

**METHOD FOR APPLYING A TITANIUM ALUMINIDE ALLOY, TITANIUM
10 ALUMINIDE ALLOY AND SUBSTRATE COMPRISING A TITANIUM ALUMINIDE
ALLOY**

The present invention relates to a method for the application
of a titanium aluminide alloy with a predominant proportion of
15 the gamma phase to a substrate, a titanium aluminide alloy
produced in accordance with such a method and a substrate com-
prising such a titanium aluminide alloy.

When repairing components made of titanium aluminide alloys
20 with a predominant proportion of gamma phase, generally the
same or related material, i.e. a filler material from the
group of titanium aluminide alloys with a predominant propor-
tion of gamma phase, is applied by means of fusion welding
processes. An example of this is the laser powder fusion weld-
25 ing process.

A significant disadvantage of the known fusion welding processes
used to date is the tendency to lead to changes in the
structure and the phase as well as the formation of pores and
30 stress cracks due to very high process temperatures and
cooling rates. Furthermore, these processes can lead to
undesired absorption of impurities such as oxygen and
nitrogen not only in the base material but also in the weld
metal.

The publication by Gizynski *et al.*, Formation and subsequent
phase evolution of metastable Ti-Al alloy coatings by kinetic

spraying of gas atomized powders, Surface & Coating Technology, 2017, 315, 240-249, discloses the use of a titanium aluminide alloy (Ti-48Al-8.5Nb-1Ta (At.%)) for the production of an oxidation protection layer on a titanium substrate (IMI-834), which is applied by means of a warm spray method. The influence of the heat treatment on the alloy powder used is examined as well. For this purpose, gas-atomized alloy powder is compared with a similar, heat-treated alloy powder. It is concluded that the heat treatment of the alloy powder does not produce any significant advantage for the warm spray method.

The first potential cold spray applications are described in the prior art.

EP 2 584 056 A1 discloses the use of the cold spray method with a powder of titanium aluminide with a gamma/alpha2 structure to form a layer of a titanium aluminide alloy with a fine structure of gamma and alpha2 phase components. The relatively large proportion of the alpha2 phase to the gamma phase present in the powder is converted by a heat treatment downstream of the coating.

EP 2 333 134 A1 describes a cold spray method in which a mixture of aluminum and titanium powder particles are sprayed together. A titanium aluminide alloy with a gamma phase is formed through a subsequent heat treatment. Such a method is also described in the publication by Novoselova *et al.*, Formation of TiAl intermetallics by heat treatment of cold-sprayed precursor deposits, Journal of Alloys and Compounds, 2007, 436, 69-77.

Sabard *et al.*, Solution heat treatment of gas atomized aluminum alloy (7075) powders: microstructural changes and resultant mechanical properties, DVS-reports, 2017, 336, 766-771,

describes a heat treatment of atomized powder particles of an aluminum alloy for the adjustment of a more ductile structure.

These publications show that cold gas coating methods are very
5 challenging. A disadvantage of the known cold spray methods is
that the titanium aluminide alloys applied by means of these
methods to the substrate with a predominant proportion of the
gamma phase do not adequately adhere to the substrates. The
coating efficiencies (deposition efficiency (DE)) that are
10 achieved with this method are therefore extremely low. In ad-
dition, the titanium aluminide alloy layers that are created
lack the corresponding physical properties such as sufficient
ductility, high mechanical strength and a low defect density
(adhesion defects, pores and cracks). Thus, these processes
15 cannot be used for the demanding coatings of, for example,
aircraft, gasoline engines, diesel engines and stationary gas
turbine components. Industrial applications in which layers of
titanium aluminide alloys with a predominant proportion of the
gamma phase are applied by means of a cold spray method to
20 substrates with at least one substrate surface made of tita-
nium aluminide alloys, which also have high proportions of the
gamma phase, are not known from prior art.

The present invention is therefore based on the object of
25 providing a method for applying a titanium aluminide alloy
with a predominant proportion of the gamma phase to a sub-
strate, said method enabling the application and production of
well-adhering and dense cold spray layers from titanium alu-
minide alloys with a predominant proportion of the gamma phase
30 on a substrate. In particular, the method according to the in-
vention is intended to enable the application of such titanium
aluminide alloys to substrates made of related or similar ti-
tanium aluminide alloys or with related or identical layers of

titanium aluminide alloy to the substrate surface. In particular, this is intended to allow for the repair of such related or similar titanium aluminide alloys as well as the substrates themselves. Another object is to provide a titanium aluminide alloy produced with the method, according to the invention, and a substrate comprising such a titanium aluminide alloy.

The invention achieves this object with a method according to the main claim. The invention relates in particular to a method for the application of a titanium aluminide alloy comprising a gamma phase proportion of at least 50% based on the overall composition of the titanium aluminide alloy on a substrate with the method comprising the following steps:

- a) Pretreatment of the substrate surface;
- b) Heat treatment of titanium aluminide powder particles at a temperature range from 600 to 1000°C in order to increase the proportion of the gamma phase;
- c) Cold spraying of the heat-treated powder particles onto the substrate or a part of the substrate to form a layer of titanium aluminide; and
- d) Thermal post-treatment of the layer of titanium aluminide applied to the substrate.

Further advantageous embodiments can be found in the sub-claims.

First, some terms used in the context of the invention are explained.

In the context of the present invention, the term "titanium aluminide alloy" is used for the finished layer of titanium aluminide alloy applied by using the method according to the invention.

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In the context of the present invention, the term "titanium aluminide layer" is, for the purpose of a better differentiation, used for the titanium aluminide layer, which is present on the substrate after the cold spraying and which has not yet
10 been thermally after-treated. According to the invention, however, this may also be a titanium aluminide alloy.

In the context of the present invention, the term "powder particles made of titanium aluminide" is, for a better differentiation, used for the powder particles made of titanium alu-
15 minide before, during and after the heat treatment. According to the invention, however, these may also be titanium aluminide alloys.

20 Titanium aluminides or titanium aluminide alloys are compounds known from prior art that comprise at least the metals titanium (Ti) and aluminum (Al). The titanium aluminides or titanium aluminide alloys preferably comprise additional alloy elements such as chromium (Cr), silicon (Si), vanadium (V), zircon (Zr), niobium (Nb), boron (B), carbon (C), tungsten (W),
25 molybdenum (Mo), yttrium (Y), cerium (Ce), hafnium (Hf), iron (Fe), nickel (Ni) or tantalum (Ta). Even small amounts of additional alloying elements can significantly improve the mechanical properties and the structural properties of the finished titanium aluminide alloys.
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Different phase constitutions may occur in titanium aluminides or titanium aluminide alloys. On the one hand, there are the

high temperature phases alpha and beta. Then there are the gamma, alpha2 and beta0 phases, which are present as intermetallic titanium aluminide compounds. The term gamma phase (gamma titanium aluminide) refers to the tetragonal gamma (TiAl) phase with an L10 structure known in the prior art. The term alpha2 phase refers in the context of the present invention to the hexagonal alpha2 (Ti3Al) phase with a D019 structure. The term beta0 refers to the ordered cubically primitive beta0 (TiAl) phase with a B2 structure. Furthermore, there are the ordered omega0 phase with a B82 structure and the orthorhombic O phase (Ti2AlNb) with a A2BC structure. Depending on the additional alloying elements and alloying contents that were selected, additional phases, which are often related with respect to their crystallography, may form. The circumstances surrounding the formation of these phases are the subject of current research.

In the context of the present invention, the term "titanium aluminide alloy with a predominant proportion of the gamma phase" refers to a titanium aluminide alloy with a gamma phase proportion of at least 50% based on the overall composition of the titanium aluminide alloy. The percentages (%) used in connection with the invention for the proportions of the individual phases are to be understood as volume percentages (vol.%).

Furthermore, it should be noted that in the context of the present invention, the method steps a) pretreatment of the substrate surface and b) heat treatment of titanium aluminide powder particles can also be carried out in the reverse order or simultaneously.

The invention has the advantage that the method according to the invention enables the application of ductile, well-adhering and dense cold spray layers from titanium aluminide alloys with a predominant proportion of the gamma phase on a substrate. In particular, the method according to the invention can be used to apply to the substrate surface titanium aluminide alloys with a predominant proportion of the gamma phase to substrates made of related or similar titanium aluminide alloys or with related or similar titanium aluminide alloy layers. As a result, the method according to the invention can be used for not only the repair of components with such titanium aluminide alloys but also the assembly of components.

The invention has recognized that the heat treatment of the titanium aluminide powder particles in a temperature range from 600 to 1000°C leads to a phase conversion of the alpha and beta phase and in particular also a conversion of the ordered alpha₂ phase to the gamma phase even before the actual coating step takes place. This significantly increases the proportion of the ductile gamma phase in the powder used. A higher proportion of the ductile gamma phase in turn has the advantage that the coating efficiency (deposition efficiency (DE)) is significantly improved in cold spray methods. In addition, the thermal aftertreatment leads to an improved adhesive strength and fatigue strength of the titanium aluminide alloys produced. Such a thermal aftertreatment is usually not carried out in the cold spray methods known from prior art. By combining the method steps according to the invention, the method according to the invention is much more efficient compared to the known cold spray coating methods.

The titanium aluminide alloy preferably has a gamma phase proportion of at least 55%, more preferably at least 60%, even

more preferably 80% based on the overall composition of the titanium aluminide alloy.

The titanium aluminide alloy preferably has a beta phase proportion of less than 10%, more preferably of less than 5%, even more preferably of less than 2% based on the overall composition of the titanium aluminide alloy.

Preferably, the titanium aluminide alloy includes the gamma phase and the alpha2 phase. It is particularly preferred that the titanium aluminide alloy is present in a phase constitution that essentially consists of the gamma phase and the alpha2 phase.

It is further preferred that the titanium aluminide alloy comprises the gamma phase and the alpha2 phase and that a ratio of the gamma phase to the alpha2 phase is present in the titanium aluminide alloy in a range from 50:50 to 99:1, more preferably from 55:45 to 90:10, and even more preferably from 60:40 to 80:20. In a particular embodiment, the ratio of the gamma phase to the alpha2 phase is 80:20 in the titanium aluminide alloy.

In a further advantageous embodiment, the titanium aluminide alloy has a composition of Ti-48Al-2Nb-2Cr (At.%).

It is further preferred that the substrate has a substrate surface made of a metal alloy. Alternatively, it is preferred that the substrate consists (substantially) of the metal alloy. The metal alloy is preferably a metal alloy selected from a titanium aluminide alloy, a nickel alloy, a titanium alloy or combinations of these alloys. A titanium aluminide alloy or

a combination of several titanium aluminide alloys is particularly preferred. A titanium aluminide alloy which comprises a predominant proportion of the gamma phase is even more preferred.

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It is further preferred that the method according to the invention is used as a method for the repair of a metal alloy already present on a substrate or for the repair of the substrate itself. In a particularly preferred embodiment, the titanium aluminide alloy already present on the substrate is a titanium aluminide alloy which has the same chemical composition as the titanium aluminide alloy which is applied to the substrate surface by means of the method.

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It is preferably provided that the pretreatment of the substrate surface is selected from the polishing, roughness blasting, high pressure water blasting, chemical etching treatment options and combinations thereof. The substrate surface is activated by the pretreatment and thus prepared for the application of the powder particles using the cold spray method. The pretreatment ensures a significantly better adhesion of the applied layer of powder particles to the substrate.

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Roughness blasting is to be understood in connection with the invention as the blasting of the substrate surface with solid particles. This causes the substrate surface to be coated to be roughened and cleaned. It is preferred that SiC, Al₂O₃ and/or a similar powder made of titanium aluminide is used as the blasting medium for the roughness blasting. The term "similar powder" refers to a powder with a chemical composition that is identical to that of the titanium aluminide alloy according to the invention.

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In chemical etching, an alkaline solution is preferably applied to the substrate surface. In an alternative preferred embodiment, the substrate surface is treated with a gas that contains fluoride ions.

It is further preferred that the pretreatment of the substrate surface comprises a polishing of the substrate surface.

It is preferred that the heat treatment be carried out in a temperature range from 620 to 900°C, more preferably from 650 to 850°C. In a preferred embodiment, the temperature is 650°C.

The phase constitution of the titanium aluminide powder particles is influenced by the heat treatment of the powder particles in a temperature range from 600 to 1000°C in such a way that the highest possible proportion of the ductile gamma phase is obtained. Higher temperatures preferably lead to a higher proportion of the ductile gamma phase. A high ductility of the powder particles is particularly advantageous for the impact of the powder particles on the substrate and thus for the entire coating process because the increased ductility of the powder particles leads to an increased plastic deformation of the powder particles when they strike the substrate surface. The increased plastic deformability then causes an increased adhesion of the powder particles to the substrate surface and an improved coating efficiency. This increase in adhesion is already noticeable not only at the start of the coating process but also during a further layer build-up.

It is preferably provided that the heat treatment of the powder particles takes place in a protective gas atmosphere or in

a vacuum. In a preferred embodiment, the protective gas is argon or a mixture of argon and a reducing gas. A particularly preferred protective gas is a mixture of argon with 4% hydrogen.

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It is further preferred that the heat treatment be carried out for a duration of 0.5 to 5 hours.

10 In a preferred variant of the invention, it is provided that the heat treatment is carried out for 1 to 3 hours in a protective gas atmosphere or in a vacuum of less than 10 mbar at a temperature range from 650 to 850°C.

15 In a preferred embodiment of the invention, it is provided that the heat treatment is carried out in a vacuum furnace.

A heat treatment in a vacuum or in a protective gas atmosphere made of argon offers the following advantages:

- 20
- The additional intake of oxygen, nitrogen or other impurities is reduced
 - The evaporation of aluminum is minimized in a protective gas atmosphere so that the homogeneity of the alloy composition is maintained
 - There is additional protection when handling the powder particles (lower risk of fire and explosion)
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- 30
- If a suitable heat treatment vessel is used, the heat treatment can be carried out in a commercially available high-temperature furnace (which reduces the processing costs compared to a vacuum furnace).

The conditions of the heat treatment are preferably selected so that no disruptive bonds develop between the powder particles. In the event that loose bonds should develop regardless,
5 it is preferred that these be broken up by a mechanical grinding and subsequent sieving process in an inert gas atmosphere.

It is further preferred that the heat treatment be carried out in an inert vessel. The use of a non-inert vessel is less
10 suitable from a technical point of view because this may result in a contamination of the powder particles and, as a rule, the vessel then cannot be reused, which increases the processing costs even further. In addition, the use of an inert vessel may avoid an undesired supply of heat. This supply
15 of heat may otherwise lead to an additional introduction of heat into the powder particles and thus result in an undesired phase and/or structural conversion.

It is further preferred that the powder particles are spherical in shape. It is further preferred that the powder particle
20 surfaces have few or no satellite formations. In particular, it is preferred that the powder particles have a size in a range from 10 to 70 μm . As the powder particle size reduces, the proportion of a brittle α_2 phase decreases further, in particular in the case of an alloy with the composition of Ti-
25 48Al-2Nb-2Cr (at.%). In addition, smaller powder particles contribute to a better coating efficiency (Deposition Efficiency (DE)). However, it should also be noted that powder particles that are too small do not sufficiently adhere to the
30 substrate. The above-mentioned shapes and sizes of the powder particles have the further advantage that they lead to a narrow speed distribution of the powder particles in the gas jet during cold spray.

In a further preferred embodiment of the invention, it is provided that the average powder particle diameter is less than 45 μm . A microstructure analysis of different powder particle fractions, for example an alloy with a composition of Ti-48Al-2Nb-2Cr (At.%), shows that the proportion of the brittle α_2 phase decreases further with such powder particle diameters.

Furthermore, it should be noted that a change in the fraction sizes of the powder particles can significantly influence the sprayability of the powder particles during the cold spray. Powder particle fractions preferred according to the invention are, for example, a powder particle fraction in which 10 vol% of the powder particles is smaller than 29 μm , 50 vol% of the powder particles is smaller than 43 μm and 90 vol% of the powder particles is smaller than 61 μm (d10/d50/d90: 29/43/61 μm), a powder particle fraction in which 10 vol% of the powder particles is smaller than 8 μm , 50 vol% of the powder particles is smaller than 13 μm and 90 vol% of the powder particles is smaller than 19 μm (d10/d50/d90: 8/13/19 μm) and a powder particle fraction in which 10 vol% of the powder particles is smaller than 18 μm , 50 vol% of the powder particles is smaller than 43 μm and 90 vol% of the powder particles is smaller than 61 μm (d10/d50/d90: 18/43/61 μm). The powder particle fraction in which 10 vol% of the powder particles is less than 18 μm , 50 vol% of the powder particles is less than 43 μm and 90 vol% of the powder particles is less than 61 μm is particularly preferred (d10/d50/d90: 18/43/61 μm).

Furthermore, in the cold spray according to the invention, the powder particles are preferably applied at 20-60 mm intervals.

A preferred delivery rate of the powder particles is around 10 to 50 g/min.

It is also preferred that a cold spray carrier gas is selected from nitrogen and a mixture of nitrogen and helium. It is further preferred that the carrier gas is preheated to a temperature of 700 to 1200°C, more preferably to a temperature of 950°C to 1100°C. A preferred gas pressure is in the range of 40 to 50 bar.

It is preferred that the temperature of the powder particles when striking the substrate is in a temperature range of 640 to 825°C. A preferred powder particle speed in cold spray is in a range of 630 to 1000 m/s. However, the powder particle temperature and the powder particle speed are not adjustable process parameters but instead result from the type of gas, the gas pressure, the gas temperature as well as the respective physical and geometric properties of the powder particles and the geometric properties of the nozzle.

It is preferred that the layer of titanium aluminide comprises the gamma phase and the alpha2 phase prior to the thermal aftertreatment. In particular, it is preferred that the layer of titanium aluminide is in a phase constitution prior to the thermal aftertreatment that substantially consists of the gamma phase and the alpha2 phase.

It is preferred that the layer of titanium aluminide has a gamma phase proportion of at least 55% prior to the thermal aftertreatment, more preferably at least 60%, even more preferably 80% based on an overall composition of the layer of titanium aluminide.

It is preferred that the layer of titanium aluminide has a beta phase proportion of less than 10% after the thermal aftertreatment, more preferably less than 5%, even more preferably less than 2% based on the overall composition of the layer of titanium aluminide.

It is further preferred that a ratio of the gamma phase to the alpha2 phase in the titanium aluminide layer prior to the thermal aftertreatment is in a range from 50:50 to 99:1, more preferably from 55:45 to 90:10, even more preferably from 60:40 to 80:20. In a particular embodiment, the ratio of the gamma phase to the alpha2 phase is 80:20 in the layer of titanium aluminide prior the thermal aftertreatment.

It is preferred, according to the invention, that the thermal aftertreatment is a hot isostatic pressing (HIP). The hot isostatic pressing preferably takes place at a temperature between 1050 and 1300°C, more preferably at a temperature of about 1200°C. It is further preferred that the hot isostatic pressing is carried out at a pressure in the range of 1000 to 3000 bar, more preferably from 1700 to 2300 bar, and most preferably at 2000 bar. In a preferred embodiment, the hot isostatic pressing is carried out for 4 hours in an argon protective gas atmosphere at a temperature of 1200°C and a pressure of 2000 bar.

Due to the relatively brittle powder particles in a layer of titanium aluminide or a titanium aluminide alloy, which mainly consists of the gamma phase, bonding defects in the form of cracks or increased porosity may occur at the interface with the substrate or during the further build-up of the coating. These defects can be healed (cracks) and closed (porosity) by hot isostatic pressing. Another advantage of hot isostatic

pressing is the improved adhesion of the layer of powder particles applied to the substrate.

5 In an alternative embodiment, it is provided that the thermal aftertreatment is a diffusion annealing, which is preferably carried out in a temperature range from 700 to 1000°C. Furthermore, the diffusion annealing preferably takes place in a vacuum of 1×10^{-6} to 1×10^{-3} mbar, more preferably 5×10^{-6} to 5×10^{-4} mbar, most preferably 5×10^{-6} to 1×10^{-4} mbar. Diffusion annealing is
10 particularly advantageous if the substrate has a certain structural state which is unsuitable for the higher temperatures used in isostatic pressing and which would result in undesirable structural changes, residual stress and geometric distortion. The diffusion annealing improves the adhesion of the applied
15 layer to the substrate surface, and any defects in the layer that have arisen are healed. The residual porosity remaining after such a heat treatment is tolerated in such a case.

The invention further relates to a titanium aluminide alloy
20 which is produced by the method according to the invention.

The invention also relates to a substrate comprising a layer of titanium aluminide alloy applied by the method according to the invention.

25 It is preferred that the substrate is an aircraft, gasoline engine, diesel engine or stationary gas turbine component.

The invention will now be described by way of example using
30 some advantageous embodiments with reference to the accompanying drawing. The figures show the following:

Figure 1: schematically a device for performing the method according to the invention according to one embodiment.

Figure 1 schematically shows a device (10) with which the method according to the invention can be carried out. A substrate (11) is shown, the substrate surface of which was pre-treated in a first step. Furthermore, a powder conveyor (12) is shown, via which the heat-treated titanium aluminide powder particles are conveyed into the device for cold spraying (13). The powder particles are preferably transported by using a carrier gas (14) consisting of nitrogen or nitrogen and helium, which is conveyed into the device at a pressure of preferably 40 to 50 bar. The carrier gas can be heated to a temperature of preferably 950 to 1100°C in an additional heating device (15). The titanium aluminide powder particles hit the pretreated substrate (10) at a powder particle speed of preferably 630 to 1000 m/s. Subsequent to the coating using cold spraying, the substrate is then thermally post-treated in a further step.

Exemplary Embodiments

The method according to the invention for the application of a titanium aluminide alloy with a predominant proportion of the gamma phase to a substrate comprises the steps detailed in the following sections. In the following, these steps are explained by using the example of the titanium aluminide alloy Ti-48Al-2Nb-2Cr (At.%).

1) Pretreatment of the substrate surface

To prepare for the cold spray coating step, the substrate surface is pretreated. It is preferred that the pretreatment of

the substrate surface be selected from the methods of polishing, roughness blasting, high pressure water blasting or chemical etching.

- 5 In a preferred variant of the roughness blasting, the conditions are selected as follows:

Grit: Al2O3

10 Grit size: F20-F150 according to the FEPA F42 standard or mesh 20-120 according to ANSI, preferably F150

Jet pressure: Pressure blasting systems: up to a maximum of 4 bar, suction blasting systems: up to a maximum of 6.5 bar

15 Blast distance: Inner diameter 10-20 mm, outer diameter 150-180 mm (depending on the component geometry)

Blast angle: 45-90°

Blast flow rate: 0.5-4 g/min (depending on the nozzle diameter and blasting material)

20 Robot velocity: at least 100 mm/s, preferably 150-300 mm/s (depending on component/parameter)

Number of transitions: 1-2

Blasting track distance: 0.5-5 mm, preferably 1.0 mm

- 25 In a preferred embodiment, the conditions for the high pressure water blasting are selected as follows:

Blasting pressure: 3500 bar

Blasting distance: 20 mm

Blasting angle: 90°

30 Robot velocity: 1.5 mm/sec

Number of transitions: 2-4

Nozzle shape: concentric

Nozzle speed: 1400 1/min

Blasting track distance: 0.5-5 mm, preferably 1.0 mm

Alkaline solutions, which activate the substrate surface and thus prepare the surface for the subsequent cold spray step, can be used for chemical etching. Alkaline rust removers in an immersion bath, for example, can also be used to activate the substrate surface.

In a particularly preferred embodiment, the chemical etching conditions are selected as follows:

Alkaline cleaner: Bonderite C-AK 4181 AERO

Temperature: 70-80°C

Cleaning time: 15-20 min

Drying: 10-30 min at 80-100°C

2) Heat treatment of the powder particles

The phase constitution of the titanium aluminide powder particles is influenced by the heat treatment in such a way that the highest possible proportion of ductile gamma phase is obtained. A high ductility of the powder particles is particularly advantageous for the impact of the powder particles on the substrate and thus for the entire coating process. This is because the increased ductility of the powder particles leads to an increased plastic deformability of the powder particles when they strike the substrate surface. The increased plastic deformation then causes an increased adhesion of the powder particles to the substrate surface and an improved coating efficiency. This increase in adhesion is already noticeable not only at the start of the coating process but also during a further layer build-up.

The above-described properties of the powder particles also lead to a uniform powder conveyance and additionally increase the ductility of the structure of the powder particles.

5 In order to obtain an optimal coating, the phase constitution of the titanium aluminide particle powder is influenced by a suitable heat treatment in such a way that the highest possible proportion of the gamma phase is obtained. For this purpose, the heat treatment of the titanium aluminide particle
10 powder is carried out in a temperature range from 600 to 1000°C.

The following conditions can be selected to obtain a phase constitution of the titanium aluminide powder particles of approximately 20% alpha² phase and approximately 80% gamma
15 phase:

Temperature: 650°C

Duration: 1 hour

20 Furnace pressure: < 10 mbar, preferably < 5 mbar (vacuum) or protective gas atmosphere

Higher temperatures lead to a higher proportion of the more ductile gamma phase. The heat treatment is preferably carried
25 out in a protective gas atmosphere or under almost vacuum conditions and in a temperature-time window so that no disruptive bonding (due to sintering processes) of the powder particles with one another can occur.

30 3) Cold spraying

The basic coating parameters for the titanium aluminide alloy Ti-48Al-2Nb-2Cr (At.%) were calculated prior to the actual performance of the cold spray.

5 Calculated powder particle sizes: d10/d50/d90: 29/43/61
 μm

Powder particle temperature when hitting the substrate:
640-723°C

10 Carrier gases: Nitrogen or a mixture of nitrogen and he-
 lium

Carrier gas pressure: about 40-50 bar

Gas preheating temperature: 750-1100°C

Coating distance: 20-60 mm

Powder particle speed: 630-1000 m/s

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Eta values are calculated for these coating conditions. The eta value is defined as the ratio of the actual powder particle speed when it hits the substrate (V_{ist}) to the critical powder particle speed (V_{crit}) and indicates when the ratio

20 $V_{ist}/V_{crit} > 1$ has been reached that a layer is being built up.

The calculated eta values of 1.11 and 1.18 indicate an adequate cold spray coating by the titanium aluminide powder particles on the substrate made of titanium aluminide alloy with a high proportion of the gamma phase.

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Following the calculations, a series of tests were carried out to confirm the calculated results. The test results show that the following coating conditions are optimal for the coating efficiency of the titanium aluminide alloy Ti-48Al-2Nb-2Cr

30 (At.%):

Powder particle fractions: d10/d50/d90: 8/13/19 μm or preferably d10/d50/d90: 18/43/61 μm

Powder particle delivery rate: 10-50g/min
Gas preheating temperature: 950-1100°C
Carrier gas: 100% nitrogen or a mixture of 75% nitrogen
and 25% helium
5 Gas pressure: about 40-50 bar
Coating distance: 20-60 mm
Robot velocity: 500 mm/s
Powder particle speed: 630-1000 m/s
Blasting track distance: 0.5-5 mm, preferably 1.0 mm

4) Thermal aftertreatment

After the cold spraying, the thermal aftertreatment takes
place. This is preferably hot isostatic pressing or alterna-
15 tively diffusion annealing. Both methods lead to improved ad-
hesion of the applied layer of powder particles to the sub-
strate.

In a preferred embodiment, the following parameters are se-
20 lected for the hot isostatic pressing:

Temperature: 1200°C
Time: 4 hours
Pressure: 2000 bar
25 Protective gas: Argon

The diffusion annealing, however, is preferably carried out at
a lower temperature in a range of 700 to 1100°C.

30 The tests show that the coating efficiencies achieved with the
method, according to the invention, are good and, surpris-
ingly, this is also the case with related or similar titanium
aluminide alloys. Among other things, this is due to the heat

treatment of the powder particles carried out prior to the actual cold spraying. In addition, a further improvement in the adhesive strength and fatigue strength of the applied titanium aluminide alloys is observed.

Claims

1. Method for the application of a titanium aluminide alloy comprising a gamma phase proportion of at least 50% based on the overall composition of the titanium aluminide alloy on a substrate, the method comprising the following steps:

a) Pretreatment of the substrate surface;

b) Heat treatment of titanium aluminide powder particles at a temperature range of 600 to 1000°C in order to increase the proportion of the gamma phase;

c) Cold spraying of the heat-treated powder particles onto the substrate or a part of the substrate to form a layer of titanium aluminide; and

d) Thermal post-treatment of the layer of titanium aluminide applied to the substrate.

2. Method according to claim 1, characterized by the titanium aluminide alloy having a gamma phase proportion of at least 55%, preferably at least 60%, even more preferably 80% based on the overall composition of the titanium aluminide alloy.

3. Method according to claim 1 or 2, characterized by the substrate having a substrate surface selected from titanium aluminide alloy, nickel alloy, titanium alloy or combinations of these alloys.

4. Method according to one of claims 1 to 3, characterized by the pretreatment of the substrate surface being selected

from polishing, roughness blasting, high pressure water blasting, chemical etching or combinations thereof.

5. Method according to one of claims 1 to 4, characterized by the heat treatment of the powder particles being carried out in a protective gas atmosphere or in a vacuum.

6. Method according to one of claims 1 to 5, characterized by the heat treatment being carried out for a period of 0.5 to 5 hours.

7. Method according to one of claims 1 to 6, characterized by the heat treatment being carried out in a temperature range from 620 to 900°C, more preferably from 650 to 850°C.

8. Method according to one of claims 1 to 7, characterized in that the heat treatment is carried out for 1 to 3 hours in a protective gas atmosphere or in a vacuum of less than 10^{-5} mbar in a temperature range of 650 to 850°C.

9. Method according to one of claims 1 to 8, characterized by the size of the powder particles being in a range from 10 to 70 μm .

10. Method according to one of claims 1 to 8, characterized by the average powder particle diameter being less than 45 μm .

11. Method according to one of claims 1 to 10, characterized by the powder particles being spherical.

12. Method according to one of claims 1 to 11, characterized by the thermal aftertreatment being a hot isostatic pressing or a diffusion annealing.

13. Method according to one of claims 1 to 12, characterized
by the titanium aluminide alloy comprising the gamma phase
and the alpha2 phase and by a ratio of the gamma phase to
5 the alpha2 phase in the titanium aluminide alloy being in
a range from 50:50 to 99:1, more preferably from 55:45 to
90:10, even more preferably from 60:40 to 80:20.

14. Titanium aluminide alloy produced by a method according to
10 any one of claims 1 to 13.

15. A substrate comprising a layer of titanium aluminide alloy
applied by using a method according to any one of claims 1
to 13.

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