

**RECHARGEABLE ALKALINE METAL AND ALKALINE EARTH ELECTRODES HAVING
CONTROLLED DENDRITIC GROWTH AND METHODS FOR MAKING AND USING THE SAME**

CROSS-REFERENCE TO RELATED APPLICATION

This utility patent application claims priority to co-pending U.S. provisional patent serial no. 61/486,946, filed on May 17, 2011, to co-pending U.S. provisional patent application serial no. 61/498,192, filed June 17, 2011, and to co-pending U.S. provisional patent application serial no. 61/565,101, filed on November 30, 2011, which are all incorporated by reference herein.

BACKGROUND

The use of Lithium metal as an anode to build a rechargeable Lithium cell or battery system with the highest anode-specific capacity has long been desired. However, the

growth of Li-metal dendrites gives rise to serious technical barriers for developing such a battery. Recently, modified versions of the Li metal battery, such as the Lithium ion battery, have been introduced with some success. However, the current modified versions possess limitations and inefficiencies that would not arise with a cell that uses Lithium metal as an anode.

Typically, a Lithium metal cell includes an anode and a cathode separated by an electrically insulating barrier or 'separator' and operationally connected by an electrolyte solution. During the charging process, the positively charged lithium ions move from the cathode, through the permeable separator, to the anode and reduce into Li metal. During discharge, the Li metal is oxidized to positively charged lithium ions which move from the anode, through the separator, and onto the cathode, while electrons move through an external load from the anode to the cathode, yielding current and providing power for the load. During repeated charges and discharges, Lithium dendrites begin to grow from on the surface of the anode. Dendritic lithium deposits, sometimes called mossy lithium, eventually tear through the separator and reach the cathode causing an internal short and rendering the cell inoperable. Lithium dendrite formation is inherently unavoidable during the charging and discharging processes of Li-metal cells. Thus, there remains a need for a lithium electrode cell system that does not suffer the effects of dendrite growth while simultaneously maintaining the cycle ability, ionic conductivity, voltage and specific capacity of the cells. The present novel technology addresses these needs.

SUMMARY

According to an aspect of the present invention, there is provided a method comprising:

- forming an electrically conductive layer comprising a dendrite seeding material in an electrolyte solution disposed between a metal-containing electrode and an electrolyte permeable separator membrane;
- growing metal dendrites from the dendrite seeding material toward the metal-containing electrode; and
- contacting metal dendrites extending from the metal-containing electrode with metal dendrites extending from the dendrite seeding material;

wherein the electrolyte solution contains metal ions.

According to another aspect of the present invention, there is provided a method comprising:

- forming an electrically conductive layer comprising a dendrite seeding material on an electrolyte permeable separator membrane in an electrolyte solution containing a metal-containing electrode, the electrolyte solution containing metal ions.

DESCRIPTION OF THE DRAWINGS

FIG. 1 is schematic view of a lithium ion cell according to a first embodiment of the present novel technology.

FIG. 2A is a perspective view of the separator of FIG. 1.

FIG. 2B is an exploded view of the separator surface of FIG. 2.

FIG. 3A is a first perspective view of a composite electrode of FIG. 1.

FIG. 3B is a second perspective view of a composite electrode of FIG. 1.

FIG. 3C is a third perspective view of a composite electrode of FIG. 1.

FIG. 3D is a fourth perspective view of a composite electrode of FIG. 1.

FIG. 4 is a perspective view of a second embodiment coin cell implementation of the present novel technology.

FIG. 5 is an enlarged elevation view of a dendrite growing from an electrode surface of FIG. 1.

FIG. 6 is an exploded view of the surface of the separator of FIG. 1 as partially coated with FNC.

FIG. 7 is a process diagram a third embodiment of the present novel technology, showing of a method to form dendrite seeding material.

FIG. 8 is a process diagram a fourth embodiment of the present novel technology, showing of a method of controlling metal dendrite growth.

FIG. 9 is a process diagram a fifth embodiment of the present novel technology, showing of a method of extending the life a cell.

FIG. 10 is a process diagram a sixth embodiment of the present novel technology, showing of a method of producing an FNC-coated separator.

DETAILED DESCRIPTION

For the purposes of promoting and understanding of the principles of the novel technology and presenting its currently understood best mode of operation, reference will now be made to the embodiments illustrated in the drawings and specific language will be used to describe the same. It will nevertheless be understood that no limitation of the scope of the novel technology is thereby intended, with such alterations and further modifications in the illustrated novel technology and such further applications of the principles of the novel technology as illustrated therein being contemplated as would normally occur to one skilled in the art to which the novel technology relates.

As shown in FIGs. 1-10, the present novel technology relates to a rechargeable lithium metal electrochemical storage cell 10 having lithium metal electrodes 20. Referring to FIG. 1, a rechargeable lithium electrode cell 10 is shown with a Li metal cathode portion 12 and Li- metal anode portion 14. Separator 50 is positioned between the anode 14 and cathode 12. Separator 50 is typically coated with a layer 80 of functionalized nanocarbons particles 40. Separator 50 includes an anode facing side 53 and a cathode facing side 52, and is typically coated with a thin or very thin film 80 of the functionalized nanocarbon (FNC) particles 40, more typically about 0.1 μm thick, and typically oriented facing the surface 70 of the Li-metal electrode 20. Gap 26 is filled with an electrolyte 25 positioned between the Li-metal electrode 20 and the FNC-coated separator 60. The functionalized nanocarbon particles 40 typically have Li^+ ions immobilized on the surface 65 of the layer 80 of nanocarbon particles 40. The FNC film 80 is electrically connected to the Li-metal electrode 20. When the Li-metal electrode 20 is charged, Li dendrites 11 extend from the

surface 70 of the Li metal electrode 20 toward the FNC-coated separator 60.

Simultaneously, dendrites 55 extend from the surface 65 of the FNC film 80 toward the surface 70 of the Li-metal electrode 20. The dendrites 55 grow in the through plane direction 94 of the Li metal electrode 20 and FNC coated separator 60.

Referring to FIG. 5, growth of dendrites 11, 55 is driven by the potential difference (ΔE) between the tip (Et) 59 and the base (Eb) 57 of the respective dendrites 11, 55. With cycling, dendrites 11, 55 continue extending toward each other; eventually, the dendrites 11, 55 touch each other and the potential difference (ΔE) dendrite 11, 55 is approximately zero because the FNC film 80 and the Li-metal electrode 20 have the same potential. Consequently, dendrite 11, 55 growth is retarded or stopped along the through plane direction 94. In the subsequent cycles, dendrites 11, 55 may grow in a direction perpendicular to the major axis of the respective dendrite 11, 55 and parallel to the plane of the Li-metal electrode 20, also referred as the in-plane direction 84, which prevents dendrites 11, 55 from piercing through permeable or selectively permeable membrane 50, as shown in FIGs. 3A-3D. Eventually, a Li secondary surface 70 may form, from the intersection of the Li dendrites 11, 55. Thus, a composite Li metal electrode 20 is formed in which an Li electrode 20 is assembled with the thin carbon layer 80.

While the lithium is typically specifically discussed herein as the electrode metal, the storage cell 10 may alternately include other alkaline earth and/or alkaline metal elements and combinations thereof as the electrode materials.

Two types of cell exemplary configurations for exploiting the Li-metal dendrite/electrode system include a symmetric cell 400 in which a Li-metal electrode 420 is used as both the anode 414 and the cathode 412, having the configuration of

Li/polymer/Li (anode/electrolyte/cathode=A/E/C), enabling Li-dendrite mechanism study or Li-polymer battery systems; and an asymmetric cell 500 in which Li metal is the anode 514 and a different material is selected for the cathode 512, such as Li/polymer electrolyte/V2O5, Li/liquid electrolyte/graphite, Li/polymer electrolyte/graphite, and Li/polymer electrolyte/FePO4. The symmetric cell 400 provides a better medium for Li-metal dendrite growth and can accelerate the cycle testing, while the asymmetric cell 500 better approximates field applications.

Dendrite growth, as shown in FIG. 5, is fundamentally unavoidable because the metallurgical characteristics of Li-metal surfaces result in surface imperfections of Li-metal electrodes after the application of either mechanical stress or the plating/stripping cycles. While configurations known in the art focus solely on stopping dendrite 11 growth, the novel cell design 10 focuses on controlling the direction of the Li-metal dendrite 11,55 growth.

As described in FIG 9, one implementation 800 of the novel electrode 20 may have a carbon-coated layer of functionalized nanocarbon particles (FNC) 80 on a separator 50 that is positioned 801 in an electrolyte 25 and grows 803 Li dendrites 11, 55 simultaneously from the surface 51 of the Li metal electrode 20 and the surface 65 of the FNC coated separator 60. An electrolyte 25 is placed 802 in the gap 26 between the electrode 20 and FNC-coated separator 60. The dendrites 11, 55 grow 803 after repeated charging and discharging 804 of the cell 10. Dendrites 11,55 contact each other 805 and when contact occurs, the dendrites 11, 55 stop extending in the through plane direction 94 due to the zero potential difference that results from contact. The control of dendrite growth direction 800 occurs by contact 805 between the FNC coated separator dendrites 55 and

the electrode dendrites 11. After multiple combinations of dendrites 11, 55 the formation 806 of a Li- secondary Li surface 70 results.

The establishment of a zeroing potential difference gives the rechargeable Li-metal electrode 20 a high specific capacity, high cycle ability, and high safety. Accordingly, the rechargeable lithium metal electrode system 10 may be implemented in many kinds of Li batteries including Li-polymer, Li-air and Li-metal oxide cells and battery systems as well as any other cells or battery systems in which Li metal anodes 14 are used, and yield benefits for electronics, electric vehicles and hybrid electric vehicles, large-scale energy storage and the like.

Typically, a challenge for developing a high specific capacity and rechargeable Lithium metal electrode 20 for different Li batteries (i.e. Li polymer, Li-air and Li-ion, etc), has been stopping electrode dendrite 11 growth during the cycling 803. The Li-metal electrode 20 has an inherent metallurgic tendency to form dendrites 11, and dendrite 11 growth is driven by the potential difference between the base 57 and the dendrite tip 59. Thus, Li electrode dendrite 11 growth is unavoidable. However, the instant system 800 incorporates, rather than avoids, the dendrite growth mechanism.

In one embodiment, a rechargeable Li-metal electrode 220 is used in other Li battery systems, such as Li-polymer and Li-air and may be fabricated by coating the FNC layers 280 on the polymer electrolyte membranes 200, which are used as the electrolyte 225 in both Li-polymer batteries and Li-air batteries. These FNC-coated polymer electrolytes 225 are typically incorporated as the interlayer 280 and assembled into a soft packed Li-air cell 285. Such polymer electrolyte membranes 260 may include those of poly(ethylene oxide) (PEO), poly(vinylidene fluoride) (PVdF), poly(acrylonitrile) (PAN),

and the other polymer electrolytes, which are widely used for both Li-polymer batteries and Li-air batteries.

Additionally, many modes of producing the FNC coated separator 60 are available. The FNC layer 80 plays a role in the novel Li-metal electrode 20 because the immobilized Li⁺ ions 30 in the FNC layer 80 serve as 'seeds' 31 for Li-metal dendrite 55 formation on the FNC layer 80. The FNC layer 80 is typically porous, allowing the FNC aggregates to be bonded 605 together by the binder network 604 to form a rigid structure 606 to hold 607 the integrity of the layer 80. The layer 80 is typically very thin with four main properties: 1) good pore structure to facilitate the passage of Li⁺ ions therethrough, 2) high electric conductivity to reduce internal impedance, 3) high coverage of Li⁺ ions 30 over the nanocarbon surface 65 for easy formation of Li metal dendrites 55, and 4) good adhesion to a polymer separator 50 or a polymer electrolyte membrane. All of these properties are similar to those for the catalyst layer in the fuel cell, (i.e. a porous layer for gas and water diffusion, electric conductivity necessitated for gas reactions, SO₃⁻ coverage for proton conduction, and good adhesion of the catalyst layer on the polymer electrolyte membrane for durability). The thinner the FNC layer 80, the less the loss of specific capacity of the Li-metal electrode 20.

The morphology of the FNC layer 80 depends on how the layer is fabricated 601. Such techniques of applying 609 the layer 80 include (1) spraying, (2) machine blade-coating, (3) brush hand-painting, and the like. Carbons may be selected from sources including carbon blacks, nanographites, graphenes, and the like. It has been found that the higher the degree of graphitization, the higher the chemical stability. The nanocarbon particles 40 may be made from carbon black, which is inexpensive, but is an amorphous

structure rather than a graphite structure. Graphene may also be used and possesses unique properties such as high electronic conductivity, high modulus, and high surface area.

The morphology of the FNC layer 80 is also influenced by the ink formulation. To make a thin carbon layer, the first step is to mix 600 the carbon source with solvents to make a uniformly dispersed suspension 603. To form such a well-dispersed carbon ink, solvent type is carefully selected based on polarity (i.e. dielectric constant) and their hydrophobicity in order to match those of the carbon aggregates and the binders. This mixture 602 is also called 'ink formulation'. The type of carbons and solvents in an ink will affect the morphology of the thin FNC layer 80. The type of binder 33 also affects the adhesion of the carbon layer 80 on the separator 50. Typically, the binder 33 has a similar chemical structure to the separator/ electrolyte membrane 50 so that they can be fused together 605 through hot pressing or other techniques to form a well-bonded interface 62 between the carbon layer 80 and the separator/electrolyte membrane 50.

The immobilized Li^+ ions 30 over the surface of nanocarbon particles 40 serve as the 'seeds' 31 for Li dendrite 55 formation on the FNC-coated separator 60. Immobilization of the Li^+ ions 30 is carried out by formation 900 of a dendrite seeding material 61, such as by diazonium reaction or similar means 902 on an appropriate 901 carbon separator 50 to chemically attach an SO_3H group 902 onto the carbon surface 65, allowing the carbon separator 50 to become functionalized 903. Then, attached SO_3H exchanges 906 with Li^+ ions 30 to immobilize the Li^+ ions 30 onto the surface 65. Thus, a dendrite seeding material 61 is formed 907. The dendrite seeding material 61 is typically carbonaceous, but may also be a metal substrate, such as Li, Na, K, Al, Ni, Ti, Cu, Ag, Au, and combinations

thereof. The seeding material 61 may also be a functionalized metal substrate, such as a self-assembled monolayer structure comprised of Au with a thiol-terminated organic molecule that contains at least one function group, such as $\text{SO}_3\text{-M}^+$, COO-M^+ , and $\text{NR}_3\text{+X-}$, an electrically conductive organic polymer, such as polyacetylene, polyphenylene vinylene, polypyrrole, polythiophene, polyaniline, and polyphenylene sulfide, or a functionalized electrically conductive organic polymer, wherein the functional groups are chemically bound to the polymer. These materials 61 may be deposited using conventional physical deposition techniques, such as mechanical layering, or physical vapor deposition techniques, such as sputtering, or the like.

The novel technology allows attachment 903 of different functional groups to the carbon surface 65, such as through the diazonium reaction and the like. In this reaction, the functional group Y is attached 903 onto the carbon surface 65 through the introduction 904 of a diazonium salt $\text{XN}_2\text{C}_6\text{H}_4\text{-Y}$ (wherein $\text{Y}=\text{Sulfonate}$, $\text{SO}_3\text{-M}^+$, Carboxylate , COO-M^+ ; and Tertiary amine, $\text{NR}_3\text{+X-}$; etc.). The attachment of different chemical groups not only provides a platform for immobilizing Li^+ ions 30 at the FNC surface 65, but also changes the surface energy of the carbon particles which can be used as a tool for adjusting the surface hydrophobicity of the carbon film 80, and is helpful for ink formulation 603.

The adhesion 609 of the FNC layer to a separator/polymer electrolyte 50 influences the cycle life of the novel Li-metal electrode 20. A good interface 62 between the FNC layer 80 and the separator/electrolyte membrane 50 is typically formed 608. This mainly depends on the network of binders 33 in the FNC layer 80 and the techniques for the formation of the interface 62. Such a catalyst layer can withstand several thousand hours of long-term durability testing due, in part to the binder 33 in maintaining 607 the FNC layer

80 bound to the separator/electrolyte membrane 50. A TEM observation of such this catalyst/membrane interface 62 would show little or no delamination after approximately 2000 hours of durability testing. Hot pressing is one of techniques for fabrication, and the parameters of the hot pressing technique (i.e. temperature, pressure, and time) allow systematic control of the process.

The morphology (i.e. surface area, pore structure, and geometry) of the FNC layer 80 on the membrane 50 has a significant impact on the performance of the novel metal electrode 20. The FNC layer 80 porosimetry 81 (i.e. pore size, pore size distribution and pore volume) is a factor in controlling the direction of dendrite growth 700 because it influences the presence 705 of metal cations 30 on the FNC membrane surface 65 and the addition 703 of the dendrite seeding material 61. The pore structure typically allows metal ions 30 to pass through smoothly during cycling 704, but not to form dendrites inside the pores that would block the diffusion of the metal ions 30. Thus, determining 701 and production 702 of an appropriate FNC layer 80 with porosimetry 81 is useful in allowing for dendrite 11, 55 presence 706 and eventual formation 707 of a secondary metal layer 70. On the other hand, the FNC layer 80 has to adhere to a separator/ electrolyte membrane 50 and the diffusion barrier (if there is any) from the formed interface 62 should be minimized.

Typically, the specific capacity of the rechargeable metal electrode 20 may be affected by varying the thickness 89 of the FNC film 80 against the thickness 29 of the Li metal electrode 20. The examples herein relate to the novel technology and various embodiments, and are not intended to limit the scope of the present novel technology to those modes and embodiments discussed herein.

Example 1:

The effect of the different carbon-coated layers on the specific capacity of the Li metal composite electrode 20 was approximately calculated and is shown in Table 1. For instance, for the carbon-coated layer 80 with the 0.1 μm thickness, the corresponding specific capacity loss of Li metal electrode 20 is only 0.026%. Even for the thick FNC film 80, 4 μm , the corresponding loss of specific capacity is only 0.53%. Thus, the effects of the carbon-coated layer 80 on the specific capacity of the Li metal electrode 20 are negligible. The thin carbon-coated layer 80 retains the advantage of the high specific capacity of Li metal electrodes.

Table 1

Thickness of Carbon Film (μm)	Thickness of Li Metal Electrode(mm)	Reduction of Li Metal Electrode Specific Capacity (%)
0.1	0.75	0.0133
1	0.75	0.1332
2	0.75	0.1332
3	0.75	0.1332
4	0.75	0.5305

Effect of thickness of carbon film on the Li metal electrode specific capacity.

Therefore, carbon has been proven to be very stable in a wide potential window. The composite Li electrode having a very thin carbon film is very stable. Carbon black may be used in many battery systems (i.e. Zn/MnO₂), in particular, Li-ion batteries (as the anode) and Li-SOCl₂ batteries (as the carbon cathode).

Referring to FIG 4, The Li metal anode 14 was assembled together with a separator 350 (thickness=25 μm) coated with a thin nanocarbon layer 80 of functionalized carbon nanoparticles 340 ($\delta=3.2 \mu\text{m}$) and a LiPFeO₄ cathode 312 into a coin cell 300 configuration

using the electrolyte of 1.2 M LiPF_6 in ethylene carbonate/ethyl-methyl carbonate (EC:EMC=3:7). A coin cell using the same components, but without the nanocarbon coating layer 380, was used as a baseline for the comparison. One concern for using such a carbon coating layer 380 is whether the addition of the FNC layer 380 on the separator 350 would result in the increased internal impedance from the carbon layer 380 blocking the pores of the separator 350, thus hindering the diffusion of Li^+ ions 330 through and, consequently, reducing the power performance of the cell 300. However, it is clear that coating the carbon layer 380 on the separator 350 did not cause an increase in the internal impedance of the cell 300, but instead gave rise to a slight impedance reduction. The Li/FNC cell 300 possesses a slightly higher discharge voltage than the baseline Li cell. Even after five hundred cycles, the same trend was observed. Noise was observed for the baseline cell, which was attributed to the formation of dendrites 355. In addition, the same phenomenon of reduction of internal impedance has been observed during the charging process.

The cell 300 was not balanced for capacity, and the capacity of the cell 300 was limited by the LiPFeO_4 cathode 312; a much higher capacity of the cell 300 is expected if an appropriate high energy density cathode is used (such as a V_2O_5 aerogel or an air cathode). The Li metal electrode 314 using an FNC layer 380 showed excellent cycleability, approximately 84% capacity after 500 cycles. The estimated capacity decay rate of the novel Li metal electrode cell 300 after the first 45 cycles is only 0.026%/cycle. Based on this decay rate, the cycle life of such a cell can typically achieve at least 500, more typically at least 725 cycles, and still more typically at least 1000 cycles, with 80% capacity (death definition of a battery in electric vehicle (EV) applications). This decay rate (0.026%/cycle) of the novel Li metal electrode 320 in the coin cell 300 may be caused by the degradation of

the LiFePO_4 cathode 312 because the coin cells 300 are sealed in ambient atmospheric pressure, which may allow the introduction of moisture into the cell 300. The moisture reacts with LiPF_6 to produce HF, which can react with LiFePO_4 , causing the degradation. Therefore, the true decay rate of the novel Li metal electrode 320 should be much lower than 0.026%/cycle if the coin cell 300 is sealed, such as inside an Argon filled glovebox.

Example 2:

Referring to FIG. 6, an FNC-coated separator 60 was examined via SEM analysis after repeated cycling. Li metal dendrites 55 were observed on the surface 65 of the FNC-coated separator 60 facing the surface of the Li metal electrode 20. Moreover, the Li dendrites 55 formed a unitary layer instead of aggregating as loosely arranged dendrites. The thickness 89 of the FNC layer 80 was measured to be about 3 μm , while the Li dendrite 70 layer was around 20 μm thick. Referring to FIG 6, and to further illustrate the function of the FNC layer 80 for inducing Li metal dendrite 55 formation, the separator 50 was coated with an FNC layer 80 on half the area of the surface, while the other half was not coated. No dendrites 55 formed on the non-coated region of the separator 50. No Li dendrites 55 were found on the opposite side of the FNC-coated separator 50. Some large size particles (50 μm or more) were observed seen underneath the separator 50; these large particles likely originated from the SEM conducting paste used to adhere the sample of the separator 50 on the SEM aluminum disc.

In another embodiment, the layer 80 formed over the electrochemical separator 50 to enable dendritic growth toward the metal anode 14 is a thin metallic layer 80. The dendrites 55 growing from the separator 50 contact dendrites 11 growing from the metal

anode 14, shorting the circuit and thus preventing the dendrites 11 growing from the anode 14 toward the separator 50 to reach and pierce the separator 50. The anode 14 is typically lithium, but may likewise be sodium or the like. The metal layer 80 on the separator 50 is typically lithium, but may also be sodium or another electrically conductive metal, electrically conducting polymer, an organometallic matrix, functionalized electrically conducting polymer, or the like. More typically, the layer 80 is a non-reactive metal, such as Ni. The metal layer 80 on the separator 50 is typically formed thin enough such that its electrical resistivity is high, typically high enough such that the layer 80 is not easily electrically or otherwise degraded. Optionally, the thin metal layer 80 may be functionalized after deposition onto the separator 50.

While the novel technology has been illustrated and described in detail in the drawings and foregoing description, the same is to be considered as illustrative and not restrictive in character. It is understood that the embodiments have been shown and described in the foregoing specification in satisfaction of the best mode and enablement requirements. It is understood that one of ordinary skill in the art could readily make a nigh-infinite number of insubstantial changes and modifications to the above-described embodiments and that it would be impractical to attempt to describe all such embodiment variations in the present specification. Accordingly, it is understood that all changes and modifications that come within the spirit of the novel technology are desired to be protected.

CLAIMS

1. A method comprising:
 - forming an electrically conductive layer comprising a dendrite seeding material in an electrolyte solution disposed between a metal-containing electrode and an electrolyte permeable separator membrane;
 - growing metal dendrites from the dendrite seeding material toward the metal-containing electrode; and
 - contacting metal dendrites extending from the metal-containing electrode with metal dendrites extending from the dendrite seeding material;
 - wherein the electrolyte solution contains metal ions.
2. *The method of claim 1, wherein contact between metal dendrites extending from the metal-containing electrode with metal dendrites extending from the dendrite seeding material substantially stops growth of the contacting metal dendrites along a major axis.*
3. The method of claim 1, further comprising:
 - intertangling metal dendrites extending from the metal-containing electrode with metal dendrites extending from the dendrite seeding material; and
 - forming a lithium layer from intertangled metal dendrites.
4. The method of claim 1, further comprising producing the dendrite seeding material, wherein the producing the dendrite seeding material comprises:
 - functionalizing a carbon separator surface with a chemically-bound anion to produce a functionalized carbon separator surface;*
 - introducing a neutral metal salt to the functionalized carbon separator surface;
 - reacting the neutral metal salt to yield a metal cation and an anion; and
 - attracting the metal cation to the chemically-bound anion.

5. The method of claim 4, wherein metal cation comprises lithium, calcium, magnesium, sodium, potassium, or a combination containing lithium, calcium, magnesium, sodium, or potassium.
6. The method of claim 5, wherein the metal cation is a functionalized metal substrate containing a functional group comprising sulfonate, carboxylate, tertiary amine, diazonium salt, or combinations thereof.
7. The method of claim 1, further comprising:
determining a desired porosimetry value for the electrolyte permeable separator membrane;
wherein the electrolyte permeable separator membrane has a porosimetry configured for growing dendrites from the electrolyte permeable separator membrane; and
wherein the growing comprises:
introducing a metal cation gradient for moving metal cations from the metal-containing electrode to the electrolyte permeable separator membrane;
immobilizing a portion of the metal cations on an electrode-facing surface of the electrolyte permeable separator membrane; and
promoting directional growth of the metal dendrites from a membrane-facing side of the metal-containing electrode and from the electrode-facing surface of the electrolyte permeable separator membrane containing the dendrite seeding material.
8. The method of claim 7, further comprising, preventing dendrite growth in a through-plane direction of the electrolyte permeable separator membrane when the metal dendrites extending from the metal-containing electrode contact with metal dendrites extending from the dendrite seeding material.
9. The method of claim 1, wherein the electrolyte permeable separator membrane is selectively permeable.

10. The method of claim 9, wherein the electrolyte permeable separator membrane includes a layer of functionalized nanocarbon particles.
11. The method of claim 1, further comprising producing the electrolyte permeable separator membrane, wherein producing the electrolyte permeable separator membrane comprises:
- mixing a particulate carbon source with a plurality of solvents to form a suspension;
 - identifying a binding element to affect adhesion of suspended carbon particles to a permeable membrane;
 - applying the binding element to the permeable membrane to define an adhesive membrane;
 - applying the suspension to the adhesive membrane; and
 - forming an interface between the suspension and the permeable membrane.
12. The method of claim 11, wherein the suspension comprises plurality of functionalized nanocarbon particles.
13. The method of claim 11, wherein the particulate carbon source comprises carbon black, graphene, graphite, nanographite, or a combination containing carbon black, graphene, graphite, or nanographite.
14. The method of claim 11, wherein the binding element and the electrolyte permeable separator membrane have a substantially similar chemical composition.
15. The method of claim 11, wherein applying the suspension is accomplished by hot pressing, spraying, machine blade coating, brush painting, or a combination of processes that includes hot pressing, spraying, machine blade coating, or brush painting.
16. The method of claim 13, wherein the suspension is uniformly dispersed.
17. The method of claim 13, further comprising maintaining adhesion of the suspension by the binding element.

18. The method of claim 1, further comprising:
coating the electrolyte permeable separator membrane with a non-reactive metal coating;
functionalizing the non-reactive metal coating to yield a functionalized non-reactive metal coating; and
positioning the electrolyte solution between the metal-containing electrode and the functionalized non-reactive metal coating;
wherein the growing comprises:
introducing a metal cation gradient from a metal electrode through the electrolyte permeable separator membrane;
immobilizing a portion of the metal cations on the functionalized non-reactive metal coating; and
promoting the growing of the metal dendrites through the electrolyte solution from the metal-containing electrode and from the electrolyte permeable separator membrane containing the dendrite seeding material.
19. A method comprising
forming an electrically conductive layer comprising a dendrite seeding material on an electrolyte permeable separator membrane in an electrolyte solution containing a metal-containing electrode, the electrolyte solution containing metal ions.
20. The method of claim 19, further comprising causing metal dendrites to grow from the dendrite seeding material toward the metal-containing electrode, and from the metal-containing electrode toward the electrolyte permeable separator membrane, wherein dendrite growth continues at least until dendrites from dendrite seeding material contact dendrites from the metal-containing electrode.