer (COC) such as a copolymer of norbornene and an olefin. While the two components have similar processing temperatures and therefore are processable, and also have similar indices of refraction and therefore are transparent, they are generally incompatible and accordingly a compatibilizing agent is utilized to stabilize the blend and to improve properties without eliminating or significantly reducing its clarity.

The preferred compatibilizer to make a clear blend of a COC and a TPU is a thermoplastic polyurethane derived from the reaction of a substantially hydrocarbon intermediate, a diisocyanate, and a chain extender. The hydrocarbon intermediate is a low molecular weight compound or a polymer having hydroxyl (preferred), amine, or carboxylic acid terminal groups thereon. When the substantially hydrocarbon intermediate is not solely a hydrocarbon but, e.g., a polyester, the number of consecutive polymer backbone carbon atoms between a non-carbon atom such as oxygen, is large, i.e., at least 20 carbon atoms, desirably at least 30 carbon atoms, and preferably at least

45 carbon atoms to 60, or 75, or 100 carbon atoms. An example of such a substantially hydrocarbon intermediate, i.e., a long chain polyester polyol Priplast[®] 3197 from Unichema. Priplast[®] 3197 is a dimerdiol dimerate prepared from dimerdiol Pripol[®] 2033 containing at least 36 carbon atoms and a dimer acid containing about 44 carbon atoms. A suitable low molecular weight hydrocarbon intermediate is Pripol[®] 2033 from Unichema, a 36 carbon atom dimerdiol. However, the intermediate is preferably solely a hydrocarbon intermediate derived from one or more dienes having a total of from 4 to 8 carbon atoms, such as butadiene, isoprene, with butadiene being preferred. The number average molecular weight of the hydrocarbon intermediate is generally from 300 or 500 to 10,000, desirably from 1,000 to 7,500, and preferably from 2,000 to 5,000. The hydrocarbon intermediate can be unsaturated but preferably is substantially hydrogenated such that at least 80%, desirably at least 90% or 95%, and preferably at least 98% or 99%, and even 100% of the carbon-carbon double bonds in the intermediate are saturated. Hydrogenation may be carried out according to any conventional process or manner such as set forth in U.S. -A- Nos. 5,393,843 or 5,405,911. When butadiene is utilized, the microstructure of the resulting polymer can be largely 1,2 structure or 1,4 structure (e.g., 15 to 85%) with a similar amount (e.g., 35 to 65%) of each generally being preferred.

Examples of hydrocarbon polyols derived from butadiene include the following:

Identification	Supplier	Description
Kraton [®] Liquid L2203	Shell	Hydroxyl-terminated polybutadiene, hydrogenated. Approximate micro-structure: 55% 1,2; 45% 1,4.
Polytail [®] H	Mitsubishi	Hydroxyl-terminated polybutadiene, hydrogenated. Approximate micro-structure: 21% 1,2; 79% 1,4.
Polytail [®] HA	Mitsubishi	Hydroxyl-terminated polybutadiene, hydrogenated. Approximate micro-structure: 83% 1,2; 17% 1,4.
Krasol [®] LBH	Kaucuk AG	Hydroxyl-terminated polybutadiene. Approximate microstructure: 65% 1,2; 35% 1,4.
Liquiflex [®] H	Petroflex	Hydroxyl-terminated polybutadiene. Approximate microstructure: 22% 1,2; 78% 1,4.

Kraton[®] L2203 is preferred in the present invention.

The term "polyol" with respect to a substantially hydrocarbon polyol intermediate is to be understood to mean that while preferably the hydrocarbon has two functional hydroxyl end groups, the same can generally range from 1.8 to 2.2 end groups per molecule.

The isocyanates utilized in the present invention are preferably diisocyanates and include aliphatic, cycloaliphatic, aromatic, alkyl-substituted aromatic diisocyanates as well as mixtures thereof. Such diisocyanates generally have the formula $R(NCO)_n$ where R is an aliphatic, an aromatic, or combinations thereof having a total of from 2 to 30 carbon atoms with from 6 or 8 to 15 carbon atoms being preferred, and an n is from 2 to 4 with 2 being highly preferred. Representative examples include ethylene diisocyanate; toluene diisocyanate; methylene bis-(4-phenylisocyanate), that is, MDI; isophorone diisocyanate; hexamethylene diisocyanate; naphthalene diisocyanate; cyclohexylene diisocyanate; diphenylmethane-3,3' dimethoxy-4,4'-diisocyanate, meta-tetramethylxylene diisocyanate (m-TMXD1), paratetramethylxylene diisocyanate (p-TMXD1), m-xylylene diisocyanate (XDI),

decane-1,10-diisocyanate, dicyclohexylmethane-4,4'-diisocyanate, as well as combinations thereof, with MDI being preferred. It is to be understood that isomers of the various diisocyanate can also be used.

The chain extenders can be either diamines, alkanolamines, or preferably diols containing a total of from 2 to 15 carbon atoms. Examples of chain extenders include ethanolamine, ethylene diamine, ethylene glycol, 1,3-propane diol, 2,3- or 1,4-butane diol, 1,5-pentane diol, 1,6-hexane diol, hydroquinone bis(2-hydroxyethyl)ether, 1,4-cyclohexanediol, diethylene glycol, dipropylene glycol, 1,4-cyclohexanedimethanol, with 2-butyl-2-ethyl-1,3-propane diol (BEPD) being preferred, and neopentyl glycol being highly preferred.

The amount of the chain extender can be zero (i.e., none) but desirably is from 3 to 30 percent by weight and preferably from 6 to 25 percent by weight based upon the total weight of chain extender and the substantially hydrocarbon intermediate utilized in the formation of the thermoplastic polyurethane compatibilizer. The amount of the chain extender and intermediate utilized, whether they contain hydroxyl groups, amine groups, is generally an equivalent excess to the amount of diisocyanate utilized. That is, the molar ratio of the diisocyanate to hydrocarbon intermediate and chain extender is generally from 0.8 to 1.05 and desirably from 0.9 to 1.01.

It is a desirable aspect of the present invention to make the thermoplastic polyurethane compatibilizer by either the random polymerization method wherein the substantially hydrocarbon intermediate, the diisocyanate and the chain extender are all added together at once and polymerized, or by the prepolymer method. The prepolymer method is preferred where the chain extender is not soluble in the intermediate as generally is the case. Thus, the prepolymer method is generally preferred wherein the isocyanate component is first partially or fully reacted with the hydrocarbon intermediate or polyol to form an isocyanate-terminated prepolymer. The same can be achieved by melt-polymerization. The partially or fully formed prepolymer can then be subsequently reacted with the chain extender.

The polymerization of the reactants forming the thermoplastic compatibilizer of the present invention can generally be carried out by melt-polymerization in a

substantially solvent-free and preferably completely solvent-free environment. The hydrocarbon intermediate is heated to a temperature of from 80°C to 160°C. The diisocyanate, such as MDI, is added and prepolymer formation commences. After a short period of time, for example a couple or several minutes whereby partial or total prepolymers have been formed, the chain extender is added and the reaction carried out to completion. This method allows ready reaction of the insoluble chain extender such as neopentyl glycol with the diisocyanate inasmuch as neopentyl glycol does not dissolve in the substantially hydrocarbon intermediate.

The formation of the compatibilizer is generally carried out in the presence of small amounts of catalysts such as organo tin catalysts, e.g., stannous octoate, a preferred catalyst; stannous oleate; dibutyl tin dioctoate; dibutyl tin dilaurate. Other organic catalysts include iron acetylacetonate, magnesium acetylacetonate. Tertiary organic amine catalysts, such as triethylamine, triethylene diamine, can also be utilized. The amount of catalyst is generally very small, from 25 to 1,000 parts per million and desirably from 40 to 500 PPM by weight based upon the total weight of the reactants.

Although various additives and fillers can be utilized as known to the art and to the literature, such as pigments, lubricants, stabilizers, antioxidants, anti-static agents, fire retardants, the same are generally not utilized in the preparation of the compatibilizer.

The thermoplastic polyurethane compatibilizer of the present invention has soft segments as well as hard segments. The soft segments are generally defined as being solely the hydrocarbon portion of the intermediate. The hard segments are defined as everything else, e.g., the reaction of the intermediate terminal group with the diisocyanate and the reaction of the chain extender with the diisocyanate. The preferred compatibilizers of the present invention desirably have high amounts of soft segments such as at least 25% or 35% by weight, desirably from 45% to 90% by weight, and preferably from 60% to 80% by weight based upon the total weight of the thermoplastic polyurethane compatibilizer excluding any additives, fillers.

The thermoplastic polyurethane compatibilizer was prepared by either the random melt polymerization method or the prepolymer method. In the random melt polymerization method, the polyol and chain extender were blended together at 100°C to 150°C or 250°C. Diphenyl-methanediisocyanate (MDI) was heated separately to 100°C to 150°C or 250°C, then mixed with the blend. The reactants were vigorously mixed for 3-4 minutes. The polymer melt was discharged into a cooled, Teflon-coated pan, cured at 70°C for 1 week, then granulated. In the prepolymer method, the polyol was heated to 100°C to 150°C or 250°C. MDI was separately heated to 100°C to 150°C or 250°C, then mixed with the polyol and allowed to react for 1-2 minutes. The chain extender was added, and the reaction continued for an additional 1.5-3 minutes. The polymer melt was then treated as described above. The melt index values were obtained by ASTM D-1238.

The essentially hydrocarbon intermediate thermoplastic urethane compatibilizers of the present invention are also described in detail in U.S. -A- No. 6,054,533 granted April 25, 2000.

When using TPU and/or TPU-IDP polymers in the blend with COC to achieve a transparent blend, other compatibilizing agents can also be utilized but are generally not preferred inasmuch as they result in a lower clarity and transmission of light. Such less desired compatibilizers are known to the literature and to the art and include block copolymers of styrene-butadiene-styrene commercially available from Shell Chemical as KRATON[®]. Other compatibilizing agents of the various maleic anhydride grafted polyolefins such as polypropylene-g-maleic anhydride and polyethylene-g-maleic anhydride. The proper selection of the compatibilizing agent will depend on the transparency desired in the blend. The less desired compatibilizers mentioned above may be used in a COC blended with TPU or TPU-IDP if less transparency is desired. The above mentioned less desired compatibilizers can also be used with COC blended with non TPU based IDPs.

The amount of the one or more compatibilizing agents of the present invention is generally low such as from 0.1 to 12 parts by weight, desirably from 0.25 to 8 parts by weight and preferably from 0.5 to 2 or 3

parts by weight for every 100 parts by weight of the one or more thermoplastic urethanes TPUs, and/or TPU-IDPs, and/or other IDP polymers and the one or more cycloolefin copolymers, COC.

Transparent-Clear Blends

As mentioned above, the COC may be blended with a TPU, TPU-IDP or other IDP polymer, and a compatibilizing agent. The clarity of the blends will depend on the proper selection of the COC to match the properties of the other polymers in the blend. Since the most preferred blends are COC blended with TPU and/or TPU-IDP, a clear blend is described below. The transparent or clear blends or alloys of the present invention are made by melt compounding the three components, i.e. the TPU and/or TPU-IDP, the COC and the compatibilizing agent, in any suitable blender such as a Banbury[®] or as in a twin screw extruder at temperatures of from 150°C to 250°C and desirably from 190°C to 220°C.

Optical properties are measured by ASTM D-1003 and include light transmission and haze values. Generally, thermoplastic compositions are utilized which have a light transmission of at least 65% or 70%, desirably at least 75% or 80% and preferably at least 82%. Suitable haze values are generally 32% and less, desirably 27% or 22% and less, and preferably 8% or 5% and less. Better optical properties are obtained when the TPU is not a TPU-IDP. Conversely, better electrostatic dissipating properties are obtained when an TPU-IDP is utilized.

While stabilizers can be utilized, desirably various additives such as antioxidants, UV inhibitors, lubricants, flame retardants are not utilized inasmuch as they reduce light transmission as well as clarity. If such additives are utilized, they generally utilize at less than 10%, desirably less than 5%, and preferably less than 3% by weight and more preferably none, based upon 100 parts by weight of the TPU and/or TPU-IDP and the COC components.

The use of a compatibilizing agent is generally required because the COC and the TPU phases have a positive free energy of mixing and are therefore immiscible. The compatibilized form of the present invention will have several advantages compared to

an uncompatibilized control. By reducing the interfacial tension and increasing interfacial adhesion, the compatibilizer helps to stabilize the desired morphology and therefore properties of the blend. A co-continuous morphology is preferred when the goal is to achieve electrical properties through the use of a TPU-IDP. Increased interfacial adhesion improves energy transfer between the phases. This helps prevent delamination or gross phase separation when the blend is subjected the shear forces seen during secondary processes such as injection molding. It also results in increased impact properties and energy to break.

The transparent clear blends of the present invention have several advantages including static dissipative properties, high flexural modulus and strength, good clarity, low extractable ions. A notable improvement of the blend is that it has low outgassing properties in that as apparent from the data below, very small amounts of gas are emitted. For example, generally less than 10 μ g off-gases/gram sample, desirably less than 5 μ g off-gases/gram sample or 3 μ g off-gases/gram sample and preferably less than 1 μ g off-gases/gram sample. Such gases include toluene, styrene, methyl methacrylate.

Outgassing is determined by placing a 500 mg sample in a 20 cm³ vial and heating for 60 minutes at 85C in a Tekmar[®] 7000 headspace analyzer. An aliquot of the headspace was then automatically removed and injected into a Hewlett Packard[®] 5890 Series II GC equipped with a flame ionization detector. A response factor for decane was measured and used to determine the amount of all three organic compounds and the total amount in the headspace. Identification of each compound was by retention time alone. Off gassing amount is reported in μ g off-gases/gram sample.

The above combination of properties result in a useful polymeric blend suitable for electronic and semi-conductor packaging, clean room components and articles, hard disc drive components and packaging, optical devices, films, or coatings. The blend composition can also be utilized in processes including injection molding, extrusion, thermoforming.

The present invention will be better understood by reference to the following examples.

EXAMPLES

The ingredients set forth in Tables 1 and 2 (Examples A through P) were blended in a Werner Pfleiderer[®] model ZSK 30 twin screw extruder under the following general conditions:

Rate: 25 – 35 lbs/hour

Rpm: 175

Temperatures (C):

Zone 1: 165

Zone 2: 175

Zone 3: 185

Zone 4: 190

Zone 5: 190

Zone 6: 185

Die: 170

Amps: 40

Specimens used for physical, electrical, and cleanliness testing were injection molded.

Luminous transmittance % was tested on a Perkin Elmer Model Lambda[®] 9 Spectrophotometer. 0.125" thick samples were scanned from 400-860 nm at a rate of 240 nm/min. The light source was a Tungsten-halogen lamp. Slit width was 2 nm and the reference sample was air. Percent transmittance is reported at 700 nm.

Optical properties are determined according to ASTM D-1003 and include light transmission and haze values.

Clear compositions were made according to the following recipes shown in Examples A through J wherein the parts listed are by weight. Additional examples are given in Table 2.

TABLE 1

EXAMPLES DESCRIPTION	Ex. A*	Ex. B	Ex. C*	Ex. D	Ex. E	Ex. F	Ex. G	Ex. H*	Ex. I*	Ex. J*
	COC 1	COC 1 /TPU-IDP Alloy w/TPU compatibilizer	COC 2	COC 2 /TPU-IDP Alloy w/TPU compatibilizer	COC 1 /TPU-IDP Alloy w/ Kraton® FG-1901X	COC 1 /58315* Alloy w/TPU compatibilizer	COC 1 /58206** alloy w/TPU compatibilizer	Estane® 58315	Estane® 58206	TPU- IDP
FORMULATIONS										
Topas® 8007	100	74			72	79	79			
Topas® 9506			100	74						
TPU-IDP		25		25						100
Estane® 58315								100		
Estane® 58206					3	20	20		100	
Kraton FG 1901X										
TPU-compatibilizer		1		1		1	1			
Molded Samples										
Plastic Tensile										
ASTM D-882-97										
Stress @ Yield psi	9450	5810	8820	5120	5200	6490	6870		4120	NA
Elongation @ Yield %	3.18	3.09	3.1	2.99	3.09	3.09	3.13		218	NA
Tensile @ Break psi	4320	2860	4510	510	1840	3330	2090	2760	3410	NA
Elongation @ Break %	6.55	37.9	18.1	30.5	51.4	51.6	40.9	535	396	NA
Modulus psi	394000	271000	381000	247000	254000	302000	330000	1940	1840	NA
Energy to break lbs-in	43.7	185	109	118	226	264	223	901	819	NA
Flex Modulus										
ASTM D-790-95(.5in/min)	397000	266000	370000	232000	237000	290000	304000	3270	3760	NA
Notched IZOD impact										
ASTM D256-93a	0.43	4.8	0.5	14.2	9.8	0.97	1.1	NA	NA	NA
Type of break	Complete	Hinged	Complete	Partial	Hinged	Hinged	Complete	NA	NA	NA
Gardner impact (in lbs/in)	192	1664	192	960	832	1024	704			
Type of failure	Brittle	Ductile	Brittle	Ductile	Ductile	Ductile	Ductile	NA	NA	NA
Glass transition temp. via DSC	78C		65C							

* not according to the invention

TABLE 1 (cont.)

EXAMPLES DESCRIPTION	Ex. A *	Ex. B COC 1 /TPU-IDP Alloy w/5740 compatibilizer	Ex. C *	Ex. D COC 2 /TPU-IDP Alloy w/5740 compatibilizer	Ex. E COC 1 /TPU- IDP Alloy w/ Kraton FG-1901X	Ex. F COC 1 /58315* alloy w/TPU compatibilizer	Ex. G COC 1 /58206** alloy w/TPU compatibilizer	Ex. H* Estane 58315	Ex. I* Estane 58206	Ex. J* TPU- IDP
Molded Plaques										
Surface Resistivity (ohm/sq)	>E12	7.80E+09	>E12	7.40E+09	7.4E+09	>E12	>E12	>E12	>E12	2.0E+07
Volume Resistivity (Ohm-cm)	>E12	2.60E+10	>E12	2.70E+10	2.50E+10	>E12	>E12	>E12	>E12	1.0E+07
Delamination?	No	No	No	No	No	No	No	No	No	No
Optical Properties										
ASTM D-1003										
Light transmission %	94.9	85	93.6	86.8	76.9	93	94.3	94.11	89.56	61.1
Haze %	3.6	16.2	4	20.3	25.5	11.9	6.6	5.04	16.91	12.7
Luminous Transmittance (%) ASTM D542	78.24	81.2	88.27	79.15	65.08	85.66	86.50	89.54	85.25	79.57
Index of refraction	1.53		1.53					1.53	1.53	1.53

*58315 is an ether based TPU commercially available from Noveon, Inc.

**58206 is an ester based TPU commercially available from Noveon, Inc.

* not according to the invention

The data in Table 2 show two commercially known IDP alloys (Examples K and L). Examples M, N and O show different levels of TPU-IDP blended with cycloolefin copolymer and a TPU compatibilizer. Example P is a COC/TPU-IDP blend without a TPU compatibilizer. The molding results show that the TPU compatibilizer is necessary to prevent delamination for high shear processing, such as molding. Delamination is indicative of a lack of compatibility in polymer blends. Low shear processing, such as often occurs in extrusion, is not as prone to show delamination.

TABLE 2

EXAMPLES	Ex. K*	Ex. L*	Ex. M	Ex. N	Ex. O	Ex. P*
DESCRIPTION	Acrylic IDP alloy	Acrylic core /shell IDP	COC 1 /TPU-IDP Alloy w/TPU compatibilizer	COC 1 /TPU-IDP Alloy w/TPU compatibilizer	COC 1 /TPU-IDP Alloy w/TPU compatibilizer	COC/TPU-IDP Blend
FORMULATIONS						
Topas [®] 8007			79	71.5	64	75
Topas [®] 9506						
TPU-IDP			20	27.5	35	25
TPU-compatibilizer			1	1	1	
ClearStat [®] C-37	100					
Bayon [®] YM-312		100				
Molded Samples						
Plastic Tensile						
ASTM D-882-97						
Stress @ Yield psi			5990	4580	3590	5200
Elongation @ Yield %			3.11	3.56	5.86	3.05
Tensile @ Break psi			1050	1870	3000	4420
Elongation @ Break %			27.3	21.7	16.9	8.71
Modulus psi			311000	245000	186000	275000
Energy to break lbs-in			124	74.6	66.3	47
Flex Modulus						
ASTM D-790-95 (.5in/min)			265000	189000	146000	247000
Notched IZOD impact						
ASTM D256-93a			12.1	12.5	2.3	5
type of break						
Gardner impact (in lbs/in)						320
Type of failure						Ductile

* not according to the invention

EXAMPLES	Ex. K*	Ex. L*	Ex. M	Ex. N	Ex. O	Ex. P*
DESCRIPTION	Acrylic IDP alloy	Acrylic core /shell IDP	COC 1 /TPU-IDP Alloy w/TPU compatibilizer	COC 1 /TPU-IDP Alloy w/TPU compatibilizer	COC 1 /TPU-IDP Alloy w/TPU compatibilizer	COC/TPU- IDP blend
Molded Plaques						
Surface Resistivity (ohm/sq)			2.5E+10	3.70E+09	6.90E+08	2.2E+10
Volume Resistivity (Ohm-cm)			4.9E+10	7.40E+09	8.30E+08	5.4E+10
Delamination?	No	No	No	No	No	Yes

* not according to the invention

The data in Table 3 below is presented to show the low offgassing of compositions of this invention as compared to two other commercially available transparent inherently dissipative polymers. The low offgassing features of the present invention are important in electronic applications to avoid damaging the electronic components. The table below shows offgassing results for Example B of the present invention and two alternate transparent, static dissipative polymers. The composition of Example B is shown in Table 1. Example K (comparative) is a blend of an acrylic based polymer and a polyether amide based IDP. It is commercially available from Cyro Industries under the tradename ClearStat® C-37. Example L (comparative) is an inherently dissipative acrylic polymer made by a core/shell process. It is commercially available from Kureha under the tradename Bayon®. Example B shows lower total outgassing and lower outgassing of each identified species. The formulations for Examples K and L are shown in Table 2.

TABLE 3

	EXAMPLE B	EXAMPLE K (Comparative)	EXAMPLE L (Comparative)
Off Gassing component (µg/g)			
Methyl Methacrylate	<0.03	12.65	1.91
Toluene	<0.02	75.77	0.08
Styrene	<0.02	7	0.15
Total including unknowns	1.32	99.4	11.54

Topas is a copolymer of ethylene and norbornene made by Ticona GmbH. The TPU-IDP, made by Noveon Inc., is a TPU derived from a polyethylene oxide intermediate, MDI, and butane diol as a chain extender with a small amount of a lithium salt and prepared according to U.S. -A- 6,140,405.

Kraton[®] FG-1901X is a maleated styrene-ethylene/butylene-styrene triblock copolymer (SEBS-MA). It contains 28% polystyrene by weight and 2% maleic anhydride.

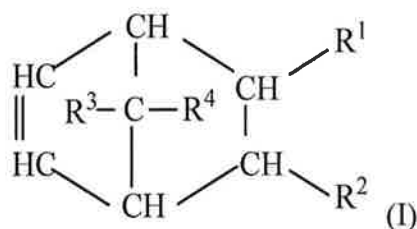
TPU compatibilizer is a compatibilizer derived from a polybutadiene intermediate, MDI, and a neopentyl glycol chain extender and made according to U.S. -A- 6,054,533.

As apparent from Table 1, improved optical properties are obtained when non-electrostatic dissipating thermoplastic urethanes are utilized, see Examples F and G. However, when a TPU-IDP is utilized as in Examples B and D, optical properties drop off but favorable surface resistivity and volume resistivity values are obtained which are not too conductive and not too resistant. Examples M, N, and O (Table 2) demonstrate the effect of TPU-IDP level on surface and volume resistivity. Resistivity decreases with increasing level of TPU-IDP. Values can be adjusted to the desired level within the static dissipative range. Tensile strength and modulus values decrease with increasing TPU-IDP level.

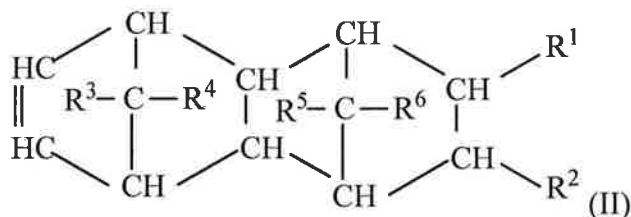
Improved impact properties (Gardner and notched izod) were also obtained when compared to the COC whether or not a TPU-IDP was utilized.

WHAT IS CLAIMED IS:

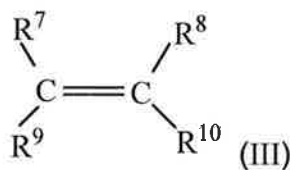
1. An electrostatic dissipating thermoplastic composition, comprising:
 - (a) a cycloolefin copolymer is derived from at least one polycyclic olefin monomer and at least one acyclic 1-olefin monomer, said polycyclic olefin monomer having the formula



or

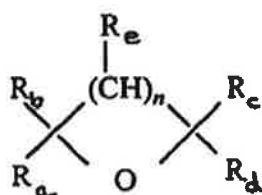


wherein each R^1 , R^2 , R^3 , R^4 , R^5 , and R^6 , independently, are the same or different, and is a hydrogen atom or a hydrocarbon radical, and wherein said acyclic 1-olefin has the formula



wherein each R^7 , R^8 , R^9 , and R^{10} , independently, are the same or different, and is a hydrogen atom or a C_6 - C_{10} aryl group or a C_1 - C_8 alkyl group.

(b) a thermoplastic polyurethane derived from one or more cyclic ether monomers, at least one diisocyanate, and at least one diol, wherein said cyclic ether monomer has the formula



wherein each R_a , R_b , R_c , R_d , and R_e , independently, is a hydrogen atom, an unsubstituted or substituted alkyl, cycloalkyl, cycloalkenyl, aryl, aralkyl or alkaryl, and wherein said substituents which can be substituted within the foregoing are OR_f , SR_f , CN or a halogen, where R_f is hydrogen, alkyl, cycloalkyl, cycloalkenyl, aryl, aralkyl, alkaryl, or carboxyl, and n is 0, 1, 2, or 4, and

(c) a compatibilizing agent comprising a thermoplastic polyurethane derived from a diisocyanate, a diol, and a substantially hydrocarbon intermediate containing at least 20 carbon atoms between non-carbon atoms in the intermediate backbone or a hydrocarbon intermediate derived from one or more dienes having a total of from 4 to 8 carbon atoms, and wherein said thermoplastic polyurethane (b) has a surface resistivity of from 1×10^6 to 1×10^{12} ohm/square as measured by ASTM D-257.

2. An electrostatic dissipating thermoplastic composition according to claim 1, wherein said cycloolefin copolymer is derived from norbornene or tetracyclododecene, wherein said acyclic 1-olefin monomer is ethylene or propylene,

wherein in said thermoplastic polyurethane (b) the cyclic ether monomer is as defined in claim 1, said diisocyanate is 1,4-diisocyanatobenzene (PPDI), 4,4'-methylenebis(phenyl isocyanate) (MDI), 4,4'-methylenebis(3-methoxy phenyl isocyanate), isophorone diisocyanate (IPDI), 1,5-naphthalene diisocyanate (NDI), phenylene-1,4-diisocyanate, toluene diisocyanate (TDI), m-xylene diisocyanate (XDI), 1,4-cyclohexyl diisocyanate (CHDI), 1,10-diisocyanatonaphthylene, and 4,4'-methylenebis-(cyclohexyl isocyanate) (H_{12} MDI), and said diol is ethylene glycol, 1,3-propanediol, propylene glycol, 1,4-butanediol, 1,3-butanediol, 1,5-pentanediol, 1,6-

hexanediol, diethylene glycol, 1,4-cyclohexane dimethanol, neopentyl glycol, hydroquinone bis(2-hydroxyethyl) ether, or combinations thereof.

3. An electrostatic dissipating thermoplastic composition according to claim 2, wherein said cycloolefin copolymer is derived from norbornene and ethylene, wherein the amount of said ethylene monomer is from 45% to 85 mole % based upon the total moles of said ethylene and said norbornene monomers, and

wherein said compatibilizing agent (c) is derived from a diisocyanate selected from ethylene diisocyanate; toluene diisocyanate; methylene bis-(4-phenylisocyanate) (MDI); isophorone diisocyanate; hexamethylene diisocyanate; naphthalene diisocyanate; cyclohexylene diisocyanate; diphenylmethane-3,3' dimethoxy-4,4'-diisocyanate, meta-tetramethylxylene diisocyanate (m-TMXD1), paratetramethylxylene diisocyanate (p-TMXD1), m-xylylene diisocyanate (XDI), decane-1,10-diisocyanate, dicyclohexylmethane-4,4'-diisocyanate, and combinations thereof, and from a diol selected from ethylene glycol, 1,3-propane diol, 2,3- or 1,4-butane diol, 1,5-pentane diol, 1,6-hexane diol, hydroquinone bis(2-hydroxyethyl)ether, 1,4-cyclohexanediol, diethylene glycol, dipropylene glycol, 1,4-cyclohexanedimethanol, and combinations thereof.

4. An electrostatic dissipating thermoplastic composition according to claim 3, wherein in said thermoplastic urethane (b) the cyclic ether is ethylene oxide, wherein said diisocyanate is MDI and wherein said diol is butane diol, wherein said electrostatic dissipating thermoplastic polyurethane has a surface resistivity of from about 1×10^8 to about 1×10^{10} ohm/square as measured by ASTM D-257, and

wherein the amount of said compatibilizing agent is from 0.25 to 8 parts by weight per 100 parts by weight of said cycloolefin copolymer and said electrostatic dissipating thermoplastic urethane.

5. An electrostatic dissipating thermoplastic composition according to claim 4, wherein the amount of said ethylene monomer in said cycloolefin copolymer is from 55 to 80 mole % based upon the total moles of said ethylene and said norbornene monomers,

wherein said compatibilizing agent is derived from butadiene as the intermediate, from MDI as the diisocyanate and from neopentyl glycol as the diol.

6. An electrostatic dissipating thermoplastic composition according to claim 5, wherein said composition comprises co-continuous phases of said cycloolefin copolymer and said electrostatic dissipating thermoplastic urethane.
7. An electronic or semi-conductor packaging material comprising the composition of any of claims 1 to 6.
8. A hard disc drive component or packaging material comprising the composition of any of claims 1 to 6.